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May 15, 2009

THIS IS TO CERTIFY THAT ANNEXED IS A TRUE COPY FROM THE RECORDS OF THIS OFFICE OF THE FILE WRAPPER AND CONTENTS OF:

APPLICATION NUMBER: 07/956,653 FILING DATE: October 02, 1992 PATENT NUMBER: 5,597,378 ISSUE DATE: January 28, 1997

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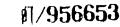
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By Authority of the Under Secretary of Commerce for Intellectual Property and Director of the United States Paten<u>t</u> and Trademark Office

P.

Certifying Officer

AT OUP ART UNI EL ASS SUBCLASS 1653 -13/92 606 301 28 -ROLÈ Ē. JERVIS, ATHERTON, CA. JAMES a . 5,190,546 THIS APPLN IS A DIV OF d7/62,243 04/09/91 PAT VERIFIED 07/252,019 09/27/88 PAT. 07/177,817 03/30/88 ABN 97/047,824 95/08/87 ABN 5,067,957 WHICH IS A DIV OF OSK. WHICH IS A CON OF WHICH IS A CON OF WHICH IS A CON OF 06/865,703 05/21/86 PAT 4,665,906 -06/541,852 10/14/83 ABN WHICH IS A CON OF. 237 POPEIGN/PCT APPLICATIONS********* VERIFIED DE NOTE - DISCLAIMER The term of this paterit subsequent to 5/19/04 has been disclaimed METEN FILING LICENSE GRANTED 10/28/92 eson priority claimed yes turn USC 119 conditions met yés turn turned Examiner's initiais FILING FEE STATE OR SHEETS COUNTRY DRWGS. TOTAL INDEP. ATTORNEY'S AS FILED 190.00 -ZÆΥ G. SHELDON ELDEN & MAK 25 S. LAKE AVENUE - 9TH FLOOR Sadena, CA 91101 ISSUE FEL IN FILE L DEVICES INCORPORATING SIM ALLOY ELEMENTS U.S. DEPT. of COMM. Pat. & TM Office - PTO-436L (rev. 10-78 TION ANCE MAILED 129129 CLAIMS ALLOWED PREPARED FOR ISSUE David J. Kenerly Print Claim ofal Claims Assistant Examiner Docket Cir 0 ROBERT A. HAFER SUE EEE 2 Date Paid S.P.E, ART UNIT 331 Primary Exa ISSUE CLASSIFICATION SSUE ATCH Subclass NUMBER . 78 6010 VANNING: The information disclosed nervice say be restricted. Unestitutized disclosure prohibited by the United States Code Title 35, Sections 122, 181 and 368. Possession outside the U.S. Patent & Trademark Office is restricted to authorized e nló and contractors only



PATENT APPLICATION SERIAL NO.

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PTO-1556

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U.S. DEPARTMENT OF COMMERCE PATENT AND TRADEMARK OFFICE <u>FEE RECORD SHEET</u>

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2	1992 5		
N. B	TRADE IN THE UNITED STATES PA	ATENT AND TRADEMARK OFFICE	
		Docket No. 9438	
		Anticipated Classification of this application:	
		Class Subclass	
		Prior application:	1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 - 1997 -
		Examiner: <u>K. Rooney</u> 3301	
		Art Unit:	- · ·
	Box Patent Application Commissioner of Patents and Tradema Washington, D.C. 20231	ırks	• •
	TRANSMITTAL OF F	ILING UNDER 37 CFR 1.60(b)	
•	WARNING: A c-i-p (continuation-in-part) cannot WARNING: Filing under 37 CFR 1.60 is permitte in the prior application.	d only if filed by the same or less than all the inventor	ors named
•		Inited States stage of an International Application re (4).	equires an
	of the new application are drawn	nay be finally rejected in the first Office action where to the same invention claimed in the earlier applic ected on the grounds or art of record in the next Offic application. MPEP § 706.07(b).	ation and
	This is a request for filing a		
	Continuation		• ·
	X Divisional		
	application under 37 CFR 1.60, of pending	prior application	
	7 (02 242	d on $\frac{4/9/91}{1}$	
	James E. Jervis	(date)	
	of		
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	(International Devices Incorporation	inventor(s)) G SIM ALLOY ELEMENTS # of invention)	
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TO THE REPORT OF A DESCRIPTION OF

NOTE: 37 CFR 1.60 permits the omission of a declaration only if the prior application was complete as set forth in 37 CFR 1.51(a), namely, the prior application comprised at least (1) a specification, including a claim or claims; (2) a declaration; (3) drawings when necessary; and (4) the prescribed filing fee. Accordingly, as presently worded, 37 CFR 1.60 does not permit this procedure to be used where the prior application is pending but only the processing and retention fee required by 37 CFR 1.21(l) is paid or where the declaration was not filed.

1. Copy of Prior Application as Filed Which is Attached

- NOTE: Under 37 CFR 1.60 practice signing and execution of the application by the applicant may be omitted provided the copy is supplied by and accompanied by a statement by the applicant or his or her attorney or agent that the application papers comprise a true copy of the prior application as filed and that no amendments referred to in the declaration filed to complete the prior application introduced new matter therein.
- NOTE: This statement need not be verified if made by an attorney registered to practice before the PTO. (37 CFR 1.60(b)).
 - I hereby verify that the attached papers are a true copy of what is shown in my records to be the above identified prior application, including the oath or declaration originally filed (37 CFR 1.60).

The copy of the papers of prior application as filed which are attached are as follows:

- X 19 page(s) of specification
- page(s) of claims
- X _____ page(s) of abstract
- [X] = 1 sheet(s) of drawing

(Also complete part 6 below if drawings are to be transferred)

 \underline{X} _____ pages of declaration and power of attorney

If the copy of the declaration being filed does not show applicant's signature indicate thereon that it was signed and complete the following:

- in accordance with the indication required by 37 CFR 60(b) my records reflect that the original signed declaration showing applicant's signature was filed on ______
- the amendment referred to in the declaration filed to complete the prior application and I hereby state, in accordance with the requirements of 37 CFR 1.60(b), that this amendment did not introduce new matter therein.

2. Amendments

- WARNING: "The claim of a new application may be finally rejected in the first Office action in those situations where (1) the new application is a continuing application of, or a substitute for, an earlier application, and (2) all the claims of the new application (a) are drawn to the same invention claimed in the earlier application, and (b) would have been properly finally rejected on the grounds or art of record in the next Office action if they had been entered in the earlier application." MPEP § 706.07(b).
 - Cancel in this application original claims ______ of the prior application before calculating the filing fee. (At least one original independent claim must be retained for filing purposes.)
 - A preliminary amendment is enclosed. (Claims added by this amendment have been properly numbered consecutively beginning with the number next following the highest numbered **original** claim in the prior application.)
- NOTE: Only amendments reducing the number of claims or adding a reference to the prior application (Rule 1.78(a)) will be entered before calculating the filing fee and granting the filing date. 37 CFR 1.60(b).
- NOTE: "When filing under Rule 1.60 retain at least one original claim from the patent application to assure a complete application." Notice of March 3, 1986 (1064 O.G. 37-38).

(37 CFR 1.60(b) [4-3]-page 2 of 8)

3. Petition for Suspension of Prosecution for the Time Necessary to File an Amendment

NOTE: Where it is possible that the claims on file will give rise to a first action final for this continuation application and for some reason an amendment cannot be filed promptly (e.g., experimental data is being gathered) it may be desirable to file a petition for suspension of prosecution for the time necessary).

(check the next item, if applicable)

There is provided herewith a Petition To Suspend Prosecution For The Time Necessary to File An Amendment (New Application Filed Concurrently).

4. Information Disclosure Statement

(check this item, if applicable)

X An information disclosure statement is submitted herewith.

5. Fee Calculation (37 CFR 1.16)

6. S

	CLAIN	IS AS F	ILED	(INCLUDING	PRELIMINARY	AMENDMENT)	
Number filed	Numb	er Extra	1	Rate	Basic Fee 37 CFR 1.16(\$690.00	a)	
Total Claims (37 CFR 1.16(c))	52 - 20=	32	×	\$ 20.00	640		
Independent Claims (37 CFR 1.16(b))	8 -3=	5	×	\$ 72.00	360		
Multiple dependent claim(s) (37 CFR 1.16(d))	, if any	5	×	\$220.00	1100		

Fee for extra claims is not being paid at this time. (37 CFR 1.16(d))

NOTE: If the fees for extra claims are not paid on filing they must be paid or the claims cancelled by amendment, prior to the expiration of the time period set for response by the PTO in any notice of fee deficiency. 37 CFR 1.16(d).

	Filing Fee Calculation	\$_2,190
mall	Entity Status	
	A verified statement that this filing is by a small entity:	
	is attached	

has been filed in the parent application and such status is still proper and desired (37 CFR 1.28(a))

Filing Fee Calculation (50% of above) \$.

NOTE: Any excess of the full fee paid will be refunded if a verified statement is filed within 2 months of the date of timely payment of a full fee then the excess fee paid will be refunded on request. 37 CFR 1.28(a).

NOTE: 37 CFR 1.28(a), last sentence states: "Applications filed under § 1.60 or § 1.62 of this part must include a reference to a verified statement in a parent application if status as a small entity is still proper and desired."

(37 CFR 1.60(b) [4-3]-page 3 of 8)

7. Drawings

WARNING: Do not check the following box if prior case is not to be abandoned.

- ☐ Transfer the drawings from the prior application to this application and, subject to item 17 below, abandon said prior application as of the filing date accorded this application. A duplicate copy of this request is enclosed for filing in the prior application file. (May only be used if signed by (1) applicant, (2) assignee of record or (3) attorney or agent of record authorized by 37 CFR 1.138 and before payment of issue fee.)
- NOTE: "A registered attorney or agent acting under the provisions of § 1.34(a), or of record, may also expressly abandon a prior application as of the filing date granted to a continuing application when filing such a continuing application." 37 CFR 1.138.
 - Transfer the following sheet(s) of drawing from the prior application to this application

NOTE: Transferred sheets must be cancelled in prior application. 37 CFR 1.88.

- A copy of the amendment cancelling these sheets of drawing in the prior application is attached
- New drawings are enclosed
 - formal
 - informal
- WARNING: DO NOT submit original drawings. A high quality copy of drawings should be supplied when filing a patent application. The drawings that are submitted to the Office must be on strong, white, smooth, and non-shiny paper and meet the standards of § 1.44. If corrections to the drawings are necessary, they should be made to the original drawings and a high-quality copy of the corrected original drawing then submitted to the Office. Only one copy is required or desired. Comments on proposed new 37 CFR 1.84. Notice of March 9, 1988 (1090 O.G. 57-62).
- NOTE: "Identifying indicia such as the serial number, group art unit, title of the inventor, attorney's docket number, inventor's name, number of sheets, etc. not to exceed 2% inches (7.0 cm.) in width may be placed in a centered location between the side edges within three fourths inch (19.1 mm.) of the top edge. Either this marking technique on the front of the drawing or the placement, although not preferred, of this information and the title of the invention on the back of the drawings is acceptable." Proposed 37 CFR 1.84(1). Notice of March 9, 1988 (1090 O.G. 57-62).

8. Priority-35 U.S.C. 119

		Prio	rity	of	application	n serial	no. in	0	/	<u>.</u>			filed	. on
		is cl	aime	d'un	der 35 U.S.C	. 119.	"'				(co	untry)		
			The 0		fied copy ha		•			•••	ation	serial n	ю.	
			The	certi	fied copy wi	il follow.								
_	Relate	e Bad	:k3	35 U.:	S.C. 120									

- Amend the specification by inserting before the first line the sentence:
 - "This is a
 - continuation
 - divisional
 - of copending application(s)
 - Serial number 0 /_____ filed
 - International Application ______ filed on ______ and which designated the U.S."

(37 CFR 1.60(b) [4-3]-page 4 of 8)

NOTE: The proper reference to a prior filed PCT application which entered t serial number and the filing date of the PCT application which design										
10. Inventorship Statement										
NOTE: If the continuation or divisional application is filed by less than all the inventors named in the prior appli- cation a statement must accompany the application when filed requesting deletion of the names of the person or persons who are not inventors of the invention being claimed in the continuation or divisional application. 37 CFR 1.60(b) [emphasis added].										
(complete appropriate items (a) and (b)))									
(a) With respect to the prior copending U.S. application for claims benefit under 35 USC 120 the inventor(s) in the										
(complete applicable item below)										
X the same										
less than those named in the prior application a following inventor(s) identified above for the prior										
(Type name(s) of inventor(s) to be deleted)										
(b) The inventorship for all the claims in this application are										
The same										
not the same, and an explanation, including the claims at the time the last claimed invention was										
11. Assignment										
The prior application is assigned of record to Raychem Corporation										
an assignment of the invention to										
· ·	· · · ·									
is attached. A separate "ASSIGNMENT COVER L NEW PATENT APPLICATION" is also attached.	ETTER ACCOMPANYING									
NOTE: "If an assignment is submitted with a new application, send two sep tion and one for the assignment." Notice of May 4, 1990 (1114 O.G. 2										
12. Fee Payment Being Made At This Time										
Not Enclosed										
 No filing fee is submitted. (This and the surch 1.16(e) can be paid subsequently). 	harge required by 37 CFR									
Enclosed										
X basic filing fee	\$ 690									
recording assignment										
(\$40.00; 37 CFR 1.21(h))	\$									
processing and retention fee										
(\$130.00; 37 CFR 1.53(d)										
and 1.21(f)	\$									
X Additional Claims	\$_1500									

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(37 CFR 1.60(b) [4-3]—page 5 of 8)

37 CFR 1.21(I) establishes a fee for processing and retaining any application which is abandoned for NOTE failing to complete the application pursuant to 37 CFR 1.53(d) and this, as well as the changes to 37 CFR 1.53 and 1.78 indicate that in order to obtain the benefit of a prior U.S. application, either the basic filing fee must be paid or else the processing and retention fee of § 1.21(I) must be paid within 1 year from notification under § 53(d).

2,190

- Method of Payment of Fees 13.
 - \overline{X} enclosed is a check in the amount of \$ 2,190
 - charge Account No. _ in the amount of \$

Total fees enclosed

- A duplicate of this request is attached.
- NOTE: Fees should be itemized in such a manner that is clear for which purpose the fees are paid. 37 CFR 1.22(b).

14. Authorization To Charge Additional Fees

WARNING: If no fees are being paid on filing do not complete this item.

- WARNING: Accurately count claims, especially multiple dependent claims, to avoid unexpected high charges if extra claim charges are authorized.
 - X The Commissioner is hereby authorized to charge the following additional fees which may be required by this paper and during the entire pendency of the application to Account No. _19-2090
 - 37 CFR 1.16 (a), (f) or (g) (filing fees)
 - 37 CFR 1.16 (b), (c) and (d) (presentation of extra claims)
- NOTE: Because additional fees for excess or multiple dependent claims not paid on filing or on later presentation must only be paid or these claims cancelled by amendment prior to the expiration of the time period set for response by the PTO in any notice of fee deficiency (37 CFR 1.16(d)) it might be best not to authorize the PTO to charge additional claim fees, except possibly when dealing with amendments after final action.
 - 37 CFR 1.17 (application processing fees)
- WARNING: While 37 CFR 1.17(a), (b), (c) and (d) deal with extensions of time under § 1.136(a) this authorization should be made only with the knowledge that: "Submission of the appropriate extension fee under 37 CFR 1.136(a) is to no avail unless a request or petition for extension is filed." [emphasis added]. Notice of November 5, 1985 (1060 O.G. 27).
 - 37 CFR 1.18 (issue fee at or before mailing Notice of Allowance, pursuant to 37 CFR 1.311(b)).
- NOTE: Where an authorization to charge the issue fee to a deposit account has been filed before the mailing of a Notice of Allowance, the issue fee will be automatically charged to the deposit account at the time of mailing the notice of allowance. 37 CFR 1.311(b)).
- NOTE: 37 CFR 1.28(b) requires "Notification of any change in status resulting in loss of entitlement to small entity status must be filed in the application . . . prior to paying or at the time of paying . . . issue fee." From the wording of 37 CFR 1.28(b): (a) notification of change of status must be made even if the fee is paid as "other than a small entity" and (b) no notification is required if the change is to another small entity.

15. Power of Attorney

The power of attorney in the prior application is to Jeffrey G. Sheldon

27,953 Reg. No.

Attorney

а.

CAREAGE STREET

The power appears in the original papers in the prior application.

(37 CFR 1.60(b) [4-3]-page 6 of 8)

b. 🕅

Since the power does not appear in the original papers, a copy of the power in the prior application is enclosed.

- c. A new power has been executed and is attached.
- d. X Address all future communications to Jeffrey G. Sheldon, Esq.
- (818) 796-4000

Sheldon & Mak

225 S. Lake Avenue - 9th Floor

Pasadena, CA 91101

(Item d may only be completed by applicant, or attorney or agent of record)

16. Maintenance of Copendency of Prior Application

(This item must be completed and the papers filed in the **prior** application if the period set in the prior application has run)

- A petition, fee and response has been filed to extend the term in the pending prior application until ______.
- NOTE: The PTO finds it useful if a copy of the petition filed in the prior application extending the term for response is filed with the papers constituting the filing of the Continuation Application. Notice of November 5, 1985 (1060 O.G. 27).
 - A copy of the petition for extension of time in the prior application is attached.

17. Conditional Petition for Extension of Time in Prior Application

(complete this item and file conditional petition in the prior application if previous item not applicable)

- a conditional petition for extension of time is being filed in the pending parent application.
- NOTE: The PTO finds it useful if a **copy** of the petition filed in the prior application extending the term for response is filed with the paper constituting the filing of the continuation application. Notice of November 5, 1985 (1060 O.G. 27).
 - A copy of the conditional petition for extension of time in the prior application is attached.

18. Abandonment of Prior Application (if applicable)

WARNING: (Do not complete this item if the application being filed is a divisional of the prior application which is not being abandoned)

- NOTE: "A registered attorney or agent acting under the provisions of § 1.34(a), or of record, may also expressly abandon a prior application as of the filing date granted to a continuing application when filing such a continuing application." 37 CFR 1.138.
 - Please abandon the prior application at a time while the prior application is pending or when the petition for extension of time or to revive in that application is granted and when this application is granted a filing date so as to make this application copending with said prior application.

(37 CFR 1.60(b) [4-3]-page 7 of 8)

19. Notification In Parent Application of the Filing of This Continuation Application

☐ A notification of the filing of this continuation is being filed in the parent application from which this application claims priority under 35 USC § 120.

I hereby declare further that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, under Section 1001 of Title 18 of the United States Code, and that such willful false statements may jeopardize the validity of the application or any patent issuing thereon.

Type or

Date Sheldon & Mak 225 S. Lake Avenue Pasadena, CA 91101

P.O. Address of Signatory

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には、日本のないないないないないです。日本のでは、

Inventor

Signatur

Assignee of complete interest

K Attorney or agent of record

Person authorized to sign on behalf of assignee

Jeffrey G. Sheldon

me of person si

anina

Tel. No.: (818) 796-4000

Filed under Rule 34(a)

Reg. No. 27,953 (if applicable)

(Complete the following if applicable)

Raychem Corporation

Type name of assignee 300 Constitution Drive

Address of assignee

Menlo Park, CA 94025

Title of person authorized to sign on behalf of assignee

Assignment recorded in PTO on _____

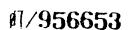
Reel _____ Frame ____

Plus ASSIGNMENT (DOCUMENT) COVER LETTER ACCOMPANYING NEW PATENT APPLICATION

(37 CFR 1.60(b) [4-3]-page 8 of 8)

U.S. Serial No:

5





MP0884-US1

Edwards Exhibit 1033, p. 12

CONFORMED COPY

MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

James E. Jervis

ABSTRACT OF THE DISCLOSURE

Medical devices which are currently proposed to use elements made from shape memory alloys may be improved by the use of stress-induced martensite alloy elements instead. The use of stress-induced martensite decreases the temperature sensitivity of the devices, thereby making them easier to install and/or remove.



MP0884-US1

BACKGROUND OF THE INVENTION

Field of the Invention

This invention relates to medical devices incorporating shape memory alloys, and to improvements therein.

- 2 -

5 Introduction to the Invention

Materials, both organic and metallic, capable of possessing shape memory are well known. An article made of such materials can be deformed from an original, heat-stable configuration to a second, heat-unstable

10 configuration. The article is said to have shape memory for the reason that, upon the application of heat alone, it can be caused to revert, or to attempt to revert, from its heat-unstable configuration to its original, heat-stable configuration, i.e. it "remembers" 15 its original shape.

Among metallic alloys, the ability to possess shape memory is a result of the fact that the alloy undergoes a reversible transformation from an austenitic state to a martensitic state with a change in temperature. This transformation is sometimes referred to as a thermoelastic martensitic transformation. An article made from such an alloy, for example a hollow sleeve, is easily deformed from its original configuration to a new configuration when cooled below the temperature at which the alloy is transformed from the austenitic state to the martensitic state.

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Edwards Exhibit 1033, p. 13

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The temperature at which this transformation begins is usually referred to as M_s and the temperature at which it finishes M_f . When an article thus deformed is warmed to the temperature at which the alloy starts to revert back to austenite, referred to as A_s (A_f being the temperature at which the reversion is complete) the deformed object will begin to return to its original configuration.

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Many-shape memory alloys (SMAs) are known to display stress-induced martensite (SIM). When an SMA sample exhibiting stress-induced martensite is stressed at a temperature above M_s (so that the austenitic state is initially stable), but below M_d (the maximum temperature at which martensite formation can occur even under stress) it first deforms elastically and then, at a critical stress, begins to transform by the formation of stress-induced martensite. Depending on whether the temperature is above or below A_s , the behavior when the deforming stress is released differs. If the temperature is below A_s , the stress-induced martensite is stable; but if the temperature is above A_s , the martensite

is unstable and transforms back to austenite, with the sample returning (or attempting to return) to its original shape. The effect is seen in almost all alloys which exhibit a thermoelastic martensitic transformation, along with the shape memory effect. However, the extent of the temperature range over which SIM is seen and the stress and strain ranges for the effect vary greatly with the alloy.

In copending and commonly assigned U.S. Patent Applicnow U.S. Patent Applicnow U.S. Patent no 4503,767 to Quin, the disclosure of which is incorporated herein by reference, a nickel/titanium/ vanadium alloy having SIM over a wide temperature range is disclosed.

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Shape memory alloys have found use in recent years in, for example, pipe couplings (such as are described in U.S. Pat. Nos. 4,035,007 and 4,198,081 to Harrison and Jervis), electrical connectors (such as are described in U.S. Pat. No 3,740,839 to Otte & Fischer), switches (such as are described in U.S. Patent No. 4,205,293), actuators, etc.

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Various proposals have also been made to employ shape memory alloys in the medical field. For example, U.S. Pat. No. 3,620,212 to Fannon et al. proposes the use of an SMA

10 intrauterine contraceptive device, U.S. Pat. No. 3,786,806 to Johnson et al. proposes the use of an SMA bone plate, U.S. Pat. No. 3,890,977 to Wilson proposes the use of an SMA element to bend a catheter or cannula, etc.

These medical SMA devices rely on the property of shape ¹⁵ memory to achieve their desired effects. That is to say, they rely on the fact that when an SMA element is cooled to its martensitic state and is subsequently deformed, it will retain its new shape; but when it is warmed to its austeniticstate, the original shape will be recovered.

However, the use of the shape memory effect in medical applications is attended with two principal disadvantages. First, it is difficult to control the transformation temperatures of shape memory alloys with accuracy as they are usually extremely composition-sensitive, although various techniques have been proposed (including the blending by founder power, metallurgy of already-made alloys of differing transformation temperatures: see U.S. Pat. No. 4,310,354 to fountain et al.). Second, in many shape memory alloys there

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is a large hysteresis as the alloy is transformed between austenitic and martensitic states, so that reversing of the state of an SMA element may require a temperature excursion of several tens of degrees Celsius. The combination of these factors with the limitation that (a) it is inconvenient to have to engage in any temperature manipulation, and (b) human tissue cannot be heated or cooled beyond certain relatively narrow limits (approximately 0° - 60°C for short periods) without suffering temporary or permanent damage is . expected to limit the use that can be made of SMA medical devices. It would thus be desirable to develop a way in which the advantageous property of shape memory alloys, i.e. their ability to return to an original shape after relatively substantial deformation, could be used in medical devices without requiring the delicacy of alloying control and/or the temperature control of placement or removal . needed by present shape memory alloy devices.

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DESCRIPTION OF THE INVENTION

Summary of the Invention

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I have discovered that if, in a medical device containing a shape memory alloy element which uses the shape memory property of that elloy, an element which shows the property of stress-induced martensite is used instead, an improved device results.

Accordingly, this invention provides a medical device intended for use within a mammalian body, or in such proximity to a mammalian body that the device is substantially at body temperature, which device comprises a shape memory alloy element, the improvement in which comprises the substitution of an alloy element which displays stress-induced martensite at said body temperature for the shape memory alloy element.



Edwards Exhibit 1033, p. 16

Brief Description of the Drawing

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Figures 1 and 2 illustrate the stress-strain behavior of an alloy which exhibits constant stress versus strain behavior due to stress-induced martensite.

Detailed Description of the Preferred Embodiments

The invention will be discussed first by introducing the concept of stress-induced martensite and the effect achievable by its use, and then by examples showing how SIM alloy elements can be substituted for conventional SMA elements in medical devices to achieve the beneficial effect of the invention.

The Figures illustrate the phenomenon of stressinduced martensite by means of stress-strain curves. In both Figure 1 and Figure 2, the alloy is at a temperature between M_s and M_d so that it is initially austenitic; and it will be assumed for the puposes of this discussion that M_s is equal to M_f , and A_s equal to A_f . Figure 1 shows the case when the temperature is below A_s , so that any martensite formed by the applied stress is stable; while Figure 2 shows the case where the temperature is above A_s , so that austenite is the only stable phase at zero stress.

In Figure 1, when a stress is applied to the alloy, it deforms elastically along the line DA. At a critical applied stress, $\sigma_{\rm M}$, the austenitic alloy begins to transform to (stress-induced) martensite. This transformation

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takes place at essentially constant stress until the alloy becomes fully martensitic at point B. From that point on, as further stress is applied, the martensite yields first elastically and then plastically (only elastic deformation is shown at point C). When the stress is released, the martensite recovers elastically to point D, at which there is zero residual stress, but a non-zero residual strain. Because the alloy is below As, the deformation is not recoverable until heating above A results in a reversion 10 to austenite. At that point, if the sample is unrestrained, the original shape will be essentially completely recovered: if not, it will be recovered to the extent permitted by the restraint. However, if the material is then allowed to re-cool to the original temperature at which it was deformed 15 (or a temperature where SIM behavior of this type is seen), the stress produced in the sample will be constant regardless of the strain provided that the strain lies within the "plateau" region of the stress-strain curve. That is, for a Strain between EB and the strain will be om. This

20 means that a known, constant force (calculable from $_{\sigma M}$) can be applied over a wide (up to 5% or more for certain Ni/Ti alloys) strain range. Thus, though this resembles the conventional shape memory effect, because the alloy shows 51M and is below A a constant force can be achieved.

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In Figure 2, when a stress is applied to the alloy, it deforms elastically along line DA, then by SIM along line AB, and by deformation of the martensite to point C, just as in Figure 1. However, the stress-strain behavior on unloading is significantly different, since the alloy is above A_{s}

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and the stable phase is therefore austenite. As the stress is removed, the alloy recovers elastically from C to D: then, at a critical stress, c , the alloy reverts to austenite without requiring a change in temperature. Thus reversion occurs at essentially constant stress. Finally if the stress is removed from the reverted austenite, it recovers elastically along line ED. The recoverable deformation associated with the formation and reversion of stressinduced martensite has been referred to as pseudoelasticity. While σ_{M} may be comparatively high, e.g. 50 ksi, σ_{Λ} is 10 usually substantially lower; e.g. less than 10 ksi; thereby creating a constant-force spring with an effective working range of about 5% ($c_{\rm H}$ - $c_{\rm A}$). The shape change available in the SMA is thus mechanically, rather than thermally, 15 actuated and controlled, permitting a greater control over a device incorporating it..

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Suitable alloy for this invention i.e. those displaying stress-induced martensite at temperatures near mammalian body temperature (35°-40°C), may be selected from known SMAs by those of ordinary skill in theart, having regard to this disclosure by tsting for the existence of the SIM effect at the desired temperature. A particularly preferred alloy is the nickel/titenium/vanadium alloy of U.S. Patent Application No. (Bocket No. HPOG73-UST), referred to previously.

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DIT The invention will now be discussed in detail by some Examples of the use of an SIM alloy.

Heart Valves Example I.

Akins, in U.S. Patent No. 4,233,69D, the disclosure of which is incorporated herein by reference, describes the use of a shape memory alloy ring to hold a sewing cuff to the body of an artifical heart valve. The ring is made in the austenstic phase, cooled to the martensitic phase, deformed, placed around the valve body, and heated or allowed to warm to cause reversion to the austenitic phase and recovery of the ring into engagement with the valve body.

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However, this technique has not found commercial acceptance. Present medical technique requires that the valve body be capable of being rotated relative to the cuff, thereby enabling the surgeon to set the rotational orientation of the valve after it has been sewn into place. This is desirable because the techniques used make it difficult to visualize or accomplish optimal orientation during initial placement.

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In order to accomplish the desired torque control to permit the desired rotation and yet ensure a firm hold of the cuff on the valve body, precise control of the pressure exerted on the valve body by the ring is needed. This is difficult because there are substantial manufacturing tolerances in the valve body which may be made, for example, of pyrolytic graphite or ceramics, etc. Because the austenite stress-strain curve is extremely steep, it is not considered practical to use the simple shape memory technique proposed by Akins. Indeed, Akins does not even address the issue of rotation of the cuff with respect to the valve body.

However, if an SIM alloy is used instead of conventional shape memory, the process may be considerably simplified.

First, if the alloy has a stress-strain curve like that of Figure 1, the alloy ring may be made just as for Akins. The ring is then expanded from its initial austenitic state by the formation of SIM. When the ring is placed about the valve body, it needs only to be heated above A_f and allowed to cool to its original temperature for the ring to

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engage the valve body with a constant force, even if the valve body has a deviation from the specified size. The torque may thus be controlled to the desired level despite manufacturing tolerances.

Second, if the alloy has a stress-strain curve like that of Figure 2, the ring may be expanded, placed over the valve body, and the stress released all at the same temperature. Because the austenitic phase is stable, the stress-induced

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martensite spontaneously reverts to austenite until recovery 10 is restrained by the ring engaging the valve body. Because the reversion to austenite takes place at constant stress, a constant force (and hence constant torque) may be obtained regardless of manufacturing tolerances. Close temperature control is not required, either; and the fact that the

patient in a heart valve replacement operation is conventionally cooled as much as 15°C or so below normal body temperature does not affect the operation of the ring.

To control the torque at a sufficiently low level, it may be desirable for the alloy ring to be other than a 20 solid ring, such as, for example, a continuous helical spring, a flat zigzag spring, etc. Such variations permit the achievement of a greater range of movement with constant force and a reduction in the force exerted by the ring on the value body, since the ring recovers in a bending mode 25 rather than in tension.

Example II. Catheters And Cannulas

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Wilson, in U.S. Patent No. 3,890,977, the disclosure of which is incorporated herein by reference, discloses a catheter or cannula (both being included hereinafter in the word "catheter") made of, or containing, an SMA element to cause all or a portion of the catheter to deploy in a useful form once introduced into a living body.

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However, again this device has not been commercialized.
Possible defects of the device which have prevented commercialization include (i) the inability to slowly emplace the catheter in a desired position when the transition temperature of the alloy is below body temperature (since the SMA element will attempt to revert to its original shape as it reaches body temperature), thus limiting the ability of the
physician to place the device carefully and precisely; or

- alternatively, if the transition temperature of the alloy is above body temperature, the requirement that the device by heated to a temperature above body temperature to cause recovery and that the device be placed so as not to change
- ²⁰ shape again when it re-cools (since the body temperature is below the transition temperature); (ii) the inability to remove the device easily: and (iii) the need for controlled temperature storage to prevent premature reversion to austenite of the SMA, with consequent shape change.
- 25 The issue of removal of a catheter is especially significant, and not addressed by Wilson. Consider, for example, a tracheal puncture catheter. This should be

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straight for easy insertion into the trachea through a puncture into the front of the neck, but should curve after insertion so that the flow of air or oxygen through the catheter passes axially down the trachea rather than impinging on the surface of the trachea and damaging it. If a shape memory catheter is used as contemplated by Wilson, it would presumably become austenitic and bend after insertion (see Figures 1a and 1b, and corresponding text, of Wilson). But removal would require either cooling to below the transition temperature (which could easily mean cooling to so low a temperature that the tracheal tissue is damaged), removal in the bent shape (presumably damaging tissue), or forcing the austenitic SMA to straighten to permit direct removal (unlikely to be satisfactory since the austenitic alloys e.g. of Ni/Ti

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may have yield strengths of 100 ksi or more, and force sufficient to cause plastic deformation would be required).

. If an SIM element is used instead, however, removal can be accomplished almost as easily as insertion. If the catheter is made in a bent shape (as in Wilson), it can be straightened by insertion of a straight pin down the catheter axis, the catheter deforming by the formation of stress-induced martensite. Insertion of the catheter into the trachea is accomplished while the catheter is straight, at whatever rate is desired (permitting easy and accurate placement), and the pin is gradually withdrawn to permit the catheter to take up its desired shape as the martensite reverts to austenite. [It is assumed here that the stress-strain curve of the alloy at the temperature of use is of the form of Figure 2, so spontaneous reversion occurs on removal of the stress induced by the pin]. When removal is desired, it may be achieved simply by the gradual insertion of the pin, straightening the catheter and permitting easy withdrawal. Insertion of the catheter into the body and pin removal may, of course, take place simultaneously if desired, as may pin reinsertion and removal of the catheter from the body.

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Example III. IUDS

Fannon et al., in U.S. Patent No. 3,620,212, the disclosure of which is incorporated herein by reference, discloses an intrauterine contraceptive device (an IUD) proposed to be formed of a shape memory alloy. The device is suggested to be deformed in the martensitic phase (the transition temperature being below the temperature of the uterus), and the deformed device insulated with, e.g., wax and inserted the Removal is contemplated only by using two 10 SMA elements in opposition, the higher temperature one being martensitic at body temperature but strong enough so that, if heated, it will overcome the lower temperature element and deform the IUD back to a removable shape. The heating contemplated is electrical. The storage problem discussed 15 in Example II also exists here, so that the device must be stored below its transition temperature.

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By the use of an SIM element, however, these disadvantages may be overcome. Again, assume that the alloy is SIM psuedgelastic, i.e. that it has the stress-strain curve of Figure 2.. Then an IUD may be formed into the desired shape in the austenitic state, and deformed by compression into a tubular placement device (the deformation being such that the strain levels lie within the "plateau" of the stressstrain curve). When the placement device is inserted into "The uterus, the IUD may be deployed by extrusion of the IUD from the placement device. Deployment is then controlled but immediate, so that the physician may satisfy himself with placement. Removal is the reversal of placement: the placement device is inserted into the uterus, the IUD deformed by withdrawal into the placement device, and the placement device withdrawn. Temperature control is not required.

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Example IV. Bone Plates

Johnson et al., in U.S. Patent No. 3,786,806, the disclosure of which is incorporated herein by reference, propose the use of Ni/Ti SMA bone plates in fracture fixation. The plate is deformed in its martensitic state, screwed to the two ends of the bone it is desired to compress together, and warmed (or allowed to warm) to the austenitic state, when the plate contracts, compressing the bone ends

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Because of the high elastic moduli of the austenitic shape memory alloys, it will be difficult to control the which may be applied by a bone plate of the type proposed by Johnson et al., and precision placement of the bone ends and elongation of the plate will be required.

If, however, an SIM pseudoelastic bone plate is used, it will be easily possible to elongate the plate and fasten it to the bone ends without requiring high precision. Because of the comparatively large (e.g. 5%) strain range at essentially constant stress, the force which will be put on the bone ends to compress them will be readily adjustable (by the size of the plate, for example) and will be insensitive to precise placement of the bone ends and/or elongation of the plate. Also, the recovery of the plate, since it is controlled by mechanical restraint, may be as gradual as desired, achieving excellent force and time control, and permitting the surgeon to make adjustments as desired.

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Example V. Marrow Nails

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Baumgart et al., in U.S. Patent No. 4,170,990, the disclosure of which is incorporated herein by reference, discloses the use of the two-way shape memory effect (where an SMA element exhibits a first shape in the austenitic state and a second in the martensitic state, and spontaneously changes between the two shapes with a change in temperature) in, inter alia, marrow nails (see Figures 1a through 1e, and corresponding text, of Baumgart et al.).

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The method proposed, however, requires the use of a wide temperature range in order to cause the phase change which is the origin of the two-way shape memory effect (5°C to 60°C for the water used to cool or heat the nail). In addition, it requires the manufacture of two-way shape memory elements, which is generally more complex than the manufacture of conventional shape memory elements; and transition temperature is required.

However, if an SIM pseudoelastic alloy element is employed, these disadvantages may be overcome. If internal tangs, which may be gripped by an inserted tool, are provided within a marrow nail of the type shown in Figure 1a of Baumgart et al., then the nail may be radially compressed by the application of stress by such a tool. When the nail is released by the tool, it will expand to fill the bone channel with a constant force (not readily available by Baumgart et al.); and it may be withdrawn by the reverse procedure.

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Example VI. Dental Arch Wire

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Andreasen, in U.S. Patent No. 4,037,324, the disclosure of which is incorporated herein by reference, proposes the use of dental arch wires made of Ni/Ti alloys instead of conventional 18-8 stainless steel wires. The wires are stated to be of lower elastic modulus and higher elastic limit than stainless steel, which is stated to be advantageous. Heat recovery of an SMA wire is also suggested as a technique for orthodonture.

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10 The technique of using the conventional shape memory effect is not believed to have found clinical application, possibly because such a technique would require rapid placement of the wire in its martensitic state to avoid premature recovery, and would result in rapid recovery with ¹⁵ extremely high forces, which would be painful for the patient.

The use of a wire which displays lower elastic modulus and higher elastic limit than stainless steel has found some application, however. Otsuka et al. in Metals Forum, v. 4, pp. 142-52 (1981) have suggested that this behavior may be the result of elasticity enhanced by cold working and martensite-to-martensite psuedoelasticity in an alloy which has a transition temperature below body temperature. The alloy, then, is martensitic rather than austenitic in its undeformed state.

While the use of an enhanced elasticity wire may offer some advantages over the more usual stainless steel wire, it remains the situation that the amount of motion in the teeth that may be produced by an arch wire without further adjustment a17

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is largely limited by the pain tolerance of the patient (since the force applied by the arch wire is proportional to the deformation of the wire). However, if an SIM pseudoelastic wire is used, it can exert a relatively constant force (chosen by the dentist to be sufficient to cause tooth movement but not painful) over a strain range of up to 5%. The load may be applied mechanically, and is thus more readily established, and no precise temperature control of the alloy is needed as would be required for the shape memory effect.

Example VII. Coil Stents and Filters

The use of tubular coiled wire stent grafts has been discussed in the medical literature since 1969. Although the coils helped maintain patency of the vessels in which they were placed, they were difficult of insertion unless narrow enough to significantly narrow the lumen of the vessel. Recently it has been proposed, see Radiology, v. 147, pp. 259-60 and pp. 261-3 (1983), the disclosures of which are incorporated herein by reference, to use SMA wire to form these tubular coils. The wire, which has a transformation temperature below body temperature, is introduced through a catheter after being straightened in its martensitic state. When the wire is heated, the coil re-forms.

Because of the difficulty of controlling the transformation temperature accurately, it has proved necessary to cool the straightened wire during insertion and/or to heat the wire to form the coil after insertion. These procedures add to the complexity of the operation.

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If an SIM pseudoelastic wire is used to form the coil, which is then isothermally deformed by loading into a catheter, then the need for temperature control is avoided. The wire remains straight when in the catheter, but re-forms the coil spontaneously when it is extruded from the catheter. Accurate placement is thus readily obtainable, since there is no urgency as might be required with a conventional shape memory effect element.

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It has similarly been proposed to use SMA wire to form a filter for emplacement by catheter in the vena cava to trap blood clots. The filter is formed in the austenitic state, the wire straightened in the martensitic state and inserted, and the filter re-forms on warming. Just as for the coil stents discussed above, the use of an SIM pseudoelastic wire would greatly simplify manufacture and insertion of such a vena cava filter, permitting accurate placement

with no need for urgency or temperature manipulation.

Example VIII. Bone Staples, Clips, etc.

Bone staples are frequently used to hold fragments of fractured bone together when the fracture is fixed, and may be used in some cases as a replacement for bone plates in the same situation. Sometimes the staples are inserted into drilled holes, sometimes merely driven into the bone directly.

It would be desirable to have a bone staple which provided ²⁵ a controlled force between the times which would tend to hold the staple in place. Shape memory alloys have been proposed for this application, but again the problem of accurate placement while operating quickly enough to prevent the shape change associated with the martensite-to-austenite transition and/or the need for temperature control complicate their use.

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If an SIM alloy is used, these disadvantages may be readily overcome. If the alloy is below A_s , it may be emplaced in the martensitic state. Brief heating will then be required to cause it to become austenitic, but on recooling to body temperature, a constant force can be achieved. If the alloy is above A_s , the staple can be held deformed by a moderate force, then released after insertion to also provide an accurately-known force. In either event, removal is easier than if the alloy is purely austenitic, as discussed above for Examples II and V, for example.

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Similarly, SIM alloy (especially alloy which is pseudoelastic, above A_s at its utilization temperature) may be used to manufacture vascular clips, etc. The alloy element here acts as a constant force spring over a wide strain range (greater than conventional elastic metals), resulting in ease of use.

From the foregoing, it is clear that, in a situation where narrow temperature differences are available or preferable, as often is the case in medical applications, mechanically constrained shape change is a much more useful solution than heat actuated shape change. It offers a degree of control heat actuation does not, it offers easier alloy composition control, it eases mating part tolerance requirements, and it offers simple mechanical reversal at minimal stress levels, all without heating, cooling or insulation complications.

It will be obvious to those skilled in the art, having regard to this disclosure, that other variations on this invention beyond those specifically exemplified here, and other medical devices making use of stress-induced martensite, may be made. Such variations are, however, to be considered as coming within the scope of this invention as limited solely by the following claims.

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I claim:

1. In a medical device intended for use within a mammalian body, or in such proximity to a mammalian body that the device is substantially at body temperature, which device comprises a shape memory alloy element, the improvement which comprises the substitution of an alloy element which displays stress-induced martensite at said body temperature for the shape memory alloy element.

2. The device of claim 1 which is a heart valve, the alloy element being a ring employed to hold a sewing cuff onto the valve body.

3. The device of claim h which is a catheter, the alloy element, being the catheter or a part thereof which causes the catheter to assume a bent shape.

4. The device of claim 3 which is a tracheal catherter.

5. The device of claim which is an intrauterine contraceptive device.

6. The device of claim 1 which is a bone plate.

7. The device of claim 1 which is a marrow nail.

8. The device of claim 1 which is a dental arch wire.

9. The device of claim 1 which is a bone staple.

10. The device of claim 1 which is a clip.

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DECLARATION AND POWER OF ATTORNEY (Page 1)

Reychem Case No.

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As a below named inventor, I hereby declare that:

my residence, post office address and citizenship are as stated below next to my name; I believe I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled: Medical Devices Incorporating SIM Alloy Elements

the specification of which

X is sttached hereto

 wes filed on
 as Application Serial No.

 and was amended on
 (if applicable).

I hereby state that I have reviewed and understand the contents of the above identified specification, including the claims, as amended by any emendment referred to above. I acknowledge the duty to disclose information which is material to the exemination of this application in accordance with 37 CFR §1.56(a).

I hereby claim foreign priority benefits under 35 U.S.C. §119 of any foreign application(s) for patent or inventor's certificate listed below and have also identified below any foreign application for patent of inventor's certificate having a filing date before that of the application on which priority is claimed:

Apolication Number	Country	Date Df Filino	Priority Cleimed
NONE			Yes_ No
		•	Yes_NO_

I hereby claim the benefit under 35 U.S.C.§ 120 of any United States epplication(s) listed below and, insofar as the subject matter of each of the claims of this application is not disclosed in the prior United States epplication in the manner provided by the first paragraph of 35 U.S.C. §112, I acknowledge the duty to disclose material information as defined in 37 CFR §1.56(a) which occurred between the filing date of the prior epplication and the mational or PCT international filing date of this application:

	PRIUR UNITED STATES APPLICATIONS										
	Application Number	Date Of Filing	•	Status							
1	NONE		Pending	Pstented _	Abandoned _						

PRIOR UNITED STATES APPLICATIONS

I hereby appoint the following attorney(s) and/or agent(s) to prosecute this application and to transact all business in the Patent and Trademark Office connected therewith:

	Herbert [. Burkard,	Reg. No.	. 24,500
and	 James W.	Peterson,	Reg. No.	26,057

Address all telephone calls to:

James W. Peterson at (415)361-5854

Pending

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Address all correspondence to:

Patent Department Maychem Corporation 300 Constitution Drive Menlo Park, CA 94025 I nereby occlare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, under 18 U.S.C. \$1001 and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

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DECLARATION AND POWER OF ATTORNEY (Pade 2)

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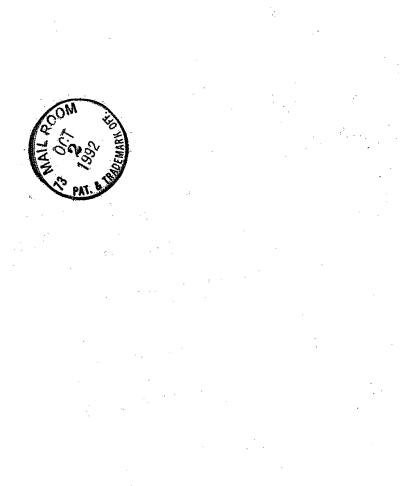
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IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of:)	Group Art	Unit:	331	
JAMES E. JERVIS)	-			-
Serial No.: 252,019))				,
Filed: September 27, 1988)	•		•	
For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS) } }	Pasadena,	Califo	rnia	
ASSOCIATE POWER	OF A	TTORNEY	· · · · ·	1 - 1. -	•

Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231

Sir:

Please recognize Jeffrey G. Sheldon, Registration No. 27,953, of Sheldon & Mak as associate attorney with power to inspect and copy the record of the above-identified application and to make corrections and additions thereto.

Please address all communications to:

SHELDON & MAK 201 South Lake Avenue, Suite 800 Pasadena, California 91101 (818) 796-4000

> <u>Hpril</u> <u>4-9-90</u> IDATE SIGNEDI

Respectfully submitted By

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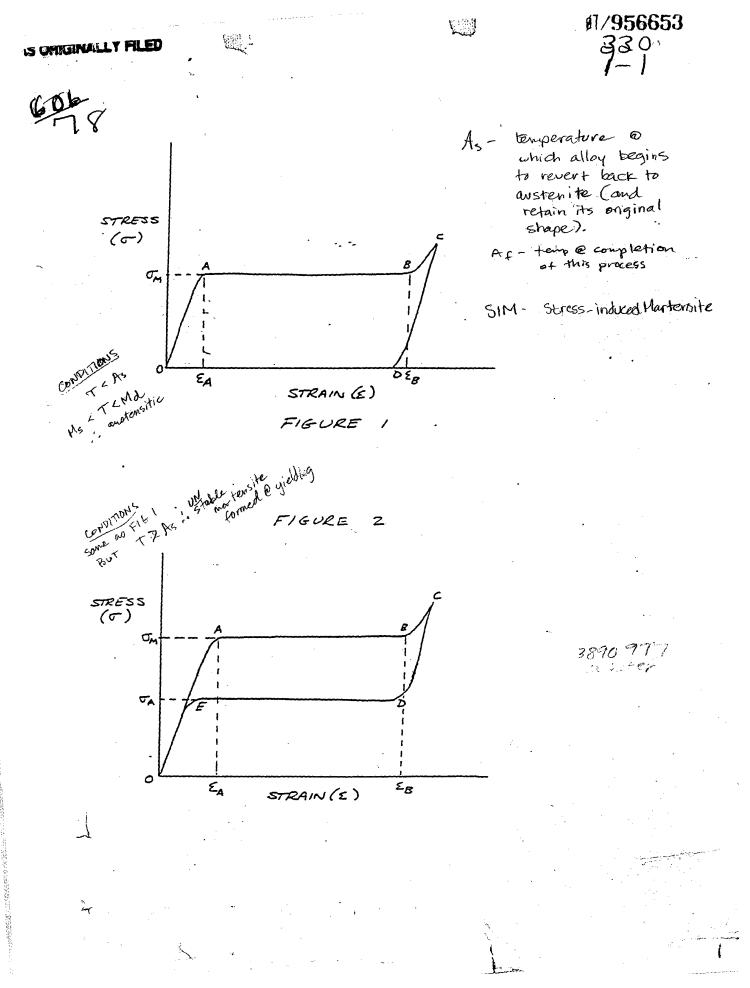
Herbert Burkard Reg. No. $\underline{24}$

I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U.S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOPE ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D.C. 2023L, ON

Date

RAYCHEM CORPORATION 300 Constitution Drive Menlo Park, California 94025 (415) 361-3338

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Edwards Exhibit 1033, p. 36

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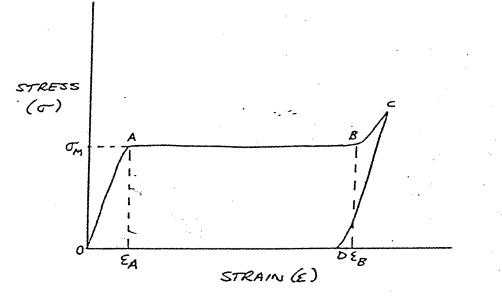
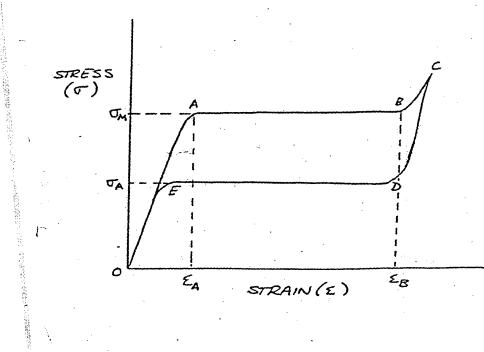


FIGURE 1

FIGURE 2



Edwards Exhibit 1033, p. 37



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IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

Group Art Unit:

Examiner:

E & Whight 12-23-92 2/Pro-A

In Re Application of:

JAMES E. JERVIS

Divisional of Serial No. 07/682,243

Filed: Herewith

MEDICAL DEVICES INCORPORATING For: SIM ALLOY ELEMENTS

PRELIMINARY AMENDMENT AND INFORMATION DISCLOSURE STATEMENT

Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231

Sir:

Please amend the above-identified patent application as

follows:

IN THE SPECIFICATION

Page 2, before the heading "Background of the

Invention", insert:

-- CROSS-REFERENCE TO RELATED APPLICATIONS

This application is a divisional of application Serial No. 682,243 filed on April 9, 1991, now U.S. Patent No.

5190546 AK

ai

, which is a divisional of Serial No. 252,019 filed on September 27, 1988, now U.S. Patent No. 5,067,957, which is a

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September 25, 1992

9438 GI continuation of Application Serial No. 177,817 filed March 30, 1988, now abandoned; which is a continuation of Application Serial No. 047,824 filed May 8, 1987, now abandoned; which is a continuation of Application Serial No. 865,703 filed May 21, 1986, now U.S. Patent No. 4,665,906; which is a continuation of Application Serial No. 541,852 filed October 14, 1983, now abandoned. Page 3, line 28, after Quin insert -- now U.S. Patent No. 4,505,767 --. Page 4, line 26, delete "power" and insert -powder --. Page 6, line 4, after "martensite," please insert Fig. 3 is a front plan view of a bone implant, namely a nail-X Page 7, line 19, delete " E_D " and insert -- E_A --. Page 7, line 19, delete "strain" and insert -stress --. Page 8, line 20, delete "theart" and insert -the art --. 2 pc4\wp51\docs\mjr\actions.pto\Raychem\7757Pre2.Amd September 25, 1992

- SEL VERSENAN AVEREL

Page 8, line 21, delete "tsting" and insert --

testing --.

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Page 8, line 24, after "(Docket No. MP0873-US1)" insert -- now U.S. Patent No. 4,505,767 --.

Page 8, between lines 24 and 25, insert the following paragraph:

-- The following table sets forth transformation

temperature data for alloys disclosed in US-4505767:

TABLE

Composition (atomic percent)

<u>Ni</u>	<u>Ti</u>	<u>v</u>	<u>M</u> s	<u>A(90)</u>
49.50	43.50	7.00	-107	-88
50.00	44.00	6.00	-96	-84
49.00	43.00	8.00	-83	-61
50.00	45.00	5.00	-42	-33
49.00	45.00	6.00	-35	-12
50.50	48.00	1.50	-32	-6
48.50	44.50	7.00	-30	-13
50.00	46.00	4.00	-11	7
48.50	45.00	6.50	-10	15
49.00	45.50	5.50	-10	14
48.00	44.25	7.75	-7	8
48.50	45.50	6.00	-5	27
41.50	38.50	20.00	-2	86
46.50	43.50	10.00	-1	50
36.25	33.75	30.00	0	42
49.50	46.00	4.50	6	35
48.00	46.00	6.00	12	36
47.75	45.75	6.50	20	54

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reduced diameter is loosely inserted into a slightly, or not at all, pre-drilled marrow channel of a bone which has been broken or fractured. By means of a heating probe the marrow nail is heated and thus expands. This achieves a relative fixing of the two bone ends along the marrow channel axis. Compression of the fracture is effected by the available muscle tension. If it should be necessary, the marrow nail may also be additionally prestretched along its longitudinal axis so that it is additionally compressed in the longitudinal direction when heated. In this case it is necessary, however, to anchor the nail at both of its ends which anchoring can be effected, for example, by sprockets or teeth on the outer surface of the nail 1 --.

IN THE CLAIMS

Cancel Claims 1 to 10.

transition --.

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or in such proximity to a mammalian body that the device is

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 $\frac{1}{2}$ substantially at body temperature, the device comprising an $\frac{1}{2}$ element which comprises a shape memory alloy which:

(a) displays stress induced martensite behavior at body temperature; and

(b) has an A(90) temperature of not more than 0°C. $102V V_{4505} - 167$

 $\Phi^{(12)}$ (12. A device as claimed in claim 11, which includes a restraint by means of which the shape memory alloy element is held in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress induced martensite.

13. A device as claimed in claim 12, in which the restraint is hollow, and the shape memory alloy element is deformed in such a way that it is compressed transversely, and is positioned within the restraint, the restraint preventing transverse expansion of the element.

14. A device as claimed in claim 13, in which the restraint is a catheter.

(15. A device as claimed in claim 13, in which the shape memory alloy element is an intrauterine contraceptive device.

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46. A device as claimed in claim-14, in which the shape memory alloy element is a filter for a blood versel.

17. A device as claimed in claim 12, in which the shape memory alloy element is tubular, and the restraint is positioned within the shape memory alloy element to deform it.

18. A device as claimed in claim 17, in which the shape memory alloy element is a tracheal catheter.

Why (19. A medical device which comprises: (a) an element for use within a mammalian body or H in such proximity to a mammalian body that the device is substantially at body temperature, the element comprising a shape memory alloy which displays stress induced martensite behavior at body temperature; and

> b) a restraint by means of which the shape memory alloy element is held in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress-induced martensite;

where in the device is adapted so that removal of the restraint from the shape memory alloy element, without change in temperature of the device, releases at least a portion of the element from its deformed configuration.

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20. A device as claimed in Claim 19, in which the restraint is hollow, and the shape memory alloy element is positioned at least partially within the restraint.

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 \mathcal{A} . A device as claimed in Claim \mathcal{A} , in which the restraint is a catheter.

A device as claimed in Claim 20, in which the shape memory alloy element is an intrauterine contraceptive device.

28. A device as claimed in Claim 21, in which the shape memory alloy element is a filter for a blood vessel.

24. A device as claimed in Claim 19, in which the shape memory alloy element is tubular, and the restraint is positioned within the shape memory alloy element to deform it.

25. A device as claimed in Claim 20, in which the shape memory alloy element is deformed in such a way that its transverse dimension is reduced, the restraint preventing transverse expansion of the element.

The device of Claim 19 wherein removal of the restraint from the shape memory alloy releases at least a portion

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September 25, 1992

 Q_{4} of the shape memory alloy element from its deformed configuration Contribution that the state of the restraint.

> 27. A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising (i) a restraining means and (ii) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stressinduced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

the restraining means engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the restraining means is adapted to be removed from the memory alloy element at a temperature greater than the As of the alloy when the device is placed within or proximate to the mammalian body to transform at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the device is adapted so that the

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transformation can occur without any change in temperature of the restraining means or the memory alloy element.

28. A medical device for treatment of a mammalian body, the device comprising (i) a hollow restraining member and (ii) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stressinduced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the device is adapted so that the transformation can

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occur without any change in temperature of the restraining member or the memory allog element.

29. The medical device of Claim 28 wherein the restraining member is a tube and the memory alloy element is axially slidable within the tube, and wherein the device is adapted so that relative axial movement between the tube and the memory alloy element extends at least a portion of the memory alloy element beyond the tube and thereby transforms the memory alloy element toward its austenic state.

element can be extruded completely out of the tube for deployment in the mammalian body.

32. A medical device for treatment of a mammalian body, the device comprising (i) a restraining member and (ii) a hollow memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

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the restraining member being within the hollow memory alloy element and engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stressinduced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the device is adapted so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

The device of Claim-31 wherein the memory alloy element is a tube and the restraining member is axially slidable within the tube, and wherein the device is adapted so that relative axial movement between the tube and the restraining means extends at least a portion of the tube beyond the restraining means and thereby transforms the tube toward its austenitic shape.

body, the device comprising (i) a restraining member and (ii) a

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hollow catheter formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stressinduced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

the restraint <u>externally or internally</u> engaging and stressing the catheter at a temperature greater than the As of the alloy so that the catheter is in its easily inserted shape;

wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from its stress-induced martensitic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the device is adapted so that the transformation can occur without any change in temperature of the restraining member or the catheter.

24. The medical device of Claim 33 wherein the catheter is a cannula.

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35. A medical device for insertion into a mammalianbody, the device comprising (i) a straight pin and (ii) a catheter formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) a straight shape when the alloy is in its stress-induced martensitic state and (ii) a curved unstressed shape when the alloy is in its austenitic state;

> the straight pin engaging and stressing the inside of the catheter at a temperature greater than the As of the alloy so that the catheter is in its straight shape;

wherein withdrawal of the pin from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from its stress-induced martensitic state to its austenitic state so that the catheter transforms from its straight shape to its curved shape, and wherein the device is adapted so that the transformation can occur without any change in temperature of the pin or the catheter.

Catheter is a tracheal insertion catheter.

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37. A medical device for insertion into a mammalian body, the device comprising (i) a hollow placement device and (ii) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

the memory alloy element being within the hollow placement device, the hollow placement device stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the memory alloy element can be extruded from the hollow placement device at a temperature greater than the As of the alloy to transform at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape alloy is Alloy of the transformation ean occur without any change in temperature of the placement device or the memory alloy element.

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38. The medical device of Claim 37 wherein the memory alloy element is an intrauterine contraceptive device.

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(39. The device of Claim 37 wherein the memory alloy element is a stent graft.

40. The device of Claim 37 wherein the memory alloy element is a filter for trapping bood clots.

At. The invention of Claim 27 wherein the transformation of the alloy occurs without any change in the state of the restraining means.

10 13 542. The invention of Claim 28, 21 or 23 wherein the transformation of the alloy occurs without any change in the state of the restraining-means.

(43). The invention of Claim 37 wherein the transformation occurs without any change in the state of the placement device.

44. The invention of Claim 35 wherein the transformation of the alloy occurs without any change in the state of the pin.

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memory alloy element exerts constant stress during its transformation.

46. The invention of Claim 35 or 35 wherein the catheter exerts constant stress during its transformation.

The invention of Claim 27, 28 or 31 wherein the

JT. The medical device of Claim 27 wherein the removal of the restraining means from the memory alloy element causes at least a portion of the alloy to transform to its austenitic state.

A8. The medical device of Claim 28, 21, 23 wherein relative movement of the restraining member and the memory alloy element causes at least a portion of the alloy to transform to its austenitic state. 24 Alloy Method

A9. The device of Claim 47 wherein the device is adapted so that engaging the restraining means with the memory alloy element after removal results in the memory alloy element transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stressinduced martensitic state.

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⁹⁴³⁸ M 50. The device of Claim 48 wherein the device is adapted so that (i) the restraining member can be completely disengaged and separated from the memory alloy element, and (ii) engaging the restraining member with the memory alloy element after separation results in the memory alloy element transforming towards its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stressinduced martensitic state.

51. The device of Claim 33 wherein the device is adapted so that (i) the restraining member can be completely disengaged and separated from the catheter, and (ii) re-engaging the restraining member with the catheter after separation results in the catheter transforming toward its easily inserted shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

52. The device of Claim 35 wherein the device is adapted so that (i) the pin can be completely disengaged and separated from the cathefer, and (ii) re-engaging the restraining means with the memory alooy element after separation results in the catheter transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

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September 25, 1992

53. The device of Claim 36 wherein the device is adapted so that (i) the placement device can be completely disengaged and separated from the catheter, and (ii) re-engaging the placement device with the catheter after separation results in the catheter transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

REMARKS

Entry of the amendments is respectfully requested. All the amendments to the specification and drawings are the same as were made in the parent application.

INFORMATION DISCLOSURE STATEMENT

Attached hereto are copies of Forms PTO-1449. These forms list all the references cited in the parent application. Copies of the references are available in the file of the parent application. It is believed that these cited references are relevant to claims pending in the present application for the same reason as discussed in the parent application. Not all of these references are relevant to the newly filed claims.

If the Examiner would like a further description, or copies of any of the references, please call the undersigned. In

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September 25, 1992

view of the bulk of the references, and the fact that they are available in the file of the parent application, Applicant does not desire to overload the Patent Office files with duplicate copies of references. However, if the Examiner has a need for the copies, applicant would be most happy to provide them.

Respectfully submitted,

Reg. No. 27,953

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SHELDON & MAK

By

0 DATED: 1992

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SHELDON & MAK 225 South Lake Avenue - 9th Floor Pasadena, California 91101 (818) 796-4000

September 25, 1992

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IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

)

In Re Application of:

Group Art Unit: 3301

JAMES E. JERVIS

Serial No. 07/956,653

Filed: October 2, 1992

For: MEDICAL DEVICES INCORPORATING) SIM ALLOY ELEMENTS)

PRELIMINARY AMENDMENT

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Dear Sir:

IN THE CLAIMS

Please add the following claim:

54. A method for compressing two ends of a mammalian bone together at body temperature, the method comprising the steps of:

(a) providing a bone plate formed from a pseudoelastic shape-memory alloy, wherein the shape-memory alloy can display reversible stress-induced martensite at about body temperature such that the shape-memory alloy has a stress-induced martensitic state and an austenitic state, the bone plate being (i) elongated when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (ii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic state and (iii) shortened when the alloy is in its stressed induced martensitic st

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stress-induced martensitic state and elongating the bone plate;

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(c) attaching the stressed and elongated bone/plate to the two ends of the bone at a temperature greater than the As of the alloy; and

(d) releasing the stress from the bone plate so that at least a portion of the alloy transforms from its stressinduced martensitic state to its austenitic state so that the bone plate compresses the two ends of the bone together at essentially constant stress.

REMARKS

The above claim was submitted in an Amendment after Allowance dated July 7, 1992, in the parent application Serial No. 07/682,243. However, the Patent and Trademark Office did not enter the Amendment in the parent case on the basis that the claim was not prosecuted during the pendency of the application.

Respectfully submitted,

SHELDON & MAK

By

Jeffrey G. Sheldon Reg. No. 27,953

DATED:

225 S. Lake Avenue - 9th Floor Pasadena, California 91101 (818) 796-4000

12/30/92

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MENEMENT COVER SHEET	
DOCKET NO. 9438	
IN RE APPERCATEORS OF James E. Jervis Paulo 44 3	5
SERIAL NO 07/956:653 FILED: October 2, 1992	
FOR: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS	

HONORABLE COMMISSIONER OF PATENTS AND TRADEMARKS

Washington, D.C. 20231

Sir:

Transmitted herewith is a paper in the above-identified application. Any necessary extension of time period set for this paper is hereby requested.

- \square No additional fee is required.
- □ The fee has been calculated as shown below:

EXTENSION FEE	RATE Non-Small Entity	RATE Small-Entity	FEE
FIRST MONTH AFTER TIME PERIOD SET	110.00	55.00	\$
SECOND MONTH AFTER TIME PERIOD SET	360.00	180.00	\$
THIRD MONTH AFTER TIME PERIOD SET	840.00	420.00	\$
FOURTH MONTH AFTER TIME PERIOD SET	1,320.00	660.00	\$

□ TOTAL EXTENSION FEE \$

FEE FOR EXTRA CLAIMS added by Amendment in this response:

	Column 1	Column 2	Column 3			
<u> </u>	Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	FEE
TOTAL CLAIMS	53	MINUS ** 52	* = 1	x 20	x 10	\$ 20
INDEPENDENT	9	MINUS *** 8	* = 1	x 72	x 36	\$ 72
First presentation of	f multiple depend	ent claim		+ 220	+ 110	\$

TOTAL FEE FOR EXTRA CLAIMS \$ 92.00

* If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3.

** If the number of Total Claims previously paid for is less than 20, write "20" in this space.

*** If the number of Independent Claims previously paid for is less than 3, write "3" in this space.

Exclosed is the fee of \$ 92.00 by Check No. 3645

Please charge Deposit Account No. 19-2090 in the amount of \$ _____

The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19-2090:

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

c (92 Date:

SHELDON By: G. Sheldon Jeffrey Reg. No .:

CERTIFICATE OF MAILING:

I hereby certify that this correspondence is being deposited with the United States Postal Service as first class in an envelope addressed to: Commissioner of Patents and Trademarks, Washington, D.C. 20231, on 12/30/92

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	EXAMINER
JEFFREY G. SHELDON	KENEALY, D
SHELDON & MAK 225 S. LAKE AVENUE - 9TH FLOOR	ART UNIT PAPER NUMBER
PASADENA, CA 91101	3301
	DATE MAILED: 02/22/93
This is a communication from the examiner in charge of your application. CoatAMISSIONER OF PATENTS AND TRADEMARKS	
/	
This application has been examined 🗖 Responsive to communication filed	on This action is made final.
	month(s),days from the date of this letter.
Failure to respond within the period for response will cause the application to become	
Part I THE FOLLOWING ATTACHMENT(S) ARE PART OF THIS ACTION:	
1. Notice of References Cited by Examiner, PTO-892. 2.	Notice re Patent Drawing, PTO-948.
3. Notice of Art Cited by Applicant, PTO-1449. 4. 5. Information on How to Effect Drawing Changes, PTO-1474. 6.	Notice of Informal Patent Application, Form PTO-152
Part II SUMMARY OF ACTION	
1. 1. Claims	are pending in the application.
Of the above, claims	
· · · · ·	
	have been cancelled.
3. Claims	
4. Claims	are rejected.
5. Claims	are objected to.
6. 11-53\$54	are subject to restriction or election requirement.
7. This application has been filed with informal drawings under 37 C.F.R. 1	.85 which are acceptable for examination purposes.
8. Formal drawings are required in response to this Office action.	
9. The corrected or substitute drawings have been received on	Under 37 C.F.R. 1.84 these drawings
are 🔲 acceptable; 🗋 not acceptable (see explanation or Notice re Pa	tent Drawing, PTO-948).
2.10. The proposed additional or substitute sheet(s) of drawings, filed on examiner; I disapproved by the examiner (see explanation).	has (have) been 🔲 approved by the
11. The proposed drawing correction, filed, has been	an 🔲 approved; 💷 disapproved (see explanation).
12. Acknowledgement is made of the claim for priority under U.S.C. 119. The deen filed in parent application, seriel no; file	
13. Since this application apppears to be in condition for allowance except fo accordance with the practice under Ex parte Quayle, 1935 C.D. 11; 453 (
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14. 🛄 Other	
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EXAMINER'S ACTIO	N
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Edwards Exhibit 1033, p. 62

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Serial No. 956653 Art Unit 331 I

The drawings are objected to under 37 C.F.R. § 1.83(a). The drawings must show every feature of the invention specified in the claims. Therefore, the IUD, the tubular structure, the catheter, the filter for a blood vessel, the tracheal catheter, etc., must be shown or the feature canceled from the claim. No new matter should be entered. It is difficult to interpret the claims without a drawings, especially in the instance of claim 25 which claims a transverse dimension without any antecedent basis and has no drawing for reference.

-2-

Restriction to one of the following inventions is required under 35 U.S.C. § 121:

I. Claims 11-53, drawn to medical devices incorporating SIM alloy elements, classified in Class 606, subclass 78.

II. Claim 54, drawn to a method for compressing two mammalian bones together, classified in Class 606 and Class 128, subclass 898.

Inventions I and II are related as process and apparatus for its practice. The inventions are distinct if it can be shown that either: (1) the process as claimed can be practiced by another materially different apparatus or by hand, or (2) the apparatus as claimed can be used to practice another and materially different process. (M.P.E.P. § 806.05(e)). In this case the apparatus as claimed can be used to practice another and

Serial No. 956653 Art Unit 331

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materially different process such as contraception or catheterization.

Because these inventions are distinct for the reasons given above and have acquired a separate status in the art because of their recognized divergent subject matter, restriction for examination purposes as indicated is proper.

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-3-

This application contains claims directed to the following patentably distinct species of the claimed invention: where the shape memory alloy is an IUD, a stent graft, a blood filter a catheter and a tracheal catheter.

Applicant is required under 35 U.S.C. § 121 to elect a single disclosed species for prosecution on the merits to which the claims shall be restricted if no generic claim is finally held to be allowable. Currently, claims 12,19 and 37 are generic.

Applicant is advised that a response to this requirement must include an identification of the species that is elected consonant with this requirement, and a listing of all claims readable thereon, including any claims subsequently added. An argument that a claim is allowable or that all claims are generic is considered nonresponsive unless accompanied by an election.

Upon the allowance of a generic claim, applicant will be entitled to consideration of claims to additional species which are written in dependent form or otherwise include all the limitations of an allowed generic claim as provided by 37 C.F.R. § 1.141. If claims are added after the election, applicant must indicate which are readable upon the elected species. M.P.E.P. § 809.02(a).

Should applicant traverse on the ground that the species are not patentably distinct, applicant should submit evidence or Serial No. 956653

Art Unit 331

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identify such evidence now of record showing the species to be obvious variants or clearly admit on the record that this is the case. In either instance, if the examiner finds one of the inventions unpatentable over the prior art, the evidence or admission may be used in a rejection under 35 U.S.C. § 103 of the other invention.

It should be noted that most of the claims subsequent to claim 41 are functional in nature and it is difficult for this examiner to comprehend what structure is being claimed with the limitations presented. Claims 49-53 all contain the term "adapted" which has little meaning in terms of patent language used for claiming a structural item.

Any inquiry concerning this communication or earlier communications from the examiner should be directed to David Kenealy whose telephone number is (703) 308-2680.

Any inquiry of a general nature or relating to the status of this application should be directed to the Group receptionist whose telephone number is (703) 308-0858.

> Robert A. Solar S.P.E.

> > ART UNIT 331

David Kenealy February 21, 1993

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×.	In re Application of:	TENT AND TRADEMARK OFFICE	#5.45g
	JAMES E. JERVIS)) Examiner:	mart
	Serial No.: 07/956,653 Filed: October 2, 1992)))	
	For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS)) Pasadena, California))	
		···	

INFORMATION DISCLOSURE STATEMENT

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Dear Madam/Sir:

Rein-

Attached hereto are PTO-1449 forms listing documents believed relevant to the subject application. It is respectfully requested that these documents be considered by the Examiner and an initialed copy of each form be returned to the undersigned.

Enclosed is check no. 3875 for \$200, the fee due under 37 C.F.R. § 1.17(p).

Enclosed please find (1) references cited by the Japanese Patent Office in a counterpart Japanese application, (2) Opposition papers and references cited to a counterpart European patent application, and (3) EPO Search Report and references cited therein.

Please note that U.S. Patent 5,190,546 is a parent of the present application, E.P.O. 0145166 is a European Patent Office counterpart application, and Japanese Patent Kokai 100956/1985 is the Japanese counterpart application. These 080 KJ 03/31/73 07756653 1 126 200.00 CK references have the same inventor as the present application. Hence, these references are not prior art.

It should be noted that the word "prior" has been deleted from the form.

It is believed that this disclosure complies with the requirements of 37 C.F.R. §§ 1.56, 1.97 and 1.98, and the Manual of Patent Examining Procedures § 707.05(b). If for some reason the Examiner considers otherwise, it is respectfully requested that the undersigned be called so that any deficiencies can be remedied.

A copy of each document is enclosed.

Some of the documents may have markings thereon. No significance is meant to be attached to the markings.

With regard to any translation provided herewith, the undersigned does not know how and who made the translations. Therefore, no representation is being made as to the accuracy of any translation.

These documents are not necessarily analogous.

Respectfully submitted,

SHELDON & MAK

3 112193

Dated:

225 South Lake Avenue Ninth Floor Pasadena, California 91101 (818) 796-4000

Encls.

PC20\AKJ\PATENT\9438-1.IDS

By Jef ĕldon Req/ 27,953

I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U. S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOT ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D. G. 20231 CN MULLIAM, A. J. J.

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March 12, 1993

US005190546A

United States Patent (19)

Jervis

[54] MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

- [75] Inventor: James E. Jervis, Atherton, Calif
- Raychem Corporation. Menio Park. [73] Assignce: Caiií.
- [21] Appl. No. 682,243
- Apr. 9, 1991 [22] Filed:

Related U.S. Application Data

- Division of Ser. No. 252,019, Sep. 27, 1938, Pat. No. [60] 5.067.957, which is a continuation of Ser. No. 177.817. Mar. 30, 1998, abandoned, which is a continuation of Ser No. 47:324, May 8, 1987, abandoned, which is a communication of Ser. No. 865,703. May 21, 1986, Pat. No. 4,665,906, which is a continuation of Ser. No. 541.852, Oct. 14, 1983, abandoned.
- 606/62; 606/68, 606/200; 606/108; 128/833
- Field of Search 606/60, 62, 67, 68, [58] 606/69-76, 78, 108: 128/833

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{11}	Patent Number:	5,190,546
[45]	Date of Patent:	Mar. 2, 1993

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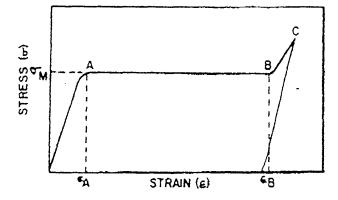
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Primary Examiner-Robert A. Hafer Assistant Examiner-Kevin G. Rooney Attorney, Agent, or Firm-Jeffrey G. Sheldon

ABSTRACT

Medical devices which are currently proposed to use elements made from shape memory alloys may be improved by the use of stress-induced martensite alloy elements instead. The use of stress-induced martensite decreases the temperature sensitivity of the devices. thereby making them easier to install and/or remove.

40 Claims, 2 Drawing Sheets



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Page 2	
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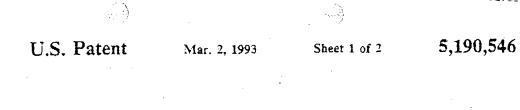
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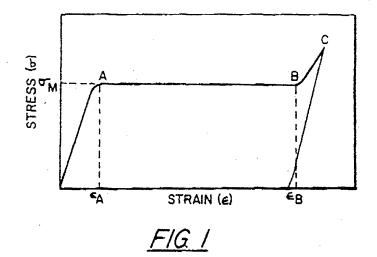
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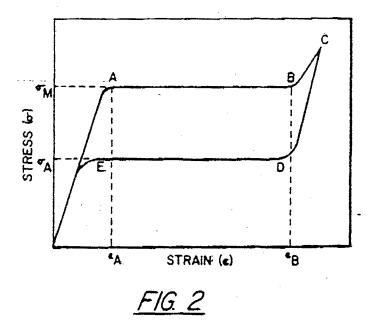
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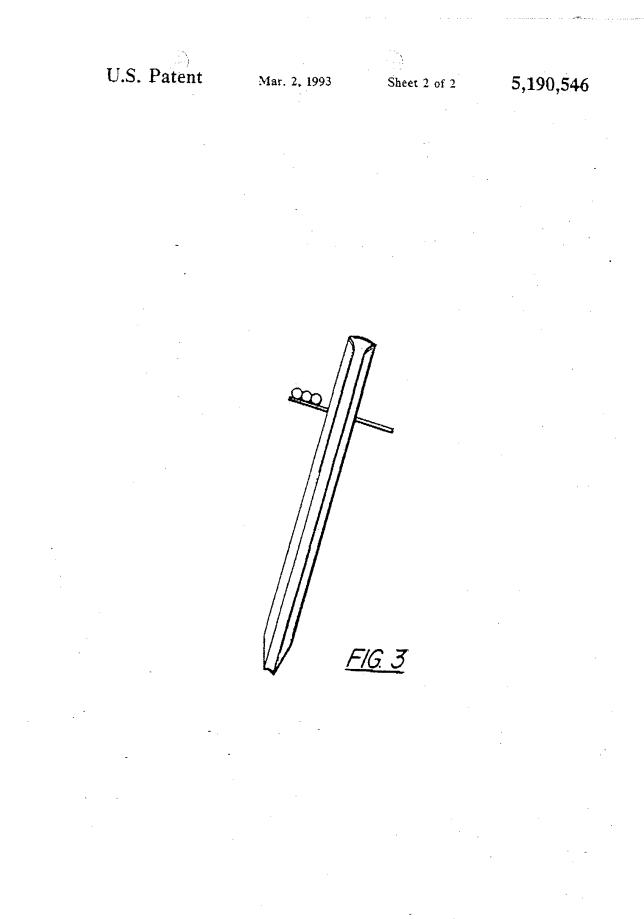


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Edwards Exhibit 1033, p. 72

5,190,546

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CROSS-REFERENCE TO RELATED APPLICATIONS

This application is a divisional of application Ser. No. 252.019 filed Sept. 27, 1988. now U.S. Pat. No. 5,067,957, which is a continuation of application Ser. No. 177,817 filed Mar. 30, 1988, now abandoned; which 10 is a continuation of application Ser. No. 047,824 filed May 8, 1987, now abandoned; which is a continuation of application Ser. No. 865,703 filed May 21, 1986, now U.S. Pat. No. 4665,906; which is a continuation of application Ser. No. 541,852 filed Oct. 14, 1983, now aban-15 doned.

BACKGROUND OF THE INVENTION

1. Field of the Invention

This invention relates to medical devices incorporat-²⁰ ing shape memory alloys, and to improvements therein.

2. Introduction to the Invention

Materials, both organic and metallic, capable of possessing shape memory are well known. An article made of such materials can be deformed from an original, ²⁵ heat-stable configuration to a second, heat-unstable configuration. The article is said to have shape memory for the reason that, upon the application of heat alone, it can be caused to revert, or to attempt to revert, from its heat-unstable configuration to its original, heat-stable 30 configuration, i.e. it "remembers" its original shape.

Among metallic alloys, the ability to possess shape memory is a result of the fact that the alloy undergoes a reversible transformation from an austenitic state to a martensitic state with a change in temperature. This 35 transformation is sometimes referred to as a thermoelastic martensitic transformation. An article made from such an alloy, for example a hollow sleeve, is easily deformed from its original configuration to a new configuration when cooled below the temperature at which 40 the alloy is transformed from the austenitic state to the martensitic state. The temperature at which this transformation begins is usually referred to as M_s and the temperature at which it finishes Mr. When an article thus deformed is warmed to the temperature at which 45 the alloy starts to revert back to austenite, referred to as As (A) being the temperature at which the reversion is complete) the deformed object will begin to return to its original configuration.

Many shape memory alloys (SMAs) are known to 50 display stress-induced martensite (SIM). When an SMA sample exhibiting stress-induced martensite is stressed at a temperature above M₃ (so that the austenitic state is initially stable), but below Md (the maximum temperature at which martensite formation can occur even 55 under stress) it first deforms elastically and then, at a critical stress, begins to transform by the formation of stress-induced martensite. Depending on whether the temperature is above or below A_i, the behavior when the deforming stress is released differs. If the tempera- 60 ture is below An the stress-induced martensite is stable; but if the temperature is above An the martensite is unstable and transforms back to austenite, with the sample returning (or attempting to return) to its original shape. The effect is seen in almost all alloys which ex- 65 hibit a thermoelastic martensitic transformation, along with the shape memory effect. However, the extent of the temperature range over which SIM is seen and the

2 stress and strain ranges for the effect vary greatly with the alloy

In the commonly assigned Quin U.S. Pat. No. 04505767, the disclosure of which is incorporated 5 herein by reference, a nickel/titanium/vanadium alloy

having SIM over a wide temperature range is disclosed. Shape memory alloys have found use in recent years in, for example, pipe couplings (such as are described in Harrison and Jervis U.S. Pat. Nos. 4,035,007 and 4,198,081), electrical connectors (such as described in Orte and Fischer U.S. Pat. No. 3,740,839), switches (such as are described in U.S. Pat. No. 4,205,293), actua-

Various proposals have also been made to employ shape memory alloys in the medical field. For example, Fannon et al. U.S. Pat. No. 3,620,212 proposes the use of an SMA intrauterine contraceptive device, Johnson et al. U.S. Pat. No. 3,786,806 proposes the use of an SMA bone plate, Wilson U.S. Pat. No. 3,890,977 proposes the use of an SMA element to bend a catheter or cannula, etc.

These medical SMA devices rely on the property of shape memory to achieve their desired effects. That is to say, they rely on the fact that when an SMA element is cooled to its martensitic state and is subsequently deformed, it will retain its new shape; but when it is warmed to its austenitic state, the original shape will be recovered.

However, the use of the shape memory effect in medical applications is attended with two principal disadvantages. First, it is difficult to control the transformation temperatures of shape memory alloys with accuracy as they are usually extremely composition-sensitive, although various techniques have been proposed (including the blending by powder metallurgy of already-made alloys of differing transformation temperatures: see Fountain et al. U.S. Pat. No. 4,310,354). Second, in many shape memory alloys there is a large hysteresis as the alloy is transformed between austenitic and martensitic states, so that reversing of the state of an SMA element may require a temperature excursion of several tens of degrees Celsius. The combination of these factors with the limitation that (a) it is inconvenient to have to engage in any temperature manipulation, and (b) human tissue cannot be heated or cooled beyond certain relatively, narrow limits (approximately 0"-60" C. for short periods) without suffering temporary or permanent damage is expected to limit the use that can be made of SMA medical devices. It would thus be desirable to develop a way in which the advantageous property of shape memory alloys, i.e. their ability to return to an original shape after relatively substantial deformation, could be used in medical devices without requiring the delicacy of alloying control and/or the temperature control of placement or removal needed by present shape memory alloy devices.

DESCRIPTION OF THE INVENTION -

Summary of the Invention

I have discovered that if, in a medical device containing a shipe memory alloy element which uses the shape memory property of that alloy, an element which shows the property of stress-induced martensite is used instead, an improved device results.

Accordingly, this invention provides a medical device intended for use within a mammalian body, or in such proximity to a mammalian body that the device is

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tors, etc.:

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substantially at body temperature, which device comprises a shape memory alloy element, the improvement in which comprises the substitution of an alloy element which displays stress-induced martensite at said body temperature for the shape memory alloy element 5

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BRIEF DESCRIPTION OF THE DRAWING

FIGS. 1 and 2 illustrate the stress-strain behavior of an alloy which exhibits constant stress versus strain behavior due to stress-induced martensite. 10

FIG. 3 is a front plan view of a bone implant, namely a nail.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

The invention will be discussed first by introducing the concept of stress-induced mattensite and the effect achievable by its use, and then by examples showing how SIM alloy elements can be substituted for conventional SMA elements in medical devices to achieve the 20 beneficial effect of the invention.

The Figures illustrate the phenomenon of stressinduced martensite by means of stress-strain curves. In both FIG. 1 and FIG. 2, the alloy is at a temperature between M_t and M_d so that it is initially austentuc; and 25 it will be assumed for the puposes of this discussion that M_f is equal to M_f and A_f equal to A_f . FIG. 1 shows the case when the temperature is below A_f , so that any martensite formed by the applied stress is stable; while FIG. 2 shows the case where the temperature is above 30 A_{fr} so that austenite is the only stable phase at zero stress.

In FIG. 1, when a stress is applied to the alloy, it deforms elastically along the line OA. At a critical applied stress, org, the austenitic alloy begins to transform 35 to (stress-induced) martensite. This transformation takes place at essentially constant stress until the alloy becomes fully martensitic at point B. From that point on, as further stress is applied, the martensite yields first elastically and then plastically (only elastic deformation 40 is shown at point C). When the stress is released, the martensite recovers elastically to point D, at which there is zero residual stress, but 2 non-zero residual strain. Because the alloy is below A, the deformation is not recoverable until heating above As results in a rever- 45 sion to austenite. At that point, if the sample is unrestrained, the original shape will be essentially completely recovered: if not, it will be recovered to the extent permitted by the restraint. However, if the matenal is then sllowed to re-cool to the original tempera- 50 ture at which it was deformed (or a temperature where SIM behavior of this type is seen), the stress produced in the sample will be constant regardless of the strain provided that the strain lies within the "plateau" region of the stress-strain curve. That is, for a strain between 55 ϵ_B and ϵ_A the stress will be σ_M . This means that a known, constant force (calculable from $\sigma_{\mathcal{H}}$) can be applied over a wide (up to 5% or more for certain Ni/Ti alloys) strain range. Thus, though this resembles the conventional shape memory effect, because the alloy 60 shows SIM and is below As a constant force can be achieved

In FIG. 2, when a stress is applied to the alloy, it deforms elastically along line OA, then by SIM along line AB, and by deformation of the mattensite to point 65 C, just as in FIG 1. However, the stress-strain behavior on unloading is significantly different, since the alloy is above A_s and the stable phase is therefore austentic. As

the stress is removed, the alloy recovers elastically from O to D; then, at a entited stress, σ_A , the alloy reverts to austenite without requiring a change in temperature. Thus reversion occurs at essentially constant stress. Finally if the stress is removed from the reverted austenite, it recovers elastically along line EO. The recoverable deformation associated with the formation and reversion of stress-induced martensite has been referred to as pseudoelasticity. While σ_M may be comparatively high, e.g. 50 ksi, σ_A is usually substantially lower, e.g. less than 10 ksi; thereby creating 2 constant-force spring with an effective working range of about 5% ($\epsilon_B - \epsilon_A$). The shape change available in the SMA is thus mechanically, rather than thermally, actuated and controlled, permitting a greater control over a device incorporating it.

Suitable alloy for this invention i.e. those displaying stress-induced martensite at temperatures near mammalian body temperature $(35^{\circ}-40^{\circ} \text{ C})$, may be selected from known SMAs by those of ordinary skill in the art, having regard to this disclosure by testing for the existence of the SIM effect at the desired temperature. A particularly preferred alloy is the nickel/titanium/vanadium alloy of U.S. Pat. No. 4,505,767 referred to previously.

The following table sets forth transformation temperature data for alloys disclosed in U.S. Pat. No. 4,505,767:

	T •	701	-
	TA	n 1	-

-			1		
		Comp	SHIDA (MOTIS	e percent)	
	Ni	Ti	V -	м	A(90)
_	49 50	43.50	7,00	- :07	- 38
	50.0C	44.00	6.00	- 96	- 34
	49 00	43.00	3.00	- 83	- 61
	30.00	45.00	5.00	- 42	- 33
	49 00	45.00	6.00	- 35	- 12
	50.50	48.00	1.50	- 32	ó
	48.30	44.50	7.00	- 50	- 13
	50.00	46.00	4 00	-11	7
)	48.50	45.00	6.50	- 10	15
	49.00	45.50	5.50	- 10	14
	+8.00	44.25	1.75	-:	8
	48.30	45.50	6.00	3	27
	41.10	38.50	20.00	- 2	36
	46.50	+3.50	10.00	- (50
	36.25	33.75	30.00	0	42
	49.50	46.00	4.50	ó	35
	48.00	46.00	6.00	12	36
	47.75	45.75	6.50	20	54
	47 30	45.50	7.00	26	38
	48.50	46.53	5.00	27	58
	45.00	+5 00	10.00	30	71
	47 50	46.50	6.00	32	71
	46.50	46.50	7.00	34	70

The invention will now be discussed in detail by some examples of the use of an SIM alloy.

EXAMPLE I

Heart Valves

Akins U.S. Pat. No. 4,233,690, the disclosure of which is incorporated herein by reference, describes the use of a shape memory alloy ring to hold a sewing cuff to the body of an artificial heart valve. The ring is made in the austenstic phase, cooled to the martensitic phase, deformed, placed around the valve body, and heated or allowed to warm to cause reversion to the austenitic phase and recovery of the ring into engagement with the valve body.

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However, this technique has not found commercial accentance. Present medical technique requires that the valve body be capable of being rotated relative to the cuff, thereby enabling the surgeon to set the rotational orientation of the valve after it has been sewn into place. This is desirable because the techniques used make it difficult to visualize or accomplish optimal orientation during initial placement.

In order to accomplish the desired torque control to permit the desired rotation and yet ensure a firm hold of 10 the cuff on the valve body, precise control of the pressure exerted on the valve body by the ring is needed. This is difficult because there are substantial manufacturing tolerances in the valve body which may be made, for example, of pyrolytic graphite or ceramics, etc. 15 Because the austenite stress-strain curve is extremely steep, it is not considered practical to use the simple shape memory technique proposed by Akins. Indeed, Akins does not even address the issue of rotation of the cuff with respect to the valve body. 20

However, if an SIM alloy is used instead of conventional shape memory, the process may be considerably simplified.

First, if the alloy has a stress-strain curve like that of FIG. I, the alloy ring may be made just as for Akins. 25 The ring is then expanded from its initial austenitic state by the formation of SIM. When the ring is placed about the valve body, it needs only to be heated above Ayand allowed to cool to its original temperature for the ring to engage the valve body with a constant force, even if 30 the valve body has a deviation from the specified size. The torque may thus be controlled to the desired level despite manufacturing tolerances.

Second, if the alloy has a stress-strain curve like that of FIG. 2, the ring may be expanded, placed over the 35 valve body, and the stress released all at the same temperature. Because the austenitic phase is stable, the stress-induced martensite spontaneously revents to austenite until recovery is restrained by the ring engaging the valve body. Because the reversion to austenite takes ŧ۵ place at constant stress, a constant force (and hence constant torque) may be obtained regardless of manufacturing tolerances. Close temperature control is not required, either, and the fact that the patient in a heart valve replacement operation is conventionally cooled 45 as much as 15° C, or so below normal body temperature does not affect the operation of the ring.

To control the torque at a sufficiently low level, it may be desirable for the alloy ring to be other than a solid ring, such as, for example, a continuous helical 50 spring, a flat zigzag spring, etc. Such variations permit the schievement of a greater range of movement with constant force and a reduction in the force exerted by the ring on the value body, since the ring recovers in a bending mode rather than in tension. 55

EXAMPLE II

Catheters and Cannulas

Wilson U.S. Pat. No. 3,890,977, the disclosure of which is incorporated herein by reference, discloses a 60 catheter or cannula (both being included hereinafter in the word "catheter") made of, or containing, an SMA element to cause all or a portion of the catheter to deploy in a useful form once introduced into a living body.

ized. Possible defects of the device which have prevented commercialization include (i) the inability to slowly emplace the catheter in a desired position when

the transition temperature of the alloy is below body temperature (since the SMA element will attempt to revert to its original shape as it reaches body temperature), thus limiting the ability of the physician to place the device carefully and precisely; or alternatively, if the transition temperature of the alloy is above body temperature, the requirement that the device be heated to a temperature above body temperature to cause recovery and that the device be placed so as not to change. shape again when it re-cools (since the body temperature is below the transition temperature); (ii) the inability to remove the device easily; and (iii) the need for controlled temperature storage to prevent premature reversion to austenite of the SMA, with consequent shape change.

The issue of removal of a catheter is especially significant, and not addressed by Wilson. Consider, for example, a tracheal puncture catheter. This should be straight for easy insertion into the trachea through a puncture into the front of the neck, but should curve after insertion so that the flow of air or oxygen through the catheter passes axially down the trachea rather than impinging on the surface of the trachea and damaging it. If a shape memory catheter is used as contemplated by Wilson, it would presumably become austenitic and bend after insertion (see FIGS. 1a and 1b, and corresponding text, of Wilson). But removal would require either cooling to below the transition temperature (which could easily mean cooling to so low a temperature that the tracheal tissue is damaged), removal in the bent shape (presumably damaging tissue), or forcing the austenitic SMA to straighten to permit direct removal (unlikely to be satisfactory since the austenitic alloys e.g. of Ni/Ti may have yield strengths of 100 ksi or more, and force sufficient to cause plastic deformation would be required).

If an SIM element is used instead, however, removal can be accomplished almost as easily as insertion. If the catheter is made in a bent shape (as in Wilson), it can be straightened by insertion of a straight pin down the catheter axis, the catheter deforming by the formation of stress-induced martensite. Insertion of the catheter into the traches is accomplished while the catheter is straight, at whatever rate is desired (permitting easy and accurate placement), and the pin is gradually withdrawn to permit the catheter to take up its desired shape as the martensite reverts to austenite. [It is assumed here that the stress-strain curve of the alloy at the temperature of use is of the form of FIG. 2, so spontaneous reversion occurs on removal of the stress induced by the pin]. When removal is desired, it may be achieved simply by the gradual insertion of the pin, straightening the catheter and permitting easy withdrawal. Insertion of the catheter into the body and pin removal may, of course, take place simultaneously if desired, as may pin reinsertion and removal of the catheter from the body.

EXAMPLE III

IUDS

Fannon et al. U.S. Pat. No. 3,620,212, the disclosure of which is incorporated herein by reference, discloses an intrauterine contraceptive device (an IUD) proposed to be formed of a shape memory alloy. The device is However, again this device has not been commercial- 65 suggested to be deformed in the martensitic phase (the transition temperature being below the temperature of the uterus), and the deformed device insulated with. e.g., wax and inserted. Removal is contemplated only

by using two SMA elements in opposition, the higher temperature one being martensitic at body temperature but strong enough so that, if heated, it will overcome the lower temperature element and deform the IUD back to the removable shape. The heating contemplated 5 is electrical. The storage problem discussed in Example II also exists here, so that the device must be stored below its transition temperature.

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By the use of an SIM element, however, these disadvantages may be overcome. Again, assume that the 10 alloy is SIM pseudoelastic, i.e. that it has the stressstrain curve of FIG. 2. Then an IUD may be formed into the destred shape in the austenitic state, and deformed by compression into a tubular placement device (the deformation being such that the strain levels lie 15 within the "plateau" of the stress-strain curve). When the placement device is inserted into the uterus, the IUD may be deployed by extrusion of the IUD from the placement device. Deployment is then controlled but immediate, so that the physician may satisfy himself 20 with placement. Removal is the reversal of placement: the placement device is inserted into the uterus, the IUD deformed by withdrawal into the placement device, and the placement device withdrawn. Temperature control is not required.

EXAMPLE IV

Bone Plates

Johnson et al. U.S. Par. No. 3,786,806, the disclosure of which is incorporated herein by reference, propose 30 the use of Ni/Ti SMA bone plates in fracture fixation. The plate is deformed in its martensitic state, screwed to the two ends of the bone it is desired to compress together, and warmed (or allowed to warm) to the austentic state, when the plate contracts, compressing the 35 bone ends together. The Johnson et al. bone plate is of generally oblong configuration, overlaps a bone fracture and is secured by two screws to one portion of the bone and by two other screws to the other portion of the bone.

Because of the high elastic moduli of the austenitic shape memory alloys, it will be difficult to constrol the amount of force which may be applied by a bone plate of the type proposed by Johnson et al., and precision placement of the bone ends and elongation of the plate 45 will be required.

If, however, an SIM pseudoetastic bone plate is used, it will be easily possible to elongate the plate and fasten it to the bone ends without requiring high precision. Because of the comparatively large (e.g. 5%) strain so range at essentially constant stress, the force which will be put on the bone ends to compress them will be readily adjustable (by the size of the plate, for example) and will be insensitive to precise placement of the bone ends and/or elongation of the plate. Also, the recovery 55 of the plate, since it is controlled by mechanical restraint, may be as gradual as desired, achieving excellent force and time control, and permitting the surgeon to make adjustments as desired.

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EXAMPLE V

Marrow Nails

Baumgart et al. U.S. Pat. No. 4,170,990, the disclosure of which is incorporated herein by reference, discloses the use of the two-way shape memory effect 65 (where an SMA element exhibits a first shape in the austenitic state and a second in the martensitic state, and spontaneously changes between the two shapes with a

change in temperature) in, inter alia, marrow nails (see FIGS. 16 through 1e, and corresponding text, of Baumgart et al.). Marrow nails according to Baumgart et al. comprise a tube of memory alloy which has been split along its longitudinal axis and which may have a circular, elliptical, clover-leaf or other rotation preventing cross section, which may also be variable along the axis of the nail. A prepared marrow nail having a reduced diameter is loosely inserted into a slightly, or not at all, pre-drilled marrow channel of a bone which has been broken or fractured. By means of a heating probe the marrow nail is heated and thus expands. This achieves a relative fixing of the two bone ends along the marrow channel axis. Compression of the fracture is effected by the available muscle tension. If it should be necessary, the marrow nail may also be additionally prestretched along its longitudinal axis so that it is additionally compressed in the longitudinal direction when heated. In this case it is necessary, however, to anchor the nail at both of its ends which anchoring can be effected, for example, by sprockets or teeth on the outer surface of the nail.

The method proposed, however, requires the use of a $_{25}$ wide temperature range in order to cause the phase change which is the origin of the two-way shape memory effect (5° C. to 60° C. for the water used to cool or heat the nail). In addition, it requires the manufacture of two-way shape memory elements, which is generally 30 more complex than the manufacture of conventional shape memory elements; and precise control of the transition temperature is required.

However, if an SIM pseudoelastic alloy element is employed, these disadvantages may be overcome. If internal tangs, which may be gripped by an inserted tool, are provided within a marrow nail of the type shown in FIG. 1a of Baumgart et al., then the nail may be radially compressed by the application of stress by such a tool. When the nail is released by the tool, it will expand to fill the bone channel with a constant force (not readily available by Baumgart et al.); and it may be withdrawn by the reverse procedure.

EXAMPLE VI

Dental Arch Wire

Andreasen U.S. Pat. No. 4,037,324, the disclosure of which is incorporated herein by reference, proposes the use of dental arch wires made of Ni/Ti alloys instead of conventional 18-8 stainless steel wires. The wires are stated to be of lower elastic modulus and higher elastic limit than stainless steel, which is stated to be advantageous. Heat recovery of an SMA wire is also suggested as a technique for orthodonture.

55 The technique of using the conventional shape memory effect is not believed to have found clinical application, possibly because such a technique would require rapid placement of the wire in its martensitic state to avoid premature recovery, and would result in rapid 60 recovery with extremely high forces, which would be painful for the patient.

The use of a wire which displays lower elastic modulus and higher elastic limit than stainless steel has found some application, however. Otsuka et al in Metals Forum, v. 4, pp. 142-52 (1981) have suggested that this behavior may be the result of elasticity enhanced by cold working and martensite-to-martensite pseudoelasticity in an alloy which has a transition temperature 9. below body temperature. The alloy, then, is martensitic rather than austenitic in its undeformed state.

While the use of an enhanced elasticity wire may offer some advantages over the more usual stainless steel wire, it remains the situation that the amount of 5motion in the teeth that may be produced by an arch wire without further adjustment is largely limited by the pain tolerance of the patient (since the force applied by the arch wire is proportional to the deformation of the wire). However, if an SIM pseudoelastic wire is 10 used, it can exert a relatively constant force (chosen by the dentist to be sufficient to cause footh movement but not painful) over a strain range of up to 5%. The load may be applied mechanically, and is thus more readily established, and no precise temperature control of the 15 alloy is needed as would be required for the shape memory effect.

EXAMPLE VII

Coil Stents and Filters

The use of tubular coiled wire stent grafts has been discussed in the medical literature since 1969. Although the coils helped maintain patency of the vessels in which they were placed, they were difficult of insertion unless narrow enough to significantly parrow the lumen 25 of the vessel. Recently, it has been proposed, see Radiology, \vee 147, pp. 259-60 and pp. 261-3 (1983), the disclosures of which are incorporated herein by reference, to use SMA wire to form these tubular coils. The wire, which has a transformation temperature below 30 body temperature, is introduced through a catheter after being straightened in its martensitic state. When the wire is heated, the coil re-forms.

Because of the difficulty of controlling the transformation temperature accurately, it has proved necessary 35 to cool the straightened wire during insertion and/or to heat the wire to form the coil after insertion. These procedures add to the complexity of the operation.

If an SIM pseudoelastic wire is used to form the coil, which is then isothermally deformed by loading into a 40 catheter, then the need for temperature control is avoided. The wire remains straight when in the catheter, but re-forms the coil spontaneously when it is extruded from the catheter. Accurate placement is thus readily obtainable, since there is no urgency as might be 45 required with a conventional shape memory effect element.

It has similarly been proposed to use SMA wire to form a filter for emplacement by catheter in the vena cava to trap blood clots. The filter is formed in the 50austenitic state, the wire straightened in the martensitic state and inserted, and the filter re-forms on warming. Just as for the coil stents discussed above, the use of an SIM pseudoelastic wire would greatly simplify manufacture and insertion of such a vena cava filter, permit-55 ting accurate placement with no need for urgency or temperature manipulation.

EXAMPLE VIII

Bone Staples, Clips, etc.

Bone staples are frequently used to hold fragments of fractured bone together when the fracture is fixed, and may be used in some cases as a replacement for bone plates in the same situation. Sometimes the staples are inserted into drilled holes, sometimes merely driven into 63 the bone directly.

It would be desirable to have a bone staple which provided a controlled force between the times which 10

would tend to hold the staple in place. Shape memory alloys have been proposed for this application, but again the problem of accurate placement while operating quickly enough to prevent the shape change associated with the martensite-to-austenite transition and/or the need for temperature control complicate their use.

If an SIM alloy is used, these disadvantages may be readily overcome. If the alloy is below A_S , it may be emplaced in the martensitic state. Brief heating will then be required to cause it to become austenitic, but on recooling to body temperature, a constant force can be achieved. If the alloy is above A_S , the staple can be held deformed by a moderate force, then released after insertion to also provide an accurately-known force. In either event, removal is easier than if the alloy is purely austenitic, as discussed above for Examples II and V.

for example. Similarly, SIM alloy (especially alloy which is

pseudoclastic, above As at its utilization temperature)
 may be used to manufacture vascular clips, et. The alloy element here acts as a constant force spring over a wide strain range (greater than conventional elastic metals), resulting in ease of use.

From the foregoing, it is clear that, in a situation where narrow temperature differences are available or preferable, as often is the case in medical applications, mechanically constrained shape change is a much more useful solution than heat actuated shape change. It of-

fers a degree of control heat actuation does not, it offers easier alloy composition control, it eases mating part tolerance requirements, and it offers simple mechanical reversal at minimal stress levels, all without heating, cooling or insulation complications.

It will be obvious to those skilled in the art, having regard to this disclosure, that other variations on this invention beyond those specifically exemplified here, and other medical devices making use of stress-induced martensite, may be made. Such variations are, however, to be considered as coming within the scope of this invention as limited solely by the following clams.

I claim:

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1. A method of forming an attachment to a bone, which comprises positioning in an aperture in the bone an element formed at least partially from a shape memory alloy which displays stress induced martensite behavior at body temperature, the element being so positioned that it is deformed by the walls of the aperture by the formation of stress-induced martensite, and thereby exerts a force outwardly on the walls of the aperture.

2. A method as claimed in claim 1, in which the aperture is formed before the element is positioned.

3. The method of claim 1 wherein the step of positioning takes place at a temperature greater than the A_S of the alloy.

4. The method of claim 1 wherein substantially all of the formation of martensite results from deformation of the element, and not from any change in temperature of the element.

5. A method for installing a bone attachment into a bone of a mammalian body such that the device is substantially at body temperature, the method comprising the steps of:

(a) providing a bone attachment device at least partly formed from a pseudoeiastic shape-memory alloy, wherein the shape-memory alloy can display reversible stress-induced martensite at about body temperature such that the shape-memory alloy has

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11 a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stressedinduced martensitic state and (ii) a different un-

stressed shape when the allow is in its austentite. 5

- state: (b) deforming the bone attachment device with a tool at a temperature greater than the As of the alloy for placing the alloy in its stress-induced martensitic state and the bone attachment device in its de- 10 formed shape;
- (c) inserting and positioning the bone attachment device into a channel in the bone at a temperature greater than the As of the alloy while the bone attachment remains in its deformed shape; and

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(d) while maintaining the temperature of the bone attachment device above the As of the alloy, releasing the bone attachment device from the tool so that at least a portion of the alloy transforms from its stress-induced martensitic state so that the bone 20 attachment device transforms from its deformed shape toward its unstressed shape to fill the bone channel and exert a constant force, wherein substantially all of the transformation occurs from releasing the bone attachment device from the tool 25 and not from any change in temperature of the bone attachment device or the tool.

6. The method of claim 5 wherein the bone attachment device is a marrow nail.

7. The method of claim 5 or 6 comprising the addi- 30 tional steps, after step (d), of (c) deforming the device with the tool at a temperature greater than the As of alloy so that at least a portion of the alloy transforms from its austenitic state so that the bone attachment device transforms into its deformed shape; and (f) with- 35 body comprising the steps of: drawing the deformed bone attachment device from the bone without changing the temperature of the device.

3. The method of claim 5 wherein the bone attachment device is a bone staple.

9. A method for installing an implant into a bone in a 40 mammalian body comprising the steps of:

(a) drilling a channel in the bone;

- (b) selecting a bone implant comprised of a pseudoelastic shape memory alloy, the alloy displaying reversible stress-induced martensite at about body 45 temperature such that it has a stress-induced martensitic state and an austenitic state, the bone implant having (i) an insertable shape suitable for insertion into the channel when the alloy is in its stress-induced martensitic state and (ii) a second 50 different shape when the alloy is in its austenitic state, the bone implant being in its second shape; and
- (c) inserting the bone implant into the channel and stressing the bone implant so that the alloy trans- 55 forms toward its stress-induced martensitic state and the bone implant transforms to its insertable shape.

10. The method of claim 9 wherein the bone implant is inserted into the channel and stressed at a temperature 50 the device and deforming the device, and not from any greater than the Ayof the alloy.

11. The method of claim 9 wherein substantially all of the transformation of the bone implant to its insertable shape occurs from inserting and stressing the bone implant, and not from any change in temperature of the 65 tine: bone implant.

12. The method of claim 9 wherein the bone implant that is selected is a marrow nail.

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12 13. The method of claim 9 comprising the additional . step, after step (c), of:

(d) causing the stress on the implant to be released so that the alloy transforms toward its austenitic state and the implant transforms toward its second shape.

14. The method of claim 9 comprising the additional step, after step (c), of:

(d) causing the stress on the implant to be released so that the implant exerts constant force in the channel.

15. The method of claim 13 or 14 including the additional step, after step (d), of stressing the implant so that the alloy transforms toward its stress-induced martens-

15 itic state and the implant transforms toward its insertable shape, and thereafter withdrawing the installed implant from the channel.

16. The method of claim 15 wherein the implant is a marrow nail.

- 17. A bone implant sized for insertion into a channel in a bone in a mammalian body, the bone implant being comprised of a pseudoelastic shape memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the bone implant having (i) an insertable shape for insertion into the bone when the alloy is in its stress-induced martensitic state and (ii) a second, different, non-insertable shape when the alloy is in its austenitic state.
- 18. The bone implant of claim 17 wherein the bone implant is a nail.

19. The bone implant of claim 18 wherein the nail has a tine.

20. A method for holding onto a bone in a mammalian

(a) drilling a hole into the bone;

- (b) selecting a bone-holding device comprised of a pseudoelastic shape memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stressinduced martensitic state and an austening state, the device having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state, the bone-holding device being in its unstressed shape; and
- (c) userting the device into the hole and deforming the device to its deformed shape so that the alloy transforms toward its stress-induced martensitic state.

21. The method of claim 20 wherein the device is a bone staple.

22. The method of claim 20 wherein the step of selecting comprises selecting a device having a tine.

23. The method of claim 20 wherein the step of inserting the device and deforming the device takes place at a temperature greater than the As of the alloy.

24 The method of claim 20 wherein substantially all of the transformation of the alloy tesults from inserting change in temperature of the device.

25. The method of claim 20 wherein the bone holding device is a bone nail.

26. The method of claim 25 wherein the nail has a

27. A method for removing from a mammalian body a medical device comprising a memory alloy element at least partly formed from a pseudoelastic shape memory

alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the device having (i) a removable shape when the alloy is in its stress-induced martensitic state and (ii) a differ- 5 ent non-removable shape when the alloy is in its austenitic state, the device being positioned in a mammalian body and being in its non-removable shape, the method comprising the steps of:

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- toward its stress-induced martensitic state and the device transforms to its removable shape, without changing the temperature of the device; and
- (b) withdrawing the transformed device from the mammalian body.

28. The method of claim 27 wherein both steps (a) and (b) take place at a temperature greater than the As of the alloy.

29. The method of claim 27 wherein the medical device is a bone implant.

30. The method of claim 29 wherein the medical device is a marrow nail.

31. The method of claim 29 wherein the medical device is a bone staple.

device is an intrauterine contraceptive device.

33. The method of claim 27 wherein the medical device is a filter for a blood vessel.

34. The method of claim 27 wherein the medical device is tubular. 30

35. The method of claim 34 wherein the medical device is a catheter.

36. The method of claim 35 wherein the medical device is tracheal catheter.

37. An article formed at least partially from a shape memory alloy which displays stress-induced martensite behavior at human body temperature, the article adapted to be positioned in an aperture in a bone of a human body so that the article is deformed by the walls (a) stressing the device so that the alloy transforms 10 of the sperture by the formation of stress-induced martensite, and thereby exerts a force outwardly on the

> wall of the aperture. 38. A bone plate for compressing two ends of a fractured mammalian bone together, the bone plate being 15 formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying teversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the bone plate being (i) elongated when the alloy is in its

> 20 stress-induced martensitic state and (ii) shortened when the alloy is in its austenitic state:

the bone plate being provided with means for securing the bone plate to both ends of a fractured bone.

39. The method of claim 9 wherein the step of stress-32. The method of claim 27 wherein the medical 25 ing of the bone implant is performed before placement of the bone implant into the channel in the bone.

40. The method of claim 20 wherein the step of deforming the device is performed before the device is inserted into the hole.

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ENGLISH TRANSLATION OF JAPANESE PATENT KOKAI PUBLICATION NO. 41546/1983

APPLICATION NO.: 140477/1981 APPLICATION DATE: SEPTEMBER 7, 1981

KOKAI DATE: MARCH 10, 1983

APPLICANT: SUWA-SEIKOSHA

INVENTORS: Susumu TANAKA, Takehiko OOSAKU, & Kumahide SUMIDA

CLAIMS:

1. A device for orthodonture, orthodontic load of which corresponds to and changed by a temperature difference between a temperature corresponding to body temperature and a temperature generated by applying a temperature stimulating material to a buccal cavity.

2. The device for orthodonture according to claim 1, which comprises an alloy mainly consisting of a metal compound of Ni-Ti.

3. The device for orthodonture according to claim 2, wherein a temperature at which reverse martensite transformation of the Ni-Ti alloys finishes is lower than the body temperature.

IN THE DESCRIPTION:

Page 282, lower left column, lines 13 to 19

The super elasticity has completely different mechanism and properties from proportional elasticity found in conventional metallic materials. Even when it is deformed by about 8 , it super elastically reverts to its original shape.

Page 283, upper left column, line 10 to upper right column, line 6

Metallic materials having super elastic effect and shape memory effect include an alloy mainly consisting of Ni-Ti metal compound, Cu-Zn, Cu-Zn-X (wherein X is Si, Sn Al or the like), Cu-Al-Ni, Au-Cd, Ag-Cd, Ni-Al. Cu-Au-Zn, Cu-Sn, etc. These alloys are super lattice alloys which exhibit martensite transformation, namely thermoelastic transformation. Super elasticity utilizes, as motive force, stress-induced martensite transformation caused in a temperature rage higher than the martensite transformation temperature of these alloys and its reverse transformation. Generally, this transformation has small hysterisis of forward and reverse transformations between a mother phase (austenite phase) and a martensite phase, and is crystallographically reversible. "Crystallographically reversible" means that not only the alloy reverts to its crystal structure of the mother phase but also the orientation of the crystal reverts to its original orientation.

ENGLISH TRANSLATION OF JAPANESE PATENT KOKAI PUBLICATION NO. 50951/1983

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APPLICATION NO.: APPLICATION DATE:	150425/1981 September 22, 1981
KOKAI DATE:	MARCH 25, 1983
APPLICANT:	SUWA-SEIKOSHA
INVENTOR:	Susumu TANAKA

CLAIMS:

1. A bracket for orthodonture, which comprises a substrate member which is attached to a tooth body and a wire supporting member which grips and support a wire, wherein the wire supporting member is imparted with elastic deformation ability.

2. The bracket for orthodonture according to claim 1 wherein the wire supporting member is imparted with deformation ability due to super elasticity.

3. The bracket for orthodonture, wherein the at least the wire supporting member is made of an alloy mainly consisting of a metal compound of Ni-Ti.

IN THE DESCRIPTION:

Page 302, lower left column, line 11 to lower right column, line 6

Metallic materials having super elastic effect and shape memory effect include an alloy mainly consisting of Ni-Ti metal compound, Cu-Zn, Cu-Zn-X (wherein X is Si, Sn Al or the like), Cu-Al-Ni, Au-Cd, Ag-Cd, Ni-Al. Cu-Au-Zn, Cu-Sn, etc. These alloys are super lattice alloys which exhibit martensite transformation, namely thermoelastic transformation. Super elasticity utilizes, as motive force, stress-induced martensite transformation caused in a temperature rage higher than the martensite transformation temperature of these alloys and its reverse transformation. Generally, this transformation has small hysterisis of forward and reverse transformations between a mother phase (austenite phase) and a martensite phase, and is crystallographically reversible. "Crystallographically reversible" means that not only the alloy reverts to its crystal structure of the mother phase but also the orientation of the crystal reverts to its original orientation.

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March 12, 1993

# Some Applications of Shape-Memory Allovs

And the LORD said unto Moses. "What is that in your hand?" And he said. "A rod."

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**OLD TESTAMENT** 

Exodus, Chapter 4:2-4

Then HE said, "Cast it on the ground." And he cast it on the ground and it became a serpent; and Moses filed from it.

And the LORD said unto Moses. "Put forth thine hand and take it by the tail." And he put forth his hand and caught it, and it became a rod in his hand.

C. M. Wayman

University of Illinois at Urbana-Champaign Urbana, Illinois

#### SUMMARY

Uses or potential uses of shape memory alloys fall into industrial, energy, and dental/medical categories. These various applications are considered after a brief discussion of the nature of the shape memory effect and other interesting properties in shape memory alloys. Most applications invoive NiTi-type and Cu-based alloys, the latter being relatively inexpensive to produce and fabricate into numerous forms.

# NATURE AND MECHANISTICS OF THE SHAPE MEMORY EFFECT

The shape-memory effect (SME) can be described as foliows: basically, an object in the low-temperature, martensitic condition when it is deformed and the stress then removed will regain its original shape when heated. Strains typically  $b-\bar{s}$  may be completely recovered. The process of regaining the original shape is associated with the reverse transtormation of the deformed martensitic phase to the higher temperature parent phase.

Many materials are now known to exhibit the shape-memory or "marmem" (martensite memory) effect: a partial list includes the alloy systems Cu-Zn, Cu-Zn-Al, Cu-Zn-Ga, Cu-Zn-Sn, Cu-Zn-Si, Cu-Al-Ni, Cu-Au-Zn, Cu-Sn, Au-Cd, Ni-Ti, Ni-Ti-X (X = ternary element), Ni-Al, and Fe-Pt. These alloys are all ordered (both parent and martensite) and exhibit a crystallographically reversible, thermoelastic martensitic transformation.

Substantial progress has recently been made in understanding the nature of SME. As is well known, a crystal of the parent phase will transform into many orientations (plates or variants) of martensite on cooling. Ideally, a single crystal of the parent phase will form 24 orientations of martensite on cooling between the M, and M, temperatures. But when this multi-orientation configuration of martensite is deformed, a single orientation of martensite eventually results because of twinning and the movement of certain martensite interfaces. It has been shown that the twins which form in the martensite are simply other orientations (variants) of martensite: thus twinning can convert one orientation of martensite to another. The same thing happens when martensite/ martensite interfaces move under stress: one orientation grows at the expense of another. In the final analysis, the single remaining orientation of martensite is the variant whose "shear" or shape deformation will permit the maximum elongation of the specimen in the direction

IOURNAL OF METALS . June, 1980

of the tensile axis.

Although the original single crystal of the parent phase transforms into many (up to 24) orientations of martensite, the reverse does not occur. Instead, the single crystal of martensite obtained from deformation below the M, temperature transforms, on heating, to a single orientation of the parent phase. This is a consequence of the relative symmetries involved and the necessity to maintain ordering. In other words, the highly symmetric (usually cubic) parent phase has many crystallographically equivalent principal axes for the lattice change (Bain distortion) which will thus lead to the many variants of martensite which are observed. On the other hand, the relatively unsymmetric martensite (e.g., monoclinic in Cu-Zn-Al alloys) does not enjoy such a multiplicity of choices, and only a single variant of the parent is usually nucleated during the reverse martensite-to-parent transformation. In essence, the single crystal of martensite "unshears" to form a single crystal of the parent, and this "unshearing" during reverse transformation restores the specimen to its original shape. This sequence is metallographically depicted in Figure 1.

The above account appears to be generally valid, irrespective of the alloy system or martensite crystal structure.

#### OTHER INTERESTING PROPERTIES OF MARMEM ALLOYS

Shape-memory alloys have interesting properties and characteristics in addition to the shape-memory effect, per se. As will be described later, excessively deformed (some-30% strain, and well beyond the limit of shape-memory recoverable strain) martensitic NiTi alloys have unusual elastic properties. When many of the martensitic Cu-based alloys are continually deformed beyond the single-crystal martensite stage, a new martensite phase is generated, i.e., a stress-induced martensite-to-martensite transformation occurs. This successive mode of martensite deformation allows recoverable strains of more than 17%. Shape-memory alloys are also excellent damping materials. The relative ease of movement of internal boundaries, such as martensite-martensite boundaries, under a small stress is strongly attenuating. Finally, a "two-way" shape memory can be programmed into various memory alloys by appropriate stress and/or thermal cycling. Once this conditioning has been achieved, a specimen will spontaneously "bend" when the parent transforms into martensite, and "unbend" to the initial shape during the reverse transformation.

## INDUSTRIAL APPLICATIONS OF SHAPE-MEMORY ALLOYS Fasteners and Couplings

One of the earliest widespread applications of SME was Raychem Corporation's (Menlo Park, Calif.) introduction of tubing or pipe couplings which shrink during heating. Typical of such NiTi-type couplings are those used for connecting aircraft hydraulic lines. The couplings are expanded ~4% in the martensitic condition at liquid-nitrogen temperature, then placed around the tubes to be joined. During warming to room temperature, they contract, producing a tight seal. The use of such fittings avoids metailurgical degradation which can result from welding or brazing, and avoids damage to the aircraft "skin." Over 300.000 such highperformance connectors have been used in U.S. Navy aircraft, with no reported failures.

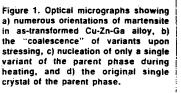
Similar NiTi-type fixtures have been used extensively for plumbing on submarines and surface ships during the past five years by the British Royal Navy, and within the past two years by the U.S. Navy for a variety of surface ships; an example is shown in Figure 2.

The size of NiTi-type fittings has been increased considerably recently, and fittings which join carbon-steel subsea pipe up to six inches in diameter have been installed successfully at dep.hs up to 300 fz using saturation diving techniques, taking a pre-chilled fitting down in the diving chamber. For broken subsea piping, SME fittings are justified by speed and ease of installation compared with other techniques and by no necessity to rely on operator skill.

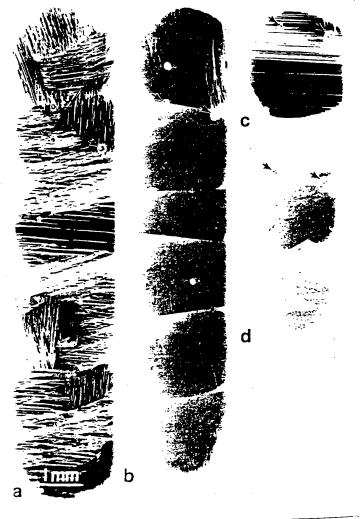
Ravchem has also developed a "Cryocon" shape-memory type electrical connector particularly suited for multiconnector electrical plugs.

Extensive research and development on NiTi and NiTiX ternary alloys is also being conducted at the Brown Boveri Research Center in Baden. Switzerland.

Raychem recently introduced a new line of Cu-based alloy heat-shrinkable fittings in addition to other fasteners and devices. These devices can be provided in the field in the (deformed) martensitic condition at room temperature and applied simply by heating them with a propane torch. Figures 3-5 are examples showing a coupling, retainer, and clamp; and Figures 6 and 7 are photographs of a clamp and a seal made of a Cu-based SME alloy. Figure 8 shows a clamp and expander, demonstrating that the engineering parameters for the Cu alloys have been well worked out. Additional fasteners, clamps, plugs, rivets, etc., will undoubtedly appear in the near future, including plugs for nuclear reactors which will eliminate welding.



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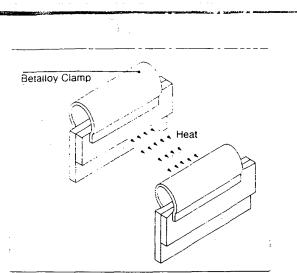


Figure 5. Clamp or crimp made from Cu-based shape-memory alloy. (Courtesy Raychem Corp.)

with heat-shrinkable NiTI-type couplings. (Courtesy Raychem Corp.)

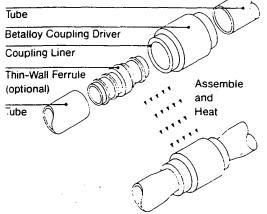


Figure 3. Tube or pipe coupling made from Cu-based shapememory alloy. (Courtesy Raychem Corp.)  $% \left( \left( \left( \left( C_{1}^{2}\right) \right) \right) \right) \right) =0$ 

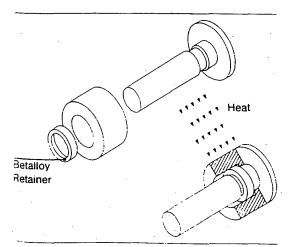


Figure 4. Retainer made from Cu-based shape-memory alloy. Courtesy Raychem Corp.)

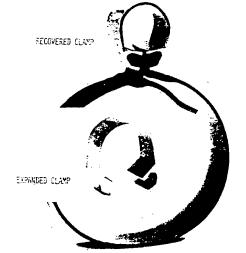


Figure 6. Clamp made from Cu-based shape-memory alloy. (Courtesy Raychem Corp.)

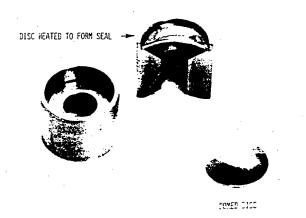


Figure 7. Disc seal made from Cu-based shape-memory alloy. (Courtesy Raychem Corp.) Numerous patents have also been filed for various SME devices not yet been marketed, such as mechanisms for platform motion, pumps for fluids, and thermal warning devices which can be attached to containers used for shipping refrigerated biological materials such as human blood.

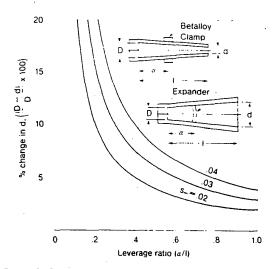


Figure 8. Clamp and expander made from Cu-based shapememory alloy. (Courtesy Raychem Corp.)

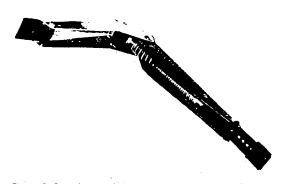


Figure 9. Greenhouse window control incorporating a Cu-based shape-memory alloy. (Courtesy Delta Memory Metal Co.)

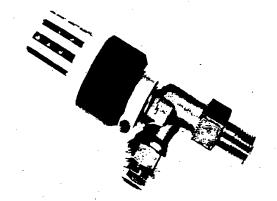


Figure 10. Thermostatic radiator valve incorporating Cu-based shape-memory alloy actuator spring. (Courtesy Delta Memory Metal Co.)

#### Thermomechanical and Thermostatic Devices

Substantial efforts in developing Cu-based shape-memory alloys have been made by the Delta Memory Meta. Company (Suffolk, England). They have developed a range of Cu-Zn-Al SME alloys with emphasis on thermostats, controls for heating and cooling equipment, automotive control devices, and actuators for equipment ranging from greenhouse windows to fire doors. Many of these prototypes have been cycled a half-million times or so with no observable fatigue, "creep," or change in deflection characteristics.

Figure 9 shows an actuator for greenhouse windows, basically a spring-loaded hinge containing a bias spring and a SME spring. Below 18°C, the SME spring is in the co: tracted (martensitic) condition and the window is close. When the temperature rises, the shape-memory spring overcomes the restraining bias spring, and at 25°C the window is fully opened. The window is automatically pulled shut when the temperature falls. As expected, the hysteresis of an unloaded actuator is some  $10-15^{\circ}$ C, but this can be compensated for by forcing the actuator to work against a bias spring. Delta has proposed similar devices to open fire doors, actuate ventilators in factories, open radiator vents in diesel trucks, and control vents in warm-air heating systems.

Another SME spring-actuator/bias-spring device is thermostatic radiator valve for residential hot-water heating systems (Figure 10). As the temperature in a room increases, the actuator expands, overcomes the force of the bias spring, and closes the port of the valve on the hot water line of the radiator system. The temperature can be adjusted by rotating the top head assembly, which alters the compression of the bias spring. Such regulators are comparatively inexpensive and have a much faster system response time. With a proper bias spring, thermal hysteresis can be held to  $1.2C^\circ$ . There is a remote-control version of these valves.

Another Delta device, an automotive clutch fan (Figur 11), uses an SME actuator in the form of a helical spru: which is biased against a set of four steel leaf springs. The SME actuator coil engages a clutch which turns an automotive engine fan when the "air-off" temperature exceeds a certain value, typically 53°C; the actuator closes the clutch plate until the temperature is under control. At low temperatures, the fan idles at ~250 rpm. At higher temperatures, the clutch fan speeds up sufficiently to cool the engine assembly. If this speed is less than engine speed, the clutch fan will slip. Thus the fan does only as much work as required and, accordingly, saves energy. Such a device has been road tested for 20.000 miles. and indications are that it would operate an additional 60,000 miles. The clutch fan was proposed in order to reduce engine noise (at idling) and fuel consumption (because it removes the energy loss from the fan when it is not required to cool the engine).

Another automotive application of Cu-type SME alloys concerns the carburetor. In this case, atmospheric pollution is minimized and fuel consumption optimized by compensating for fuel viscosity. A simple jet made of a Cu-Zn-Al SME alloy is inserted in a Stromberg-type carburetor. As the fuel warms, an orifice reduces in size and thus meters the correct fuel volume. Figure 12 shows the performance of such a  $\mu$ Note particularly the reduction in CO emission at hter fuel temperatures.

A further automotive application, also concerning the carburetor, is to use a Cu-type SME actuator to close a "cold start" choke at a predetermined time after an engine has started. The SME actuator is energized by heat from an electrical source, causing the actuator to close the choke.

Figure 13 shows a room-temperature thermostat control designed by Delta. This control consists of an SME actuator spring and bias spring mounted to a standard microswitch. with a simple adjustment for temperature. A similar thermostatic principle is involved in the use of an SME elemen: switch off electrically-operated tea kettles once the way boils. If the kettle boils drv or is ... of filled, the  ${\rm SME}$  element is heated, switching off the kettle.

Other examples of Cu-Zn-Al SME devices are the tubular and coll-type torsion actuators shown in Figures 14 and 15 in the "closed" and "open" positions. A somewnat similar line of Cu-Zn-Al SME devices is being developed by N.V. Bekaert in Zwevegen, Belgium, but they have not yet been introduced in the U.S.

The above discussion emphasizes many existing thermo-

Inechanical and thermostatic applications of Cu-based SME alloys. More will surely tollow, considering the inexpensiveness of such brasses and their ease of fabrication. However, such alloys are subject to aging effects and cannot cyclically operate indefinitely when the "upper" temperature is  $\sim 150^{\circ}$ C or more. The nature of the aging effects which cause deterioration of the shape memory remains to be determined. Higher operating temperatures are expected in the tuture through alloy development.

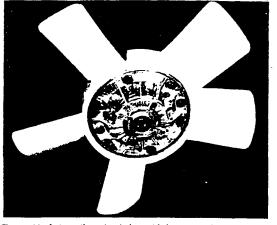


Figure 11. Automotive clutch fan with heat-energized actuator made from a Cu-based shape-memory alloy. (Courtesy Delta Memory Metal Co.)

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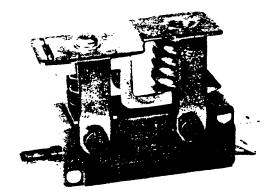


Figure 13. Room-temperature thermostat control using a Cubased shape-memory alloy spring. (Courtesy Delta Memory Metal Co.)

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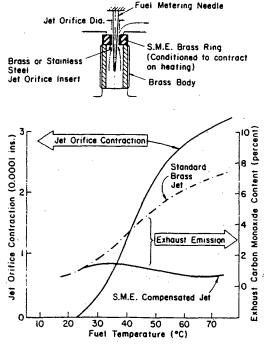


Figure 12. Carburetor jet assembly with variable orifice controlled by a Cu-based shape-memory alloy. Note the reduction of CO emission by using the SME compensated jet. (Courtesy Delta Memory Metal Co.)

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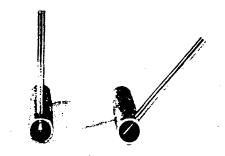


Figure 14. Tubular torsion actuator made from Cu-based shapememory alloy. (Courtesy Delta Memory Metal Co.)

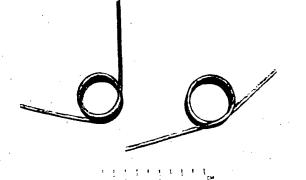


Figure 15. Coil-type torsion actuator made from Cu-based shape-memory alloy. (Courtesy Delta Memory Metal Co.)

## **Recording Pen Drive Unit**

The Foxboro Company (Foxboro, Mass.) has developed a simplified servomechanism to drive recording pens and indicating-pointer assemblies. The servodrive unit (Figure 16) contains a NiTi wire maintained under tensile stress. The input signal is converted to a current which is induced into the SME wire, the thus-heated wire changes its length and moves a connected lever. This device eliminates many moving parts and is extremely reliable. Over 500,000 such units have been produced since their introduction in 1972.

# ENERGY APPLICATIONS

That large stresses are generated during the shape-memory effect has been known for some time. For example, in NiTi alloys stresses as high as 100,000 psi are created by the reverse transformation of the deformed martensite to the memory configuration during heating. Such stresses are an order of magnitude higher than those necessary to deform the martensite at lower temperatures. Thus, heat can be used to create a mechanical force which can do work. Figure/ 17 illustrates the principle involved.

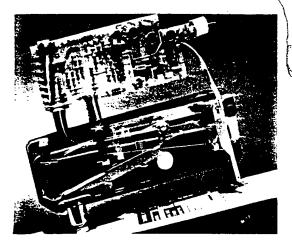


Figure 16. Servodrive unit using NiTi wire actuator. (Courtesy Foxboro Co.)

Numerous heat engines built in the past few years typical: operate between two fixed temperatures (usually maintained by two water reservoirs) and have modest efficiencies of  $4-6^{c_0}$  when operated at room temperature and above. They are consequently well suited to extract heat from "low-grade" energy sources such as industrial coolant water, discharge water from nuclear reactors, geothermal sources, and solar heated masses.

# DENTAL AND MEDICAL APPLICATIONS

# Orthodontic Dental Arch Wires

Unitek Corporation (Monrovia, Calif.) now extensive: markets an orthodontic dental arch wire made from a NiTalloy. This arch wire, attached to bands on the teeth, is novreplacing the traditional stainless steel arch wire in many cases. Since its introduction about  $2^4$ : years ago, over 5.000 of the estimated 6.500 orthodontists in the U.S. have used this device for straightening teeth.

In contrast to the other SME devices mentioned in this report, the dental arch wire is used in the martensitic condition. Because the martensitic wires have been plastically deformed to more than 30%, however, they exhibit an unusual springback and rubberlike character. After a 90° bend test, the coldworked NiTi wires will almost completein unbend, in contrast to similar stainless steel wires which remain bent at a 45° angle. Figure 18 shows the results of comparative bend tests, and Figure 19 shows how the NiTi arch wires are tied into malposed teeth.

It is claimed that using NiTi arch wires offers advantages, such as fewer arch wire changes during treatment, a greater working range (and thus fewer arch wire adjustments), less patient discomfort, and shorter treatment time.

# **Blood** Clot Filters

A vena cava filter using a NiTi alloy has been evaluated by Dr. M. Simon of the Harvard Medical School and other colleagues. The device they propose is a new method r trapping "wandering" blood clots. A chilled, initial straight NiTi wire (martensitic condition) assumes a complex filter shape as it warms to body temperature (parent phase condition) after being placed into the vena cava (a large vein which returns blood to the heart) by a catheter inserted in a vein in the arm. A straight martensitic wire obtains a complex cross sectional shape once ejected from the catheter into the warm body, a process schematically shown in Figure 20.

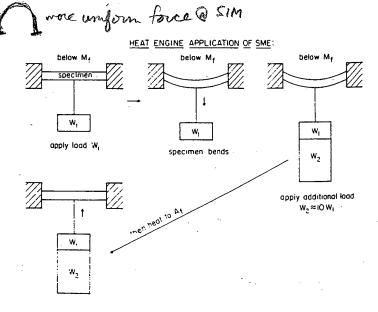


Figure 17. Schematic of heat engine application of a shape-memory alloy.

Edwards Exhibit 1033, p. 92

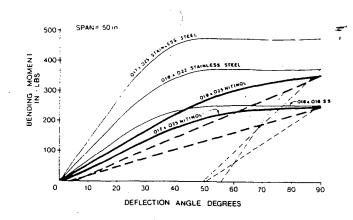


Figure 18. Bend tests comparing coldworked stainless steel and coldworked NiTi martensitic alloy. (Courtesy G.F. Andreasen.)

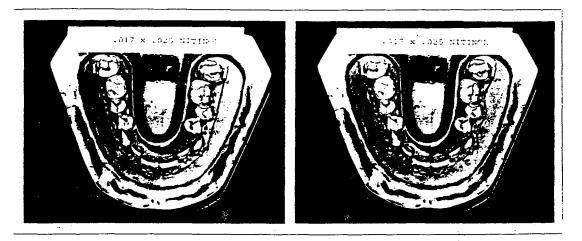


Figure 19. NiTi dental arch wire before (above) and after (below) being fastened to malposed teeth. (Courtesy G.F. Andreasen.)

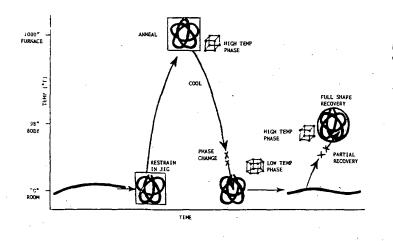


Figure 20. Schematic of vena cava filter operation. (Courtesy M. Simon.)

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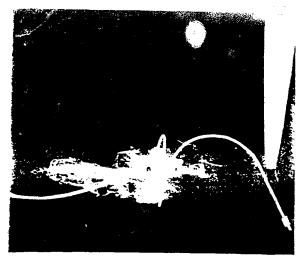


Figure 21. Radiograph showing placement of vena cava filter in a dog. (Courtesy M. Simon.)

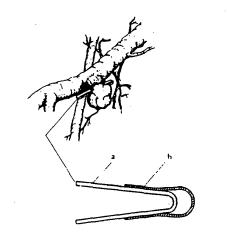


Figure 22. Composite aneurism clip; a.) of Ag; b.) of NiTi. (Courtesy T. Honma.)

Figure 23. Prototype artificial heart using NiTi contractile elements. (Courtesy P.N. Sawyer.)



Such devices can prevent  $95^{\circ}c$  of clots formed in the legs, peivis, or thighs, and which later dislodge, from reaching the heart (lung region and causing a pulmonary embolism. The NiTi filter offers several advantages, such as avoiding anticoagulant drugs (dangerous when internal bleeding may occur) and surgery, both risky procedures, and requiring only a local anesthetic. Such devices promise greater safety, simplicity, and speed of introduction.

Experiments on dogs have been very encouraging and those involving humans are anticipated soon. Figure 21, a radiograph of an NiTi filter implanted into the vena cava of a dog, clearly shows the trapped embolus at one side of the filter.

#### Intracranial Aneurism Clips

Aneurism clips or clamps used to tie off unwanted bulges which form in arteries have to be easily applied and removed. After experimenting with different versions of metallic aneurism clips, Honma and colleagues in Japan reported on a basic silver clip straddled by a supplementary clip made of NiTi. Such composite clips, as shown in Figure 22, met all mechanical conditions for practical use and could be removed easily by local heating.

#### **Artificial Hearts**

Dr. Philip Sawyer and associates at the State University of New York conceived the intriguing idea of using NiTi SME alloys to act as prosthetic muscles when heat pulsed, and they have proposed using NiTi wire strands as a contractile artificial muscle for an artificial heart. They claim that NiTi alloys are a potentially practical means of obtaining proximate contractility of the chamber wall and a satisfactory beginning towards developing an artificial muscle "skin-activated" cardiac chamber. The wire strands were initially constructed using "muscle groups" which were anchored to the exterior of the chamber in various configurations to attempt to replicate the contractility of the left heart ventricle.

Such artificial hearts are envisioned to be activated by electrical heating and programmed timing cycles involving various groups of contractile elements. Using NiTi elements attached to an elastomer chamber, significant pumping speeds were obtained. Such devices have pumped water up a 160 cm gradient 12-15 times per minute.

Sawyer and colleagues suggest that the next critical step is an evaluation *in vivo* following the implantation of such devices in dogs and calves. This is a novel, exciting possibility for applying an SME alloy. Figure 23 shows a prototype artificial heart with elastomer pumping elements activated by bands of NIT wires.

#### **Orthopedic Devices**

Workers at the Polytechnic Institute of New York (see Castleman et al.) have suggested the manufacture of bone plates from NiTi alloys for the compression fixation of bone fractures. A "preprogrammed" NiTi implant would be fastened to a fractured bone on each side of the fracture. Raising the plate temperature locally some 10-15F° above body temperature would cause the shape memory to contract the plate and fit the ends of the fractured bone securely together. Such a process would involve a much simpler surgical procedure than is now common, using implants made of vitallium and other allovs with 'stati. dimensions. However, recognizing the possible bioincon patibility of NiTi alloys, these workers carried out a tissuresponse study by implanting NiTi bone plates in dogs. No adverse tissue reactions. loss of implant material to surrounding tissue, corrosion effects, or contamination of body organs by implant materials were found, leading to the conclusion that NiTi alloys are sufficiently compatible with dog tissue to warrant further investigation as a biomaterial.

Another evaluation of NiTi alloys for orthopedic implants was conducted by Dr. James Hughes of the University of Mississippi. He also prepared bone plates (Figure 24) and

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confirmed the biologic acceptability of NiTi alloys. Hughes aiso suggested a new type of hip prosthesis and intrameduljary rod. Instead of cement to hold the stem of the prostnesis i:, place, the metallic components within the bone are firmin fixed by the gripping segments of NiTi in the stem of the prosthesis.

Another proposed orthopedic device involves using NiTi ailovs in a Harrington rod for straightening a bent spinal column.

# SOME FURTHER APPLICATIONS

A few additional applications of SME allows can be briefly mentioned. A U.S. patent has been obtained for a variety of intrauterine contraceptive devices (IUDs) fabricated from NiTi SME alloys. A Cu-Zn SME alloy has been proposed for manufacturing integrated circuit packages with SME used in making contacts. Finally, researchers at the Polytechnic Institute of New York have developed "blind plugs" of NiTi to be used as remotely activated, internally placed seals for old gas lines under streets in New York City.

#### CLOSURE

The previous discussion suggests that, from a metallurgical point of view. shape-memory alloys are reasonably well understood. Some of the many existing and proposed applications of this new class of materials have been described, and this admittedly nonexhaustive discussion indicates a remarkable variety of new things that can be done with these materials. Since the shape-memory effect is an intrinsic consequence of martensitic transformations, this type of phase change, once thought to be of interest only for quenched steels, takes on new dimension and importance. Many new developments and applications are expected now that the Biblical "serpent" has been tamed.

## ACKNOWLEDGMENTS

I would like to thank many friends and colleagues for providing information and photographs. They are too numerous to mention individually, but their contributions are identified in the text. I am particularly indebted to Dr. Tsugio Tadaki for translating this report into Japanese: thanks are also due Dr. Morris Simon for providing the Biblical description of the shape-memory effect. I wish to acknowledge the support of the National Science Foundation, the Materials Research Laboratory at the University of Illinois, and the Army Research Office.

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Figure 24. Radiograph showing bone plate made from NITi alloy attached to a bone. (Courtesy James Hughes.)

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Experimental Archaures, Camorna, 2006, Brochure on NiTi Dental Archaures. Foxboro Corporation, Foxboro, Massachusetts. 02035. Technical Information Sheets on Nitimol Fen Druce Unit

#### ABOUT THE AUTHOR

C.M. Wayman received a PhD in metallurgy from Lehigh University in 1957; he then joined the University of Illinois where he has been Professor of Metallurgy since 1964. An acknowledged expert in the field of martensitic transformation, he has written a book on the crystallography of martensitic transformations, contributed chapters to 10 others.

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and has authored or co-authored some 170 papers on martensitic transformations, thin films, and electron

In 1969 he was Visiting Professor at the University of Cambridge. England, and also NATO Lecturer. Guggenheim Fellow. and an Overseas Fellow of Churchill College. In 1978, he received the TMS-AIME Mathewson Gold Medal for "outstanding contributions to the study of thermoelastic martensitic transformations in alloys." A member of numerous committees in technical societies, he is also a noted lecturer and has presented keynote lectures on martensitic transformations, the shape-memory effect and its applications.



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	JAMES E. JERVIS	Examiner: David Kenealy
	Serial No.: 07/956,653 ) Filed: October 2, 1992 )	
	For: MEDICAL DEVICES ) INCORPORATING SIM ALLOY ) ELEMENTS )	Pasadena, California 40 M

# RESPONSE TO OFFICE ACTION

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Dear Madam/Sir:

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In response to the Office Action of February 22, 1993, please amend the above-identified patent application as follows:

# IN THE SPECIFICATION

In the Cross-reference section of the Preliminary Amendment, in the blank line after "U.S. Patent No.," please insert --5,190,546--.

Page 6, line 4, delete the language --Fig. 3 is a front plan view of a bone implant, namely a nail.--, as inadvertently added by the preliminary amendment. (There is no Figure 3 in the application as filed.)

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Page 6, line 4, after "martensite." please add:

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--Figure 3 is a side elevation view of a partial section of a catheter of the present invention in a stressed configuration.

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Figure 4 is a side elevation view of the catheter of Figure 3 in an unstressed configuration.--

# IN THE DRAWINGS

Please add Figures 3 and 4 to the drawings.

# IN THE CLAIMS

Please amend the following claim:

25. (once amended) A device as claimed in claim 20, the device having a transverse dimension, [in which] wherein the shape memory alloy element is deformed by the restraint, in such a way that its transverse dimension is reduced [, the restraint preventing] and transverse expansion of the element <u>is</u> obstructed.

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# RESPONSE TO RESTRICTION REQUIREMENT

In response to the Office Action mailed on February 22, 1993, applicant elects, the claims of Group I as defined by the Examiner, namely claims 11-53.

# PROVISIONAL ELECTION

The Office Action also requires a provisional election between the claims directed to an IUD, a stent graft, a blood filter, a catheter, and a tracheal catheter.

Applicant provisionally elects the catheter, with traverse. The generic and species claim directed to catheters are: 11-14, 17-21, 24-37, and 41-53.

Applicant respectfully submits that no election is required between tracheal catheters and catheters, because "tracheal catheters" are as their name suggests, merely species of catheters, namely those catheters that are inserted into the throat of a patient. This relationship is evident from the fact that Claim 36, which is directed to tracheal catheters, depends upon Claim 35 which is directed to the generic catheters. Thus, Applicant has included Claims 18 and 36, which are directed to tracheal catheters, in the list defined above.

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Applicant respectfully traverses this election requirement. Restriction is proper only when two or more independent and distinct inventions are claimed in one application. 35 U.S.C. § 121; 37 C.F.R. § 1.142. The IUD, stent graft, blood filter, and catheter devices are not independent because they have common design and operational features. The relationship between the devices is evident from the structure of the claims. Many of the claims to these devices were drafted as claims which are dependent upon independent claims which recited common structural and operational features. For example, Claim 14 (directed to a catheter), claim 15 (IUD), Claim 16 (blood filter), and Claim 18 (tracheal catheter) are all dependent on Claim 11 which recites common structural features. Similarly, Claims 38 (IUD), 39 (stent graft) and 40 (blood filter) are all dependent upon independent Claim 37 which has common design elements. Thus, these devices are connected in both design and operation.

Furthermore, if the Examiner requires election, is it the Examiner's position that these devices are patentably distinct, i.e., novel and non-obvious over each other as claimed? If they are not patentably distinct as claimed, a restriction requirement is improper. M.P.E.P. § 802.01.

Finally, Applicant wishes to direct the Examiner's attention to U.S. Patent No. 5,190,546, in which allowed Claim nos. 29-36 are directed to methods of using a bone implant,

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marrow nail, bone staple, IUD, blood filter, tubular device, catheter, and a tracheal catheter, respectively. Since these claims directed to separate devices were allowed in the parent application, Applicant respectfully submits that these claims should all be examined here.

# **REMARKS**

Figures 3 & 4 show a catheter with a tubular structure in the stressed and unstressed configurations, respectively. These drawings were added pursuant to the request of the Examiner. Both Figures 3 and 4 are merely copies of the drawings shown in U.S. Patent 3,890,977 to Wilson (Figures 1a and 1b in the Wilson patent). These figures are expressly incorporated by reference in the specification, on page 12, lines 7-8. Hence, no new matter is added.

Since applicant has provisionally elected the catheter species, applicant believes that it is not necessary at the present time to submit drawings showing the IUD or the filter for a blood vessel. Applicant will submit such drawings if and when it is determined that the generic claim is allowable.

# CLAIM 25

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As per the Examiner's request, claim 25 was amended to provide antecedent basis for "transverse dimension."

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# CLAIMS 41-44

The Office Action also objects to claims subsequent to claim 41 on grounds that these claims are "functional in nature, and it is difficult for this Examiner to comprehend what structures being claimed with the limitations being presented."

Applicant respectfully submits that claims 41-44 merely impose a limitation that the restraining means, placement device, or pin, does not change in state. Such limitations are allowed in claims. <u>See, In re Bankowski</u>, 318 F.2d 778, 138 U.S.P.Q. 75 (C.C.P.A. 1963) (limitation that virus propagating medium is "devoid of avian tissue" held not objectionable); <u>Johns-Manville Corp. v. Guardian Indus. Corp.</u>, 586 F.Supp. 1034 (E.D. Mich. 1983), <u>aff'd</u>, 770 F.2d 178 (Fed. Cir. 1985), (limitation in claim of "without using hot gas blast continuation" allowed); <u>In re</u> <u>Duva</u>, 387 F.2d 402, 156 U.S.P.Q. 90 (C.C.P.A. 1967) (limitation that read "absent sufficient CN ions to prevent decomposition" held proper).

# USE OF "ADAPTED"

The Office Action also states that the term "adapted" present in claims 49-53 has "little meaning in terms of patent language used for claiming a structural item." This rejection is respectfully traversed.

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A. M. Martin, M. M. M. Martin, and A. Martin

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Applicant has identified over 25 issued patents where the term "adapted" has been used for claiming a structural item. A partial listing of these issued patents includes: U.S. Patent 4,386,477 ("tube adapted to be inserted"); U.S. Patent 4,386,486 ("adapted to fit within the eye of an anchor insert. . .adapted to engage the anchor insert. . .adapted to receive"); U.S. Patent 4,386,646 ("adapted to be mounted"); U.S. Patent 4,431,111 ("adapted to engage"); U.S. Patent 4,431,155 ("adapted to receive").

Thus, applicant respectfully submits that the term "adapted" has widespread use for claiming a structural item, and consequently, should be allowed by the Examiner.

# **CONCLUSION**

For the reasons given above, entry of the amendments and allowance of the claims is respectfully requested.

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Respectfully submitted,

SHELDON & MAK

Dated:

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Ninth Floor Pasadena, CA 91101 (818) 796-4000

By Jeffrey G. Sheldon Reg. No. 27,953

225 S. Lake Avenue "EXPRESS MAIL" MAILING LABEL NO. TB309073644US. DATE OF DEPOSIT: March 22, 1993. I HEREBY CERTIFY THAT THIS PAPER OR FEE IS BEING DEPOSITED WITH THE UNITED STATES POSTAL SERVICE "EXPRESS" MAIL POST OFFICE TO ADDRESSEE SERVICE UNDER 37 CFR 1.10 ON THE DATE INDICATED ABOVE AND IS ADDRESSED TO THE COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D.C. 20231 BY MELISSA ALEXANDER Melisse (chander

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Fig. 4

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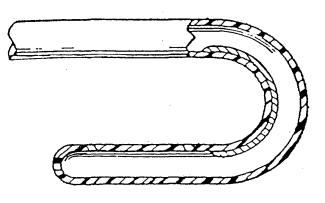


Fig. 3

Edwards Exhibit 1033, p. 104



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Edwards Exhibit 1033, p. 105

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IN RE APPLACEMON OF: JAMES E, JERVIS

SERIAL NO.: 07/956,653 FILED: OCTOBER 2, 1992

FOR: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

HONORABLE COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231

Sir:

Transmitted herewith is a paper in the above-identified application. Any necessary extension of time period set for this paper is hereby requested.

- No additional fee is required.
- $\Box$  The fee has been calculated as shown below:
- □ EXTENSION FEE

/		Non-Small Entity	RATE Small-Entity	FEE
	FIRST MONTH AFTER TIME PERIOD SET	110.00	55.00	\$
	SECOND MONTH AFTER TIME PERIOD SET	360.00	180.00	\$
	THIRD MONTH AFTER TIME PERIOD SET	840.00	420.00	\$
	FOURTH MONTH AFTER TIME PERIOD SET	1,320.00	660.00	\$

TOTAL EXTENSION FEE \$

FEE FOR EXTRA CLAIMS added by Amendment in this response:

	Column 1	Column 2	Column 3			
	Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	FEE
TOTAL CLAIMS		MINUS **	* =	x 20	x 10	\$
INDEPENDENT		MINUS ***	* =	x 72	x 36	\$
First presentation o	f multiple depende	nt claim		+ 220	+ 110	\$

TOTAL FEE FOR EXTRA CLAIMS \$

- * If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3.
- ** If the number of Total Claims previously paid for is less than 20, write "20" in this space.
- *** If the number of Independent Claims previously paid for is less than 3, write "3" in this space.

Enclosed is the fee of \$______by Check No.______

Please charge Deposit Account No. 19-2090 in the amount of \$ _____

The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19-2090:

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

Date:

SHELDO By: Reg. No.

"EXPRESS MAIL" mailing label number <u>TB309073644US</u> Date of Deposit <u>March 22, 1993</u>

I hereby certify that this paper is being deposited with the United States Postal Service "Express Mail Post Office to Addressee" service under 37 C.F.R. § 1.10 on the date indicated above and is addressed to the Commissioner of Patents and Trademarks, Washington, D.C. 20231.

Signature

225 South Lake Avenue Suite 900 Pasadena, California 91101 (818) 796-4000 (213) 681-9000

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Melissa Alexander Typed or Printed Name of Person Mailing Paper or Fee

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	In re Application of: $\bigcirc \bigcirc \lor$	Group Art Unit: 3301	Prior art
is.	JAMES E. JERVIS	) Examiner:	· · · · · ·
	Serial No.: 07/956,653		ter (
·	Filed: October 2, 1992		
	For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS	) ) ) Pasadena, California	

# SUPPLEMENTAL INFORMATION DISCLOSURE STATEMENT

) Pasadena, California

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Dear Madam/Sir:

ELEMENTS

Attached hereto is a PTO-1449 form listing documents believed relevant to the subject application. It is respectfully requested that this document be considered by the Examiner and an initialled copy of this form be returned to the undersigned.

It should be noted the word "prior" has been deleted from the form.

It is believed that this disclosure complies with the requirements of 37 C.F.R. § 1.56 and the Manual of Patent Examining procedures § 707.05(b). If for some reason the Examiner considers otherwise, it is respectfully requested that the undersigned be called so that any deficiencies can be remedied.

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These documents are not necessarily analogous art.

Enclosed is check no. 3929 for \$200, the fee due under 37 C.F.R. § 1.17(p).

Respectfully submitted,

Sheldon

SHELDON & MAK

Jeffrey G. Shel Reg. No. 27,953

193 Dated:

225 South Lake Avenue Ninth Floor Pasadena, California 91101 (818) 796-4000

Encls.

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I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U.S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOPE ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D.C. ON 4-2-93Melissa alfanda

BY: MELISSA ALEXANDER

Ву

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This is a communication from the COMMISSIONER OF PATENTS	e examiner in charge of your a	application.			06/24/93	
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Part II SUMMARY OF A	CTION					
1. 1. Claims	- 54				are pending in the application	
Of the abo	ve, claims15,110	122,25 3	8-40 and	_57 are	withdrawn from consideration	ı.
2. 🗹 Claims 1-	10				_ have been cancelled.	
3. 🗌 Claims					are allowed.	
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4. 🗹 Claims	-14, 17 -21, 24	•			are rejected.	:
5. 🛛 Claims	<u> </u>				_ are objected to.	
6. 🗋 Claims				re subject to restrict	on or election requirement.	
7 The application h	as been filed with informa					
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8. Formal drawings	are required in response t	to this Office action.				
					F.R. 1.84 these drawings	÷ .
are acceptal	ble. 🗌 not acceptable (s	ee explanation or No		ng, PTO-948).		
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The following is a quotation of the first paragraph of 35 U.S.C. § 112:

The specification shall contain a written description of the invention, and of the manner and process of making and using it, in such full, clear, concise, and exact terms as to enable any person skilled in the art to which it pertains, or with which it is most nearly connected, to make and use the same and shall set forth the best mode contemplated by the inventor of carrying out his invention.

The specification is objected to under 35 U.S.C. § 112, first paragraph, as failing to provide an adequate description of the invention. There is no basis in the specification for the variable "A(90)". In the claims it is not understood what limitation such a variable intends to claim. Although there is an A(90) column in applicant's description of the Quinn 4,505,767 device in the 9/25/92 preliminary amendment, applicant does not describe the characteristics that the variable represents nor the units of measurement.

Claims 11-14,17 and 18 are rejected under 35 U.S.C. § 112, first paragraph, for the reasons set forth in the objection to the specification.

Claims 13,14,33 are rejected under 35 U.S.C. § 112, second paragraph, as being indefinite for failing to particularly point out and distinctly claim the subject matter which applicant regards as the invention.

Claim 13 is indefinite because it contains language concerning transverse compression when there has been no transverse dimension set forth. Without drawings it is difficult to understand such

limitations.

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Claim 33 uses incorrect alternative language at line 10 where applicant claims, "the restraint externally OR internally engaging...".

It is requested that applicant use the standard "means for" language when claiming a structure that has a particular function and for which applicant desires broad coverage. For example, in claim 12, a preferable way of claiming the restraint is, "a restraint <u>means for</u> holding the shape memory alloy element in a deformed configuration...". This type of claim language should be used throughout the claims whenever applicant desires a function to have structural weight.

In response to applicant's remarks in the 3/22/93 Office response, this examiner was not stating that "adapted for" language is forbidden in any patent claim. However, such language does not carry with it the patentable weight that positively reciting a structure or its characteristics carries. Applicant is welcome to use "adapted for" language in the claims, but please note that this examiner gives little structural weight to such language.

The drawings are objected to under 37 C.F.R. § 1.83(a). The drawings must show every feature of the invention specified in the claims. Therefore, the restraint, the catheter, the tracheal catheter and the straight pin, must be shown or the feature canceled from the claims. No new matter should be entered.

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An election of species restriction is typically accomplished by applicant selecting one of the figured embodiments. Presently applicant does not have a drawing for each of the different embodiments, but has selected to prosecute the group of claims that are directed towards the catheter. It has been determined that claims 11-14,17-21,24-37 and 41-53 all depend from a generic claim. The claims for a tracheal catheter will be included.

The following is a quotation of 35 U.S.C. § 103 which forms the basis for all obviousness rejections set forth in this Office action:

A patent may not be obtained though the invention is not identically disclosed or described as set forth in section 102 of this title, if the differences between the subject matter sought to be patented and the prior art are such that the subject matter as a whole would have been obvious at the time the invention was made to a person having ordinary skill in the art to which said subject matter pertains. Patentability shall not be negatived by the manner in which the invention was made.

Subject matter developed by another person, which qualifies as prior art only under subsection (f) or (g) of section 102 of this title, shall not preclude patentability under this section where the subject matter and the claimed invention were, at the time the invention was made, owned by the same person or subject to an obligation of assignment to the same person.

Claims 11-14,17-21,24-37 and 41-53 are rejected under 35 U.S.C. § 103 as being unpatentable over Schreck in view of Wilson. Schreck discloses the use of a restraint means, 3 and 7, for holding a shape memory catheter, 4, in a predetermined configuration. A portion of the restraint, 7, is hollow so that the catheter is held within the restraint and compressed "transversely". The catheter displays "stress induced martensite

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behavior" at body temperature. It is not known whether the catheter has an A(90) temperature of not more than 0 degrees C due to the insufficiency of the specification. Wilson teaches the use of varying compositions in a shape memory alloy such that the temperature range where the alloy is in its transitional phase can vary between -396 and +331 degrees F. It is believed that the range of transitional temperatures being claimed would have been an obvious characteristic to have included in the material traits of Schreck's catheter in light of the teaching of Wilson because if one had desired a catheter that would return to its original shape at body temperature after being deformed at a temperature below 0 degrees C, one could have looked to Wilson to see that such a temperature range in a shape memory alloy is well known in the art.

-5-

Wilson also teaches the use of a curved shape being desirable at the catheter's unstressed state so that it can be placed in certain body regions with greater ease. It is also believed to have been obvious to have had the Schreck device's unstressed state be in a curved configuration instead of just at a larger diameter because if one had wanted to place the device at angle in the human body, one could have looked to Wilson to see how this is done in a shape memory catheter. Item 3 of Schreck could be obviously replaced by a straight pin for design reasons if a fenestration already existed in the area that the catheter is desirably inserted.

It is requested that applicant define more structural

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limitations that allow applicant's device to function in the manner described in the claims. Some of applicant's limitations are strictly functional, such as: "can be extruded from the device at a temperature..", "the hollow placement device stressing the memory alloy at a temperature...", "transformation of the alloy occurs without any change in the state..." and more. What are the structural limitations that allow the device to function in these manners and how are they different from the prior art?

Any inquiry concerning this communication or earlier communications from the examiner should be directed to David Kenealy whose telephone number is (703) 308-2680.

Any inquiry of a general nature or relating to the status of this application should be directed to the Group receptionist whose telephone number is (703) 308-0858.

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MICKEY YU PRIMARY EXAMINER ART UNIT 331

David Kenealy June 13, 1993

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MENDMENT COVER S	HEET
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IN RE APPLICATION OF: JAMES E. JERVIS	340-114 331
SERIAL NO.: 07/956,653	FILED: October 2, 1992
FOR: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS	I A RECEIVED
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Washington, D.C. 70931	DEC 1 4 1993
Sir: NOV	GROUP 330
Transmitted her with is a paper in the above-identified application. Any necessar	y extension of time period set for this paper is b
hereby requested. 300	

No additional fee is required. []

[X] The fee has been calculated as shown below:

[X] EXTENSION FEE

	Non-Small Entity	Small-Entity	FEE
FIRST MONTH AFTER TIME PERIOD SET	110.00	55.00	\$
SECOND MONTH AFTER TIME PERIOD SET	360.00	180.00	\$360.00
THIRD MONTH AFTER TIME PERIOD SET	840.00	420.00	\$
FOURTH MONTH AFTER TIME PERIOD SET	1,320.00	660.00	\$

[X] TOTAL EXTENSION FEE \$360.00

[] FEE FOR EXTRA CLAIMS added by Amendment in this response:

-	Column 1	Column 2	Column 3		·····	
-	Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	FEE
TOTAL CLAIMS	46	MINUS **	* = 3	x22	x 10	\$.66.00
INDEPENDENT	8	MINUS ***	* = 1	x 7.4	x 36	\$ 74.00
First presentation of	f multiple depende	ent claim		+ 220	+ 110	\$

TOTAL FEE FOR EXTRA CLAIMS \$132.00

- * If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3.
- ** If the number of Total Claims previously paid for is less than 20, write "20" in this space. ***
- If the number of Independent Claims previously paid for is less than 3, write "3" in this space.
- Enclosed is the fee of \$500.00 by Check No. U 51 [X]

Please charge Deposit Account No. 19-2090 in the amount of \$ []

The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19,0090: [X]

HOR

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

Date:

SHELDQN & MAK By: Jeffrey G. Sheldon, Reg. No.: 27,953

CERTIFICATE OF MAILING: I hereby certify that this correspondence is being deposited with the U.S. Postal Service as first class mail in an envelope addressed to: Commissioner of Patents and Trademarks, Washington, D. C. 20231 on November 24, 1993

Date Signed:

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By

225 SOUTH LAKE AVENUE, SUITE 800 07956653 SHELDON & MAK PASADENA, CALIFORNIA 91101 (838) 796-4000 - Direct Line: (818) 356-1201

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ह	for the UNITED STATES PATENT AN	438 HP0884-US7
	In re application of:	) Group Art Unit: 3301 attach)
	JAMES E. JERVIS	) Examiner: KENEALY, D ) RECENCED
	Serial No. 07/956,653	DEC 1 4 1993
	Filed: October 2, 1992	) GROUP 330
	For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS	) Pasadena, California
· · ·	AMENDMENT Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231 Sir:	I HEREBY CERTIFY THAT THIS COURSE OF DEMO IS BEING DEFORTED WITH THE V.S. POSTAL SERVICE AS FROM THE THE V.S. POSTAL ADDREAMED TO A COURSE OF THE TWE DWELCOE ADDREAMENTED TO A COURSE OF THE TAKENTS AND TRADEMARKE, YOU THANK E. D. 20231 ON NOVEMBER 24, 1993 Manufacture (DATE SIGNED)
	In response to the Office Ac	tion of June 24, 1993.

please amend the above-identified application as follows:

## IN THE SPECIFICATION

At page 8, between lines 24 and 25, and after the table added by the preliminary amendment filed with the application,

please insert:

--The A(90) temperature is the temperature at which the transformation from the martensitic phase to the austenitic phase is 90% complete.--

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120 WP 12/13/93 07956653	after the description of Figures 3

and 4 added by the Amendment of March 22, 1993, please insert:

--Fig. 5 is a tracheal catheter, which is curved in its unstressed configuration, partially straightened by a straight pin restraint.

Fig. 6 shows an IUD formed at least partly from a pseudoelastic shape-memory alloy being restrained in a deformed shape by a restraining tube.--

At page 13, line 24, before "When", insert/--The IUD  $\partial / \partial h$ as a longitudinal dimension and a transverse dimension.--

## IN THE CLAIMS

Please amend the claims as follows:

12. (Amended) A device as claimed in Claim 11, which includes a restraint [by means of which] <u>holding</u> the shape memory alloy element [is held] in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress-induced martensite.

13. (Amended) A device as claimed in claim 12, in which the restraint is hollow, and the shape memory alloy element has a transverse dimension and a longitudinal dimension, and wherein the shape memory alloy element is deformed [in such a way that it is] by being compressed transversely, and is positioned

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within the restraint, the restraint preventing transverse expansion of the element.

(Amended) A medical device which comprises: (a) an element for use within a mammalian body or in such proximity to a mammalian body that the device is substantially at body temperature, the element comprising a shape memory alloy which displays stress-induced martensite behavior at body temperature; and

(b) a restraint [by means of which] <u>holding</u> the shape memory alloy element [is held] in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body the deformation occurring through the formation of stress-induced martensite;

wherein [the device is adapted so that] the element is sufficiently deformed that removal of the restraint from the shape memory alloy element, without change in temperature of the device, releases at least a portion of the element from its deformed configuration.

25. (Amended) A device as claimed in claim 26, in which the shape memory alloy element <u>has a transverse dimension</u> and a longitudinal dimension, and wherein the shape memory alloy <u>element</u> is deformed [in such a way that] by its transverse dimension [is] <u>being</u> reduced, <u>and wherein</u> the restraint [preventing] <u>prevents</u> transverse expansion of the element.

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26. (Amended) The device of claim 19, wherein <u>the</u> <u>shape memory alloy element is sufficiently deformed that</u> removal of the restraint from the shape memory alloy releases at least a portion of the shape memory alloy element from its deformed configuration without change in state of the restraint.

27. (Amended) A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising: [(i) a restraining means and (ii)]

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) [the] restraining means engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the <u>alloy is selected so that removal of the</u> restraining means [is adapted to be removed] from the memory alloy element at a temperature greater than the As of the alloy when the device is placed within or proximate to the mammalian body. [to transform] transforms at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape toward its

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unstressed shape, [and wherein the device is adapted so that the transformation can occur] without any change in temperature of the restraining means or the memory alloy element <u>being required</u> for the transformation of the alloy.

28. (Amended) A medical device for treatment of a mammalian body, the device comprising: [(i) a hollow restraining member and (ii)]

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) a hollow restraining member with the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the [device is adapted] <u>alloy</u> <u>is selected</u> so that the transformation can occur without any

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change in temperature of the restraining member or the memory alloy element.

(Amended) The medical device of claim 28 wherein the restraining member is a tube and the memory alloy element is axially slidable within the tube, and wherein the [device is adapted so] <u>memory alloy element is sufficiently long</u> that relative axial movement between the tube and the memory alloy element extends at least a portion of the memory alloy element beyond the tube and thereby transforms the memory alloy element toward its austenitic state.

Claim 31, line 21, change "device is adapted so" to -alloy is selected--.

Claim 32, line 3, change "device is adapted so" to --tube is sufficiently long--.

Claim 33, Tine 10, delete "externally or internally"; in lines 19-20; and change "device is adapted" to --alloy is selected--.

Claim 35, line 18, change "device is adapted" to --alloy is selected--.

Claim 37, line 21, change "device is adapted to --alloy is selected--.

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Claim 49, lines 1-2, change "device is adapted" to --alloy is selected--.

AST. (Amended) The device of claim AST wherein [the device is adapted so that] (i) the restraining member can be completely disengaged and separated from the memory alloy element, and (ii) the alloy is selected so that engaging the restraining member with the memory alloy element after separation results in the memory alloy element transforming towards its deformed shape by reversion to its stress-induced martensitic state.

F. (Amended) The device of claim 23 wherein [the device is adapted so that] (i) the restraining member can be completely disengaged and separated from the catheter, and (ii) the alloy is selected so that reengaging the restraining member with the catheter after separation results in the catheter transforming toward its easily inserted shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

A 52. (Amended) The device of claim 35 wherein [the device is adapted so that] (i) the pin can be completely disengaged and separated from the catheter, and (ii) <u>the alloy is</u> <u>selected so that</u> reengaging the restraining means with the memory alloy element after separation results in the catheter transforming toward its deformed shape by reversion of at least a

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portion of the alloy from its austenitic state to its stressinduced martensitic state.

53. (Amended) The device of claim 36 wherein [the device is adapted so that] (i) the placement device can be completely disengaged and separated from the catheter, and (ii) <u>the alloy is selected so that</u> reengaging the placement device with the catheter after separation results in the catheter transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stressinduced martensitic state.

Please add the following claims to the application:

3555. The medical device of claim 23 wherein the restraint externally engages the catheter. 15

5632 15 The medical device of claim 33 wherein the

restraint internally engages the catheter.

F4 51:56. A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising (i) a restraint, and (ii) an element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite by wirtue of being above its A and above its M and

martensite by virtue of being above its  $A_s$  and above its  $M_s$  and below its  $M_d$  at about body temperature;

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such that it has a stress-induced martensitic state and an austenitic state, the element having (1) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

wherein the restraint is capable both of stressing the element and of being at least partially removed from the element while the device is within or proximate to the said body at the said body temperature and the element is therefore at an operating temperature greater than the  $A_s$  and  $M_s$  and below the  $M_d$ of the alloy,

portion of the alloy to transform from its stress-induced martensitic state to its austenitic state so that the element spontaneously transforms from its deformed shape toward its unstressed shape,

and such transformation can occur without a change in temperature of the restraint or of the element from the operating temperature.

### IN THE DRAWINGS

Please approve new Figures 5 and 6 accompanying this amendment.

#### REMARKS

Claims 11-56 are pending in this application. Claims 54-56 are added by this amendment. Claim 15, 16, 22, 23, and 38-40 are withdrawn from consideration. Claims 11-14, 17, 18, and

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33 were rejected under 35 U.S.C. § 112. The drawings were objected to under 37 C.F.R. § 1.83(a). All of the claims originally submitted were rejected under 35 U.S.C. § 103. Reexamination, reconsideration, and allowance are respectfully requested.

It is noted the Examiner referred to claim 54 in the Office Action. It is believed this was in error in that there was no claim 54 in the application prior to this amendment.

Entry of the amendments is respectfully requested. No new matter is added by the amendments in that they are clearly supported by the specification. In particular, the amendments to the claims are to meet some objections raised by the Examiner. For example, new claims 54 and 55 result from deletion of the alternative language that appeared in claim 33. Claim 56 is similar to a claim undergoing prosecution in Europe based on the present application.

New Fig. 5 is based on Fig. 4, with a pin inserted as described in the specification. Fig. 6 is based on drawings in U.S. Patent No. 3,620,212, which was incorporated by reference at page 13, lines 2-5. The new drawings have been added pursuant to the request of the Examiner to include in the application a drawing showing all features claimed. Since these features are claimed, and the claims are supported by the specification, no new matter is added.

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The reference explaining the significant of A(90), which was added to page 8 of the specification, is based on the amendment and declaration filed in great-grandparent Application Serial No. 177,817, filed March 30, 1988, in response to an Office Action filed August 11, 1988. Relevant portions of that amendment and the declaration are provided herewith for the convenience of the Examiner. As is made clear that amendment and declaration, A(90) is an inherent property of the materials listed in the table.

## REJECTIONS AND OBJECTIONS UNDER 35 U.S.C. § 112

The specification was objected to as failing to provide an adequate description of "A(90)". By this amendment, that objection has been obviated. This also obviates the rejection of claims 11-14, 17, and 18 under 35 U.S.C. § 112, based on the same grounds.

Claim 13 was objected to because of reference to transverse compression. This objection has been obviated by providing an antecedent basis in claim 13 for a transverse dimension. Moreover, the specification has been amended to make it clear that an IUD has a transverse and longitudinal dimension, which is inherent in such a device.

The rejection to claim 33 has been obviated by removing the alternative language.

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Applicant has used "means for" language where appropriate. However, in view of recent decisions by the Federal Circuit, where it is unclear exactly what scope the Federal Circuit will apply to "means for" language, Applicant has elected to use broad language, such as "restraint".

With regard to the use of "adapted for" language, Applicant respectfully disagrees with the Examiner's position that "adapted for" language cannot be considered in determining the patentability of the claimed invention. Nevertheless, to avoid controversy, Applicant has revised the claims, without prejudice, to eliminate the "adapted for" language.

In view of the foregoing remarks, removal of the objections and rejections under 35 U.S.C. § 112 is respectfully requested.

## OBJECTION TO THE DRAWINGS

The drawings were objected to for not showing every feature of the invention specified in the claims. The drawings have been amended to specify all such features. Accordingly, approval of the accompanying drawings and removal of the objection are respectfully requested.

## REJECTION UNDER 35 U.S.C. § 103

All the claims under consideration were rejected under 35 U.S.C. § 103 as being unpatentable over Schreck U.S. Patent

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No. 4,411,655 in view of Wilson U.S. Patent No. 3,890,977. This rejection is respectfully traversed.

This rejection assumes that Schreck's catheter displays "stress-induced martensite behavior". As detailed below, Schreck teaches no such thing. Thus, no prima facie case of obviousness has been made, and the rejection should be withdrawn.

As is made clear throughout the specification, a key feature of the claimed invention is a medical device that comprises an alloy in its stress-induced martensitic state. By way of background, there are two techniques available for transforming an appropriate alloy into the martensitic state. The first technique, which is the common technique, is cooling the material so that martensite forms at  $M_s$  under no stress. By the second technique, the same material, martensite can form above  $M_s$  if stress is applied, thereby forming "stress-induced martensite". The present invention is directed to use of the unique properties of stress-induced martensite, not martensite formed by cooling.

Contrarily, Schreck only teaches conventional martensite formed by cooling. In particular, Schreck first describes a "one way" mode where "a martensite phase transformation . . . occurs as the specimen is cooled through the appropriate temperature range  $(M_s \rightarrow M_f)$ ." (Column 3, lines 18-20)

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Thus, it is clear that in Schreck's "one way" mode, martensite is formed through conventional cooling.

Schreck then goes on to discuss a "two way" or reversible mode. Again, however, martensite is clearly formed by cooling. Schreck states "if a shape-memory alloy is deformed beyond a minimum stained (sic, strained) value <u>while in its</u> <u>martensite phase</u>, the original "parent" shape will be recovered on heating the specimen above the phase change as in the 'one way' effect." (emphasis added; Column 3, lines 32-37) Thus, clearly, Schreck is referring to deformation <u>after</u> the alloy is in its martensite phase, and not the use of stress or deformation to put the alloy into its martensite phase. Thus, Schreck has no teaching of stress-induced martensite.

Moreover, by no stretch of the imagination, can Schreck's elastomeric sleeve be considered to be a restraint that causes stress-induced martensite transformation. As Schreck states at column 2, lines 21-25 "the shape memory alloy cannula may advantageously be encased in elastomeric sleeve whose lumenal diameter is stretched by the SMA material as it equilibrates to the desired temperature." Thus, the elastomeric sheath is deformed by the dilating cannula, not the reverse. There is no suggestion that the sheath could possibly deform the cannula and cause phase transformation, i.e., the formation of stress-induced martensite. Indeed, there is no mention of stress-induced martensite anywhere in Schreck.

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Moreover, as specifically recited in many of the claims, such as claims 19, 27, 28, 31, 33, 35, 37, and 56, and the claims dependent therefrom, the changes that take place in the memory alloy element occur "without any change in temperature of the restraining means or the memory alloy element be required for the transformation of the alloy." (This is language quoted from claim 27). This is clearly not what is taught by Schreck, who specifically requires the temperature of the shape-memory alloy change to dilate the cannula (see for example, column 2, line 7-16 of Schreck). This is a significant advantage of Applicant's invention in that recovery is achieved <u>without</u> temperature change. Schreck does not have this feature.

Therefore, for the foregoing reasons, and, in particular, because no prima facie case of obviousness has been made, removal of the rejection under 35 U.S.C. § 103 is respectfully requested.

## INFORMATION DISCLOSURE STATEMENT

Applicant has obtained a translation of Oonishi, <u>Clinical Magazine: Orthopedic Surgery</u>, 32, page 1180 (1981) which was listed as one of the references cited to the Examiner as part of the information disclosure statement that accompanied the original application. A copy of the translation is enclosed.

Applicant appreciates the fact that the Examiner considered some of the references cited. However, the Examiner

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has not acknowledged considering references that were cited as part of a supplemental information disclosure statement that was mailed April 2, 1993 (copy enclosed), nor references cited in the original application (pages 1-7 of PTO-1449 forms). It is respectfully requested that the Examiner acknowledge consideration of all these references. Copies of these seven pages are provided herewith for the convenience of the Examiner.

If for some reason the Examiner does not have copies of any of these references, please notify the undersigned and courtesy copies will be hand carried to the Examiner. Applicant respectfully requests the Examiner to initial all of the attached forms and return them to the undersigned.

### CONCLUSION

For reasons detailed above, it is believed that the present application is in condition for allowance. If for some reason the Examiner considers otherwise, it is respectfully requested that a telephone call be placed to the undersigned to resolve any remaining difficulties. It is the undersigned's experience that such telephone calls can often lead to early disposal of applications.

Respectfully submitted,

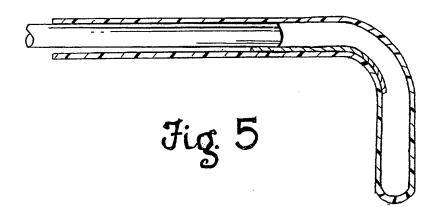
SHELDON & M. Sheldon, Reg. No. 27,953 By

Date

225 South Lake Avenue, Suite 900 Pasadena, California 91101 - Phone: (818) 796-4000

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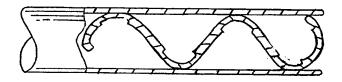


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# United Stater Patent [19]

## Quin

### [54] NICKEL/TITANIUM/VANADIUM SHAPE MEMORY ALLOY

- [75] Inventor: Mary P. Quin, Redwood City, Calif.
- [73] Assignee: Raychem Corporation, Menlo Park. Calif.
- [21] Appl. No.: 541.844
- [22] Filed: Oct. 14, 1983
- [51] Int. CL¹ ..... C22C 19/00; C22C 30/00
- - +20/441
- [58] Field of Search ..... 148/402, 11.5 F, 11.5 N; 420/442, 441

## [56] References Cited

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3.174.851	3/1965	Buehler et al
3.351.463	11/1967	Rozner et al
3,558,369	1/1971	Wang et al
3.620.212	11/1971	Fannon et al 128/130
3.740.839	6/1973	Otte et al
3.753.700	8/1973	Harrison et al 148/402
3,786,806	1/1974	Johnson et al 128/92 D
3.832.243	8/1974	Donkersloot et al 148/402
3.890.977	6/1975	Wilson
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4.035.007	7/1977	Harrison
4.144.057	3/1979	Melton et al 148/11.5 N
4.198.081	4/1980	Harrison
4.205.293	5/1980	Melton et al
4.310.354	1/1982	Fountain et al 75/211

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### OTHER PUBLICATIONS

Alloys Index, vol. 8, 1981, p. E-758, "Ti48Ni43V9", Alloys Index, vol. 9, 1982, p. E-871, "48Ti-43Ni-9V", U.S. Patent Application Ser. No. 541.852, Applicant: James Jarvis.

Buehler et al., (Mater, Des. Eng., pp., 82-83, (Feb. 1962)).

Wang et al., J. App. Phys., V. 36, pp. 3232-3239, (1965). Wasilewski et al., Met. Trans., v. 2, pp. 229-238, (1971). U.S. Naval Ordinance Laboratory Report NOLTR 64-235, (Aug. 1965).

Honma et al., Res. Inst. Min. Dress. Met. Report No. 622, (1972).

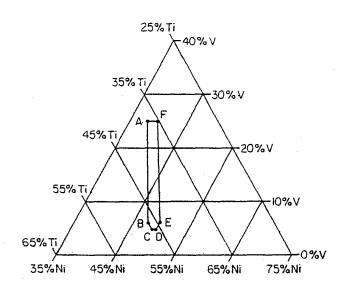
Kovneristii et al., Proc. 4th Int. Conf. on Titanium, v. 2, pp. 1469-1479.

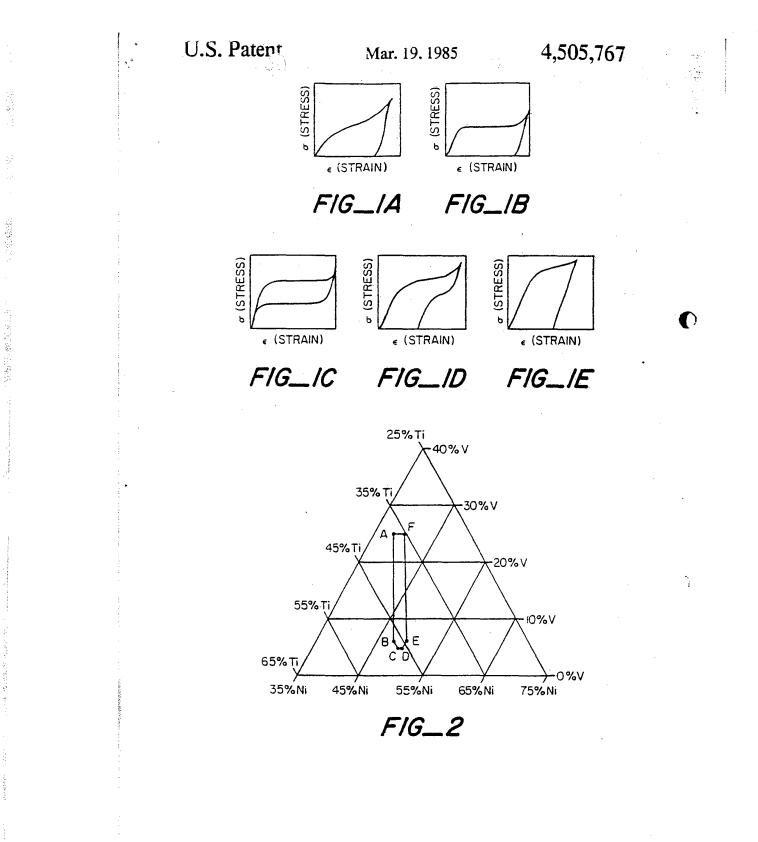
Primary Examiner—Peter K. Skiff Attorney, Agent. or Firm—Ira D. Blecker: James W. Peterson: Herbert G. Burkard

#### [57] ABSTRACT

Nickel/titanium alloys having a nickel/titanium atomic ratio between about 1:02 and 1:13 and a vanadium content between about 4.6 and 25.0 atomic percent show constant stress versus strain behavior due to stressinduced martensite in the range from about 0° to 60° C.

#### 8 Claims, 6 Drawing Figures





Edwards Exhibit 1033, p. 137

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#### NICKEL/TITANIUM/VANADIUM SHAPE MEMORY ALLOY

#### BACKGROUND OF THE INVENTION

#### Field of the Invention

This invention relates to nickel/titanium shape memory alloys and improvements therein.

#### Introduction to the Invention

Materials, both organic and metallic, capable of posessing shape memory are well known. An article made of such materials can be deformed from an original, heat-stable configuration to a second, heat-unstable configuration. The article is said to have shape memory for the reason that, upon the application of heat along, it can be caused to revert, or to attempt to revert, from ats heat-unstable configuration to its original, heat-stable configuration, i.e. it "remembers" its original shape.

20 Among metallic alloys, the ability to possess shape memory is a result of the fact that the alloy undergoes a reversible transformation from an austenitic state to a martensitic state with a change in temperature. This transformation is sometimes referred to as a thermoelas-25 the martensitic transformation. An article made from such an alloy, for example a hollow sleeve, is easily deformed from its original contiguration to a new configuration when cooled below the temperature at which the alloy is transformed from the austenitic state to the 30 martensitic state. The temperature at which this transformation begins is usually referred to as M, and the temperature at which it finishes Mr. When an article thus deformed is warmed to the temperature at which the alloy starts to revert back to austenite, referred to as 35  $A_{1}$  (A_f being the temperature at which the reversion is complete) the deformed object will begin to return to its original configuration.

Shape memory alloys (SMAs) have found use in recent years in. for example, pipe couplings (such as are described in U.S. Pat. Nos. 4.035.007 and 4.198.081 to 40 Harrison and Jervis), electrical connectors (such as are described in U.S. Pat. No. 3.740.839 to Otte and Fischer), switches (such as are described in U.S. Pat. No. -2.05.293), actuators, etc.

Various proposals have also been made to employ ⁴⁵ shape memory alloys in the medical field. For example, U.S. Pat. No. 3.620,212 to Fannon et al. proposes the use of an SMA intrauterine contraceptive device, U.S. Pat. No. 3.786.806 to Johnson et al. proposes the use of an SMA bone piate, U.S. Pat. No. 3.890,977 to Wilson ⁵⁰ proposes the use of an SMA element to bend a catheter or cannula, etc.

These medical SMA devices rely on the property of shape memory to achieve their desired effects. That is to say, they rely on the fact that when an SMA element ⁵⁵ is cooled to its martensitic state and is subsequently deformed, it will retain its new shape; but when it is warmed to its austenitic state, the original shape will be recovered.

However, the use of the shape memory effect in med- 60 ical applications is attended with two principal disadvahtages. First, it is difficult to control the transformation temperatures of shape memory alloys with accuracy as they are usually extremely composition-sensitive, although various techniques have been proposed 65 including the blending by powder metallurgy of already-made alloys of differing transformation temperajures: see U.S. Pat. No. 4.310.354 to Fountain et al.).

Second, in many shape memory alloys there is a largehysteresis as the alloy is transformed between austenitic and martensitic states, so that reversing of the state of an SMA element may require a temperature excursion of several tens of degrees Celsius. The combination of these factors with the limitation that human tissue cannot be heated or cooled beyond certain relatively narrow limits without suffering temporary or permanent damage is expected to limit the use that can be made of SMA medical devices.

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In copending and commonly assigned U.S. patent application (Ser. No. 541.844, filed 10/14/83) to Jervis, the disclosure of which is incorporated herein by reference, it is proposed that the stress-induced martensite (SIM) properties of shape memory alloys be employed in SMA medical devices, rather than the use of the shape memory effect.

When an SMA sample exhibiting stress-induced martensite is stressed at a temperature above M₅ (so that the austenitic state is initially stable), it first deforms elastically and then, at a critical stress, begins to transform by the formation of stress-induced martensite. Depending on whether the temperature is above or below An the behavior when the deforming stress is released differs. If the temperature is below An the stress-induced martensite is stable; but if the temperature is above A₅, the martensite is unstable and transforms back to austenite. with the sample returning (or attempting to return) to its original shape. The effect is seen in almost all alloys which exhibit a thermoelastic martensitic transformation, along with the shape memory effect. However, the extent of the temperature range over which SIM is seen and the stress and strain ranges for the effect vary greatly with the alloy. For many purposes, it is desirable that the SIM transformation occur at a relatively constant stress over a wide strain range, thereby enabling the creation of, in effect. a constant force spring.

Various alloys of nickel and titanium have in the past been disclosed as being capable of having the property of shape memory imparted thereto. Examples of such alloys may be found in U.S. Pat. Nos. 3,174,851 and 3,351,463.

Buehler et al (Mater. Des. Eng., pp.82-3 (Feb. 1962); J. App. Phys., v.36, pp.3232-9 (1965)) have shown that in the binary Ni/Ti alloys the transformation temperature decreases dramatically and the yield strength increases with a decrease in titanium content from the stoichiometric (50 atomic percent) value. However, lowering the titanium content below 49.9 atomic percent has been found to produce alloys which are unstable in the temperature range of 100° C. to 500° C., as described by Wasilewski et al., Met. Trans., v.2, pp. 229-38 (1971). The instability (temper instability) manifests itself as a change (generally an increase) in Mr between the annealed alloy and the same alloy which has been further tempered. Annealing here means heating to a sufficiently high temperature and holding at that temperature long enough to give a uniform. stressfree condition, followed by sufficiently rapid cooling to maintain that condition. Temperatures around 900° C. for about 10 minutes are generally sufficient for annealing, and air cooling is generally sufficiently rapid. though quenching in water is necessary for some of the low Ti compositions. Tempering here means holding at an intermediate temperature for a suitably long period (such as a few hours at 200°-400° C.). The instability thus makes the low titanium allovs disadvantageous for

shape memory application ) where a combination of high yield strength and reproducible M, is desired.

Although certain cold-worked binary mackeletitanium alloys have been shown to exhibit SIM, these alloys are difficult to use in practice because, in order to 5 obtain the appropriate M, to give SIM properties at physiologically acceptable temperatures, the alloys must have less than the stoichtometric titanium content. These binary alloys then are (1) extremely compositionsensitive in M, as referred to above for shape memory; (2) (2) unstable in M, with aging and sensitive to cooling rate: and (3) require cold-working to develop the SIM. so that any inadvertent plastic deformation is not recoverable simply by heat-treatment: new cold-working is required.

Certain ternary Ni/Ti alloys have been found to overcome some of these problems. An alloy comprising 47.2 atomic percent nickel, 49.6 percent titanium, and 3.2 atomic percent nickel, 49.6 percent titanium, and 3.2 atomic percent iron (such as disclosed in U.S. Pat. No. 3,753,700 to Harrison et al.) has an  $M_{4}$  temperature 29 near  $-100^{\circ}$  C, and a yield strength of about 70,000 psi. While the addition of iron has enabled the production of alloys with both low  $M_{4}$  temperature and high yield strength, this addition has not solved the problem of instability, nor has it produced a great improvement in  $\frac{19}{100}$  the  $M_{4}$  temperature to compositional change.

U.S. Pat. No. 3.558.369 shows that the M, temperature can be lowered by substituting cobalt for nickel, then iron for cobalt in the stoichiometric alloy. Howyou ever, although the alloys of this patent can have low transformation temperatures. they have only modest yield strengths (40.000 psi or less).

U.S. Naval Ordnance Laboratory Report NOLTR 64-235 (August 1965) examined the effect upon hard-35 ness of ternary additions of from 0.08 to 16 weight percent of eleven different elements, including vanadium, to stoichiometric Ni/Ti. Similar studies have been made by, for example, Honma et al., Res. Inst. Min. Dress. Met. Report No. 622 (1972) and Proc. Int. :0 Conf. Martensitic Transformations (ICOMAT "9), pp. 259-264: Kovneristii et al., Proc. 4th Int. Conf. on Titanium. v. 2. pp. 1469-79 (1980); and Donkersloot et al., U.S. Pat. No. 3.832.243, on the variation of transformation temperature with ternary additions, also including vanadium. These references, however, do not describe any SIM behavior in the alloys studied.

It would thus be desirable to develop an alloy which exhibits stress-induced martensite in the range from  $0^{\circ}$ to 60° C, which is preferably of low composition sensitivity for ease of manufacture.

#### DESCRIPTION OF THE INVENTION

#### Summary of the Invention

I have discovered that the addition of appropriate 55 amounts of vanadium to nickel/titanium shape memory alloys permits the production of workable alloys exhibiting stress-induced martensite in a physiologically acceptable temperature range, when in the fully annealed condition (i.e. no cold working is required to produce 60 the desired mechanical properties).

This invention thus provides a shape memory alloy consisting essentially of nickel, titanium, and vanadium within an area defined on a nickel, titanium, and vanadium ternary composition diagram by a hexagon with 65 its first vertex at 38.0 atomic percent nickel, 37.0 atomic percent titanium, and 25.0 atomic percent vanadium; its second vertex at 47.6 atomic percent nickel, 46.4 atomic

percent titaniu () and 6.0 atomic percent vanadium; its third vertex at +9.0 atomic percent nickel, 46.4 atomic percent titanium, and 4.6 atomic percent vanadium; its fourth vertex at 49.8 atomic percent nickel, 45.6 atomic percent titanium, and 4.6 atomic percent vanadium; its fifth vertex at 49.8 atomic percent nickel, 44.0 atomic percent titanium, and 6.2 atomic percent vanadium; and its sixth vertex at 39.8 atomic percent nickel, 35.2 atomic percent titanium, and 25.0 atomic percent vanadium.

#### BRIEF DESCRIPTION OF THE DRAWING

FIGS, 1A through 1E are typical stress-strain curves for shape memory alloys at various temperatures.

FIG. 2 is a nickel/titanium/vanadium ternary composition diagram showing the area of the alloy of this invention.

#### DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

FIGS. 1A through 1E are typical stress-strain curves for shape memory alloys at various temperatures. Ignoring, for the moment, the difference between M, and Ms and between A, and As the behavior of a shape memory alloy may be generally seen to fit with one of these Figures.

In FIG. 1A, T is below M., The alloy is initially martensitic, and deforms by twinning beyond a low elastic limit. This deformation, though not recoverable at the deformation temperature, is recoverable when the temperature is increased above A., This gives rise to the conventional shape memory effect.

In FIG. 1B. T is between M, and M_d (the maximum temperature at which martensite may be stressinduced), and below A., Here, though the alloy is initially austenitic, stress results in the formation of martensite permitting ready deformation. Because the alloy is below A,, the deformation is again not recoverable until heating to above A, results in the transformation back to austenite If the sample is unrestrained, the original shape will be completely recovered: if not, it will be recovered to the extent permitted by the restraint. However, if the material is then allowed to re-cool to the temperature of deformation, the stress produced in the alloy is constant regardless of the strain provided that the strain lies within the "plateau" region of the stress-strain curve. This means that a known, constant force (calculable from the height of the stress plateau) can be applied over a wide (up to 5% or more) strain range.

In FIG. 1C, T is between M, and M_d, and above A., Here, the stress-induced martensite is thermally unstable and reverts to austenite as the stress is removed. This produces, without heating, what is, in effect, a constantforce spring acting over a strain range which can be about 5%. This behavior has been termed stressinduced martensite pseudoelasticity.

FIG. 1D shows the situation where T is near  $M_d$ . Although some stress-induced martensite is formed, the stress level for martensite formation is close to the austentic yield stress of the alloy and both plastic and SIM deformation occur. Only the SIM component of the deformation is recoverable.

FIG. 1E shows T above  $M_{d}$ . The always-austenitic alloy simply yields plastically when stressed beyond its elastic yield point and the deformation is non-recoverable.

an behavior shown in these The type of stress FIGS. 1A through 1E will hereafter be referred to as Athrough E-type behavior.

Constant stress over a wide strain range is desirable

For a ser, of samples, stress-strain curves were measured at temperatures between - 10° and 60° C, to determine the existence of stress-induced martensite hehavior.

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TABLE	1
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45 0	410	[44)	- 32						СD		
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49.5	45.5	50	-13	8	c		C			D	
50.0	46.()	40	- 11*		B		D		D		
48.5	45.0	n.5	- 10	8		в		C C		D	
49.0	45.5	5.5	- 10	В		8		С	• •	C D	
48.0	44 25	7 75			A B		C		СD		
48.5	45.5	n ()	- 5	A	8		в		C.		
41.5	38.5	20.0	- 2	A	4		8	•	в		BC
46.5	43.5	10.0	- 1		. А		8		С		
36 25	33.75	30.0	43*		`.A		٦		B		в
49 5	46.()	45	n*		8		B		Ð٠		
0.84	46.0	a ()	12	4	A/B	В	в	Ð	в	в	° D
47 75	45 75	0.5	. 20		А		4		н		8
47.5	45.5	-0	26		А		А		в		8
48.5	46.5	5.0	27		А		А		в		B
45.0	45.0	10.0	30				4	A/B	В		B
47.5	46.5	5.0	32				А	8	В	в	B
46.5	46.5	7.0	34		А		A	-	B		-
48.25		5.5	10		A		A		B		В

*Alloys with an asterisk beside the M-temperature are not within the scope of the invention, even though the M, temperature is in the correct range

mechanical behavior for many medical applications. Such a plateau in the stress-strain curve of these alloys occurs over limited temperature ranges above M, and below Ma.

Such properties are useful for medical products when 40 they occur at temperatures between 0° C. and 60° C., and particularly at 20° C. to 40° C. It has been discovered that certain compositions of Ni/Ti/V alloys exhibit B- or C-style behavior in this temperature range.

Shape memory alloys according to the invention may 45 conveniently be produced by the methods described in, for example, U.S. Pat. Nos. 3,753,700 and 4,144.057. The following example illustrates the method of preparation and testing of samples of shape memory alloys.

#### EXAMPLE

Commercially pure titanium and vanadium and carbonyl nickel were weighed in proportions to give the atomic percentage compositions listed in Table I (the total mass for test ingots was about 330 g). These metals 55 were placed in a water-cooled copper hearth in the chamber of an electron beam melting furnace. The chamber was evacuated to 10-5 Torr and the charges were melted and alloyed by use of the electron beam. The resulting ingots were hot swaged and hot rolled in 60 air at approximately 850° C. to produce strip of approximately 0.025 inch thickness. Samples were cut from the strip, descaled, vacuum annealed at 850° C. for 30 minutes, and furnace cooled.

The transformation temperature of each alloy was 65 determined (on an annealed sample) as the temperature at the onset of the martensite transformation at 10 ksi stress, referred to as M₅ (10 ksi).

It can be seen from Table I that alloys with an M. higher than -40° C. but lower than 20° C. show predominantly B- and C-type behavior at 20° and 40° C. This M₅ criterion is not sufficient to ensure a flat stressstrain curve at the desired temperatures, however, A vanadium content of at least 4.6 atomic percent is also necessary, since alloys with 1.5 and 4.0 atomic percent V show D- and E-type behavior at 20° C. and 40° C. The sample with a V content of 4.5 at % shows D-type behavior at 40° C., although B-type at 0° and 20° C.

Since the alloy with an M₃ of -42* C. has D-type behavior at 0° C., it is expected that alloys with an M, below -40° C. will show D- or E-type behavior in the ⁵⁰ temperature range of interest. while alloys with an M. above 20° C. show A-type behavior over at least half

Too much vanadium also leads to undesirable properties, since an alloy with 30 atomic percent vanadium shows a lesser degree of SIM elongation and a much higher vield strength for the SIM transformation than alloys of lower vanadium content. This alloy also showed A-type behavior at 20° C. despite an  $M_c$  of  $-3^\circ$ C. Such an alloy, with a nearly 1:1:1 composition ratio, is probably not treatable as a Ni/Ti type alloy.

The claimed composition range, based on these data, is shown in FIG. 2, and the compositions at the vertices given in Table II.

<b>T A</b>	BI	т.	н
- I A	<b>n</b> 1	. –	

		ADGE II	
	Atomic P	ercent Compositio	ns
Point	Nickel	Titanium	Vanadrum
4	38.0	37.0	25.0

Such an alloy would be marginally useful.

the 0°-60° C. range.

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		-7		
	TABL	continued	i	
	Atomic Per	cent Compositio	n×	
Prim	Nickei	Franium	Vanadium	
н	÷- v	1n 4	,	- :
1 ¹	2943	1n 4	÷n	
Ð	20 4	-5 n	÷n	
Ξ	29 a	() شد	~ <u>:</u>	
F	14 K	15.2	25.0	

The lines AB and BC represent the upper limit of M. expected to allow the desired behavior, i.e. 20° C. The line AB corresponds approximately to a Ni:Ti atomic ratio of 1.13. The line CD corresponds to the lower limit of vanadium composition: alloys having less vana- 15 dium do not exhibit B- or C-type behavior in the desired temperature range even if of the correct Me. The lines DE and EF represent the lower limit of M_s giving the desired behavior, i.e. -40° C. The line EF corresponds approximately to an Ni:Ti atomic ratio of 1.02. Finally, 20 25.0 atomic percent vanadium. the line FA represents the upper limit of vanadium content for the desirable SIM properties.

Presently preterred alloys include a region consisting essentially of 47.6-48.8% at % Ni, 45.2-46.4 at % Ti, remainder V around 48.0% Ni, 46.0% Ti, 6.0% V. 25 which alloy has B-type behavior from 10° to 50° C .; and a region having an Ni:Ti atomic ratio between about 1.07 and 1.11 and a vanadium content between 5.25 and 15 atomic percent, which shows C-type behavior at 20° 30 C. and/or 40° C.

In addition to the method described in the Example. alloys according to the invention may be manufactured from their components (or appropriate master alloys) by other methods suitable for dealing with hightitanium alloys. The details of these methods, and the 35 precautions necessary to exclude oxygen and nitrogen either by melting in an inert atmosphere or in vacuum. are well known to those skilled in the art and are not repeated here.

Changes in composition cann occur during the electron-beam melting of alloys: the technique employed in this work. Such changes have been noted by Honma et al., Res. Inst. Min. Dress. Met. Report No. 622 (1972), and others. The composition ranges claimed as a part of  $_{45}$ this invention are defined by the initial commonstions of alloys prepared by the electron-beam method. However, the invention includes within its scope nickel/titanium/vanadium alloys prepared by other techniques which have final compositions which are the same as 50 nickel, 45.2 and 46.4 atomic percent titanium, and the the final compositions of alloys prepared here.

Alloys obtained by these methods and using the materials described will contain small quantities of other elements, including oxygen and nitrogen in total amounts from about 0.05 to 0.2 percent. The effect of 55 these materials is generally to reduce the martensitic transformation temperature of the alloys.

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this invention are not-workable and The allovs exhibit stress-induced martensite in the range of  $0^{\circ}$  to o0° C, in the fully annealed condition.

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We ciaim:

I. A shape memory alloy consisting essentially of nickel, titanium, and vanadium within an area defined on a nickel, titanium, and vanadium ternary composition diagram by a hexagon with its first vertex at 38.0 atomic percent nickel, 37.0 atomic percent titanium, and 10 25.0 atomic percent vanadium; its second vertex at 47.6 atomic percent nickel. 46.4 atomic percent titanium, and 6.0 atomic percent vanadium; its third vertex at 49.0 atomic percent nickel. 46.4 atomic percent titanium, and 4.6 atomic percent vanadium; its fourth vertex at 49.8 atomic percent nickel. 45.6 atomic percent titanium; and 4.6 atomic percent vanadium: its fifth vertex at 49.8 atomic percent nickel, 44.0 atomic percent titanium, and b.2 atomic percent vanadium; and its sixth vertex at 39.8 atomic percent nickel. 35.2 atomic percent titanium, and

2. The alloy of claim 1 which has an Ni:Ti atomic ratio between 1.07 and 1.11 and a vanadium content between 5.25 and 15 atomic percent.

3. The alloy of claim 1 which consists essentially of between 47.6 and 48.8 atomic percent nickel, 45.2 and 46.4 atomic percent titanium, and the remainder vanadium.

4. A shape-memory article comprising a shape-memory alloy consisting essentially of nickel, titanium, and vanadium within an area defined on a nickel, titanium. and vanadium ternary composition diagram by a hexagon with its first vertex at 38.0 atomic percent nickel, 37.0 atomic percent titanium, and 25.0 atomic percent vanadium: its second vertex at 47.6 atomic percent nickel, 46.4 atomic percent titanium, and 6.0 atomic percent vanadium; its third vertex at 49.0 atomic percent nickel. 46.4 atomic percent titanium, and 4.6 atomic percent vanadium; its fourth vertex at 49.8 atomic percent nickel, 45.6 atomic percent titanium, and 4.6 atomic percent vanadium: its fifth vertex as 49.8 40 atomic percent nickel, 44.0 atomic percent titanium, and 6.2 atomic percent vanadium: and its sixth vertex at 39.8 atomic percent nickel, 35.2 atomic percent titanium, and 25.0 atomic percent vanadium.

5. The article according to claim 4 which has an Ni:Ti atomic ratio between 1.07 and 1.11 and a vanadium content between 5.25 and 15 atomic percent.

6. The article according to claim 4 which consists essentially of between 47.6 and 48.8 atomic percent remainder vanadium.

7. The article according to claim 4 exhibiting stressinduced martensite.

8. The article according to claim 4 exhibiting stressinduced martensite in the range of 0° to 60° C. when in the fully annealed condition.

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# Variation in the Shape Recovery Temperature in Ni-Ti Alloys

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#### SUMMARY

## 1. INTRODUCTION

In the near-equiatomic Ni-Ti alloy, certain processing factors significantly affect the shape recovery temperature  $T_{\rm f}$  and the extent of shape recovery while others do not. The lowest recovery temperature and the best shape recovery were obtained by annealing between 450 and 500 °C. Varying the annealing temperature resulted in the largest change in these two parameters, causing an increase of 20 °C in  $T_{\rm f}$  and a decrease of 13% in the shape re-Increasing the maximum strain (in the range between 0% and 8%) induced an increase of 10 °C in  $T_f$  at 500 °C with larger increases at other annealing temperatures but without any effect on the extent of shape recovery. Altering the annealing time had a relatively small effect. A recovery-resistant physical constraint raised the effective  $T_{f}$  but if the constraint was maintained at a sufficiently high temperature the transformation occurred, plastic deformation resulted and permanent partial loss of shape memory was suffered.

 $T_t$  extrapolates to nearly the same temperature, at zero strain, for all the annealing temperatures near 500 °C. Similarly, the final temperature of the transformation from martensite to  $\beta$  phase (zero strain) is the same for all the annealing temperatures. This strongly suggests that each annealing temperature yields a high temperature B2 phase with a slightly different structure which interacts differently with stress.

Although this study is based on only two batches of Ni-Ti wires of similar composition,

* expected that qualitatively similar variations with processing factors occur in all Ni-Ti alloys which exhibit shape memory.

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Near-equiatomic Ni-Ti alloys (Nitinol) exhibit shape memory characteristics [1 - 3]. The memory is such that, given the proper conditions, Nitinol objects can be restored thermally to their original shape after being deformed from that shape. The shape recovery takes place over a range of temperatures on heating with maximum or, ideally, complete recovery being reached at a specific temperature  $T_f$ .

The high temperature phase in Ni-Ti alloys is reported to be of a CsCl (B2) type, and the martensite is reported to be a monoclinic distortion of the B19 structure [4]. There is also a higher order phase transition from the B2 phase which occurs below a critical temperature  $T_{\rm R}$ . Among many other effects the transition from the B2 phase is characterized by a rapid increase in electrical resistance [4, 5], extra diffraction spots at the one-third positions of the B2 reciprocal lattice [4, 6] and a rhombohedral distortion of the B2 lattice [5, 6]. This transition is pre-empted by the martensitic transformation when  $M_s > T_R$ (where  $M_s$  is the martensitic transformation temperature); application of factors which lower  $M_s$  relative to  $T_R$ , such as thermal cycling of the specimen in the transformation range or compositional variations [7], may be necessary to see both transformations. We define the structure below  $T_{\rm R}$  as the R phase and refer to the structure encompassing both the B2 and the R phases as the  $\beta$  phase. Thus martensite may form from the B2 or the R phase depending on the relative temperatures  $T_{\rm R}$  and  $M_{\rm s}$  [5].

The primary mechanism for the shape memory effect is generally considered to be the interaction of stress with the martensitic

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transformation [5, 8 - 11]. Depending on the deformation temperature with respect to  $M_s$ , one or more of the following three mechanisms may be operative in Ni–Ti alloys [12]: (1) stress-induced martensite formation from the  $\beta$  phase; (2) reorientation of thermally induced martensite variants through twininterface motion; (3) variation in twin thickness within individual variant "plates". The B2  $\neq$  R transition has been found to contribute to the two-way shape memory [5]. Our recent results [13] show that this transition is probably also a primary shape memory

mechanism, although the effect is smaller in magnitude than that due to the martensitic transformation. When  $T_{\rm R}$  is below the final temperature  $A_{\rm f}$  of the transformation from martensite to  $\beta$  phase, the final shape recovery temperature will be determined by the reversion of the last plate of martensite (ideally the first one formed). This situation is generally observed in Ni-Ti alloys. Thus we expect  $T_{\rm f}$ to be closely related to  $A_{\rm f}$ .

The shape memory properties of Ni-Ti alloys are very sensitive to the chemical composition of the alloy. Buehler and Wang [14] have shown that  $T_{\rm f}$  varies in binary Ni-Ti alloys from about -50 to 166 °C. Eckelmever [15] has also reported that  $T_f$  showed a steep increase (80 °C) as the titanium content changed from about 49.7 to 50.4 at.% and remained constant for further increases in titanium content. Substituting a small percenage of a third element for either nickel or titanium can cause either an increase or a decrease in  $T_{t}$  [15]. The shape memory properties are also sensitive to the methods by which the alloy is processed and heat treated [16]. Cross et al. [17] have performed a study on the shape memory response of Ni-Ti rod, wire and foil of varying diameter or thickness. They selected a final annealing temperature of about 500 °C as the optimum for shape memory and reported on the shape recovery as a function of material form (rod, wire or foil) and the amount of strain. The variation in  $T_{\rm f}$  was not documented. Besides the references cited above, we are not aware of any published results which deal quantitatively with the effect of other factors on  $T_f$ and shape recovery.

Variables of concern in utilizing the shape memory phenomenon can be divided into two groups: the first group, associated with shape memory in general, includes the final annealing temperature, the annealing time, the amount of strain and the effect of thermal cycling; the second group relates to parameters intrinsic to practical applications such as shape change constraints caused by a storage device and manual handling. Although these were not expected to have as drastic an effect on the recovery temperature as the composition, there are potential applications where strict temperature requirements demand close knowledge and control of all parameters which may together cause unacceptable variations.

#### 2. EXPERIMENTAL PROCEDURES

The materials used in this study were drawn wires approximately 0.020 in in diameter. Two batches numbered A383 and V4865 were chemically analyzed to be of compositions 50.13at.%Ti-49.96at.%Ni and 50.12at%Ti-49.86at.%Ni respectively with a reported estimated error of 0.1%. (The A383 wire was supplied by the Nitinol Technology Center, Naval Surface Weapons Center, White Oaks, MD. It was strand annealed at 500 °C in the as-received condition. The V4865 wire was purchased from the Timet Corporation. In the as-received condition it is totally martensitic.) The procedures used for specimen preparation, electrical resistance measurement and shape recovery experiments were as described in ref. 5. Figure 1 illustrates schematically the shape memory geometry for these experiments. As a shape change parameter we chose the included angle  $\theta$  between the two arms of the U-shaped specimen. The "fractional shape recovery" was taken to be  $(180 - \theta)/180$ .

The measurements of the detailed shape recovery, including precise  $T_f$  values, were made as a function of annealing temperature and annealing time and for various "straighten-

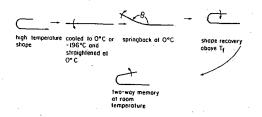


Fig. 1. A schematic sequence of the shape memory experiment.

**?**" strains between 2% and 8%. Earlier empirical observations [16, 17] have indicated that an annealing temperature in the vicinity of 500 °C yields good shape recovery behavior in Ni–Ti alloys. Thus our investigation centered on this temperature region. All specimens, unless otherwise specified, were air cooled after the high temperature anneal by placing the jig over an air vent.

The specimen was straightened at 0 °C. At this temperature the V4865 wire is essentially 100% martensite. For A383 wire the initial state depends on the lowest cooling temperature prior to deformation: (a) 0 °C, 100% R phase; (b) -196 °C, a mixture of R phase and martensite.

#### 3. RESULTS AND DISCUSSION

Although the A383 and V4865 alloys have similar atomic compositions, differences exist in their transformation behavior [5]. For example, in the range of annealing temperature and strain investigated,  $T_{\rm f}$  lies between 30 and ^A°C for the A383 alloy and between 60 and C for the V4865 alloy. The different recovery temperatures for the two materials are also reflected in the low  $M_s$  value (below -10 °C) observed for the A383 alloy compared with that (about 30 °C) for the V4865 alloy. The zero-strain values of  $M_s$  and  $A_f$  as a function of annealing temperature for an annealing time of 30 min are listed in Table 1.  $A_{\rm f}$  is constant for all the annealing temperatures in each material. Although  $M_s$  is also constant for the V4865 alloy, changing the

annealing temperature from 500 to 540 °C raises  $M_s$  from -25 to -12 °C for the A383 alloy. It was also observed that 100% shape recovery was attained for the A383 alloy for all strains tested compared with 94% for the V4865 alloy at the higher strains of 5% - 8% and annealed at 500 °C. On repeating shape memory cycles up to 10 times, the A383 alloy still achieves 100% recovery while the incomplete shape recovery in the V4865 alloy deteriorates with cycling. The differences in the transformation behavior between the two materials are not clear. Neither optical microscopy nor X-ray diffraction in these wire specimens revealed any noticeable difference in the structure or phases initially present. Processing variables prior to the final anneal (such as in wire drawing etc.) and the impurity content probably cause the differences.

A typical shape recovery curve of a U-shaped specimen consists of an immediate elastic spring-back and then a slow recovery followed by a narrow temperature region of rapid recovery. A description of shape recovery mechanisms has been presented earlier [5]. In this paper we shall concentrate on the experimental variations of  $T_f$  and the percentage of shape recovery.

# 3.1. Variation in the shape recovery temperature

Although the A383 and V4865 alloys show differences in the transformation behavior, both wire batches none the less exhibit similar variations in  $T_f$  with annealing temperature and with percentage strain (for a fixed annealing time of 30 min). As shown in Fig. 2,  $T_f$ 

# TABLE 1

The variations in  $T_{\rm R}$ ,  $M_{\rm s}$  and  $A_{\rm f}$  at zero strain as a function of annealing temperature for an annealing time of 30 min

Specimen batch	Annealing temperature (°C)	$T_{\mathrm{R}}$ (°C)	<u>M</u> _s (°C)	$A_{f}$ (°C)
A383	500	28	-25	31
at a f	520	25	-20	31
· . ·	540	20	-12	31.5
V4865	400	45	31.5	59
	450	40	31.5	60
	500	37 ± 2	31	60
2	550	$37 \pm 2$	31	58
``````````````````````````````````````	600	37 ± 2	31	61

These values of  $T_{\rm R}$ ,  $M_{\rm s}$  and  $A_{\rm f}$  were determined from the electrical resistance measurements on the first thermal cycle after cooling to room temperature from the high temperature annealing treatment.

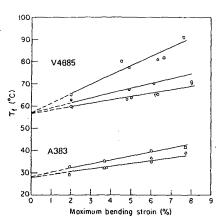


Fig. 2. Variation in the shape recovery temperature  $T_{\rm f}$  with percentage bending strain at different annealing temperatures for an annealing time of 30 min for the V4685 alloy ( $\Box$ , 600 °C;  $\triangle$ , 550 °C;  $\circ$ , 500 °C;  $\nabla$ , 450 °C;  $\circ$ , 400 °C) and the A383 alloy ( $\Box$ , 540 °C;  $\triangle$ , 520 °C;  $\circ$ , 500 °C).

increases linearly with increasing strain between 2% and 8% strain with a change of approximately 9 °C for annealing temperatures near 500 °C. Larger increases are caused in samples of the V4865 alloy annealed at other temperatures. External strain (stress) raises  $M_s$  and  $A_f$  in general, so this result is not unexpected. For a fixed strain,  $T_f$  has a minimum for annealing temperatures between 450 and 500 °C, as shown in Fig. 3. Similar variations occur for strains between 2% and 8%.

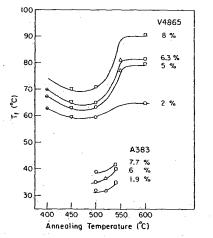


Fig. 3. Variation in shape recovery temperature  $T_{\rm f}$  with annealing temperature at different bending strains for an annealing time of 30 min.

Thus a final anneal near 500 °C yields a low recovery temperature.

The effect of changing the annealing time is shown in Fig. 4 for the V4865 alloy. With the annealing temperature fixed at 500 °C and with 6.3% strain,  $T_t$  increases from 65 to 67.5 °C as the annealing time is increased from 30 min to about 1.5 h and remains constant for longer annealing times up to 24 h. (The specimen annealed for 10 min was quenched in water at room temperature.) The perfection of the shape recovery decreases slightly, from 94% to 91%, with increasing annealing time.

Using A3S3 wire, other parameters were studied to determine their effect on  $T_{\rm f}$ . Cooling to 0 °C (100% R phase) or -196 °C (a mixture of R phase and martensite) prior to straightening at 0 °C resulted in a variation in  $T_{\rm f}$  of only 1 °C. This means that  $A_{\rm f}$  is the same whether the induced strain is a result of stress-induced martensite formation and/or reorientation of martensite variants. Two different cooling rates after the final anneal, air cooling and quenching in iced water, showed less than 1 °C difference in  $T_{\rm f}$ . This indicates that any structural changes which may occur on cooling to the straightening temperature are not diffusion controlled. On cycling the same specimen through the shape memory cycle up to 10 times the  $T_f$  values showed a scatter of 1.5 °C. Changing the straightening temperature from 0 to 15 °C and to 24.5 °C (the A383 alloy is 100% R phase at all these temperatures) does not shift  $T_{\rm f}$  by more than 1 °C. The immediate recovery at the higher straightening temperatures is such that the specimen goes directly to the shape

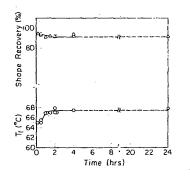


Fig. 4. Variations in  $T_f$  and shape recovery with annealing time (batch V4865; 500 °C anneal; 6.3% strain).

It was attained at the same temperature by a specimen that was straightened at 0  $^{\circ}$ C. Thus these processing parameters seem to have relatively little effect on the recovery behavior.

We now turn to the effect on the wire of physical constraint (as in a storage or application container) while the temperature is raised above the normal  $A_{f}$ , after it has been straightened and placed in the container. The normally occurring shape change is partially constrained by the container. This might result in plastic deformation which will impair shape recovery. In our experiments an initially U-shaped specimen (A383 wire; air cooling at 500 °C for 30 min; 6.0% strain) was straightened at 0 °C and was placed inside a rigid tube  $\frac{1}{R}$  in inside diameter, which only allows a small fraction of the shape recovery to occur. The specimen was then heated in water to various temperatures before being pulled from the tube into the water. These experiments are depicted in Fig. 5. Figure 5, curve 1, illustrates the case when the specimen was not constrained. The corresponding electigal resistance curve is also shown. (It should 

surements here are for the entire length of the wire (approximately 2.5 in) while only about 20% of the wire was actually deformed as part of the U bend. Thus the resistivity changes are underrepresented.)

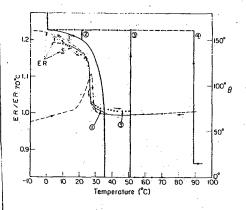


Fig. 5. Effect of a recovery resistant constraint on  $T_{\rm f}$ and shape recovery (batch A383; 500 °C anneal for 30 min; 6% strain): ---, electrical resistance (ER) 4 for strain; curves 1 and 3, correspondence between the electrical resistance and  $\theta$  changes.

After straightening, the electrical resistance is lower than in the zero-strain curve (broken line) because of the stress formation of martensite, which has a lower intrinsic electrical resistance than the R phase. Between 27 and 37 °C the martensite electrical resistance is higher than that of the R or B2 phase, so curve 1 lies above the zero-strain curve. Above 37 °C ( $A_f$ ) the transformation of martensite to the B2 phase is completed and the electrical resistance curves show the same temperature variations. If the intermediate temperature is below  $T_{\rm f}$  (about 36 °C), the specimen immediately reverts to the identical (partially recovered) shape that was attained at the same temperature by a specimen which has not been constrained. Under these conditions the ultimate  $T_f$  did not vary from the original value by more than 1.5 °C (Fig. 5, curve 2). If the intermediate temperature was above  $T_{\rm f}$ but below about 90 °C, an instantaneous and 100% recovery was obtained. Figure 5, curve 3, illustrates the case when the intermediate temperature is 51 °C. The corresponding electrical resistance curve showed that the electrical resistance remained constant between 37 and 51 °C, indicating that a volume fraction of martensite was prevented from reverting to the B2 phase by the constraint. Once the specimen had been pulled from the rigid tube, the electrical resistance immediately attained the value of the B2 phase at that temperature, so the instantaneous shape recovery was accompanied by the simultaneous reversion of the remaining martensite to the B2 phase. Thus the extra stress caused by the constraint raises the effective  $T_{\rm f}$ . At a temperature of about 90 °C, only 85% recovery was achieved (Fig. 5, curve 4). Further, only incomplete shape recovery was achieved on further cycling at lower intermediate temperatures. Thus at a sufficiently high temperature the external stress due to the constraint is unable to prevent the transformation in the material, which then results in permanent plastic deformation.

This important result indicates that the equilibrium  $T_{\rm f}$  may be exceeded significantly while a shape memory device is constrained without impairing shape recovery. These data also stress the fact that any handling technique likely to cause plastic deformation in the wire should be avoided.

# 3.2. Variation in the percentage of shape recovery

In the A383 wire, 100% recovery was achieved in all the specimens and conditions tested. The V4865 wire, however, did not show similarly perfect behavior. The variation in percentage shape recovery with annealing temperature and strain is shown in Fig. 6. At a fixed annealing temperature the elastic spring-back is largest at 2% strain while the total shape recovery (spring-back and shape memory) decreases from 100% at 2% strain to approximately 94% for strains between 5% and 8% at an annealing temperature of 500 °C. These results agree qualitatively with those reported by Cross et al. [17] on Ni-Ti alloys with a final annealing temperature of 500 °C. Their tests have shown that shape recovery deteriorates with increases in strains up to 18%.

At different annealing temperatures the spring-back is essentially constant for a fixed strain, while the extent of shape recovery has a maximum of 94% between 450 and 500  $^{\circ}$ C with a slight decrease to 90% at 400  $^{\circ}$ C and a larger decrease to 81% at 550 and 600  $^{\circ}$ C. Thus a final annealing temperature near 500  $^{\circ}$ C gives the best shape recovery as well as a low recovery temperature.

# 3.3. Further consideration of the annealing treatment

 $T_{\rm f}$  extrapolates to nearly the same temperature at zero strain for all annealing temperatures near 500 °C (Fig. 2). Similarly,  $A_{\rm f}$  at zero strain is the same for all the annealing

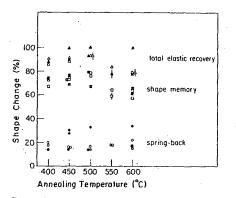


Fig. 6. Percentage shape change as a function of annealing temperature and strain (batch V4865; maximum strains of 8% ( $^{(A)}$ ), 6.3% ( $^{(\Delta)}$ ), 5% ( $^{(\Delta)}$ ) and 2% ( $^{(A)}$ ); annealing time, 30 min).

temperatures ( $A_f$  is about 60 °C for the V4865 alloy and 31 °C for the A383 alloy). This strongly suggests that each different annealing temperature yields a high temperature B2 phase with a slightly different structure (degree of order, defect structure etc.) that interacts differently with stress to yield different (finite strain)  $T_f$  values at different strains.

Consideration must therefore be given to the possibility that the annealing temperature affects such structural factors as the degree of long- and short-range order in the mutual arrangement of the nickel and titanium atoms and that subsequent transformations are affected by these factors, particularly in the presence of strain. Similarly, the effect of annealing on defect structures must be considered. For example, the Ni-Ti alloys are subject to contamination by oxygen and nitrogen, which form oxides and nitrides as inclusions. Although inclusions are an integral part of the microstructure in Nitinol alloys [17], their atomic composition, structure and morphology have not been determined. The inclusions may affect the phase transformation and shape changes because of the internal stress modifications which they cause, as a result of interaction with external stress. The influence of annealing temperature on the effect of the inclusions is not clear. A clarification of these effects may lead to a better understanding of the changes caused by the annealing treatment.

#### 4. CONCLUSIONS

We established that certain processing factors affect  $T_f$  and the extent of shape recovery significantly while others do not. The lowest recovery temperature  $T_{f}$  and the best shape recovery is obtained by annealing between 450 and 500 °C. Varying the annealing temperature results in the largest change in these two parameters, causing an increase of as much as 20 °C in  $T_f$  and a decrease of 13% in shape recovery for the V4865 alloy if the annealing temperature is increased much above 500 °C. Changing the maximum strain (between 2% and 8%) induces a 10 °C variation in  $T_f$  at 500 °C with larger increases at other annealing temperatures but without an effect on the extent of shape recovery. Altering the annealing time has a relatively small effect. At 500 °C and 6.3% strain there is only a maximum of 3 °C change in  $T_f$  and a 4% loss in shape recovery for the V4865 alloy. Recovery-resistant stress caused by an external constraint raises the effective  $T_f$  but, if maintained at a sufficiently high temperature, the plastic deformation accompanying the transformation under constraint causes permanent partial loss of shape memory.

Although this study is based on only two batches of Nitinol wires of similar compositions, it is expected that qualitatively similar variations occur in all Ni-Ti wires that exhibit shape memory.

#### ACKNOWLEDGMENTS

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We wish to acknowledge the strong impetus this work provided by Dr. Morris Simon,

Litts work provided by Dr. Morris Sinion, Litts work provided by Dr. Morris Sinion, medical application of these materials [18] and the interaction and collaboration with him and his colleagues Dr. Edwin Salzman, Dr. David Freiman, Dr. Aubrey Palestrant and Mr. Martin Prince. We also greatly appreciate the efforts of Dr. David Goldstein, Nitinol Technology Center, Naval Surface Weapons Laboratory, in supplying the Nitinol wire.

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# Lur Dwyerschen Skoliosenoperation mittels Drähten aus Memory-Legierungen

Sine experimentelle Studie

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-75 (1978)

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# In DWYER'S Scoliosis Operation Using Iemory Alloy Wire

summary. In Dwyer's spinal column correction a itanium cable is stretched from vertebra to vertebra vertebra and secured to each vertebra with c. As and clips. It is suggested to replace the titanium vire with wire consisting of the memory alloy NiTi. This will permit the prestretched wire to be tensioned vy heating it to 60 deg C after it has been fixed at its nds in the vertebrae.

The functional principle of the NiTi memory wire vas demonstrated in an experiment carried out on a vastic model. This consists of 8 plastic vertebrae with 0mm sides which are connected by interposed vedge-shaped soft rubber discs, giving the model a urved shape. Memory wire prestretched by 7% is led hrough eylets on the convex side and fixed at the nds. On being heated (electrically in this experiment or the sake of simplicity), the wire shortens, righting he model so that it assumes a straight shape.

The authors also describe in detail the manuacture of the alloy, i.e. the melting and shaping perations, as well as the properties of the material, hat is, the stress-strain and strain-temperature reitionships and the transformation temperatures, as 'ell as mechanical problems.

usammenfassung. Bei der Dwyerschen Wirbelsäulenorrektur wird mittels einer Spezialspannzange ein itankabel von Wirbelkörper zu Wirbelkörper ver-

årdruckanfragen an: Prof. Dr. F. Baumgart (Adresse siehe ben) spannt und an jedem einzelnen mittels Schrauben in Agraffen festgeklemmt. Hier wird vorgeschlagen, das Titankabel durch einen Draht aus Memory-Legierung NiTi zu ersetzen. Dies ermöglicht das Spannen des vorgereckten Drahtes durch einfache Erwärmung auf etwa 60°C, nachdem dieser an den Enden in den Wirbelkörpern verankert wurde.

Die Wirkungsweise des NiTi-Drahtes wurde in einem prinzipiellen Versuch an einem Kunststoffmodell demonstriert. Das Modell besteht aus 8 Kunststoffwirbeln von 30mm Kantenlänge, die durch dazwischengeklebte keilförmige Weichgummischeiben verbunden sind. Dadurch erhält das Modell eine gekrümmte Form. Der auf der konvexen Seite durch Ösen geführte und an den Enden verankerte Memory-Draht ist um 7% vorgereckt. Er wird in diesem Versuch der Einfachheit halber elektrisch aufgeheizt, verkürzt sich und richtet das Modell in eine gerade Form aus.

In diesem Zusammenhang werden auch ausführlich die Herstellung der Legierung, d.h. Erschmelzen und Umformen, die Eigenschaften, d.h. Spannungs-Dehnungs-Verhalten, Dehnungs-Temperatur-Verhalten und Umwandlungstemperaturen sowie mechanische Probleme beschrieben.

Die Skoliose ist eine Dreh-Seitverbiegung der Wirbelsäule, die auch mit den heute bekannten und angewandten Operationsverfahren nur sehr schwer zu korrigieren ist. Mit den meisten derzeit üblichen Operationsmethoden läßt sich die Seitverbiegung in erstaunlich gutem Maße korrigieren. Die Torsion

hingegen ist bisher noch nicht oder nur unbefriedigend zu beherrschen. Wird eine dorsale Spondylodese nach Harrington durchgeführt, so resultiert meist ein lordosierender Effekt, wogegen bei der ventralen Wirbelsäulenkorrektur nach Dwyer sich die Kyphose nicht oder nur schwer vermeiden läßt. Dieser kyphosierende Effekt ist im Dorso-Lumbalbereich, wo die Dwyer-Technik auch am erfolgreichsten ist, nur erwünscht. Besonders vorteilhaft ist diese Technik bei rigiden thorakolumbalen und lumbalen Wirbelsäulenverbiegungen Erwachsener sowie bei Lähmungsskoliosen. Sie bilden somit die besten Erfolgschancen für die Dwyersche Operationsmethode. Bei den Skoliosen und Lordosen, die durch Meningomyelozelen oder andere Fehlbildungen im Bereich der dorsalen, lumbalen Wirbelsäule bedingt sind, ist sie sogar die Methode der Wahl.

Um nun die aufwendige und ohnehin schon schwierige Operationsmethode zu vereinfachen, wurde von uns ein Verfahren zur Korrektur der Skoliose im Modell erprobt, das auf die Grundidee von Dwyer zurückgeht.

## 1. Operationstechnik im Experiment

Nach dem Ausräumen der Bandscheiben wird bei der Dwyerschen Operationsmethode ein Titaniumkabel durch Schraubenköpfe geführt, die in der Mitte ein Loch haben. Diese Schrauben wiederum sitzen in sogenannten Agraffen, die den Wirbelkörper wie eine Klammer fest umschließen. Diese Agraffen haben die Aufgabe, ein seitliches Abrutschen oder Ausbrechen der Schrauben während des Spannvorganges zu verhindern. Bei der Methode nach Dwyer wird das Titaniumkabel mit einer Spezialspannzange von Wirbelkörper zu Wirbelkörper verspannt und jeweils durch Zusammenquetschen des Schraubenkopfes blokkiert.

In unserem weiter unten ausführlich beschriebenen Modellversuch wurde das Titaniumkabel durch einen NiTi-Draht ersetzt, der lediglich an den Neutralwirbeln fixiert wurde. An den übrigen Wirbelkörpern wird der NiTi-Draht durch Ösen geführt, um ein Ausgleiten der Korrektur aus der gewünschten Richtung zu vermeiden. Die Korrekturwirkung wird durch Erwärmen des Drahtes erreicht und ist stufenlos steuerbar. Die genaue Versuchsbeschreibung des Modellversuches ist weiter unten angegeben.

Wir glauben, das Operationsverfahren nach Dwyer durch die Verwendung von NiTi-Drähten noch weiter vereinfachen und verbessern zu können.

Weiterhin wäre auch eine postoperative Nachkorrektur durchaus noch denkbar, da die NiTi-Drähte ja nicht in ihrer vollen Zugkapazität ausgeschöpft werden müssen. Es erscheint z. Z. nicht ausgeschlossen, das Metall z. B. induktiv soweit wieder aufzuheizen, daß sich eine gewisse Nachkorrektur evtl. auch noch post operationem womöglich unter Bildwandlerkontrolle durchführen ließe.

 $(\mathbf{x}_{i})^{T}$ 

Neben schon anderen von uns beschriebenen Einsatzmöglichkeiten dieses neuen Implantat-Werkstoffes sehen wir in der Verwendung von NiTi-Drähten bei der Dwyerschen Operation eine erhebliche technische Verbesserung einer vielfach mit Erfolg erprobten Operationsmethode. Der Vorteil gegenüber der herkömmlichen Spanntechnik ist die stufenlose Spanntechnik des Drahtes, die eine schonende Korrektur der Wirbelsäule, auch in Etappen, ermöglicht.

# 2. Herstellung der NiTi-Legierung

Die Legierung NiTi kann grundsätzlich sowohl im Elektronenstrahlofen als auch in einem Vakuuminduktionsofen erschmolzen werden. Da beim Elektronenstrahlofen zu einer bestimmten Zeit immer nur ein geringer Anteil des zu erschmelzenden NiTi-Volumens flüssig ist, kann es vorkommen, daß beim ungleichmäßigen Abschmelzen des Ausgangsmaterials der Gußblock inhomogen wird. Die Inhomogenitäten des Gußblockes können auch durch mehrfaches Umschmelzen im Elektronenstrahlofen nicht beseitigt werden. Legierungsschwankungen von  $\pm 1\%$ zwischen Kopf und Fuß des Ingots waren bei den durchgeführten Versuchen keine Seltenheit.

Günstiger gestaltet sich die Erschmelzung des NiTi im Vakuuminduktionsofen. Probleme der Inhomogenität treten hier praktisch nicht auf, da die Schmelze während des Herstellungsprozesses stetig gut durchgemischt wird.

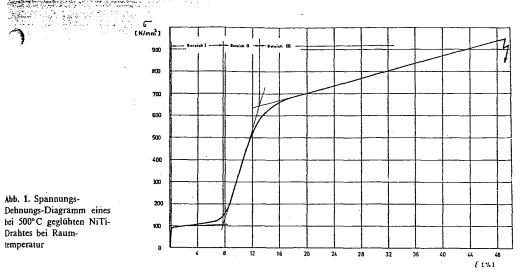
Nachdem der Gußblock erstarrt und abgekühlt ist, wird er gedrittelt. Das anschließende Warmwalzen selbst bereitet, solange man nicht zu weit von der stöchiometrischen Zusammensetzung entfernt ist, keine Schwierigkeiten. Die Blocksegmente werden in einem elektrischen Kammerofen auf 900°C erwärmt, wobei darauf zu achten ist, daß das Material gut durchgewärmt ist.

Nach jedem Walzstich ist es empfehlenswert, die Bramme im Ofen wieder aufzuwärmen. Gewalzt werden kann mit einem Walzgrad von  $\varphi_g \approx 0.25$ . Höhere Walzgrade können mit großer Wahrscheinlichkeit erreicht werden, wurden aber versuchsmäßig noch nicht erprobt.

Der dem Warmwalzen folgende Kaltwalzvorgang gestaltet sich etwas aufwendiger als der Warmwalzvorgang. Die warmgewalzte Platte kann in den ersten drei Stichen mit einem Umformgrad von  $\varphi \approx 0.04$  gewalzt werden, wobei nach jedem Stich eine Zwischenglühung zu erfolgen hat. Bei dieser Zwischenglühung wird der in der Niedertemperaturphase vorliegende Werkstoff über seine A_s-Temperatur erwärmt, und die Umformung wird durch den einsetzenden Memory-Effekt weitgehend wieder rückgängig gemacht. Pro Walzzyklus (Kaltwalzen, Zwischenglühung, Abkühlung an Luft) wird dadurch nur ein sehr geringer Umformgrad erreicht. Danach kann der Umformgrad langsam gesteigert werden, ohne daß mit einem Versagen des Werkstoffes gerechnet werden muß.

Die Probleme der Drahtherstellung sind ähnlich wie beirr Kaltwalzen. Die warmgewalzten, geschmiedeten oder stranggepreßten Rundstäbe können zunächst nur mit einem sehr kleinen Ziehgrad gezogen werden. Hat der Draht bereit mehrere Züge hinter sich, kann der Ziehgrad gesteigert werden igart et al.- Zur Dwyerschen Skoliosenoperation mittels Drähten aus Memory-Legierungen





# 3. Eigenschaften der erschmolzenen NiTi-Legierung

Um Aussagen über die Eigenschaften der NiTi-Legierung machen zu können, wurden Drähte von 2 mm  $\emptyset$  aus dem Material bei verschiedenen Temperal Å geglüht und dann in der Zerreißmaschine bis zu vorgegebenen plastischen Dehnungen gereckt. Diese verformten Drähte wurden dann bis über die Umwandlungstemperatur erwärmt und die Umwandlungstemperaturen und die Rückdehnung gemessen.

# 3.1. Das Spannungs-Dehnungs-Diagramm

Abbildung 1 zeigt das typische Spannungs-Dehnungs-Diagramm der NiTi-Legierung. Klar erkennbar gliedert sich die Kurve in drei Bereiche. Vor dem ersten Bereich erfolgt ein linearer Anstieg der Spannung, der durch die elastische Dehnung der Probe hervorgerufen wird.

# Die drei Bereiche lassen sich wie folgt deuten:

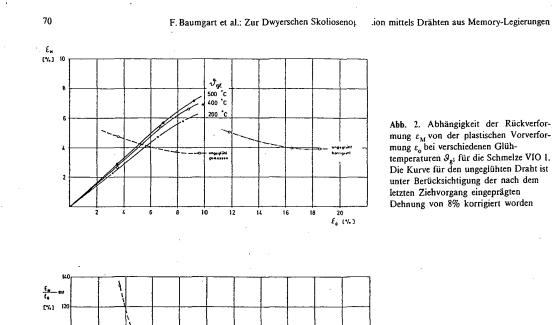
Bereich I. Sobald eine kritische Spannung überschritten wird, wird das in der Niedertemperaturphase vorliegende, martensitische Gefüge bei günstig orientierten Kristallen in "Verformungsmartensit" umgewandelt. Da für die Bildung des "Verformungsmartensits" nur eine geringe Energie erforderlich ist, wird die Spannung bei größeren Dehnungen nur geringfügig erhöht, so daß die Kurve hier einen sehr flachen Anstieg zeigt.

Bereich II. Die Bildung von "Verformungsmartensit", auf die im Bereich I der größere Anteil der Dehnung zurückzuführen war, s* stzt weitgehend abgeschlossen. Durch die nun verstärkt zenden Versetzungsbewegungen kommt es zu Versetzungsreaktionen, durch die die Gleitwege verkürzt und die Versetzungen aufgestaut werden. Mit größer werdender Dehnung nimmt die Versetzungsdichte stark zu, was eine erheblich größere Fließspannung als im Bereich I erfordert. Der Werkstoff verfestigt.

Bereich III. Sobald die Versetzungsdichte einen bestimmten Wert erreicht hat, können Schraubenversetzungen quergleiten. Dadurch werden die Versetzungen wieder frei beweglich, und die Kurve nimmt wieder einen flacheren Verlauf. Dieser Bereich wird auch als der "Bereich der dynamischen Erholung" bezeichnet.

## 3.2. Der Memory-Effekt

Erwärmt man einen plastisch verformten Draht, bis sich das in der Niedertemperaturphase vorhandene, martensitische Gefüge in ein austenitisches umgewandelt hat (Hochtemperaturphase), so nimmt er seine ursprüngliche Gestalt weitgehend wieder an. In Abbildung 2 ist die Abhängigkeit der Rückverformung von der Vorverformung und der Glühtemperatur aufgetragen. Bei den Proben handelt es sich um Drähte, die im letzten Stich um ca. 8% kaltverformt wurden. Aus Abbildung 2 ist ersichtlich, daß die Rückverformung mit steigender Glühtemperatur ansteigt. Bei einer Glühtemperatur von 500°C scheint die Rückverformung unabhängig von der danach erfahrenen plastischen Dehnung am größten zu sein. Das läßt sich dadurch erklären, daß bei dieser Temperatur bereits alle Eigenspannungen, die durch den Herstellungsprozeß in die Proben eingebracht wurden, weitgehend abgebaut sind. Bei niedrigen Glühtemperaturen (s. z. B. 200°C) dürfte das noch nicht der Fall sein, was zu



500 °C 400 °C 200 °C

Abb. 3. Abhängigkeit der prozentualen Rückverformung  $\varepsilon_M / \varepsilon_0$  von der plastischen Vorverformung  $\varepsilon_0$  bei verschiedenen Glühtemperaturen für die Schmelze VIO 1. Korrektur der für die ungeglühte Probe gemessenen Werte

1

einer Behinderung der Rückdehnung führt. Trägt man die Rückdehnung in Prozent von der plastischen Vorverformung des Drahtes auf, so erhält man Abbildung 3. Wie aus diesem Diagramm ersichtlich, beträgt die Rückdehnung bis zu einer Vorverformung von ca. 7% etwa 80%. Werden 7% Vorverformung überschritten, verringert sich der Rückdehnungsanteil. Dieser Effekt ist zwanglos aus dem Spannungs-Dehnungs-Diagramm erklärbar. Bis etwa 7% plastischer Dehnung resultiert die Verformung zum überwiegenden Teil aus der Bildung von "Verformungsmartensit", der bei Überschreitung der Umwandlungstemperatur in eine austenitische Konfiguration übergeht. Bei größeren plastischen Verformungen erfolgt diese in zunehmendem Maße durch das Wandern von Stufenversetzungen. Der dadurch aufgebrachte Verformungsanteil ist irreversibel.

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## 3.3. Die Umwandlungstemperatur

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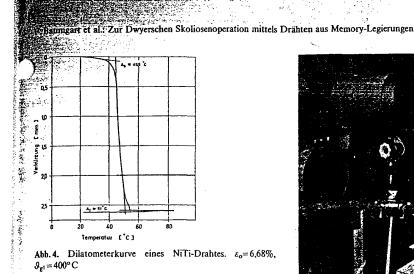
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Nimmt man von einem plastisch verformten NiTi-Draht die Dilatometerkurve  $\varepsilon = f(\vartheta)$  auf, so erhält man Abbildung 4. Aus diesem Diagramm können die Umwandlungstemperaturen wie folgt abgelesen werden:

entsprechend Abbildung 2

 $A_s \approx 45,5^{\circ}$ C,  $A_f \approx 51^{\circ}$ C. Bei der in Abbildung 4 gezeigten Kurve handelt es sich um einen NiTi-Draht, der bei 400°C geglüht und um 6,68% plastisch gedehnt worden war.

Trägt man die mit Hilfe des Dilatometers bestimmten Umwandlungstemperaturen für verschiedene plastische Dehnungen in Abhängigkeit von der Glühtemperatur auf, so erhält man die in Abbildung 5 dargestellten Kurven. Ausgehend vom ungeglühten Zustand nimmt die Umwandlungstemperatur mit steigender Glühtemperatur rasch ab. Bei einer Glühtemperatur von 200 bis 300°C zeigen die Kurven ein schwaches Minimum, um dann mit höher werdender Glühtemperatur wieder leicht anzusteigen. Aus diesen Diagrammen kann abgeschätzt werden, innerhalb welcher Bereiche die Umwandlungstemperatur durch eine entsprechende Glühbehandlung variiert werden kann.



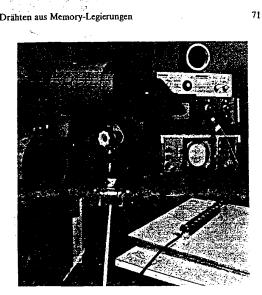
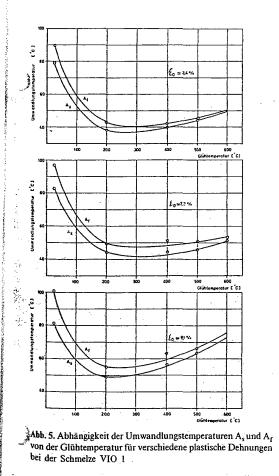


Abb. 6. Versuchsaufbau



#### 4. Modellversuch: Dwyer-Operation

Wie bereits unter Punkt 1 dargelegt, besteht die Möglichkeit, eine Wirbelsäulenkorrektur mittels eines Spanndrahtes vorzunehmen. Es erscheint nun sehr sinnvoll, bei dieser recht aufwendigen Operationstechnik den Spanndraht durch einen Memory-Draht zu ersetzen und dadurch die Operation wesentlich einfacher zu gestalten. In diesem Fall wird ein um Eo vorgereckter Memory-Draht im Bereich der Wirbelsäulenverkrümmung an den beiden äußeren Wirbeln befestigt. Durch die anschließende Erwärmung bis über die Umwandlungstemperatur nimmt der Draht seine ursprüngliche Gestalt näherungsweise wieder an und richtet dadurch die deformierte Wirbelsäule wieder gerade. Das Richten der Wirbelsäule kann dabei durch die Wahl einer geeigneten Aufheizgeschwindigkeit sehr schonend erfolgen.

Eine prinzipielle Demonstration dieser Möglichkeit wurde an einem einfachen Modell im Krupp-Forschungsinstitut in Essen durchgeführt. Abbildung 6 zeigt die Versuchsanordnung. Das Modell besteht aus kubischen Kunststoffelementen, die die Wirbel symbolisieren. Sie sind durch dazwischengeklebte Weichgummischeiben miteinander verbunden, die durch ihre Keilform die Verkrümmung des Wirbelsäulenmodells erzeugen. Der Memory-Draht von 2,8 mm Durchmesser ist um  $\varepsilon_o = 7\%$  bleibend vorgereckt, durch in den "Wirbeln" sitzende Ösen gezogen und an den Enden mit Schraubklemmen verankert. Dabei ist darauf zu achten, daß sich der Draht möglichst reibungsfrei durch die Ösen bewegen kann. Prinzipiell

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# Edwards Exhibit 1033, p. 158

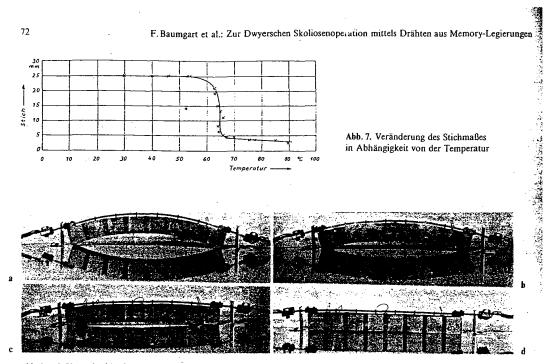


Abb. 8a-d. Versuchsablauf

besteht auch die Möglichkeit, den Draht an den in den Wirbeln befestigten Ösen festzulegen. Diese Methode bringt dann keinen Vorteil, wenn die Wirbelsäule zwischen den beiden äußeren Befestigungspunkten gleichmäßig gerichtet werden soll, d.h. wenn die verkrümmte Wirbelsäule Kreisbogenform aufweist.

Günstig wirkt sich die Festlegung des Memory-Drahtes an allen Wirbeln nur dann aus, wenn der Memory-Draht sich zwischen den einzelnen Fixpunkten um unterschiedliche Wege verkürzen soll, was bei einer vom Kreisbogen stark abweichenden Verkrümmung interessant sein könnte. In diesem Fall sind die einzelnen Drahtabschnitte zwischen den Befestigungspunkten getrennt zu beheizen und die Wärmezufuhr in dem Moment zu unterbrechen, in dem partiell die gewünschte Verkürzung eingetreten ist. Bei dem hier durchgeführten Modellversuch wird der Memory-Draht nur an den beiden äußeren Wirbeln befestigt.

Die Erwärmung des vorgereckten Drahtes erfolgt über eine stufenlos regelbare Widerstandsheizung. Zur Bestimmung der Temperatur wird eine Infrarot-Thermovisionskamera eingesetzt, die es gestattet, berührungslos die Temperatur innerhalb des Gesichtsfeldes zu bestimmen.

Hierzu wird die vom Objekt ausgehende Infrarotstrahlung punktförmig abgetastet, auf einen Detektor fokussiert und in ein elektrisches Signal umgewandelt. Dieses Signal wird auf einem Oszillografen dargestellt, wobei die Abstufung der Grautöne der Temperaturverteilung des Objektes entspricht. Diese Kamera wird übrigens auch in der Medizin eingesetzt, z.B. zur Krebsfrüherkennung. Für die kontinuierliche Registrierung der Temperatur sind am Draht zusätzlich 3 Thermoelemente angebracht, die an einen Punktdrucker angeschlossen sind.

Die ursprüngliche Verformung des Modells gemessen als Stichmaß in der Mitte — betrug 25 mm. Bei ca. 60°C wurde die Umwandlungstemperatur erreicht, und der Memory-Effekt setzte ein. Diese Umwandlungstemperatur kann durch eine geringe Änderung der Legierungszusammensetzung sowohl nach oben als auch nach unten verschoben werden, vgl. Buehler, Wang. In Abbildung 7 ist die Veränderung des Stichmaßes in Abhängigkeit von der Temperatur aufgetragen. Die Gesamtverformung von 25 auf 4 mm Stichmaß wird bei einer mittleren Drahttemperatur von ca. 65°C erreicht.

Die Verformungsgeschwindigkeit kann über die Wärmeeinbringung in weiten Grenzen gesteuert werden. Im vorliegenden Fall beträgt die interessierende Zeit 15 min, sie kann im Bedarfsfall verlängert oder auch wesentlich verkürzt werden. Eine weitere Steigerung der Temperatur von 67°C auf 90°C erbringt dann keine Vergrößerung des Memory-Effektes mehr. Abbildung 8 zeigt das Modell in verschiedenen Versuchsphasen. المردية المحجز وتوقيق THEFMOVIS 100

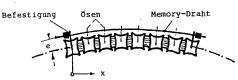


Abb. 10. System Wirbelsäule-Memory-Draht

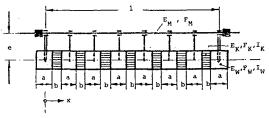


Abb. 11. Rechenmodell

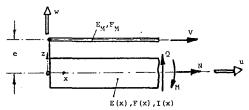


Abb. 12. Bezeichnungen

### 5. Mechanische Probleme

## 5.1. Grundlagen, System und Belastungen

Die Grundlagen für die Berechnung von Memory-Bauteilen wurden von Baumgart, Bensmann und Hartwig ausführlich dargelegt. Hier geht es um die Berechnung eines Systems aus einem einachsig beanspruchten Memory-Draht und einem als elastisch angenommenen Balken (Wirbelsäule) mit veränderlichen Material- und Querschnittswerten (Abb. 10). Von funktionellen Belastungen soll bei dieser Untersuchung zunächst abgesehen werden. Der einzig wirkende Lastzustand ist der durch den Memory-Effekt hervorgerufene Vorspannzustand des Memory-Drahtes. Die Zusammenziehung des Memory-Drahtes bewirkt eine Krümmungsänderung und eine Kompression der Wirbelsäule.

Der Berechnung wird das idealisierte System gemäß Abbildung 11 zugrunde gelegt. Es stimmt mit dem im Experiment untersuchten System praktisch überein. Der Memory-Draht ist nur an den beiden Enden

Abb. 9. Temperaturverteilung am Wirbelsäulenmodell. a Thermogramm in der Draufsicht, b Temperaturverteilung längs des Memory-Drahtes

Durch den Einsatz mehrerer Memory-Drähte kann eine beliebig starke Wirbelsäulenverkrümmung korrigiert werden. Der Mechanismus ist dabei der gleiche wie bei dem hier beschriebenen Modellversuch.

Die mit der Infrarotkamera gemessene Temperaturverteilung ist in den folgenden Abbildungen 9a und b dargestellt. Im Thermogramm (Abb. 9a) ist in der Draufsicht der erwärmte Memory-Draht als helle Linie zu erkennen. Die Köpfe der Ösenschrauben zeichnen sich als einzelne dunkle (kältere) Punkte ab. Die Wärmeableitung in die Kunststoffwürfel ist gering. In der Umgebung der eingeschraubten Ösen ist zwar eine gewisse Temperaturerhöhung festzustellen, die jedoch auf einen recht engen Bereich beschränkt ist. In Abbildung 9b ist die Temperaturverteilung in Amplitudenform längs einzelner Bildzeilen wiedergegeben. Die Temperatur stellt sich über die gesamte Länge recht gleichmäßig ein, so daß davon auszugehen ist, daß der Memory-Effekt auf der gesamten Länge des Drahtes ausgenutzt wird.

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fest mit dem Stab verbunden, dazwischen wird er in Ösen geführt. Für eine genauere Berechnung müßte die Veränderlichkeit der Querschnitte zusätzlich berücksichtigt werden.

## 5.2. Berechnung

Die Berechnung lehnt sich an die von Baumgart, Bensmann und Hartwig durchgeführte Untersuchung eines Stabsystems aus einem Memory-Stab und einem elastischen Stab an.

Für die Überlegungen wurden die in Abbildung 12 eingetragenen Bezeichnungen benutzt. Es wird davon ausgegangen, daß das System als schwach gekrümmter Balken betrachtet werden kann.

Das Biegemoment M als Funktion der Koordinate x ist beim Wirken der Vorspannkraft V im Memory-Draht bei einer konstanten Exzentrizität c des Drahtes gegenüber der Schwerachse des Stabes:  $M = -V \cdot e = \text{const.}$ 

Die Längskraft N ist

N = -V = const.

während keine Querkraft Q vorhanden ist: Q = 0

Dabei ist angenommen, daß die Stützung des Memory-Drahtes nicht in äquidistanten (reibungsfreien) Ösen, sondern kontinuierlich erfolgt.

Legt man die Querschnittswerte und die Kombination von 8 Kunststoff, wirbeln" mit 7 Gummi, knorpel"-Scheiben gemäß Abbildung 11 zugrunde, dann ergibt sich die für die Erzeugung der Vorspannkräfte V erforderliche einzuprägende Relativverkürzung  $\Delta u$  im Memory-Draht zu

$$\Delta u = \int_{0}^{l} \frac{Vdx}{E_{M}F_{M}} + \int_{0}^{l} \frac{Vdx}{E(x)F(x)} + \int_{0}^{l} \frac{Ve^{2}dx}{E(x)J(x)}$$
$$\Delta u = V \left[ \frac{l}{e_{M}F_{M}} + \frac{7a}{E_{w}} \left( \frac{l}{F_{w}} + \frac{e^{2}}{J_{w}} \right) + \frac{7b}{E_{K}} \left( \frac{l}{F_{K}} + \frac{e^{2}}{J_{K}} \right) \right]$$

Für die Verschiebung  $\Delta w$  in der Mitte des Balkens gegenüber der Anfangsauslenkung relativ zur Sehne hat man in gleicher Weise bei Annahme eines zur Stelle x = l/2 symmetrischen Systems

$$\Delta w = -Ve \int_{0}^{\pi} \frac{x dx}{E(x)J(x)}$$
$$\Delta w = -\frac{Ve}{8} \left| \frac{a(49a + 48b)}{E_w J_w} + \frac{b(50a + 49b)}{E_K J_K} \right|$$

Füreinen zahlenmäßigen Überschlag wurden die folgenden Werte eingesetzt:

 $E_w = 1.000.000 \text{ N/cm}^2 = 10.000 \text{ N/mm}^2$  (Kunststoff, Holz)  $F_w = 30 \cdot 30 = 900 \text{ mm}^2$ 

$$J_w = \frac{1}{12} \cdot 30^4 = 67.500 \text{ mm}^4$$

 $E_{K} = 0.43$  N/mm² (in einem Druckversuch gemessener Materialwert)

 $F_{K} = 900 \text{ mm}^{2}$ 

 $J_{K}^{A} = 67.500 \text{ mm}^{4}$ 

 $a^{\prime} = 30 \text{ mm}$ 

b = 8 mm

e = 19,5 mm

 $E_{\mu} = E_1 = 35.000 \text{ N/mm}^3$ 

 $\pi = \pi = 0^2$ 

$$F_{M} = \frac{1}{A} \cdot 2,8^{2} = 6,16 \text{ mm}^{2}$$

Man sieht leicht, daß die Anteile aus den "Wirbeln" gegenüber denen aus dem "Knorpel" wegen des erheblichen Unterschiedes der Elastizitätsmoduli in diesem Falle zu vernachlässigen sind. Man erhält:

$$\Delta u = V \left[ \frac{7 \cdot 38}{35\,000 \cdot 6,16} + \frac{7 \cdot 30}{10\,000} \left( \frac{1}{900} + \frac{19,5^2}{67\,500} \right) + \frac{7 \cdot 8}{0,43} \left( \frac{1}{900} + \frac{19,5^2}{67\,500} \right) \right]$$
  
$$\Delta u = V \left[ 0,0012 + 0,0001 + 0,8783 \right] \frac{\text{mm}}{\text{N}} = V \cdot 0,8796 \frac{\text{mm}}{\text{N}}$$
  
and  
$$\frac{\Delta u}{I} = \frac{V}{F_M} \cdot 204 \cdot 10^{-4} \frac{\text{mm}^2}{\text{N}}$$
  
$$\Delta w = -V \cdot \frac{19,5}{8} \left[ \frac{30\,(49 \cdot 30 + 48 \cdot 8)}{10\,000 \cdot 67\,500} + \frac{8\,(50 \cdot 30 + 49 \cdot 8)}{0,43 \cdot 67\,500} \right]$$

Der Draht ist um 7% bleibend vorgereckt. Man hat also eine Anfangsdehnung von  $\varepsilon_0 = 0.07$ .

 $\Delta w = -V \cdot (0,0002 + 1,2711) = -V \quad 1,2713 \frac{\text{mm}}{\text{N}}$ 

Man kann sich leicht überlegen, daß

$$\frac{\Delta u}{l} = \varepsilon_M (\sigma, \varepsilon_0) = \frac{V}{F_M} \cdot 204 \cdot 10^{-4} \frac{\mathrm{mm}^2}{\mathrm{N}} = \sigma \cdot 204 \cdot 10^{-4} \frac{\mathrm{mm}^2}{\mathrm{N}}$$

sein muß. Dabei ist  $v_M(\sigma, \varepsilon_o)$  eine für eine bestimmte Legierung aus Versuchen ermittelbare Kurvenschar für die Memoryrückdehnung als Funktion der Lastspannung  $\sigma$  und der Anfangsdehnung  $\varepsilon_o$ . Wir nehmen für diesen Fall an, daß die Spannung  $\sigma$  sehr klein ist, so daß keine Behinderung des Memory-Effektes erfolgt und daß bei dieser Legierung bei Lastfreiheit ( $\sigma = 0$ ) eine 80%ige Rückverformung erfolgt, d.h.

 $\varepsilon_M (\sigma, \varepsilon_o) = 0.8 \varepsilon_o = 0.056$ Damit läßt sich leicht errechnen

$$\sigma = \frac{0.056}{204 \cdot 10^{-1}} \quad \frac{N}{mm^2} = 2,75 \frac{N}{mm^2}$$

und eine Vorspannkraft von

 $V = \sigma F_M = 2,75 \cdot 6,16 = 16,9 \text{ N}$ 

Damit ergibt sich eine Stichänderung von

 $\Delta w = -16.9$  1,2713 = -21,5 mm

Dies stimmt recht gut mit den gemessenen Werten überein.

Der nach Abschluß der Erwärmung verbleibende Spannungszustand sieht also im Versuchsmodell grö-Benordnungsmäßig so aus:

im Memory-Draht

$$\sigma = 2,75 \frac{N}{mm^2}$$
 Zugspannung

in der "Wirbelsäule"

$$\sigma = \frac{V}{F} - \pm \frac{Ve}{W} = -V \cdot \left(\frac{1}{a^2} \pm \frac{6e}{a}\right) = -\frac{V}{a^2} \cdot \left(1 \pm 6\frac{e}{a}\right)$$
$$= -\frac{16.9}{900} \left(1 \pm 6 \cdot \frac{19.5}{30}\right) = \begin{cases} -0.0920 \text{ N/mm}^2, \text{ Druck,} \\ +0.0545 \text{ N/mm}^2, \text{ Zug.} \end{cases}$$

Man sicht, daß die Beanspruchungen sehr gering sind, was auf die große Nachgiebigkeit der Gummischeiben zurückzuführen ist. Auf der ursprünglich

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Baumgart et al.: Zur Dwyerschen Skoliosenoperation mittels Drähten aus Memory-Legierungen

konkaven Seite treten Zugspannungen auf, während e dem Memory-Draht zugewandte Seite Druckspannungen aufweist.

# 6. Abschließende Bemerkungen

Der beschriebene Versuch zeigt, daß es möglich ist, Krümmungsänderungen an der Wirbelsäule durch Memory-Spanndrähte vorzunehmen. Die Vorteile bei dieser neuartigen Technik sind das Entfallen jeglicher Spannvorrichtung und die schonende Auslösung durch Erwärmen des Drahtes. Der Draht braucht nur an den beiden Enden der Spannstrecke hinreichend verankert zu werden, an den Zwischenwirbeln genügen u. U. leichte Führungsösen, so daß der Umfang des Eingriffs reduziert wird. Die bei dieser Maßnahme auftretenden Spannungen in der Wirbelsäule sind relativ gering.

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# 苦労しました! ゴム金属

----超弾性 Ni-Ti 合金をメガネフレーム用ワイヤへ応用開発-----

古河北気工業 中央研究所 鈴木 雄一

 メカネのフレームは構造的にいくつかのタイプに分け ちれるが、これらの内で、レンズに溝(グルーブ)を切 ってワイヤをかけ、レンズを吊って固定するタイプのも のは、軽くて、視野が広いなど、機能的にきわめてすぐ れたフレームである。デザイン的な面からも、レンズ下 部のリムがほとんど見えないため、いわゆるリムレスフ レームと同じ外観を有し、誰にもフィットしやすいファ ッション性の高いフレームであるといわれている。特 に、プラスチックレンズと組み合わせると、軽量という 利点が生かされ、より効果的である。プラスチックレン ズで高いシェアをもつ、㈱諏訪精工舎が新しいメガネフ



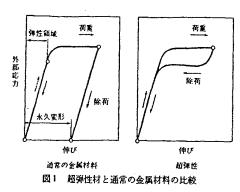
レームの開発に当って、このタイプのフレームに着目し たのは主にこのような理由からであった。

ところが、ワイヤで吊るタイプのフレームはワイヤの 材質に問題があった。従来、このタイプ のフレーム に は、通常の金属製のワイヤ (たとえば、ニッケルクロム 合金)や、合成樹脂のワイヤなどが用いられていたが、 いずれも、レンズを拭く時や、冬場レンズが縮んだ時に 外れやすいとか、長期間使用するとのびてしまうとか、 いろいろと問題があり、その機能上の利点を生かせない のが実状であった、レンズを吊るワイヤはレンズの収縮 やフレームの変形に追従する弾力と、長時間の使用に対 してもクリープなどでゆるまない耐久性を備えていなけ ればならない、通常の金属バネ材では前者の、合成樹脂 では後者の点で十分でなかった。ワイヤは、さらに、キ ズがつきにくく、耐食性にすぐれ、外観的にも美しい、 といったメガネフレーム材としての一般的な要件も満足 しなくてはならない、これらの多くの条件を満たすワイ ヤ材を選択することは困難な課題であった。

㈱諏訪精工舎では、新しいフレームを開発するに際して、まず、このワイヤ材の選定に力を注ぎ、いくつかの材質については実際に試作をするなど、積極的な調査検討を行なった。経過の詳細は割愛するが、結局、超弾性Ni-Ti 合金のワイヤがあらゆる面から最適であるという結論にいたった。

1. ゴムのように元にもどる超弾性合金

超弾性とは、弾性限界をはるかにこえ、降伏領域にお よぶ変形が、変形応力を除くと変形前のひずみゼロの状態にもどってしまう現象である(図1)。通常の金属材料では弾性限以下のひずみは除荷時に完全に元にもどる が、ひずみが弾性限をこえると応力を除いてもひずみは 完全に消失せず永久変形が残る。したがって、たがだか



0.5%程度のひずみしか元にもどらない。超弾性合金で は降伏点をこえ,降伏領域の終点近くまで変形しても, たとえば Cu-Al-Ni 合金の単結晶では実に10%をこえ る変形ひずみが除荷時に完全に元にもどってしまう。超 弾性といわれる所以である。超弾性はこのような変形挙 動から,擬弾性,ゴム弾性ともよばれる。

超弾性を示す合金は、現在、Au-Cd、Cu-Al-Ni, Ni-Ti 合金など+数種類が知られているが、それらの合金 の多くは単結晶でないと超弾性を示さない、工業上、実 用的な多結晶材では変形ひずみが超弾性ひずみにいたる 前に、つまり、降伏点をこえる前に粒界で破断してしま う. 多結晶ですぐれた超弾性特性を示すのは、今のとこ ろ、Ni-Ti 合金のみである。また、この合金はチタン並 みのきわめて良好な耐食性をもち、応力腐食割れの心配 もないこともあって、超弾性材料として最も有望視され ている、メガネフレーム材としては、さらに、軽量で、 金属光沢面の美観にすぐれているなどの利点があげられ る。

ところが、開発を始めた当時、Ni-Ti 合金の超弾性に 関する研究報告は、同じ合金の形状記憶効果にくちべて 極端に少なく、わずかに、本間¹⁾ と Wasilewski²⁾ が独立 に発表した2件の報告があるだけであった。この2つの 報告は簡単なもので、データも少なかったため、超弾性 Ni-Ti 合金の開発は、まず合金組成を見つけることから 始めなければならなかった。

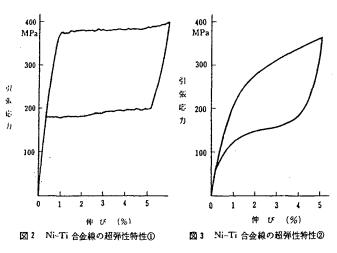
一方, 超弾性現象については, Au-Cd, Cu-AI-Ni 合 金などで1950年代 から 多くの研究が 行なわれてきてお り, 1970年代のはじめにはこの現象が形状記憶効果と同 じく熱弾性型マルテンサイト変態によっておこること, マルテンサイト変態温度と密接な関係があることなどが 判明していた. これらの理論によれば, Ni-Ti 合金もマ ルテンサイト逆変態温度 (Ar 点)を使用温度よりやや低 い温度に設定してやれば超弾性特性を示すことが予想さ れた。しかし、Ni-Ti 合金のマルテンサイト変態温度は 組成 (Ni wt%) が0.1% ずれただけで、約10℃も変化す るため、変態温度の制御が非常にむずかしいことが知ら れており、所定の変態温度をもつ合金の調製が実用化へ の第1の関門であった。

## 2. 超弾性 Ni-Ti 合金の細線を作る

当社では以前から,超弾性や形状記憶効果とはまった く別に,Ni-Ti 合金の高温相(母相)の特徴であるすぐ れた耐食性と耐摩耗性に着目し,すでに昭和45年にこれ を改良した摺動部材用合金(FAEDIC-NT®)の商品^ル に成功していた.FAEDIC-NT は主に化学ブラント ^ル 器の部品として製造販売を続けているが,最近では,原 子力発電所の炉水浄化系ポンプ部品として採用され,高 い評価を受けている.これらFAEDIC-NT の製造技術 特に溶解鋳造技術はほとんど超弾性材に応用できたの で,均質で健全な合金鋳塊を作ることは比較的容易であ った.マルテンサイト変態温度の制御,つまり合金:) 制御はかなり離航したが,示差走査熟量計(DSC)によ る測定精度の向上,溶解法の改良などにより,±5℃以 内にコントロールできるまでになった.

超弾性合金線製造の第2の関門は伸線加工であった. Ni-Ti 合金は金属間化合物でありながら冷間で塑性加工 ができるまれな材料であるといわれているが、実際にこ れを加工してみると、加工硬化が非常に大きくなか。 加工がむずかしい材料であることがわかる.形状記憶効 果を示す組成範囲の Ni-Ti 合金もかなり伸線加工 が む ずかしいが、超弾性材は Ni 量がわずかに多いだけなの にこれより大幅に加工が困難である.このよう な材料 を、焼付、ビビリ、断線などなしに能率的に伸線するた めには、ダイス形状、潤滑、熱処理(中間焼なまし)、整 直など多くの点に高度な技術が必要であった.

これらの技術のベースになったのは、当社で主にエレ クトロニクス関係向けに製造販売しているチダン細線の 製造技術および設備であった。フレーム用ワイヤの場合、 超弾性特性だけでなく、通常より高い線径精度と真直度 が要求されていたが、数次の設備および加工条件の改良 により、図2のような特性の Ni-Ti 細線を作るこ ' *i できた。現在では、線径が0.1 mm までの超弾性 Ni ii 線を精度よく製造することができる。これらのNi-Ti 線は熱処理のやり方によって図3のようなややなだらか な超弾性特性をもたせることができる。



# 3. ビスタ「リムライトフレーム」

20年にはないで、このでのこと。

超弾性 Ni-Ti 線材の開発と併行して、㈱諏訪精 工舎 ではこれを使用したフレームの開発が進められた. 超弾 性線をフレーム用ワイヤとして実用するためには,たと えば,好ましい光沢を与える磨き方,弾力を適正に整え る熟処理法など多くの問題を解決しなければならなかっ たが,特に苦労したのは,ワイヤ端部をフレーム上部に 固定する方法であった.ワイヤと他の部品が溶接できれ ば問題はないのであるが,これは Ni-Ti 合金の性質か ら不可能であった.ろう付は可能ではあるが,繰り返し 変形に弱くなり良い結果は得られなかった.このため時 計製造で培った精密加工技術を応用して機械的な固定法 をいろいろと改良した結果,カシメとネジ止めを併用す る方式で解決することができた(図4).また,レンズに 溝(グループ)を切る新し

い装置もあわせて開発され た.

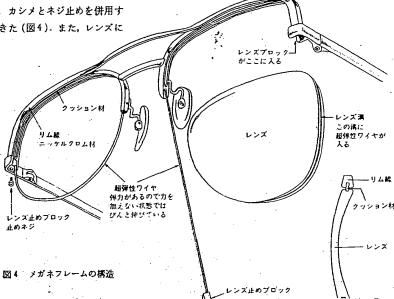
このようにしてできあが ったフレーム (ビスタ「リ ムライトフレーム」^ゆ) は, レンズが伸び縮みしたり, 強い力がかかっても, 超弾 性ワイヤがレンズをしっか り保持し, また, 長期間の 使用にもゆるまないなど、このタイプの フレームのこれまでの欠点をすべて解決 したといってよいであろう、デザインは 写真に示すとおりで非常に好評である。

# 4. バネ材として広い応用分野

超弾性 Ni-Ti 合金は通常の金属 材料 と比較すると桁違いに大きな復元能力を もつので、メガネフレームば かりで な く、パネ材として大きな期待がもたれて いる。最近は線材以外に板材、条材に対 する要求がふえており、これらの内では マイクロモーター、コンピューター周辺 機器などの小型精密機器関係が有望であ

る 超弾性パネは本質的に非線形パネであるため、当然 この非線形性を生かした使い方が考えられるが、通常の パネと同じ設計法が適用できないので、現在は、パネ加 工,熱処理法などとともに超弾性パネ素子(コイルパネ、 ヘリカルパネなど)の設計法が検討されている。

超弾性 Ni-Ti 合金の特殊な用途として医療機器 への 応用がある. Ni-Ti 合金は耐食性が良いうえに,生体に 対する適合性 (パイオコンパティビリティー)が非常に 良く,生体に埋め込んで使用できるので、インプラン ト材としての応用が考えられている.形状記憶効果 Ni-Ti 合金の医療への応用研究は割合と進んでいるが,超弾 性については研究が始められたばかりである.現在, 骨



の固定,結束などの整形外科関係の研究が,形状記憶効 果の応用と併行して国立大阪南病院と大阪府立工業技術 研究所で,歯列矯正をはじめとする歯科関係の研究が東 京医科歯科大学で進められており,その成果の一部はす でに学会で発表されている、

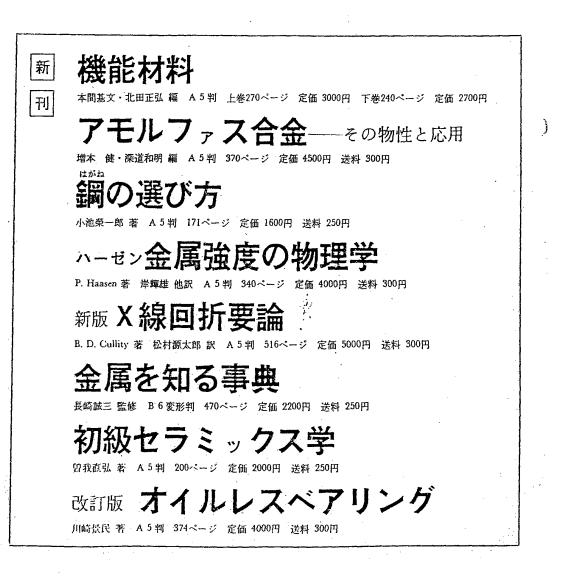
一方,大学の金属系研究室を中心に,Ni-Ti 合金の超 弾性に関する基礎的な研究が行なわれており,最近の宮 崎ら"の報告に見られるように,その変形挙動,温度依 存性などが次第に解明されつつある。

超弾性 Ni-Ti 合金は,小規模な製造が始められ たば

かりで、まだ実用化の例も少ないが、5%をこえるひず みがゴムのように完全に元にもどるという性質は、金属 バネ材として画期的なものであり、今後の広範な応用が 期待される、

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# 新しい超弾性 NiTi 矯正用ワイヤーの研究

# (第1報) 引張および曲げ試験*

# 渡辺勝 久**

# Studies on New Superelastic NiTi Orthodontic Wire

#### (Part 1) Tensile and Bend Test

#### Katsuhisa WATANABE

# Division of Metallurgy, Institute for Medical and Dental Engineering, Tokyo Medical and Dental Univ. Tokyo (Director: Prof. Ishi Miura)

NiTi alloy has attracted the most public interest in recent years. This alloy has two unique properties, one is "shape memory effect" and the other is "outstanding elasticity". NiTi wire with the property of outstanding elasticity suggested it could be useful in orthodontics.

Recently new NiTi alloy which had other characteristics of "superelasticity" was developed. In this study, new superelastic NiTi wire was examined to evaluate for use in orthodontics, in tensile and bend tests, compared with stainless steel, Co-Cr alloy and work hardened NiTi wire.

The results obtained are as follows;

(1) New superelastic NiTi wire showed unique stress-strain curve with a plateau from the strain of 2% to 5%, the unique deformation behavior was caused by stress induced transformation, and returned to almost zero strain as stress was reduced.

(2) New superelastic NiTi wire showed an elongation of about 11%. In the cyclic tensile tests to the strain of 8%, new superelastic NiTi wire showed little permanent deformation of 0.5% after 10 cycles.

(3) In bend tests, new superelastic NiTi wire showed lower than half load compared with stainless steel and Co-Cr alloy wires. Its permanent deformation was very little after 2 mm deflection.

(4) The load-deflection curve of new superelastic NiTi wire showed almost constant load in the wide range of deflection.

The results of this study indicate that new superelastic NiTi wire must be considered as a promising candidate for orthodontic arch wire.

## I. 緒

歯科矯正臨床において、不正咬合を積極的に矯正治療

言

- * 本論文の要旨は第40回歯科理工学会学術講演会(昭和56 年5月17日)において発表した。
   原稿受付 昭和56年9月28日
- ** 東京医科歯科大学医用器材研究所金属材料部門(指率 三 浦維四教授) (千代田区神田駿河台 2-3-10)

する方法として、フルバンドシステムが一般的に使用さ れている。しかし、従来のフルバンドシステムに使用す るアーチワイヤーは、ステンレススチールや Co-Cr 基 合金のものが主に用いられてきたが、これらは弾性率が 高く、少量の変形で永久ひずみが生じるという欠点があ る。このため、口腔内への装着に際しては、ワイヤーを 歯列に適合させて屈曲する必要があり、適切な矯正力を 得るためには、アーチワイヤーの屈曲とデザインに熟練

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### と工夫を必要としている^{1~31}.

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の日にと感謝で弱い

最近, NiTi ワイヤーが注目をあつめている. 米国で は非常に弾力性のある NiTi ワイヤーが市販され、これ に関する矯正学的また 理工学的研究が G.F. Andreasen らにより報告されている1~6.

NiTi 合金は普通の合金とは異なり、Ni と Ti が原子 比で1:1の割合で結びついた金属間化合物であるが、 特異な性質を持っており、加工が可能で、しかも形状記 憶効果という 特殊な現象を示す^{7.81}. これは 一定の温度 以下の温度で変形した後、これを逆変態温度以上に加熱 すると変形前の元の形状に戻る現象である。 逆変態点以 上の高温相は CsCl 型体心立方構造になっており、機械 的性質および耐食性に優れている. そこで逆変態温度を 室温以下に抑えさらに加工を加えることによって弾性を 増すことができる、現在市販されている NiTi 合金ワイ ヤーはこの加工硬化型のワイヤーであると考えられる". また、この NiTi 合金は組成や加工および熱処理により 大きく性質が変化するという特徴があり、適当な条件の 下では、超弾性という特殊な性質を示すことが判ってき 1-10).

本研究は、新しく開発された超弾性という特殊な性質 を示す NiTi ワイヤーについて、引張および歯列を模し た3点曲げ試験を行い、理工学的見地から、矯正用ワイ ヤーとしての考察を試みた. なお従来から用いられてい るステンレススチール、 Co-Cr 基合金 および 加工硬化 NiTi ワイヤーについても 同様に実験を行ない 比較検討 した.

# II. 実験材料および実験方法

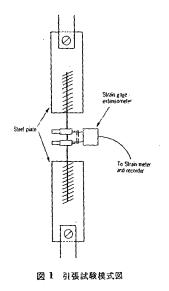
実験材料は ø 0.4 mm (0.016")の超弾性型 NiTi ワ イヤー,加工硬化型 NiTi ワイヤー,2種類の市販ステ ンレススチールワイヤー、3種類の市販 Co-Cr 基合金 ワイヤーを使用した. なお Co-Cr 基合金については、 メーカーの 指示に 熱処理を 施すことが 記されているた め、ここでは500℃の電気炉内で1分間の熱処理を行っ た.

#### 1.a 引張試験

本研究では ø 0.4 mm の細い ワイヤーの 引張試験を 行うために、図1のように 20×80mm の鉄板の間にエ ポキシ樹脂接着剤 (Araldite Standard, Ciba-Geigy) に よりワイヤーを接着した。 鉄板には ¢8mm の穴をあ け、特別に作製した専用チャックにより試験機に装着を して試験した.

試験機は万能試験機 Instron Model 1102 を使用し、 クロスヘッドスピード 1 mm/min にて行った. 同時に 正確な 応力-ひずみ 曲線 を 求めるために, 精密伸び計 (Instron Strain Gage Extensometer G 51-17 MA ゲー で変位させ荷重を測定できるように設計を行った. 支点

ジ長 10 mm)を図1 に示すように 試験片部に 取付け, 引張荷重に対応する伸びを測定し、レコーダーで自動記 録した.以上の方法および 測定によって 応力-ひずみ曲 線を求めた

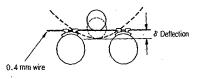


Lb 引張繰返し試験

上記の引張試験と同様な方法により NiTi ワイヤーを 一定量引張った後に、除荷を行い応力が0となるまで戻 すという操作を繰返して試験した. 前記の引張試験の結 果から判断して、超弾性型 NiTi については伸び8% ま で、 加工硬化型 NiTi では伸び 4.8% までの試験を 10 回繰返した.

#### 曲げ試験

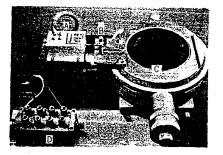
矯正用ワイヤーの曲げに対する変形量と荷重の関係を 調べて比較を行うために、新しくワイヤー曲げ試験機を 作製して試験を行った、この曲げ試験機は犬歯から小臼 歯部分の歯列矯正を想定して、図2に示すように、直径





7mm の金属製丸棒を、中心間距離 14mm で植立固定 し、支点間に渡したワイヤーの中央を直径5mmの丸棒

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図3 矯正ワイヤー用に試作した微小3点 曲げ試験機 A:ロードセル B:3点曲げ試験部 C:移動ステージ D:ゲージボックス

の丸棒には、ダイレクトボンディング用金属ブラケット (0.018"×0.025" slot) を 瞬間接着材により接着固定し て、ワイヤーの支持点とした. なお曲げ試験機は図3の ように万能投影機のステージ (Nikon 社製) に支点部分 を固定し、ステージを移動させることにより変位を与え ることができるようにした. ステージの移動はマイクロ メーターにより 1/1000 mm まで読取ることが可能で、 変位は 50 µ/30 sec で行った. 荷重はロードセル (東洋 ボールドウィン社、容量 1000 gf) によりレコーダーで 自動記録した (図3). 試験は一定の変位量に達するま でワイヤーに変位を加えて行き、引続いて変位を減少さ せ荷重が 0 となるまでの一連のサイクルを測定した. 試験の条件は、表 1 に示した通りである. ワイヤーの

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試験条件	Bracket 種類	Bracketでの固定	変位量
(a)	Single	無	2 m m
(b)	Single	有	2 mm
(c)	Siamese	無	2 mm
(d)	Siamese	有	2. ?~3 mm

支持点として、Single type と Siamese type の 2 種類の ダイレクトボンディング用 金属ブラケット (小臼歯用) を使用し、ブラケットの固定には歯科矯正用リングレッ トを用いた。

# III. 実験結果

l.a 引張試験

引張試験での 応力-ひずみ曲線は、 図4に示した. ス テンレススチールは弾性率が 17~20×10³ kg/mm² と高

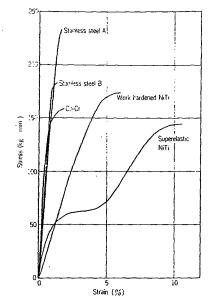


図4 各種ワイヤーの応力-ひずみ曲線

く、伸びは 1.2~1.6%、 引張強さはステンレススチー ルA が 233.3 kg/nm²、ステンレススチール B が 182 kg/ mm² であった. 次に Co-Cr 基合金は 弾性率 20~23× 10³ kg/mm²、 熱処理を行わないものは 引張強さ 150~ 185 kg/mm²、 伸びは 2.4~3% であったが、 熱処理を 行うと引張強さ 160~200 kg/mm²と 上昇をし、反対に 伸びは 1.8~2% 程度と減少した .

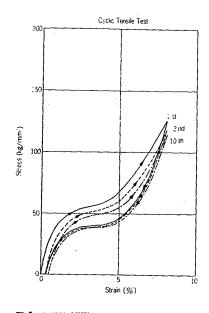
以上のワイヤーに比べ NiTi ワイヤーは明らかに低い 弾性率と大きな伸びを示した.加工硬化型 NiTi は弾性 率 5~6×10³ kg/mm² とフレキシブルで、ほぼ直線的な 挙動を示し約 6% まで伸びた.一方超弾性型 NiTi は他 種のワイヤーとは異なり、中央に 平担な 部分をもつ 応 力-ひずみ曲線を示した.すなわち伸び 2% までは弾性 率 8×10³ kg/mm² を示すが、2% を過ぎると 5% あた りまで応力は 増加せず、ほぼ一定の値となった.伸び 5% を過ぎてから再び応力が増加しはじめ右上がりの曲 線を示し、伸び 11% で破断した.

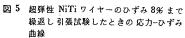
1.b 繰返し引張試験

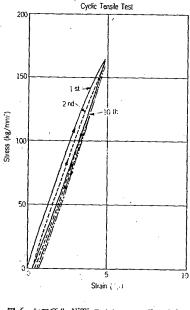
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NiTi ワイヤーに対して行った 繰返し引張試験の 結果 はそれぞれ、図5と図6に示した. 超弾性型 NiTi は図 5のように、引張試験において観察された曲線途中の平 担部が、荷重を減少していく際にも現れた. 1回目の永 久変形は 0.3% であり、10 回繰返した後の永久変形は 0.5% と、ほとんど変らなかった.加工硬化型 NiTi は 10 回繰返した後は永久変形 0.75% であった.

Edwards Exhibit 1033, p. 170







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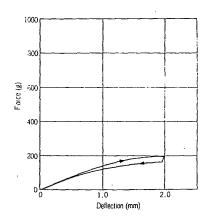
図 6 加工硬化 NiTi ワイヤーのひずみ 4.8% まで繰返し引張試験したときの 応力-ひ ずみ曲線

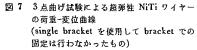
# 2. 曲げ試験

条件 (a) (single bracket 使用, bracket でのワイヤ -の固定をしない.)

この条件での実験結果は 図7~図 13 に変位量と荷重 で示した. 超弾性型 NiTi は変位量が1mm を超えると, 荷重の増加率が低下し,曲線は次第に平担となった. 変 位 2mm での荷重は 200g であった. 変位量を 2mm から 漸次減少させた時にも,変位量 2mm から 1mm までは荷重の変化が少なく,永久変形は 0.01mm 以下 であった.

加工硬化型 NiTi は変位に対する荷重の増加がはぼ直





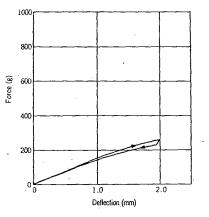


図83点曲げ試験による加工硬化 NiTi ワイ ヤーの荷重-変位曲線 (single bracket を使用して bracket で の固定は行わなかったもの)

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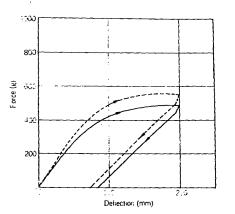


図93点曲げ試験による Co-Cr 基合金 ワイ ヤー A の荷重-変位曲線 破線で表わしたものは熱処理を施したワ イヤー

(single bracket を使用して bracket での固定は行わなかったもの)

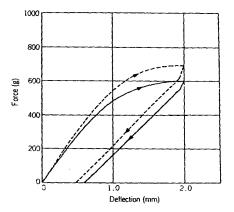


図 10 3 点曲げ試験による Co-Cr 基合金ワイ ヤーBの荷重-変位曲線 破線で表わしたものは 熱処理を施した ワイヤー (single bracket を使用して bracket で の固定は行わなかったもの)

線を示し、 変位 1.7 mm を 過ぎたあたりからやや増加 率が低下を示した. そして変位 2 mm で荷重 258 g を 示した. 変位減少時も直線的に荷重を減じ,永久変形は 0.030 mm であった.

一方、Co-Cr 基合金は変位 2.0 mm における 荷重と 永久変形量にかなりの差があったが、3種ともほとんど 同様の曲線を示した。点線で示したものが熱処理を行っ た試料である。最初は直線的に荷重が上昇したが、途中

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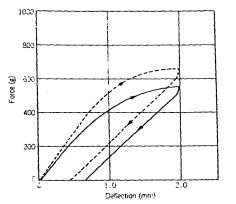
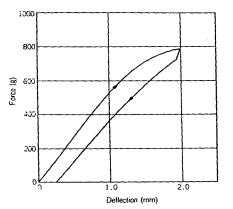
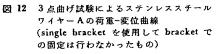


図11 3 点曲げ試験による Co-Cr 基合金ワイ ヤーCの荷重-変位曲線 破線で 表わしたものは 熱処理を施した ワイヤー

(single bracket を使用して bracket で の固定は行わなかったもの)





から荷重の増加率は低下し始めた。2 mm 変位での荷重 は 580~700g であった。 変位を減少していくと荷重は ほぼ直線的に減少し, 永久変形は 0.45~0.8 mm とか なり大きな値を示した。

ステンレススチールは Co-Cr 基合金と似たような曲線を示した. ステンレススチール A は変位 2 mm での 荷重は 785 g, 永久変形は 0.23 mm であったが, ステン レススチール B は変位 2 mm で荷重 690 g, 永久変形は 0.63 mm と大きな差を示した. しかし,両方のワイヤー とも変位を減少していくと、S字に近い直線を示した.

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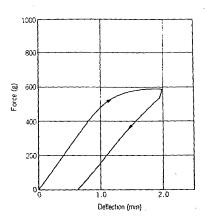
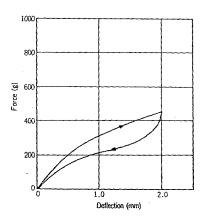
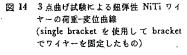


図 13 3点曲げ試験によるステンレススチール ワイヤーBの荷重-変位曲線 (single bracket を使用して bracket で の固定は行わなかったもの)

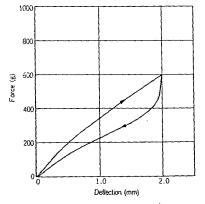


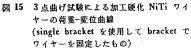


条件 (b) (single bracket 使用、リングレットにより ワイヤーを固定)

これらの結果は図14~図20 に示した. 超弾性型 NiTi は変位開始後に変位約 0.4 mm から徐々に荷重の増加率 が低下し始め、変位約 0.7 mm からは 荷重の増加率が 一定となった. 2 mm 変位での荷重は 460gを示した. 変位を減少させると 1.5 mm から 0.5 mm の間では、 ほぼ一定の荷重の部分が あらわれた.永久変形は 0.030 mm を示した.

加工硬化型 NiTi は超弾性型と同様に変位 0.5 mm で 増加率の変化を示した。しかし、超弾性型に比べると変





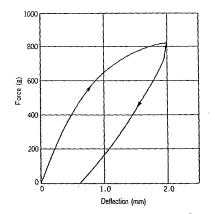


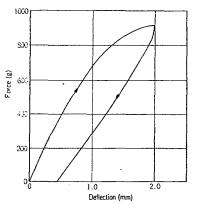
図 16 3 点曲げ試験による熱処理を施した Co-Cr 基合金ワイヤーAの荷重-変位曲線 (single bracket を使用して bracket で ワイヤーを固定したもの)

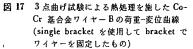
化は少なかった. 変位 2 mm では 600 g の荷重を示した. 変位を減少させると,最初急厳に 荷重が 低下したが,その後は,ほぼ一定に 荷重も 減少し,永久変形は0.038 mm であった.

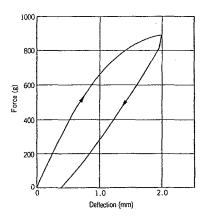
Co-Cr 基合金は条件(a)とほぼ同様な荷重の増加を示 した. 2 mm 変位における 荷重は 800g から 900g で あった. 変位を 減少すると、 荷重は 直線的な減少はせ ず、減少率を徐々に低下させる傾向を示した. また永久 変形は 0.38 mm から 0.6 mm であった.

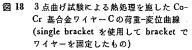
ステンレススチール B は引張試験や条件 (a) において Co-Cr 基合金に似た性質を示したが、この試験でも Co-

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Cr 基合金に近い曲線を示した. 2 mm 変位での 荷重は 840 g, 永久変形は 0.63 mm であった. 一方ステンレス スチールA は変位 1.7 mm でロードセルの容量 1000 g に達したため、2 mm まで変位を 与えることが できな かった.

条件 (c) (siamese bracket 使用, bracket での固定 をしない.)

この結果は図 21 に示した. 超弾性型 NiTi は変位 1 mm 程度から荷重の増加率が鈍くなり,変位 1.5 mm を過ぎると荷重の増加率は非常に低下した. この挙動は 変位を減じたときにも 現われ,変位 2 mm から 1 mm

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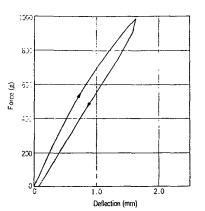


図193点曲げ試験によるステンレススチール ワイヤーAの荷重-変位曲線 (single bracket を使用し bracket でワ イヤーを固定したもの)

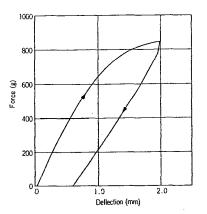


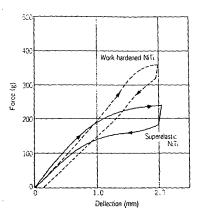
図 20 3 点曲げ試験によるステンレススチール ワイヤー B の荷重-変位曲線 (single bracket を使用し bracket でワ イヤーを固定したもの)

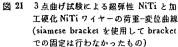
までは荷重の変化が少ない曲線となった。一方、加工硬 化型 NiTi はほぼ直線的な挙動を示し、超弾性型とは異 なる曲線を示した。

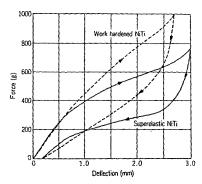
条件 (d) (siamese bracket 使用, bracket 部でリン グレットによりワイヤーを固定した。)

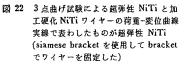
この条件での結果は図 22 に示した.変位約 0.5 mm までは超弾性型 NiTi のほうがやや高い荷重を示した. しかし、超弾性型はその後荷重増加率が低下して、曲線 はやや平担な傾向を示し出した.加工硬化型 NiTi は途 中やや荷重増加率が低下したものの、ほぼ直線的に荷重 が増加し、変位 2.7 mm で荷重は 1000g まで達した.

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変位を減少してゆくと、両ワイヤーとも最初急に荷重を 減じたが、 超弾性型は変位 2.5 mm から荷重の変化が 少なくなり、変位 0.8 mm までは 非常にゆるやかな荷 重の減少を示した.一方、加工硬化型は急激な荷重の減 少を示した後は、直線的な荷重の減少を示した.

#### IV. 考

寥

超弾性型 NiTi ワイヤーの歯科矯正用アーチワイヤー としての適性を検討するため、引張試験、3点曲げ試験 を行ない理工学的な見地から検討を試みた.その結果、 超弾性型 NiTi ワイヤーは非常に興味ある特性を示し、 歯科矯正用ワイヤーとして優れた性質を有することが示 唆された.以下に、現在矯正ワイヤーとして用いられて いるステンレススチール、Co-Cr 基合金、加工硬化型 NiTi ワイヤーと比較しながら、超弾性 NiTi 合金ワイ ヤーの特性について見ていくことにする.

1. 引張試験

まず引張試験における 応力-ひずみ曲線図から 見てい くことにする。一般に 金属材料の 応力-ひずみ線図は, まず連続的な弾性変形とそれに引続いて起る塑性変形か ら成っている、弾性変形の範囲では外力を除くとひずみ は消失して変形前の形状に戻る.弾性限を超えるとひず みは急に増加し始めるが、弾性範囲を超えたひずみは応 力を除いても、 永久変形として残留する。 矯正用ワイ ヤーとしての使用を考えた場合、利用できる範囲はこの 弾性限度内であり、この弾性限度のひずみ量が大きく取 れるほど優れた矯正用ワイヤーということになる、また 応力-ひずみ曲線の勾配,すなわち 弾性率も 矯正用ワイ ヤーとして重要な要因である.図4においてステンレス スチールや Co-Cr 基合金ワイヤーのように 弾性率が高 ければ、わずかなひずみ量に対して大きな応力を生じて しまい、また逆にほんのわずかにひずみ量が減少した場 合でも応力は大きく減少し、歯科矯正においては荷重の 変動が大き過ぎ、そのためにはループを作るとか様々な 工夫が必要になってくる^{11,12)}.

加工硬化型 NiTi ワイヤーの場合, 応力-ひずみ線図 は直線的であるが、弾性限度は約4% であり,ステンレ ススチールや Co-Cr 基合金よりも 広いひずみの範囲ま で応用でき,しかも弾性率が約1/4 と低く,ひずみ量が 大きくなっても応力はあまり増加しないという特徴があ る.すなわち矯正用ワイヤーとしてはより望ましい材料 ということになると考えられる.

新しく開発された超弾性型 NiTi ワイヤーの場合,他 の種類のワイヤーとは、異なった応力-ひずみ曲線を示 す.伸び1%までは直線的な弾性変形を示した後に応力 はほぼ一定値となり、それが約4%まで続いた後に再び 応力は上がり始め約8%まで直線領域を示して、10%あ たりから再び曲り始めて伸び 11% で破断に至る. 伸び 1% までの直線部分は通常の弾性変形によるものである が、1%以上の変形は、一般の金属材料に見られるような 転位のすべりによる塑性変形やそれに伴なう加工硬化と はまったく異なったメカニズムによるもの、すなわち応 力誘起変態という特殊な変形機構による見かけ上の降伏 現象である100. このことは図5に示した繰返し引張試験 の結果から明らかである。すなわち超弾性型 NiTi ワイ ヤーを伸び8% まで引張った後に伸びを減少させていく と、増加のときと同様に応力がほぼ一定となる部分があ らわれ、ほぼ元の位置まで戻ってくる.一定となる応力 はひずみの増加時に比べて若干小さい値となるが、この

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特徴のある変形挙動を超弾性と称している.

超弾性は, 最近注目を あびている 形状記憶効果と 同 様,熱弾性型マルテンサイト変態すなわち変態温度と逆 変態温度の差の少ないマルテンサイト変態に起因するこ とが知られるようになった。形状記憶効果というのは、 一定の変態温度以下で変形させた後に、逆変態温度以上 に加熱すると変形前の元の形状に復帰する現象を言う. 変態点以下での変形は一般の金属材料とは異なり転位の *すべりによる変形ではなく,応力が加わると結晶が部分 的にマルテンサイト変態を起しながら変形が進行する. これを逆変態以上の温度に加熱すると、マルテンサイト 変態していた部分がもとの CsCl の結晶構造にもどるた めに、 全体として 元の形状に 復帰するという ものであ る^{7~99}. 超弾性型 NiTi 合金の場合は逆変態温度 (Af 点) を室温よりも低くしてあるために、室温では形状記憶効 果が起らないが、荷重をかけていくと応力により変態が 誘起され、見掛け上は変形が起るが、 周囲の温度が Af 点以上なので温度を加えなくても応力を減じるとすぐ逆 変態が起り元に戻るというわけである10. このようにし て伸び1%から4%までは変形が進むが、4%以上とな ると次第にこの応力誘起変態も起り難くなり、さらに変 形させるためにはより多くの応力を必要とするようにな る. 伸び 4%から8% にかけて直線的に応力が増加する のはこのためであると考えられる.

超弾性型 NiTi 合金は 破断するまでの 伸びが約 11% ある. しかも繰返し引張試験において 8% 前後の変形で もほとんど元に戻り、 10 回の繰返しの後においても生 じた永久変形量は0.5% というこれまでの金属材料では 到底想像できないような大きな復元力を持っている. ま た現在市販 されている 加工硬化型の NiTi に 比較 する と、1% までの少ないひずみ量がそれ以上増すと応力が ほとんど一定となるという特質を持っている. この超弾 性型 NiTi の矯正用ワイヤーへの応用を考えた場合, 変 位が次第に減じた場合でもほぼ一定の復元力を有すると いう特徴があり、現在市販されている NiTi ワイヤーよ りも一歩進んだ材料であると考えられる. 引張試験でこ のような結果が得られたことから、次に実際の矯正用ワ イヤーとして必要な曲げ試験について検討することとし た.

2. 曲げ試験

フルパンドシステムにより歯科矯正治療を行う時, アーチワイヤーの弾力は個々の歯に複雑に作用して歯の 移動が行われる。しかし複雑な力も局所的に考えると, 3点曲げ試験に置き換えることができると考えられる。 矯正用ワイヤーの曲げに対する性質を試験するとき,な るべく臨床に近づけた結果を得るために,特に小臼歯部 分を想定して試験機を考案した。今回の曲げ試験では, 3 点曲げ試験を行い、ワイヤー中央部に変位を与えその 時の荷重を測定した.また最大変位を与えた後、引続き 変位を減少させてゆき、歯の移動に伴う荷重の減弱とい う変化も同時に記録し、一連のサイクルとして測定し た.

条件 (a) (single braket を使用して bracket での ワイヤーの固定をしない.)

single bracket はその支持面がワイヤーと平行して一 致するため、ワイヤーの中央部分に変位を与える時、支 点間の距離に変化を生じないという特徴がある. このた め、この条件では支点となる bracket でのワイヤーの固 定は行わず基本的な3点曲げ試験により、ワイヤーの性 質の比較を行った.

超弾性型および加工硬化型 NiTi は曲げ試験での永久 変形が非常に少なかった.ステンレススチール および Co-Cr 基合金の場合は、3 点曲げにより変位を与えてい くと、ある荷重まで増加した後に塑性変形を生じるた め、荷重が増加しなくなるところがある。このため変位 を 2mm より減少させると、塑性変形したぶんだけ永 久変形として残留することになり、直線的に荷重が減少 する.しかも変位量 2mm のときの荷重は かなり高い 値を示し、このままの弾力を矯正力として使用するには 高過ぎる値と考えられる¹³⁾.

超弾性型 NiTi の場合,変位を増加しても荷重の増加 は除々に少なくなるという Co-Cr 基合金と似た 曲線を 示した(図7). しかし変位を減少してゆくと, Co-Cr 基合金のように直線的な荷重の減少は示さず,変位を増 した時と同じ曲線を示した.荷重の増加が徐々に低下し 平担な曲線となるのは、塑性変形が生じているのではな く,応力誘起変態による超弾性の発現のためである. 一 方,加工硬化型 NiTi はほぼ直線的に荷重が上昇し,ま た変位を減少していくと,直線的に荷重が減少している (図 8).

条件 (b) (single bracket を使用して リングレット によりワイヤーを固定した.)

アーチワイヤーとして 実際に 口腔内で 使用するとき は、ワイヤーをブラケットに固定して用いる、つまりワ イヤーの 弾力を歯に 伝えるためには、 ブラケットでの 固定が必要となる. この固定のためには金属製の細い線 (結紮用線)を用いることが多いが、結紮の強さによりブ ラケット部分のワイヤーに加わる力が異なると、ブラケ ットでのワイヤーの滑り等に差が生じ、結果に差が出る 原因となる.このため、固定にはリングレットを使用し、 条件を一定に揃え、しかも実際の使用状態に近い条件で 比較検討した.

超弾性型 NiTi は変位を加えてゆくと約 0.5 mm か ら荷重の増加率が低下するが、その後は一定の増加率で

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前重が増し、加工デー型 NiTi とやや似た曲線を示し た.変位を減少し、 もやや似た曲線であったが、超弾 性型 NiTi は変位量が 1mm 前後において 超弾性によ り発現したものと考えられる平担な部分があらわれた. この超弾性が発現したと考えられる部分では、変位量が 変っても荷重の変化が少ないため、実際に歳の移動を行 うとき、歯が移動してもワイヤーの矯正力が減弱しない ことが考えられる、矯正力として望ましい力は持続する 力である²¹. 超弾性型 NiTi を利用することにより、今 までの アーチワイヤーでは 得られなかった 歳の 移動に 伴っても減弱の少ない矯正力が得られると考えられる.

支点を固定することにより、どのワイヤーの試験結果 も条件(a)より荷重が著しく上昇している.特にステン レススチールAは 2mm 変位させる前に 1000g に 達 してしまった.また Co-Cr 基合金も同様に高い荷重を 示した、あまり強い力は逆にスムーズな歯の移動を阻害 すると言われている¹⁴⁾.このためアーチワイヤーの屈曲 を行い、各種のループ等により緩和で持続性のある矯正 力を得られるように細工が行われている^{11,13)}.超弾性 型 NiTi ワイヤーの場合は、2mm の変位を与えたときの 荷重は Co-Cr 基合金の半分以下で、また永久変形も非 常に少ない、しかも超弾性による減弱の少ない力を示す ので、あらかじめ理想的な歯列の形態にワイヤーを作っ ておくことにより、スムーズな歯の移動が達成されるの ではないかと考えられる.

条件 (c) (siamese bracket を使用して bracket での固定をしない。)

この条件では 支持の ツメが 4本に 分かれた siamese bracket を使用した。このブラケットはツメが 曲面上に あるために、ワイヤーの曲げを行うに従い、ごくわずか に支点間の距離が縮小する特徴がある。

超弾性型 NiTi は変位を減少してゆくと、やはり超弾 性による平担な部分を持つ曲線を示した。一方加工硬化 型 NiTi は直線的な変化を示し、荷重の変化の大きな曲 線を示した。

条件(d)(siamese bracket を使用して, bracket 部で リングレットによりワイヤーを固定した.)

この条件では今までワイヤーの変位を 2 mm までで 終了していたものを, 超弾性型は 3 mm までの 変位に した. 試験結果は条件 (b) と同様に超弾性型 NiTi と加 工硬化型 NiTi はやや似た曲線を示した. これは, ワイ ヤーを固定することにより, ブラケットにおける ワイ ヤーの 滑りが 制限されたことなどが考えられる. しか し, 変位を減少する時, 変位 2 mm から 1 mm までの 荷重の減少を比較すると, 加工硬化型 NiTi が 220g で あるのに対して 超弾性型 NiTi は 100g と少ないこと がわかる. 固定を行うことにより超弾性の現われる部分

が明確ではない 超弾性型 NiTi が持続的な力を発揮 することは図 22 により明らかである。

以上の結果から超弾性型 NiTi の歯科矯正用ワイヤー としての適性を考えると、次のようなことがいえる. 超 弾性型 NiTi は従来の矯正ワイヤーにはない. 広い変位 量にわたって一定の荷重が得られるという超弾性と、大 きな変形を与えても永久変形が非常に少ないという特徴 をもっている. これらを利用してアーチワイヤーとして 使用すると、矯正力としては理想的ということができる 持続的な力が得られることが考えられる.

#### V. 結 論

新しく開発された超弾性型 NiTi 合金ワイヤーについ て引張および曲げ試験を行ない、歯科矯正用ワイヤーと しての適性について検討した。比較のために現在歯科矯 正用として用いられているステンレススチール、Co-Cr 基合金、および加工硬化型 NiTi ワイヤーについても合 わせて検討を試みた。との研究で得られた主な結果はお よそ次のとうりである。

1) 引張試験において、超弾性型 NiTi ワイヤーは一 般の金属材料とは かなり 異なった応力-ひずみ曲線を示 した. ひずみ 2% から 5% の間では、ひずみ量は増加す るが応力はほぼ一定となり、この範囲においてはひずみ を減少させていった場合でもやや低い応力で一定値を示 すという特殊な変形挙動を示した.

2) 超弾性型 NiTi の破断までの伸びは約 11% であ るが、伸び 8% までの繰返し引張試験の後でも永久変形 は 0.5% とわずかであった、これらの特殊な変形挙動は 応力誘起変態に基づくものである。

3) 曲げ試験において、 超弾性型 NiTi ワイヤー は Co-Cr 基合金や ステンレス スチールワイヤーと 比較し て、大きな変位量を与えても荷重が少なく、また永久変 形が非常に少ないという特性を示した.

4) 超弾性 NiTi ワイヤーは曲げ試験において、変位 量が 減少しても 荷重の変化は 少なく。 広い変位量にわ たって一定の荷重が得られるという特性を示した.

以上より超弾性 NiTi は変化の少ない持続的な力を発 揮することができ、しかも永久変形が少ないという特徴 をもつことから、矯正用ワイヤーとして非常に優れた性 質をもつものと考えられる。

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同種移植による抗腫瘍抵抗性 同種移植による抗腫瘍抵抗性 になったのはよほどに古い時期のことと思われる。一時期流行 をみた白血球輪血もある種の同種移植であろうが、抗種馬抵抗 住の獲得に至るのに必要な条件および抗種傷性の成立限序その ものがまったく不明であったため、現在ではあまり行なわれな くなっているようである。

immunological xcnogenization として追及を 始めた、マウスに比べてラットでは免疫遺伝学 そのものが決成熱なこともあって、必ずしも十 分な解析がなされたとはいえない現状である。 小林ちによれば、同類特徴による抗腫瘍抵抗性

TINI 合金は強度や耐化性にすぐれ、熱弾性 型マルテンサイト変態をし、記憶処理を行たえ デ ば「形状記憶効果 (shape memory efficet)」 が得られる、また、変態温度 (Af 点) より少 し上の温度で変形させても、応力を除けばゴム

のように元の形状にもどる「級州性的性質」がある。特に、応 力誘起マルテンサイト辺礁とその逆辺礁に起因するものを「超 弾性(superclasticity)」とよぶ、超弾性も形状記憶効果も、 ともに逆変態が形状回復の原動力になっていて、そのどちらが 起きるかは変形温度と Af 点との相対関係で定まる。二つの 現象にとってもっとも基本的なことは、逆辺礁が結晶等的にま ったく可逆的に行なわれているということである。すなわち、 マルテンサイトが母和の方位を覚えているのである。形状記憶 合金と並んで、超弾性 TiNi 合金を 整形外科領域に 応用すれ ば、従来の金属の話欠点を改善しうる。

超弾性 TiNi 線は 不妨角線よりもたわみやすく。 除荷すれ ば完全に元の形状にもどるために,従来の捻り法による結合は 不可能である。したがって、TiNi 線を 同種金属パイプに 通 し、パイプをかしめる「かしめ法」を応用した。「かしめ法」 を従来の不防傾線に応用しても従来の捻り法の2倍以上の都結 力が得られ。すぐれた締結法である。 海結操作として、まず、 金属パイプを通した TiNi 線を 馬蹄形種情器具を 用いて強い 限力でもって称めつけた後、我々が試作した先端をV 狸にした 「かしめ器」で金属パイプと TiNi 線を圧迫変形させ、TiNi 線がパイプ内ですべらないように圧着する。

超弾性 TiNi 線のひっ張り 試験結果では、 曲線の形は温度 に著しく依存しているが、この合金のマルテンサイト変態の特 性温度 (Ms, Mf, As, Af) と密接に関係している。Ms より の成立には反応のメモリーがなく、特異性もみられないことか ち、NK 細胞によって抗動防抵抗性が発現する可能性を示較 している。ウイルスおよびハブテンによる xenogenization の 試入もみられるようである。

私たちはマウスの系において阿種体植の組み合わせをかた。 どのような 11-2 抗原の組み合わせで体積を行なうと強い抗種 将抵抗性が 誘導されるかを決定した。 11-2 抗原をコードする 知伝子領域内の運復域がそれぞれ異なる C 57 10/10系 comgenic マウスを体植の donor として用いて、 どの運復域が異 なることが必要なのかをみると、K、A、B および J 運復域が異 いて、どの運復域がそれ でいて、どの運復域がよ

一同種件値による抗腫與低抗性がヒトの悪性種間に対して試み ちれる目がくるかもしれないが、現在はなおいっそうの状態実

> 陸を必要としている負傷であると思われる。ヒ トへの応用は社会的および法的掲述を含んでい るようである。

> > (群馬・千木良正機)

上の温度でひっ張ったとき、弾性変形に伴う直 控領域に続いて陡快が起こり、みかけのうえで は塑性変形している。しかし、この変形は応力 誘起マルテンサイト資源によるもので、このマ ルテンサイトは AF より上では応力セッの状態

で熱力学的に不安定であるから、除荷すれば進度無が起こって 肉相にもどる。この合金はその進度無が結晶学的に可逆的だか ち、肉相の元の方位にもどって塑性ひずみが完全に消失する。 しがたって、手術時に増殖性 TNN 線を強く締結すれば、術 後、骨接合部の骨敷収により接合部がわずかに弛んでも、応力 誘起マルテンサイト変態により操が弛むことがない。

従来の AO プレートによる 骨接合を行なえば、皮質骨の海 協骨化の結果、プレート除去後に 再骨折の 危険性があり、ま た、プレートの折損など AO プレートの 問題点は周知のとお りである。この AO プレート材に超弾性 TiNi 合金を加いれ ば、生物学的になじみがあるばかりでなく、TiNi 合金に従来 の金属に比較して、骨の弾性に近づき、生体力学的にもなじみ があり、また外力によりプレートが変形しても、外力が除かれ ると完全に元の形状にもどり、皮質量の海陽骨化やプレートの 折損が起こりにくくなるであるう、また、人工問題に応用すれ ば、特に替モメントを使用しない場合、ステムの弛みや、折損 が減少するであるう、

形状記憶効果と超弾性を同時に応用すれば、さらに応用範囲 が拡大され、要多と合金である。現在、大阪府立工業技術研究 所および古河電気工業(性)の協力を得て、応用開発をすずめて いる。
(大阪・大西啓靖)

超弾性的性質のある TiNi 合金



チュトピック

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# <経験と考察>

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Yoshimasu Donishi, Clinical Ma zine: Orthopaedic Surg y, 32 at p. 1180 (1987)

# 骨 Paget 病におけるウナギカルシトニンの治療効果

Effects of synthetic cel calcitonin in patients with Paget's disease of bone

# 武藤芳照 岩田 久 富田明夫 杉浦 邕 梅沢健司[•]

# はじめに

骨 Paget 病は 1877 年 Sir James Paget¹⁰ により詳 細に報告された骨弦患である。本症の本態は骨組織の remodeling の過程。すなわち骨形成と骨吸取が病的に 亢進した状態であるとされているが、その病因について はいまだ定識が得られていない。そのために治療も対症 的なものにならざるをえず、種々な薬剤の投与が試みら **療を行ない。その効果について検討したので報告する。** 

例

# 症 1. 臨床像

症例は表1に示す5症例である。男1例、女4例で、 平均年令は58.6才である。主訴は頭痛、腰痛、股関節 痛などであるが、顔貌の落しい変化や頭部の変形を他人 に指摘されたり、他科にて骨N線像の異常陰影を発見さ



n. 内外版の肥厚(症例3)。

b. Cotton wool appearance (症例4).

図 1. 頭蓋骨X線像。

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れてきた。1967年カルシトニン(以下 CT と略す)が水 症に有効であると報告されて以来¹⁾、水症の CT 治療に 別する報告があい次いでなされるようになった。 筆者 Paget 招 5 症例に対して合成ウナギ CT による治

れて当科へ紹介された例もある.

診断は、血液・尿生化学検査にて血清 alk. p-asc 値 の上昇と尿中 OH-proline 量の増加、骨X線像上異状、 限局性の骨硬化像と骨梁の 祖大、不整化(症例2,3, 5)や頭蓋骨の内外板の肥厚(症例1,3), cotton wool appearance (症例4), osteoporosis circumscripta (症 例5)などの特布な所見(図1)、骨生検組織像で既存 骨梁の破壊吸取後の新旧骨質の接合によるモザイク構造 (図2)、^{60m}Tc 骨 scintigram での 異常集積像(図3) などにより行なった。

c. Osteoporosis circumscripta (症 例 5 ).

Key words: Paget's disease of bone, calcitonin

Y. Mutoh (現:東京大学教育学部体育学性規教育学科助教
 ⁵. H. Iwata: 名音斯大学教形外科 (Dept. of Orthop. g., Nagoya University School of Medicine, Nagoya):
 A. Tomita (新術): 阿大学第一州科:S. Sugiura (部長).
 K. Unizawa (部長): 前立网站病院警形外科.

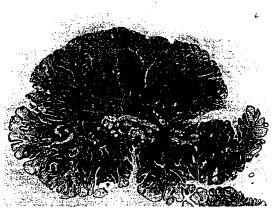


Figure 3. Case 1. Photomicrograph of the polypectomy specimen. Note the branching core of smooth muscle fibers in a treelike pattern in the lamina propria (H&E, original magnification ×17).

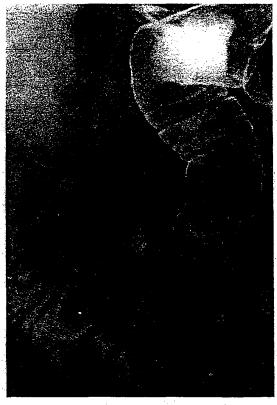


Figure 4. Case 2. Radiographic appearance of a multilobulated polyp in the second to third portion of the duodenum; the head measures  $25 \times 20$  mm in size, and the stalk measures 53 mm in length.

mucocutaneous  $\mu$ -gmentation and family history, and suggested that these were incomplete forms of the syndrome. A recent report⁴ gave some support to their concept by the finding of a solitary hamartoma in a resected specimen of ileum.

Paterlini et al.⁵ in 1983 succeeded in performing endoscopic polypectomy of multiple jejunal polyps in a patient with Peutz-Jeghers syndrome who had undergone surgical segmental resection of the jejunum. In the literature, there have been only two cases of hamartoma detected endoscopically in the distal duodenum.^{2,6} One polyp was removed by endoscopic polypectomy.²

Because of the difficulty of inserting the conventional duodenofiberscopes into the distal duodenum and upper jejunum, endoscopic excision of polypoid lesions in these areas has rarely been reported. We recently described the advantages of jejunal endoscopy with a long duodenofiberscope,³ which was successfully used for the excision of these solitary hamartomas in the distal duodenum.

1

Hiroaki Tanaka, MD Mitsuo lida, MD Norio Kohrogi, MD Toshiyuki Matsui, MD Youich Yasunami, MD Takashi Yao, MD Kenjirou Nakamura, MD Masatoshi Fujishima, MD Departments of Internal Medicine II, Surgery I, and Pathology II Faculty of Medicine Kyushu University Fukuoka, Japan

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- Iida M, Yamamoto T, Yao T, Fuchigami T, Fujishima M. Jejunal endoscopy using a long duodenofiberscope. Gastrointest Endosc 1986;32:233-6.
- 4. Nakamura T. Polyps of the small intestine: pathology and practice. Saishin-Igaku 1981;36:68-79 (in Japanese).
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- Giardiello FM, Welsh SB, Hamilton SR, et al. Increased risk of cancer in the Peutz-Jeghers syndrome. N Engl J. Med 1987;316:1511-4.

#### Variable stiffening device for colonoscopy

#### To the Editor:

It is apparent to most endoscopists who perform colonoscopy on a regular basis that it is sometimes difficult to perform colonoscopy where there is formation of loops in the sigmoid, transverse colon, or even the descending colon. The more redundant the colon and the less stiff the scope, the more frequent this problem would be. Some endoscopists have already used various devices such as biopsy forceps to

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GASTROINTESTINAL ENDOSCOPY

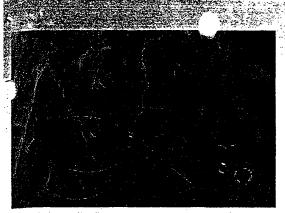


Figure 1. Prototype variable stiffening device SVSC-1 (Wilson-Cook Medical, Inc.) for colonoscopy.

stiffen the colonoscope. As the colonoscope becomes older with more frequent uses and more torqueing, the insertion tube becomes more flexible; it then becomes more difficult to perform colonoscopy since the critical stiffness is lost.

Various ways to introduce stiffening into the scope, including external stiffeners such as splints, have not received widespread acceptance. Patients with redundant sigmoid colons, redundant colons in general, long colons, and large dilated atonic colons, and those with megacolon are especially difficult to treat. To provide for a stiffening device, it was thought best that the stiffener might have the additional luxury of being variably stiff. This would allow different degrees of stiffness for different settings and different enploscopists.

For the past 2 years, we have been experimenting with different types of cables to be used through the biopsy forceps of a standard colonoscope. A prototype stiffening device, SVSC-1 (Wilson-Cook Medical Inc., Winston-Salem, N.C.) through an Olympus CF10L colonoscope was used in 25 patients (Fig. 1). The patients were selected when routine colonoscopy was not easily performed.

The SVSC-1 was successful in providing colonoscopy to the cecum in 22 of 25 patients (88%). It was thought that the stiffening device was helpful in proceeding with more rapid colonoscopy where the procedure might have been prolonged without the use of the stiffener.

Although we do not have a control group, it was thought subjectively that the stiffener provided a great improvement in forward motion in cases where the colonoscope was thought to be too limp or the colon itself was thought to be too redundant.

This type of variable stiffener can be expanded to other types of endoscopy, including in the very flexible upper endoscopes that require further stiffening to intubate the pylorus and even in intubation of other structures that are routinely performed by the gastrointestinal endoscopist (e.g. choledochoscopy, stent placement, etc.). It is thought that a variable stiffening device will provide many more applications in the future in all fields of endoscopy.

Michael J. Sullivan, MD East Tennessee University School of Medicine Kingsport, Tennessee

VOLUME 36, NO. 6, 1990

# Pyelo-choledochal ti ula secondary to pancreatic carcinoma

#### To the Editor:

The presence of spontaneous fistulization between the biliary system and the gastrointestinal tract is an uncommon complication of neoplasms. Fistulizations are exceptional when one of the organs is not digestive.

We encountered a 55-year-old man, who was a heavy smoker and moderate drinker. He was admitted to the hospital with a 1-month history of upper abdominal pain, progressive jaundice, dark urine, and pale stools.

Abdominal sonography displayed dilation of the intrahepatic and common bile ducts and gallbladder. In addition, CT demonstrated a mass in the head of the pancreas and dilation of the main pancreas duct, and the common bile duct was sharply interrupted at the level of the pancreatic mass.

ERCP was performed and showed a long, irregular stenosis in the intrapancreatic portion of the common bile duct and a small fistula between the bile duct and the right renal pelvis. The ureter was filled with the contrast that was introduced into the common bile duct. There was a significant dilation of the common hepatic and intrahepatic bile ducts (Fig. 1). A subsequent abdominal radiograph showed the bladder to be filled with contrast. Urinalysis confirmed the presence of bile.

The patient underwent surgical exploration that disclosed unresectable pancreatic carcinoma. A choledochojejunostomy was performed. The patient died 1 month later.



Figure 1. ERCP showing fistula between the common bile duct and the right renal pelvis (arrow).

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	In vo the Apolication of	、	SN 956653	
	In re the Application of:	)	Group Art Unit: 331	
	James E. Jervis	) ) }	Examiner: C. Sam	
	Serial No. 177,817	) )	Raychem Corporation 300 Constitution Drive	
÷	Filed: March 30, 1988	) )	Menlo Park, CA 94025	
	For: Medical Devices	)		
	Incorporating SIM Alloy	í		
	Elements	Ś	August 11 1000	
	ETEMETICS	1	August 11, 1988	

## RESPONSE TO OFFICE ACTION

Hon. Commissioner of Patents and Trademarks Washington, D.C. 20231

Sir:

This is a response to the office action dated August 5, 1988. Reexamination is requested in view of the following amendments and remarks.

## REQUEST FOR WITHDRAWAL OF FINALITY OF REJECTION

The present application was filed as a continuation application claiming priority under 35 USC \$120 from US patent applications nos. 06/541852 and 07/047824. As filed, the application contained claims identical to those contained in application serial no. 07/047824 and rejected in an office action in that application dated September 30, 1987. It was the intention to amend the present application, by submitting new claims in a preliminary amendment.

I hereby certify that this correspondence is being deposited with the United States Postal Service as first class mail in an envelope addressed to Commissioner of Patents and Trademarks, Washington, D.C. 20231, on sterad top Shilan Santi ...

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In order to provide the Examiner with all the information that the might require in order to consider the issues presented in the amendment, it was thought prudent to submit a declaration under 37 CFR §1.132 by a technical expert. This was completed and signed at the beginning of August.

- 2 -

It is sincerely regretted that compilation of the papers to be submitted has meant that the preliminary amendment has not been filed before receipt of the first office action.

In accordance with MPEP 706.07 (e), it is requested that the finality of the rejection of this application be withdrawn. The general policy of the Patent Office of rejecting a continuation application in the first office action in the situation outlined in MPEP 706.07 (b) is recognized. However, in the present case, the delay in filing a preliminary amendment arose from efforts to make more clear the issues presented in the amendment.

Furthermore, the rationale behind the policy of final rejecting applications is to further the interest of the public that prosecution of an application be confined to as few actions as is consistent with a thorough consideration of its merits (see MPEP 706.07). It is respectfully submitted that the examination, to which the present application was subjected for the purposes of the outstanding office action, did not involve the consideration of merits not considered previously in connection with the application from which priority is claimed. The merits are now presented for consideration, unfortunately and regretably after receipt of the first office action. In view of the bona fide reasons for the delay in filing the amendment and of the nature of

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the examination to which the application has been subjected, it is respectfully requested that the finality of the rejection be withdrawn and that the amendments now submitted be entered and considered. While refusal to enter the amendments would be contrary to the interest of the applicants, it is believed that it would also be contrary to the interest of the public since the required further continuation application would prolong prosecution of this case yet further. It is believed that this should be avoided, and that, in the present circumstances, to do so by withdrawing the finality of the rejection is consistent with the provisions of the MPEP and the Code of Federal Regulations.

- 3 -

#### AMENDMENTS

The Examiner is requested to amend the specification as follows:

#### In the Description:

Page 2, before the heading "Background of the Invention", insert:

# -- CROSS-REFERENCE TO RELATED APPLICATIONS

This application is a continuation of copending commonly assigned application serial no. 047,824, filed May 8, 1987, which is a continuation of application serial no. 865,703, filed May 21, 1986 now US Patent No. 4665906, which is a continuation of application serial no. 541852 filed October 14, 1983 now abandoned. --

- 4 -

Page 3, line 28, after Quin insert --now U.S. Patent No. 4,505,767--.

Page 4, line 26, delete "power" and insert --powder--.
H Page 7, line 19, delete "E_D" and insert --E_A--.
A S Page 7, line 19, delete "strain" and insert --stress--.
A D Page 8, line 20, delete "theart" and insert --the art--.
D Page 8, line 21, delete "tsting" and insert --testing--.
A B Page 8, line 24, after "(Docket No. MPO873-US1)" insert !
A Page 11, line 17, delete "by" and insert --be--.
A D Page 13, line 9 delete "it".

All Page 14, line 12, delete "whch" and insert --which--. All Page 15, line 17, delete "transition" and insert -transition --.

AB Page 8, between lines 24 and 25, insert the following paragraph:

--The following table sets forth transformation temperature data for alloys disclosed in US-4505767:

......

# - 5 -

# TABLE

Composition (atomic percent)

) .

Ni	Ti	<u>v</u>	Ms	A(90)
49.50	43.50	7.00	-107	-88
50.00	44.00	6.00	-96	-84
49.00	43.00	8.00	-83	-61
50.00	45.00	5.00	-42	-33
49.00	45.00	6.00	-35	-12
50.50	48.00	1.50	-32 -	-6
48.50	44.50	7.00	-30	-13
50.00	46.00	4.00	-11	7
48.50	45.00	6.50	-10	15
49.00	45.50	5.50	-10	14
48.00	44.25	7.75	-7	8
48.50	45.50	6.00	-5	27
41.50	38.50	20.00	-2	86
46.50	43.50	10.00	-1	50
36.25	33.75	30.00	0	42
49.50	46.00	4.50	6	35
48.00	46.00	6.00	12	36
47.75	45.75	6.50	20	54
47.50	45.50	7.00	26	58
48.50	46.50	5.00	27	58
45.00	45.00	10.00	30	71
47.50	46.50	6.00	32	71
46.50	46.50	7.00	- 34	70

# In the Claims:

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Cancel claims 1 to 10.

Add new claims 11 to 28 as follows:

-- 11. A medical device for use within a mammalian body, or in such proximity to a mammalian body that the device is substantially at body temperature, the device comprising an element which comprises a shape memory alloy which:

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(a) displays stress induced martensite behavior at body temperature; and

(b) has an A(90) temperature of not more than 0°C.

- 6 -

12. A device as claimed in claim 11, which includes a restraint by means of which the shape memory alloy element is held in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress induced martensite.

13. A device as claimed in claim 12, in which the restraint is hollow, and the shape memory alloy element is deformed in such a way that it is compressed transversely, and is positioned within the restraint, the restraint pre-venting transverse expansion of the element.

14. A device as claimed in claim 13, in which the restraint is a catheter.

15. A device as claimed in claim 13, in which the shape memory alloy element is an intrauterine contraceptive device.

16. A device as claimed in claim 14, in which the shape memory alloy element is a filter for a blood vessel.

17. A device as claimed in claim 12, in which the shape memory alloy element is tubular, and the restraint is positioned within the shape memory alloy element to deform it.

18. A device as claimed in claim 17, in which the shape memory alloy element is a tracheal catheter.

- 7 -

19. A medical device for use within a mammalian body or in such proximity to a mammalian body that the device is substantially at body temperature, the device comprising an element which comprises a shape memory alloy consisting essentially of nickel, titanium and vanadium within an area defined on a nickel, titanium, and vanadium ternary composition diagram by a hexagon with its first vertex at 38.0 atomic percent nickel, 37.0 atomic percent titanium, and 25.0 atomic percent vanadium; its second vertex at 47.6 atomic percent nickel, 46.4 atomic percent titanium, and 6.0 atomic percent vanadium; its third vertex at 49.0 atomic percent nickel, 46.4 atomic percent titanium, and 4.6 atomic percent vanadium; its fourth vertex at 49.8 atomic percent nickel, 45.6 atomic percent titanium, and 4.6 atomic percent vanadium; its fifth vertex at 49.8 atomic percent nickel, 44.0 atomic percent titanium, and 6.2 atomic percent vanadium; and its sixth vertex at 39.8 atomic percent nickel, 35.2 atomic percent titanium, and 25.0 atomic percent vanadium.

20. A medical device which comprises:

(a) an element for use within a mammalian body or in such proximity to a mammalian body that the device is substantially at body temperature, the element comprising a shape memory alloy which displays stress induced martensite behavior at body temperature; and

(b) a restraint by means of which the shape memory alloy element is held in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress-induced martensite.

- 8 -

21. A device as claimed in claim 20, in which the restraint is hollow, and the shape memory alloy element is deformed in such a way that it is compressed transversely, and is positioned within the restraint, the restraint preventing transverse expansion of the element.

22. A device as claimed in claim 21, in which the restraint is a catheter.

23. A device as claimed in claim 21, in which the shape memory alloy element is an intrauterine contraceptive device.

24. A device as claimed in claim 22, in which the shape memory alloy element is a filter for a blood vessel.

25. A device as claimed in claim 20, in which the shape memory alloy element is tubular, and the restraint is positioned within the shape memory alloy element to deform it.

26. A method of medical treatment which comprises:

(a) providing a device comprising an element which comprises a shape memory alloy which displays stress induced martensite behavior at body temperature, the element being restrained in a

deformed configuration against a stress which induces the formation of stress induced martensite;

(b) positioning the device so that the shape memory alloy element is within a mammalian body or in such proximity to a mammalian body that the element is substantially at body temperature; and

- 9 -

 (c) allowing the element to transform from the deformed configuration, the transformation occurring substantially at body temperature.

27. A method as claimed in claim 26, in which the shape memory alloy element is held in the deformed configuration by a restraint, and the method includes the step of removing the restraint to allow the element to transform from the deformed configuration.

28. A method as claimed in claim 26, in which transformation of the shape memory alloy element causes one or more parts of the body in contact with the element to be displaced.--

#### REMARKS

## Amendments to the description:

The description has been amended by incorporating a cross-reference to related applications and to correct typographical errors.

Information concerning alloys disclosed in US-4509767 has been incorporated in the description on page 8. The

disclosure of US-4505767 is incorporated in the present specification by the reference thereto on page 3 at line 28. Specifically, several alloys are disclosed in Table I in column 6 of the incorporated document, each alloy having characteristic thermomechanical properties. In particular, the Ms temperature (which is the temperature at which transition of the alloy from martensitic phase to austenitic phase starts) is specified for each alloy. Each alloy inherently also has a characteristic A(90) temperature (which is the temperature at which the transformation from martensitic phase to austenitic phase is 90% complete). It is appropriate to define alloys which are preferred for use in the device of the present invention in terms of their A(90) temperature since, for the alloy to be capable of being deformed by the formation of martensite under stress, it is necessary for the alloy initially to be at least partially, preferably completely, in the austenitic phase. Like the  $M_S$  temperature, the A(90) temperature is an inherent characteristic of the alloys disclosed in US-4505767, so that the disclosure therein of alloys having the compositions set forth in Table I represents also the disclosure of alloys having the A(90) values set forth in the table to be included on page 8 of the description.

- 10 -

The table incorporated on page 8 sets out  $M_S$  and A(90) data for the alloys disclosed in US-4505767. The data is the subject of a 37 CFR §1.132 declaration submitted herewith. It is believed that amendment of the description by inclusion of the data set out in the table does not involve the addition of subject matter.

Amendments to the claims:

# IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re the Application of:	) Group Art Unit: 331
James E. Jervis	) Examiner: C. Sam
Serial No. 177,817	) Raychem Corporation ) 300 Constitution Drive
Filed: March 30, 1988	) Menlo Park, CA 94025
For: Medical Devices	) )
Incorporating SIM Alloy Elements	) ) August 1, 1988

# DECLARATION UNDER 37 CFR §1.132

Hon. Commissioner of Patents and Trademarks Washington, D.C. 20231

Sir:

I, Thomas W. Duerig of 41618 Mission Creek Drive, Fremont, CA 94539 hereby declare as follows:

 I hold a BS Degree in physics from Lehigh University, and ME and Ph.D Degrees in metallurgy from Carnegie-Mellon University. I have worked in the field of shape memory alloys for eight years.

2. For the last five years, I have been employed by Raychem Corporation of 300 Constitution Drive, Menlo Park, California 94025-1164 in its metals division to develop inter alia shape memory alloys and devices employing such alloys.

3. US Patent No. 4505767 relates to shape memory alloys which consist essentially of nickel, titanium and vanadium

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and which were developed by Mary P. Quin who, at the time, was also employed by Raychem Corporation in the metals division.

4. The compositions, and aspects of the thermomechanical behavior, of certain Ni-Ti-V alloys are disclosed in Table I in column 6 of US-4505767. The disclosed alloys have characteristic martensite-austenite transformation temperatures of which the  $M_S$  temperatures are set forth in the table.

5. The table which forms part of this declaration sets forth  $M_S$  and A(90) temperatures for alloys disclosed in Table I of US-4505767. The A(90) temperature is the temperature at which the transformation from the martensite phase to the austenite phase is 90% complete. Like the  $M_S$ temperature, the A(90) temperature is an inherent characteristic of the alloys disclosed in US-4505767, so that the disclosure therein of alloys having the compositions set forth in Table I represents also the disclosure of alloys having the A(90) values set forth in the table below.

5. The data included in the table below have been compiled by me from technical records of the metals division of Raychem Corporation, which were compiled during the development of the alloys which form the subject of US-4505767.

I declare that all statements made herein of my own knowledge are true, and that all statements made on information and belief are believed to be true, and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment or both, under the provisions of 18 U.S.C. 1001 and that such willful false statements may jeopardize the validity of any patent issuing upon the application.

FURTHER DECLARANT SAYETH NOT.

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Thomas

Declared at Menlo Park, California on <u>Aug 9</u>, 1988.

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# TABLE

Composition (atomic percent)

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<u>Ni</u>	Ti	<u>v</u>	Ms	<u>A(90)</u>
49.50	43.50	7.00	-107	-88
50.00	44.00	6.00	-96	-84
49.00	43.00	8.00	-83	-61
50.00	45.00	5.00	-42	-33
49.00	45.00	6.00	-35 -	-12
50.50	48.00	1.50	-32	-6
48.50	44.50	7.00	-30	-13
50.00	46.00	4.00	-11	7
48.50	45.00	6.50	-10	15
49.00	45.50	5.50	-10	14
48.00	44.25	7.75	· -7	8
48.50	45.50	6.00	-5	27
41.50	38.50	20.00	-2	86
46.50	43.50	10.00	-1	50
36.25	33.75	30.00	0	42
49.50	46.00	4.50	6	35
48.00	46.00	6.00	12	36
47.75	45.75	6.50	20	54
47.50	45.50	7.00	26	58
48.50	46.50	5.00	27	58
45.00	45.00	10.00	30	71
47.50	46.50	6.00	32	71
46.50	46.50	7.00	34	70



Examiner: D. Kenea

Group No. 3301

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of:

James E. Jervis

Serial No.: 07/956,653

Filed: October 2, 1992

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

Commissioner of Patents and Trademarks Washington, D.C. 20231

#### STATUS INQUIRY

- 1. More than 3 months have passed since the filing of a response on 11/24/1993. No further communication has been received from the Patent and Trademark Office.
- 2. Kindly advise the undersigned of the present status of this application, by checking the appropriate box on the next page. A stamped return-addressed envelope is provided.

Respectfully submitted,

SHELDON & MAK

1148 Date:

Jef/firte∦ G. Sheldon Reg. No. 27,953

225 South Lake Avenue 9th Floor Pasadena, California 91101 (818) 796-4000

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# STATUS INQUIRY REPLY

APPLICATION SERIAL NO. 07/956,653 IS CURRENTLY

ASSIGNED TO GROUP	AND AWAITS:
□ ACTION BY THE EXA	MINER

APPLICANT'S RESPONSE TO THE OFFICE ACTION
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	SERIAL NUMBER   FILING BATE 92	JERV IFURST NAMED INVENTOR	J ATTOTNEY DOCKET NO.	
	· · ·	F3M1/0307	KENEALY, D	: ٦
	JEFFREY G. SHELDON		EXAMINER	
	SHELDON & MAK 225 S. LAKE AVENUE - PASADENA, CA 91101	9TH FLOOR	ABEANIT	]
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		· · · ·	DATE MAILED:	:
This CON	is a communication from the examiner in charge MISSIONER OF PATENTS AND TRADEMARKS	of your application.		
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			<b>—</b>	•
	••	esponsive to communication filed on		
	ened statutory period for response to this a to respond within the period for response w	ction is set to expire month(s), ill cause the application to become abandon	days from the date of this letter. ed. 35 U.S.C. 133	ту тичк •
Pärt I	THE FOLLOWING ATTACHMENT(S) AR	E PART OF THIS ACTION:	· · · ·	
1. [			e re Patent Drawing, PTO-948.	
3. L 5. [	Ξ		e of Informal Patent Application, Form PTO-152	· .
	SUMMARY OF ACTION			
	V Claims 1-57		in the second	
		16, 22, 23, 37-40, 43, 54	are pending in the applicatio	:
- ·		120123, 31- 10,43, 54	are withdrawn from consideration	<b>1.</b>
2. [			have been cancelled.	2 .
3. L		71/ 7, 11, 11/ 5-	are allowed.	
4. [	$\underline{Y}_{\text{Claims}} = \underbrace{H}_{\mathcal{L}} \underbrace{H} \underbrace{H} \underbrace{H} \underbrace{H} \underbrace{H} $	24-36, 41,42,44-5	3,55-57 are rejected.	
5. L	_] Çlaims	······································	are objected to.	*
	Claims		are subject to restriction or election requirement.	
- ≫d -7.8	This application has been filed with info	rmal drawings under 37 C.F.R. 1.85 which a	re acceptable for examination purposes.	
8.1	Formal drawings are required in respon			
9.		ve been received on (see explanation or Notice re Patent Drawin	Under 37 C.F.R. 1.84 these drawing g, PTO-948).	ŝ
	The proposed additional or substitute s	heet(s) of drawings, filed on	has (have) been 🖸 approved by the	
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11. L		has been 🖅 appr	oved; 🔲 disapproved (see explanation). xopy.has: 🔲 been received: 🗂 not been received.	•
12.[_	Acknowledgement is made of the claim been filed in parent application, seria		opymas 🛄 been received: 🛄 not been received	•
13. [		condition for allowance except for formal ma arte Quayle, 1935 C.D. 11; 453 O.G. 213.	tters, prosecution as to the merits is closed in	
14.	Other		•	·.
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Serial No. 956653 Art Unit 331

Applicant should note that preliminary amendment B filed 1/5/93 added a claim 54 (subsequently restricted). Applicant's additional claims 54-56 have been renumbered as claims 55-57. Applicant should refer to these claims as claims 55-57 in any later communications.

The following is a quotation of the first paragraph of 35 U.S.C. § 112:

The specification shall contain a written description of the invention, and of the manner and process of making and using it, in such full, clear, concise, and exact terms as to enable any person skilled in the art to which it pertains, or with which it is most nearly connected, to make and use the same and shall set forth the best mode contemplated by the inventor of carrying out his invention.

The specification is objected to under 35 U.S.C. § 112, first paragraph, as the specification, as originally filed, does not provide support for the invention as is now claimed. Applicant's declaration defining the A(90) temperature as the temperature at which 90% of the material has been transformed from martensite to austenite is not sufficient to now allow a claim for a specific value for the A(90) temperature being 0 degrees C. Nowhere in applicant's specification does he state the criticality of such a temperature for the A(90) temperature, nor does he ever state a preferred range in which the A(90) temperature should fall.

Claims 11-14,17 and 18 are rejected under 35 U.S.C. § 112, first paragraph, for the reasons set forth in the objection to the specification.

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Serial No. 956653 Art Unit 331

Claim 25 is rejected under 35 U.S.C. § 112, second paragraph, as being indefinite for failing to particularly point out and distinctly claim the subject matter which applicant regards as the invention. It cannot be discerned from the drawings or the specification how the SIM material inside a catheter can be deformed in a transverse dimension. If this claim is directed towards an IUD, it should have been restricted from this application. Examiner cannot find support for such a deformation of the catheter SIM invention.

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Claims 11-14,17-21,24-27,31-36,41,42,44-53 and 55-57 are rejected under the judicially created doctrine of obviousness-type double patenting as being unpatentable over claim 3 of U.S. Patent No. 4,665,906. Although the conflicting claims are not identical, they are not patentably distinct from each other because they both are directed towards a restraint in combination with an SIM medical device.

Claims 28-30 rejected under the judicially created doctrine of obviousness-type double patenting as being unpatentable over claim 3 of U.S. Patent No. 4665906 in view of Wilson. To have made the restraint out of SIM material and the catheter out of non-SIM material is well known in the art as shown by Wilson.

A patent may not be obtained though the invention is not identically disclosed or described as set forth in section 102

The following is a quotation of 35 U.S.C. § 103 which forms the basis for all obviousness rejections set forth in this Office action:

Serial No. 956653 Art Unit 331

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of this title, if the differences between the subject matter sought to be patented and the prior art are such that the subject matter as a whole would have been obvious at the time the invention was made to a person having ordinary skill in the art to which said subject matter pertains. Patentability shall not be negatived by the manner in which the invention was made.

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Subject matter developed by another person, which qualifies as prior art only under subsection (f) or (g) of section 102 of this title, shall not preclude patentability under this section where the subject matter and the claimed invention were, at the time the invention was made, owned by the same person or subject to an obligation of assignment to the same person.

Claims 19,20,26-28,41,42,45,47,48,49,50, and 57 are rejected under 35 U.S.C. § 103 as being unpatentable over the Wayman article titled, "Some Applications of Shape Memory Alloys", of applicant's disclosure (referred to as Wayman). Wayman discloses an SIM alloy that can be used in orthodontic dental wires. The wire is placed in a restraint (bands placed on the user's teeth) and exhibits SIM behavior at body temperature inside the mouth of the patient. This SIM behavior cause the wire to act more like a spring and retain pressure or tension on the treated teeth. When the wire is taken out of the mouth it is assumed that the wire will return to its austenitic state and straighten out of the stress induced martensite state. The Wayman disclosure of the orthodontic device does not specifically disclose the use of SIM material for the wire, but the disclosure does describe the use of SIM as natural property of SMA's in general. It is not believed to have been outside of the scope of the skilled artisan to have used a wire with an Af temperature at about body temperature such that this Serial No. 956653 Art Unit 331

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"super-elasticity" region can be taken advantage of in an orthodontic device as described later in the Wayman article.

Claims 21,29 rejected under 35 U.S.C. § 103 as being unpatentable over Wayman in view of Wilson. Wayman discloses the use of SIM alloys that exhibit SIM behavior at body temperature. Wilson teaches the use of SMA's in a catheter. It would have been obvious to have used the Wayman alloy in a catheter because if one had desired to have a catheter that could be bent and instantly retain its original unbent shape when the stress induced martensite transforms back to austenite, one could have looked to Wayman to see that such a material was available.

#### RESPONSE TO APPLICANT'S REMARKS

Applicant's remarks have been considered but are deemed moot in light of the new rejections set forth. This action has not been made final since the new rejection was not necessitated by applicant's amendments.

Any inquiry concerning this communication or earlier communications from the examiner should be directed to David Kenealy whose telephone number is (703) 308-2680.

David J. Kenealy March 7, 1994

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UNITED STATES DEPARTMENT OF COMMERCE Patent and Trademark Office

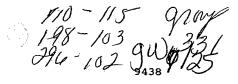
Address: COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231

SERIAL NUMBER FILING DATE	FIRST NAMED APPLICANT	AT	TORNEY DOCKET NO.
95453			
r	7	EXA	
		ART UNIT	PAPER NUMBER
			12
EV A MINED	INTERVIEW SUMMARY RECO	DATE MAILED:	
All participants (applicant, applicant's representative, PTO perso			
1) Mr. Sheldon			
2) Exr. Kenealy	(4)		
Date of interview6/30/94			
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Гуре: 🛛 Telephonic 🛛 Personal (сору is given to 🗌 арј	plicant 🔲 applicant's representative)		
Exhibit shown or demonstration conducted: 🛛 Yes 📴 No.	If yes, brief description:		
	<u></u>	· · · · · · · · · · · · · · · · · · ·	
Agreement DY was reached with respect to some or all of the c	laims in question. 🗌 was not reach	ed.	
Claims discussed:			
Identification of prior art discussed: Way Mc	an and Wilsor	<b>ר</b>	-0
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Description of the general nature of what was agreed to if an agr			
Examiner dances that the	e indusion of	a limitation	n reavining
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the restraining means to en	gage and stres	s the m	emory
Examiner agrees that the the restraining means to en alloy element at a t	temperature below	w body te	mperature.
will define over the	Wayman ortic	le ac an	led in the 3/7/94
	I	·	reject
A fuller description, if necessary, and a copy of the amendm ttached. Also, where no copy of the amendments which would			
Inless the paragraphs below have been checked to indicate to NOT WAIVED AND MUST INCLUDE THE SUBSTANCE OF	the contrary, A FORMAL WRITTEN THE INTERVIEW (e.g., items 1-7)	NRESPONSE TO THE	LAST OFFICE ACTION IS W
ast Office action has already been filed, then applicant is given		provide a statement of the	
It is not necessary for applicant to provide a separate reco		left with th	e examiner (except
Since the examiner's interview summary above (includir requirements that may be present in the last Office acti response requirements of the last Office action.	ng any attachments) reflects a comple on, and since the claims are now allow	te response to each of t	he objections, rejections and
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	Examiner	's Signature	<
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IN THE UNITED STATES PATENT	AND TRADEMARK OFFICE
In re Application of: )	Group Art Unit: 3301
JAMES E. JERVIS	Examiner: KENEALY, D.
Serial No.: 07/956,653	Pasadena, CA UL FOR
Filed: October 2, 1992	Pasadena, CA GROUP
For: MEDICAL DEVICES ) INCORPORATING SIM ALLOY ) ELEMENTS	#13/2007
Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231	REQUEST FOR EXTENSION OF THE IS OF MATCH L. AUTHORITY OF THE PRIMATY ECOMPLET POR 1/2000
AMENDME	NT Clerk, Group 350
sir:	H.C.P. D ATTY Noticed D
In response to the Office	Action of March 7, 1994,

please amend the above-identified application as follows:

# IN THE CLAIMS

Please amend Claim 16, 19, 23, 27, 28, 30, 32, 33, 42, -

and 57 as follows:

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Claim 16, line 1, change "14" to --13--.

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k	Claim 19,	line 10,	after	"body" insert	in its
deformed 090 BA 07/12/94 090 BA 07/12/94		ion	1 102	296.00 CK	A
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(Twice Amended) A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising:

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy the alloy displaying reversible stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) restraining means engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element <u>can be positioned within</u> or in proximity to the mammalian body while the memory alloy <u>element</u> is in its deformed shape;

wherein the alloy is selected so that removal of the restraining means from the memory alloy element at a temperature greater than the As of the alloy when the device is placed within or proximate to the mammalian body, transforms at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape toward its unstressed shape, without any change in temperature of the restraining means or the memory alloy element being required for the transformation of the alloy.

28. (Twice Amended) A medical device for treatment of a mammalian body, the device comprising:

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(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) a hollow restraining member with the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element <u>can be positioned within or in proximity</u> to the mammalian body while the memory alloy element is in its Ideformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

Claim 30, line 1, change "27" to --29--.

Claim 32, lines 5 and 7, change "means" to --member--.

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(Twice Amended) A medical device for insertion into a mammalian body, the device comprising (i) a restraining member and (ii) a hollow catheter formed at least party from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

the [restraint] <u>restraining member</u> engaging and stressing the catheter at a temperature greater than the As of the alloy so that the catheter is in its easily inserted shape <u>so</u> that the catheter can be inserted into the mammalian body;

wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from its stressinduced martensitic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the catheter.

Claim 42, line 3, change "means" to --member--.

57. (Amended) A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising (i) a restraint, and (ii)

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an element formed at least partly from a pseudoelastic shapememory alloy,

the alloy displaying reversible stress-induced martensite by virtue of being above its  $A_s$  and above its  $M_s$  and below its  $M_d$  at about body temperature;

such that it has a stress-induced martensitic state and an austenitic state, the element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

wherein the restraint is capable both (i) stressing the element for placement of the element in its deformed shape in or in proximity to the mammalian body, and (ii) of being at least partially removed from the element while the device is within or proximate to the [said] body at the [said] body temperature and the element is therefore at an operating temperature greater than the A, and M, and below the  $M_d$ of the alloy,

such removal of the restraint causing at least a portion of the alloy to transform from its stress-induced martensitic state to its austenitic state so that the element spontaneously transforms from its deformed shape toward its unstressed shape,

and such transformation can occur without a change in temperature of the restraint or of the element from the operating temperature.

Please add claims 58-65 to the application.

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I 58:34 medical device for treatment of a mammalian body, the device comprising:

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) a hollow tubular restraining member with the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;

wherein the memory alloy element is axially slidable within the tube, and wherein the memory alloy element can be extruded completely out of the tube for deployment in the mammalian body to transform at least a portion of the alloy from its stress-induced martensitic state towards its austenitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

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A medical device which comprises:

(a) a tubular element for use within a mammalianbody or in such proximity to a mammalian-body that the device is substantially at body temperature, the tubular element comprising a shape memory alloy which displays stress-induced martensite behavior at body temperature; and

(b) a restraint within the tubular element holding and deforming the tubular shape memory alloy element in a deformed configuration to allow it to be positioned within or in proximity to a mammalian body, the deformation occurring through the formation of stress-induced martensite;

wherein the tubular element is sufficiently deformed that removal of the restraint from the tubular shape memory alloy element, without change in temperature of the device, releases at least a portion of the tubular element from its deformed configuration.

Why the device comprising (i) a restraining member and (ii) a hollow memory alloy element formed at least partly from a pseudoelastic shape-memory alloy the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

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the restraining member being within the hollow memory alloy element and engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element can be positioned within or in proximity to the mammalian body while the memory alloy element can be positioned within or in proximity to the mammalian body while the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

5. The device of Claim 50 wherein the memory alloy element is a tube and the restraining member is axially slidable within the tube, and wherein the tube is sufficiently long that relative axial movement between the tube and the restraining member extends at least a portion of the tube beyond the restraining member and thereby transforms the tube toward its austenitic shape.

SUF5 62. A medical device for insertion into a mammalian body, the device comprising (i) a restraining member and (ii) a hollow catheter formed at least partly from a pseudoelastic PCANVPSINGSVARB-3.AMD 8 shape-memory alloy, the alloy displaying reversible stressinduced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

the restraining member engaging and stressing the catheter at a temperature greater than the As of the alloy so that the catheter is in its easily inserted shape so that the catheter can be inserted into the mammalian body; and

wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from its stressinduced martensitic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the catheter.

34 30 53. The invention of Claim 58, 50, or 61 wherein the transformation of the alloy occurs without any change in the state of the restraining member.

64. The device of Claim 62 wherein the device is adapted so that (i) the restraining member can be completely disengaged and separated from the catheter, and (ii) re-engaging the restraining member with the catheter after separation results

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in the catheter transforming toward its easily inserted shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

#### REMARKS

Claims 11-64 are pending in this application, Claims 1-10 have been canceled, and Claims 15, 16, 22, 23, 25 (after this amendment), 37-40, 43, and 54 have been withdrawn from consideration. All of the claims originally submitted and examined were rejected. Claims 58-64 are added by this amendment. Re-examination, reconsideration, and allowance are respectfully requested.

Entry of the amendments to the claims originally submitted is respectfully requested. Claims 16, 23, and 30 have been amended to correct their dependency; Claims 19, 27, 28, 33, and 57 have been amended to make it clear that the because of the restraint, the shape memory alloy element can be positioned within or in proximity to a mammalian body while it is in its deformed configuration. Claims 32 and 42 have been amended to use the correct antecedent term, namely, "restraining member."

New claim 58 is claim 30 rewritten in independent form, rewritten as if it depended from Claim 29 (which it should have in the first place).

Claim 59 is Claim 24 rewritten in independent form.

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Claims 60-64 correspond to claims 31, 32, 33, 42, and 51, respectively, where the independent claims include the limitation that the memory alloy element can be positioned within or in proximity to the mammalian body while it is in its deformed shape as a result of the restraining member.

# REJECTION UNDER 35 U.S.C. 112

Claims 11-14, 17, 18, 25 were rejected under 35 U.S.C. § 112. With regard to the rejection under 35 U.S.C. § 112, of Claim 20, applicant agrees that Claim 25 should be withdrawn from consideration. The rejection of the remaining claims is respectfully traversed.

Claims 11-14, 17, and 18 were rejected under the first paragraph of 35 U.S.C. § 112 on the basis that the specification, as originally filed, does not provide support for the invention as it is claimed. In particular, the Office Action contends that the definition in the specification of the A(90) temperature and including a specific value of 0° C is not supported by the specification as originally filed. This rejection is respectfully traversed.

Both the Federal Circuit and the CCPA have held that including in the claims an inherent limitation of a composition described in the patent application is permissible. <u>See</u> <u>Kennecott Corp. v. Kyocera International, Inc.</u>, 5 USPQ 2d 1194, 835 F.2d 419 (Fed. Cir. 1987); <u>In Re Nathan</u>, 140 USPQ 601, 321

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F.2d 1005 (CCPA 1964). In the <u>Kennecott</u> decision, the Federal Circuit held that it is permissible to add to the claims the limitation that a ceramic body has a predominantly "equitaxed microstructure", even though the original application did not disclose such a structure, because the structure was inherent in the original disclosure. The Court stated:

> "The disclosure in a subsequent patent application of an inherent property of a product does not deprive that product of the benefit of an earlier filing date. Nor does the inclusion of a description of that property in later-filed claims change this reasonable result."

5 USPQ 2d at 1198.

Similarly, in <u>Nathan</u>, the CCPA held that it is permissible to include in the claims the limitation that a particular compound had an alpha orientation, even though that was not in the original disclosure, because it was "merely a statement of an inherent property of the steroids as disclosed in appellants' original disclosure."

If it is the examiner's position that applicant was required to indicate in the disclosure that 0°C is a preferred temperature for the A(90) temperature of the alloys, that contention is respectfully traversed. The requirement of 35 U.S.C. § 112, first paragraph, is: "The specification shall contain a written description of the invention . . . ". There is nothing in section 112 that requires the applicant to specify

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Edwards Exhibit 1033, p. 217

that a particular limitation is preferred or critical; 112 only requests that the invention as claimed be described. <u>See for</u> <u>example, In Re Eickmeyer</u>, 202 USPQ 65 (CCPA 1979), where the CCPA held that a specification supported adding to the claims the limitation, "at least about 56° C". In that case, the patent office contended that the description requirement of 35 U.S.C. § 112 was not satisfied because applicant had not disclosed that 56° C was a minimum or critical lower limit for operation of the process. The CCPA reversed that rejection noting that the appellant was entitled to claim a range of temperatures below 56°C and above 56°C, stating:

> We are not persuaded that there is any requirement for the appellant to demonstrate the *criticality* of a lower limit to meet the description requirement. (Emphasis original.)

202 USPQ at 663.

The examiner's attention is also directed to the Federal Circuit decision in <u>Vas-Cath, Inc. v. Mahurkaas</u>, 35 F.2d 1555 (Fed. Cir. 1991), where the Federal Circuit held that a design patent application provided a sufficient basis under 35 U.S.C. § 112 for a utility patent. Certainly, if a design application, which contains substantially nothing but drawings, can support a utility patent application, applicant's disclosure herein supports the claims presented.

For these reasons, removal of the rejection under 35 U.S.C. § 112 is respectfully requested.

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## DOUBLE PATENTING REJECTION

Certain claims were rejected under the judicially created doctrine of obviousness-type double patenting. Applicant, without prejudice, to expedite prosecution, is willing to file a terminal disclaimer on resolution of the remaining issues present in this application.

### ALLOWABLE CLAIMS

Upon filing of the terminal disclaimer, it is believed that Claims 24, 30-56, 35, 36, 42 (31, 33), 44 (35), 45 (31), 46, 48 (31, 33), 50 (48 (31, 33)), 51, 52, and 53 are allowable. Moreover, all of the new claims, are likewise allowable. In particular, Claims 58 and 59 are claims that are believed to be allowable based on the prior office action, rewritten in independent form. Claim 60-64 are narrower versions of claims containing allowable subject matter.

## REJECTION UNDER 35 U.S.C. SECTION 103

Claims 19, 20, 26-28, 41, 42, 45, 47-50, and 57 were rejected under 35 U.S.C. 103. The rejection was based upon the Wayman article alone. In addition, Claims 21 and 29 were rejected as unpatentable over Wayman in view of Wilson. These rejections are respectfully traversed as applied to these claims as you presented.

It is believed that the claims as originally submitted are allowable over Wayman alone or Wayman in combination with Wilson. However, to further distinguish the claimed subject

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matter over these two references, certain of the claims have been amended. In particular, all of the claims rejected under 35 U.S.C. § 103 now state that the restraining member stresses the SIM memory allow element so that the memory allow element can be positioned within or in proximity to the mammalian body while it is in a deformed shape, something not suggested by these references.

In particular, it is contended in the office action that Wayman suggests or teaches the use of dental wires using SIM (Stress-Induced Martensite) alloys. However, Wayman's device is clearly of the kind which is inserted into the body in an <u>unstressed</u> state, and is subsequently stressed <u>within</u> the body and fixed in the stressed state, so as to apply corrective pressure to a misaligned tooth.

The Wayman pressure-applying device does not require or suggest the presence of a restraint, which is essential to the claimed invention, for holding the SIM device in the deformed state <u>during insertion</u> into the body. There is no need in Wayman for a restraint to be applied to the dental wire before it is positioned in the body, since it is unstressed during insertion. Indeed, it is difficult to imagine how a pre-insertion restraint could be used with the dental wires, given their elongated form.

The claims presented specifically require that the restraint stresses the SIM memory alloy element so that it can be "positioned within or in proximity to the mammalian body while the memory alloy element is in its deformed shape." (See, for example, Claim 27.) The claimed device is constructed so that,

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after the deformed device held in the restraint has been positioned within the body, the restraint can be removed to allow the SIM element to recover to its unstressed shape.

The claimed devices with the insertion restraint are highly advantageous in surgical techniques whereas the unstressed shape, to which the SIM element reverts on removal of the restraint within the body, is important for the therapeutic effect. This ability to release the unstressed shape after placing the SIM element where it is needed for treatment is the antithesis of deforming and stressing the element within the body, which would be essential to the use of an SIM dental wire.

Thus, the claimed "shape-forming" device claims should be regarded as a different class of medical devices from the "pressure-applying" Wayman device. The problems encountered in surgical use of the shape-forming devices generally are quite different from those of the pressure-applying devices, and it must be recognized that each special surgical procedure has its own unique set of problems.

Part of the claimed invention is the recognition of the problem in prior art devices that rely on temperature change. Note that Claim 27, for example, specifically states that the change in shape of the SIM member can occur "without any change in temperature of the restraining means or the memory alloy element".

Wilson does nothing to remedy the deficiencies of Wayman. Wilson, in fact, teaches away from the invention, in that Wilson requires a change in temperature of the catheter for

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it to be effective. This is completely contrary to what is claimed by applicant. In effect, Wayman teaches away from the use of SIM material in combination with a restraint.

Further, there is no suggestion in the references to place an SIM element <u>inside</u> a restraint as specified in Claims 20, 21, 28, 29, and claims dependent therefrom. The most Wayman teaches is an SIM element engaged with a device on a tooth, and Wilson does not teach the use of any restraint. In the Wilson device, the catheter 10 does not restrain the rod 16 in any configuration. Rather, the shape of the rod is dependent only upon temperature. Thus, neither reference suggests the use of an external restraint for maintaining an SIM element in a deformed configuration.

### CONCLUSION

Thus, for these additional reasons, it is respectfully submitted that these claims are allowable.

SHELDON & MAK, INC.

G. Sheldon

Reg. No. 27,953

Date: June 24, 1994

SHELDON & MAK 225 South Lake Avenue Suite 900 Pasadena, California 91101 (818) 796-4000 (213) 681-9000

PC4\WP51\JGS\9438-3.AMD

FILED: October 2, 1992



MENDMENT COVER SHEET

ICATION OF: JAMES E. JERVIS

NO.: 07/956,653

FOR: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

HONORABLE COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231

Sir:

Transmitted herewith is a paper in the above-identified application. Any necessary extension of time period set for this . paper is hereby requested.

□ No additional fee is required.

D The fee has been calculated as shown below:

E EXTENSION FEE	RATE Non-Small Entity	RATE Small-Entity	FEE
FIRST MONTH AFTER TIME PERIOD SET	110.00	55.00	\$110.00
SECOND MONTH AFTER TIME PERIOD SET	360.00	180.00	\$
THIRD MONTH AFTER TIME PERIOD SET	840.00	420.00	\$
FOURTH MONTH AFTER TIME PERIOD SET	1,320.00	660.00	\$

**TOTAL EXTENSION FEE \$ 110.00** IN

FEE FOR EXTRA CLAIMS added by Amendment in this response:

1 Column 2	Column 3
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	Column 1	Column 2	Column 3			
	Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	FEE
TOTAL CLAIMS	55	MINUS 46	* = 9	x 22	x 11	\$198.00
INDEPENDENT	12	MINUS 8	* = 4	x 74	x 37	\$296.00
First presentation of	of multiple depende	ent claim		+ 230	+ 115	\$

TOTAL FEE FOR EXTRA CLAIMS \$ 494.00

If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3.

** If the number of Total Claims previously paid for is less than 20, write "20" in this space. ***

If the number of Independent Claims previously paid for is less than 3, write "3" in this space.

M Enclosed is the fee of \$604.00 by Check No. 5012

Please charge Deposit Account No. 19-2090 in the amount of \$

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The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19-2090:

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

SHELDON & MAK By: Jeffrez G//Skeldon Reg. No.: 29,753

Date: June 24, 1994

225 South Lake Avenue Suite 900 Pasadena, California 91101 (818) 796-4000 (213) 681-9000

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### Edwards Exhibit 1033, p. 223

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IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re a	pplication of:
JAMES E	. JERVIS
Serial	No.: 07/956,653
Filed:	October 2, 1992
For:	MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

Group Art Unit: 3301 Examiner: KENEALY, D.

Pasadena, CA

### SUPPLEMENTAL AMENDMENT

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Sir:

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In response to the Office action of March 7, 1994, and as a supplement to the Amendment of June 24, 1994, please amend the above-identified application as follows:

IN THE CLAIMS Please amend claims 19, 27, 28, 57, 60, and 62 as

follows:

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-1-

Jul. 28. 1994

19. (Thrice Amended) A medical device which comprises:

 (a) an element for use within a mammalian body or in such proximity to a mammalian body that the device is substantially at body temperature, the element comprising a shape memory alloy which displays a stress-induced martensite behavior at body temperature; and

(b) a restraint holding the shape memory alloy element in a deformed configuration <u>at a temperature less than the body</u> <u>temperature of the mammal</u> [to allow it to be positioned] <u>for</u> <u>positioning the shape memory alloy element</u> within or in proximity to a mammalian body in its deformed configuration, the deformation occurring through the formation of stress-induced martensite;

wherein the element is sufficiently deformed that we have a sufficiently deformed that we have a sufficient of the restraint from the shape memory allow element, without change in temperature of the device, releases at least a solution of the element from its deformed configuration.

(Thrice Amended) A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising:

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying

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Jul. 28, 1994

Edwards Exhibit 1033, p. 226

reversible stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and

(b) restraining means engaging and stressing the memory alloy element at a temperature <u>less than the body</u> <u>temperature of the mammal and</u> greater than the As of the alloy [so that] for positioning the memory alloy element [can be positioned] within or in proximity to the mammalian body while the memory alloy element is in its deformed shape;

wherein the alloy is selected so that removal of the restraining means from the memory alloy element at a temperature greater than the As of the alloy when the device is placed within or proximate to the mammalian body, transforms at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape toward its unstressed shape, without any change in temperature of the restraining means or the memory alloy element being required for the transformation of the alloy.

(a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying

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PATENT 9438

reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state , and (ii) a different unstressed shape; and

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(b) a hollow restraining member with the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature <u>less than the body temperature of the</u> <u>memmal and greater than the As of the alloy [so that] for</u> <u>positioning the memory alloy element [can be positioned] within</u> or in proximity to the mammalian body while the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

33 57. (Twice Amended) A medical device suitable for placement within or proximate to a mammalian body for treatment PC3\PT0\9438-4.AMD -4- Jul. 28, 1994 of the mammalian body, the device comprising (i) a restraint, and (ii) an element formed at least partly from a pseudoelastic shape-memory alloy,

the alloy displaying reversible stress-induced martensite by virtue of being above its  $A_s$  and above its  $M_s$  and below its  $M_d$  at about body temperature;

such that it has a stress-induced martensitic state and an austenitic state, the element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

wherein the restraint is [capable both] (i) stressing the element at a temperature less than the body temperature of the mammal for placement of the element in its deformed shape in or in proximity to the mammalian body and (ii) is capable of being at least partially removed from the element while the device is within or proximate to the body at the body temperature and the element is therefore at an operating temperature greater than the  $A_s$  and  $M_s$  and below the  $M_d$  of the alloy,

such removal of the restraint causing at least a portion of the alloy to transform from its stress-induced martensitic state to its austenitic state so that the element spontaneously transforms from its deformed shape toward its unstressed shape,

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PATENT 9438

and such transformation can occur without a change in temperature of the restraint or of the element from the operating temperature.

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34 60. (Amended) A medical device for treatment of a mammalian body, the device comprising (i) a restraining member and (ii) a hollow memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

the restraining member being within the hollow memory alloy element and engaging and stressing the memory alloy element at a temperature <u>less than the body temperature of the mammal and</u> greater than the As of the alloy [so that] <u>for positioning the</u> memory alloy element [can be positioned] within or in proximity to the mammalian body [while the memory alloy element can be positioned within or in proximity to the mammalian body] while the memory alloy element is in its deformed shape;

wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stressed-induced martensitic PC3(PTO)(9438-4.AMD -6- Jul. 28, 1994

Edwards Exhibit 1033, p. 229

PATENT 9438

state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member of the memory alloy element.

(Amended) A medical device for insertion into a numar mammalian body, the device comprising (i) a restraining member and (ii) a hollow catheter formed at least partly from a pseudoelastic shape-memory allow, the alloy displaying reversible numar stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

the restraining member engaging and stressing the catheter at a temperature less than the body temperature of the human <u>mamma</u> and greater than the As of the alloy so that the catheter is in its easily inserted shape [so that the] for inserting the human catheter [can be inserted] into the mammalian body, and

wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from it stress-induced -7-

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人名英格兰姓氏 化分子

martensitic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the catheter.

### REMARKS

Applicant wishes to thank the Examiner for the courtesy shown to Applicant's attorney, the undersigned, during the interview on June 30, 1994.

As indicated during the interview, and as stated on the Examiner's Interview Summary Record:

"Examiner agrees that the inclusion of the limitation requiring the restraining means to engage and stress the memory alloy element at a temperature below body temperature will define over the Wayman article . . . ."

In accordance with that statement, the independent claims pending in this application that were subject to rejection under 35 U.S.C. § 103 (namely claims 19,27, 28, and 57) have been

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Edwards Exhibit 1033, p. 231

so amended. In addition, other claims, namely claims 60 and 62, which were not so rejected, have been amended.

As indicated in the interview, for the purposes of a response, Applicant has assumed that Wayman teaches an SIM dental wire. However, as appreciated by the Examiner in the interview, Wayman's SIM dental wire is not stressed until it is in the mouth, and thus Wayman does not teach or suggest a restraining means "engaging and stressing the memory alloy element at a temperature less than the body temperature of the mammal."

Therefore, for this reason and in addition the reasons presented in the prior amendment, allowance of all of the claims is respectfully requested.

Applicant wishes to remind the Examiner of two comments made by Applicant's attorney during the interview. In particular, it was noted that at page 12, line 3, the word is "equiaxed" and not "equitaxed". It was also noted that at page 15 of the amendment, the paragraph bridging pages 15 and 16, in the first sentence, the claims being referred to were the claims rejected under 35 U.S.C. § 103. Not all of the claims presented include the limitations specified in that sentence, but only the

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claims that were being argued with regard to the rejection under § 103.

Another typographical error has been noted. At page 11, line 9, the reference is to claim 25 and not claim 20.

Respectfully submitted,

SHELDON & MAK

Dated: July ____, 1994

In By: REY G. SHELDON

JEFFREY G. SHELDON Regis. No. 27,953

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	MENDME	NT COVER SHEET	
and a star and a star and a star a The star a st			DOCKET NO. 9438
IN RE APPLICATION OF	JAMES E. JERVIS		RECE
SERIAL NO.: 07/956,653	FILED: OCTOBER 2, 199	2	
FOR: MEDICAL DEVICE	S INCORPORATING SIM ALI	LOY ELEMENTS	JUL 29 94
HONORABLE COMMISSI PATENTS AND TRADE Washington, D.C. 20231	~ ~ ~ ~	$\mathcal{X}$	GROUP: 330
Sir:	Ke	ncaly	
Transmitted herewith is a pa	per in the above-identified applic	ation. Any necessary extension of tim	e period set for this paper is

N No additional fee is required.

hereby requested.

[] The fee has been calculated as shown below:

[] EXTENSION FEE	RATE Non-Small Entity	RATE Small-Entity	FEE
FIRST MONTH AFTER TIME PERIOD SET	110.00	55.00	\$
SECOND MONTH AFTER TIME PERIOD SET	360.00	180.00	\$360.00
THIRD MONTH AFTER TIME PERIOD SET	840.00	420.00	\$
FOURTH MONTH AFTER TIME PERIOD SET	1,320.00	660.00	\$

[X] No extension fee is necessary because the original amendment was filed with the extension fee already paid.

[] FEE FOR EXTRA CLAIMS added by Amendment in this response:

	Column 1	Column 2	Column 3		<u> </u>	
	Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	, PEE
TOTAL CLAIMS		MINUS **	* =	x 22	x 11	\$
INDEPENDENT		MINUS ***	* =	x 74	x 37	\$
First presentation o	f multiple depende	nt claim		+ 230	+ 115	\$

TOTAL FEE FOR EXTRA CLAIMS \$ _

If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3. *

** If the number of Total Claims previously paid for is less than 20, write "20" in this space.

*** If the number of Independent Claims previously paid for is less than 3, write "3" in this space.

[] Enclosed is the fee of \$___ __ by Check No.

Please charge Deposit Account No. 19-2090 in the amount of \$ []

The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19-2090: [X]

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

7 (25/mg Date:

SHELDON & MAK By Sheldon, Reg. No. 27,953

SHELDON & MAK 225 SOUTH LAKE AVENUE, SUITE 400 PASADENA, CALIFORNIA 91101 (818) 796-4000 - Direct Line: (818) 356-1201

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JEFFREY G. SHELDON	F 30117 10.		T UNIT PAPER NUMBER	
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225 S. LAKE AVENUE - PASADENA, CA 91101	91H FLOOR	3:	301	$\left\{ \right\}$
		DATE MA	ILED:	
This is a communication from the evention in	charge of your application		10/31/94	/
This is a communication from the examiner in COMMISSIONER OF PATENTS AND TRADE			,	43
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This application has been examined	Responsive to commun	ication filed on $\frac{7/29}{2}$	74 This action is made fin	al.
A shortened statutory period for response to the	nie action is ont to ounits	3 month/a)	days from the date of this letter.	
Failure to respond within the period for response				÷.
Part I THE FOLLOWING ATTACHMENT(S	) ARE PART OF THIS ACTI	ON:		
_		_	· ·	2 A
1. Notice of References Cited by Exa			nan's Patent Drawing Review, PTO-94	8.
<ol> <li>Notice of Art Cited by Applicant, P[*]</li> <li>Information on How to Effect Draw</li> </ol>		4. LI Notice of Inform	al Patent Application, PTO-152.	1
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Part II SUMMARY OF ACTION	1			,
1. 🔯 Claims (	64	<u> </u>	are pending in the applicatio	n. :
Of the above, claims	(		are withdrawn from consideration	4.
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,				1 and the second
3. Claims	·		are allowed.	1
4. Claims 11-64			are relected	
5. Claims			are objected to.	<u> </u>
6. 🛄 Claims		are subject to	restriction or election requirement.	10 ].
7. This application has been filed with in	formal drawings under 37 C.	F.R. 1.85 which are acceptable t	or examination purposes.	ų. S
	, -		· • •	.»
<ol> <li>Formal drawings are required in response</li> </ol>	onse to this Office action.			4
<ol> <li>The corrected or substitute drawings i are acceptable; ont acceptable</li> </ol>			der 37 C.F.R. 1.84 these drawings	
	(See explanation of Notice C	in Dransman's Patern Drawing n	eview, F 10-940j.	
10. The proposed additional or substitute examiner; disapproved by the examiner		n has (have	) been Dapproved by the	
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11. The proposed drawing correction, filed	اا	nas been 🔲 approved; 🔲 disa	pproved (see explanation).	4.
12. Acknowledgement is made of the claim				<b>i</b> - 🔨
been filed in parent application, see	rial no	_ ; filed on	•	
13. Since this application apppears to be accordance with the practice under Ex			ion as to the merits is closed in	<b>1</b> : 1
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EXAMINER'S ACTION

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Serial No. 956653 Art Unit 331

2.

The following is a quotation of the first paragraph of 35 U.S.C. § 112:

The specification shall contain a written description of the invention, and of the manner and process of making and using it, in such full, clear, concise, and exact terms as to enable any person skilled in the art to which it pertains, or with which it is most nearly connected, to make and use the same and shall set forth the best mode contemplated by the inventor of carrying out his invention.

The specification is objected to under 35 U.S.C. § 112, first paragraph, as the specification, as originally filed, does not provide support for the invention as is now claimed. Applicant's declaration defining the A(90) temperature as the temperature at which 90% of the material has been transformed from martensite to austenite is not sufficient to now allow a claim for a specific value for the A(90) temperature being 0 degrees C. Nowhere in applicant's specification does he state the criticality of such a temperature for the A(90) temperature, nor does he ever state a preferred range in which the A(90) temperature should fall.

Claims 11-14,17 and 18 are rejected under 35 U.S.C. § 112, first paragraph, for the reasons set forth in the objection to the specification.

Claim 25 is withdrawn since it is drawn to a non-elected species of invention.

Claims 11-14,17-21,24-27,31-36,41,42,44-53,55-57,59 and 60-64 are rejected under the judicially created doctrine of obviousnesstype double patenting as being unpatentable over claim 3 of U.S. Serial No. 956653 Art Unit 331

Patent No. 4,665,906. Although the conflicting claims are not identical, they are not patentably distinct from each other because they both are directed towards a restraint in combination with an SIM medical device.

-3-

Claims 28-30,58 rejected under the judicially created doctrine of obviousness-type double patenting as being unpatentable over claim 3 of U.S. Patent No. 4665906 in view of Wilson. To have made the restraint out of SIM material and the catheter out of non-SIM material is well known in the art as shown by Wilson.

Claims 19,20,21,26-29,41,42,45,47,48,49,50,57 and 63 are rejected under 35 U.S.C. § 112, second paragraph, as being indefinite for failing to particularly point out and distinctly claim the subject matter which applicant regards as the invention. Applicant's amended language requires the element to be restrained in a deformed configuration at a temperature <u>less than</u> body temperature. The claim also states that this deformation occurs through the formation of stress-induced martensite at that temperature less than body temperature. However, the device is earlier claimed as exhibiting stress-induced martensite behavior <u>at</u> body temperature.

### **RESPONSE TO APPLICANT'S REMARKS**

Applicant's amended claim language is not believed to place the case in condition for allowance. Applicant is invited to discuss the claim language with the examiner to determine possible amendment to the claims to overcome the indefiniteness rejection. Serial No. 956653 Art Unit 331

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With regard to the rejection under 35 USC 112 1st paragraph, it should be noted that if applicant believes that the A(90) temperature being greater than 0 degrees C is an inherent quality of any SIM material, then the only true limitation that claim 11 has is that the material exhibits stress induced martensite at body temperature. Examiner does not consider the value of A(90) temperature being 0 degrees C to be inherent. Thus this limitations is considered to be a new limitation presently set forth in the claims with no basis in the original specification.

Applicant's amendment necessitated the new grounds of rejection. Accordingly, THIS ACTION IS MADE FINAL. See M.P.E.P. § 706.07(a). Applicant is reminded of the extension of time policy as set forth in 37 C.F.R. § 1.136(a).

A SHORTENED STATUTORY PERIOD FOR RESPONSE TO THIS FINAL ACTION IS SET TO EXPIRE THREE MONTHS FROM THE DATE OF THIS ACTION. IN THE EVENT A FIRST RESPONSE IS FILED WITHIN TWO MONTHS OF THE MAILING DATE OF THIS FINAL ACTION AND THE ADVISORY ACTION IS NOT MAILED UNTIL AFTER THE END OF THE THREE-MONTH SHORTENED STATUTORY PERIOD, THEN THE SHORTENED STATUTORY PERIOD WILL EXPIRE ON THE DATE THE ADVISORY ACTION IS MAILED, AND ANY EXTENSION FEE PURSUANT TO 37 C.F.R. § 1.136(a) WILL BE CALCULATED FROM THE MAILING DATE OF THE ADVISORY ACTION. IN NO EVENT WILL THE STATUTORY PERIOD FOR RESPONSE EXPIRE LATER THAN SIX MONTHS FROM THE DATE OF THIS FINAL ACTION.

Any inquiry concerning this communication or earlier communications from the examiner should be directed to David Kenealy whose telephone number is (703) 308-2680.

W

David J. Kenealy October 31, 1994

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Sir:							
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Edwards Exhibit 1033, p. 239

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UNITED STATES PATENT AND TRADEMARK OFFICE

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In re application of:

JAMES E. JERVIS

Sir:

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Serial No. 07/956,653

Filed: October 2, 1992

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS Pasadena, California

Group Art Unit: 3301

Examiner:

DECLARATION

Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231 I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U. S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOPE ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D. C. 20231 ON

9, 1990 (DATE SIGNED)

I, Dr. John D. Harrison, declare:

#### BACKGROUND

1. As established in my attached resume, I am an expert metallurgist, with special knowledge of alloys exhibiting martensite/austenite transformation characteristics.

2. In particular, I received a B.S. and M.S. in Metallurgy from Pennsylvania State University and an Sc.D. in Metallurgy in 1958 from the Massachusetts Institute of Technology.

3. My professional experience includes almost 30 years as a metallurgist associated with Raychem Corporation where, among other projects, I worked on the research and development which lead to commercial shape memory alloy products, including those based upon titanium/nickel alloys.

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4. I left Raychem in 1981 to start my own consulting business. In my consulting business I have continued my work in titanium/nickel alloys, as exhibited by the list of publications provided in my curriculum vitae. For example, since 1989 I have had two publications dealing with shape memory effects and titanium/nickel alloys. Also, I was co-editor on a chapter on "Metals" in *Electronic Materials and Processes Handbook*.

5. In my position as a consultant, one of the companies I consult for is Raychem, from whom I also receive a pension as a result of my retirement. I am charging Raychem Corporation for the time expended in preparing this declaration at my normal billing rates.

6. In addition to the publications listed in my attached resume, I am also the inventor or co-inventor of the following U.S. patents, which relate to shape memory alloys:

<u>Patent No.</u>	Title
3,753,700	HEAT RECOVERABLE ALLOY
4,035,007	HEAT RECOVERABLE METALLIC COUPLING
4,198,081	HEAT RECOVERABLE METALLIC COUPLING
4,337,090	HEAT RECOVERABLE NICKEL/TITANIUM ALLOY WITH IMPROVED STABILITY AND MACHINABILITY
4,565,589	NICKEL/TITANIUM/COPPER SHAPE MEMORY ALLOY

There are also foreign patents corresponding to at least some of these U.S. patents.

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## MATERIALS REVIEWED

7. In preparation for this declaration, I reviewed the above-identified patent application, the Office Action of October 31, 1994, for this application, and U.S. Patent No. 4,505,767 to Quin ("Quin"), which is incorporated by reference into the above-identified patent application.

### CONCLUSIONS

8. I wish in this declaration to correct some misconceptions that appear in the Office Action of October 31, 1994.

9. A(90) temperature is an inherent property of any alloy that can transform from the martensitic phase to the austenitic phase. It represents the temperature of such an alloy, annealed under standard conditions, where the transformation from the martensitic phase to the austenitic phase is 90% complete. Just like water has a freezing point of 0°C at standard pressure, each alloy that exhibits SIM behavior inherently has a fixed A(90) temperature, which is a function of its specific composition, when annealed under standard conditions.

10. The A(90) of an alloy is dependent upon its composition. At least one and possibly more of the alloys described in the Quin example and prepared using standard annealing conditions, i.e., those described in the example of the Quin patent, has an A(90) temperature "of not more than 0°C", a limitation that appears in claim 11. Stating it in another way,

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at least one, but definitely not all, of the alloys in the hexagonal area in Fig. 2 of the Quin patent inherently has an A(90) of not more than 0°C, when annealed under standard conditions.

11. Accordingly, claim 11 has at least two important limitations on the shape memory alloy that is used to form the medical device of claim 11, namely the alloy has to display SIM behavior at body temperature, and has to have an A(90) of not more than 0°C. These properties clearly are not satisfied by all shape memory alloys.

12. As the table added to the application (and which appears in issued U.S. Patent No. 5,067,957) clearly shows alloys within the scope of the Quin patent have A(90) that ranges from at least -88°C to 86°C.

13. A person of ordinary skill in the art, by reading the above-identified patent application, will recognize that the preferred shape memory alloy for use in at least some medical devices according to the present invention has an A(90) of not more than 0°C. This is because the application makes it clear the invention requires "an alloy element which displays stressinduced martensite at said body temperature for the shape memory alloy element." (page 5, lines 29-31). It is evident that an alloy with an A(90) of not more than 0°C will assure good SIM effect in the body temperature range. A(90) above 0°C would give a mix of SIM and SME or all SME in the body temperature range. An A(90) far below 0°C would give permanent deformation in the body temperature range.

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I further declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true, and further that these statements were made with the knowledge that willful false statements and the like are punishable by fine or imprisonment, or both, under § 1001 of Title 18 of the United States Code, and that such willful false statements may jeopardize the validity of this application and any patent or patents resulting therefrom.

Respectfully submitted,

Date: 25 February 1995

By: John D. Harrison

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# **CURRICULUM VITAE**

JOHN D. HARRISON, SC.D. Watsonville, California 95076-5333

### **PROFESSIONAL EXPERIENCE**

## October 1981 to Present

## JACK HARRISON, INC. (formerly John D. Harrison, Metallurgist)

## December 1966 to October 1981

# RAYCHEM CORPORATION, Menlo Park, California

First metallurgist at the corporation; primary activity was the research and development which led to shape memory alloy products; most intimately involved with TiNi alloys, extensively involved with copper-base memory alloys; participated in initial laboratory development, first production, product introduction to customers, trouble-shooting field problems; alloy development, thermomechanical processing, melting techniques from gasfired crucibles through induction, electron beam and plasma; strategy for proprietary protection and patents; corporate-wide metals selection, corrosion studies, failure analysis.

## September 1959 to November 1966

# WESTINGHOUSE RESEARCH LABORATORIES, Churchill Borough, Pennsylvania

Research into solidification phenomena; special topics included interface morphology of ice during freezing from saline solutions, embrittlement of chromium-copper castings and seeded growth of selenium single crystals under high pressure.

### August 1958 to August 1959

## FRITZ-HABER-INSTITUT DER MAX-PLANK-GESELLSCHAFT, Berlin, German

Post-doctoral study of internal oxidation in alloys.

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### **EDUCATION**

# MASSACHUSETTS INSTITUTE OF TECHNOLOGY

Sc.D., Metallurgy, 1958 Thesis Advisor: Professor Carl Wagner

## PENNSYLVANIA STATE UNIVERSITY

M.S., Metallurgy & Ensign, USNR 1953 B.S., Metallurgy 1952

## MILITARY EXPERIENCE

## July 1953 to July 1955

U.S. Navy, Ensign, then Lt. j.g. USS LST 32 Home port: Naples, Italy

September 1949 to June 1953

NROTC Midshipman

# LIST OF PUBLICATIONS

K.N. Melton & J.D. Harrison, "Corrosion of TiNi," Proceeding of Shape Memory & Superelasticity Technologies Conference, Asilomar, March 6-10, 1994, in publication.

J.D. Harrison & D.E. Harrison, co-editors, Chap. 5, "Metals," *Electronic Materials & Processes Handbook*, second edition, editors: C.A. Harper and R.M. Sampson (McGraw-Hill, New York, 1993), p 5.1-5.69.

J.D. Harrison, "Measurable Changes Concomitant with the Shape Memory Effect Transformation," *Engineering Aspects of Shape Memory Alloys*, editors T.W. Duerig, K.N. Melton, D. Stoeckel & C.M. Wayman, (Butterworth, Boston, 1990) p. 106-111.

C.M. Wayman & J.D. Harrison, "The Origins of the Shape Memory Effect," JOM 41 no. 9 26 (September 1989).

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J.D. Harrison & D.E. Hodgson, "Use of TiNi in Mechanical and Electrical Connectors," *Shape Memory Effects in Alloys*, edited by A.J. Perkins (Plenum Press. New York, 1975), p. 517-523.

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J.D. Harrison & D.E. Harrison, "Etch Pit Studies on Single Crystals of Hexagonal Selenium Grown from the Melt at High Pressures," *The Physics of Selenium and Tellurium*, edited by W.C. Cooper, (Pergamon Press, New York, 1969), p 115-134.

J.D. Harrison, "Seeded Growth of Selenium Crystals under High Pressure," Journal of Applied Physics 39 no. 8 3672 (July 1968).

J.D. Harrison, "Measurement of Brine Droplet Migration in Ice," Journal of Applied Physics 36 3811 (December 1965).

J.D. Harrison, "Solute Transpiration Pores in Ice," *Journal of Applied Physics* 36 326 (January 1965).

J.D. Harrison & W.A. Tiller, "Ice Interface Morphology and Texture Developed During Freezing," *Journal of Applied Physics* 34 3349 (November 1963).

J.D. Harrison & W.A. Tiller, "Controlled Freezing of Water," *Ice and Snow Properties, Processes and Applications*, edited by W.D. Kingery (Technology Press, Cambridge, Massachusetts, 1963) p 215.

J.D. Harrison & W.A. Tiller, "The Controlled Solidification of Aqueous Solutions," Desalination Research Conference Proceedings, NAS-NRC Publication 942 312 (1963).

J.D. Harrison & W.A. Tiller, "Optimum Conditions for Zone Refining," *Trans. AIME* 221 649 (June 1961).

J.D. Harrison & C. Wagner, "The Attack of Solid Alloys by Liquid Metals and Salt Melts," Acta Met. 7 722 (November 1959).

I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U. S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOGE ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D. C. 20231 ON

arch 9, 995 (DATE SIGNED)

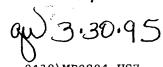
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In re application of:

JAMES E. JERVIS

Serial No. 07/956,653

Filed: October 2, 1992

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

### AMENDMENT

Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231

Sir:

This is submitted in response to the final rejection of October 31, 1994. It is believed this response places this application in condition for allowance, and such allowance is respectfully requested.

IN THE CLAIMS

Cancel claim 54.

### REMARKS

Claims 11-53 and 55-64 are in this application with claim 25 being withdrawn as being directed to a non-elected species of the invention. All the claims presented were rejected under 35 U.S.C. § 112 and/or for double-patenting. Reexamination, reconsideration, and allowance are respectfully requested.

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### DOUBLE-PATENTING REJECTION

All the claims under consideration have been rejected for double-patenting. Applicant respectfully disagrees with the Examiner that a terminal disclaimer is appropriate with regard to at least some of the claims subject to the double-patenting rejection. However, to expedite issuance of this application, a terminal disclaimer is submitted herewith.

## SPECIES CLAIMS

Claims 15, 16, 22, 23, 25, 37-40, and 43 were withdrawn from consideration as being directed to a non-elected species. However, for the reasons detailed below and in view of the terminal disclaimer, the generic claims are allowable in this application, and thus the species claims should also be allowed.

# REJECTION OF CLAIM 25

Claim 25 complies with 35 U.S.C. § 112. As the Examiner noted, at least the IUD version of the invention supports claim 25. Applicant recognizes that claim 25 is not under consideration. However, since a generic claim is allowable, claim 25 should be allowed.

## REJECTION UNDER 35 U.S.C. § 112

### REGARDING TEMPERATURE LIMITATIONS

Claims 19-21, 26-29, 41, 42, 45, 47-50, 57, and 63 were rejected under 35 U.S.C. § 112, second paragraph for being indefinite. In particular, the Examiner was not clear as to how

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the element can be restrained in a deformed configuration at a temperature less than body temperature, while the claim also requires that the element display stress-induced martensite behavior at body temperature.

The undersigned placed a telephone call to the Examiner regarding this rejection in December, 1994, and it is believed that in view of the telephone conversation and the remarks made by the undersigned, the rejection will be withdrawn.

In particular, the two limitations referred to by the Examiner are two limitations on the claimed device, which are complimentary, and not inconsistent. The first limitation requires that the element which is used within a mammalian body be a shape memory alloy (SMA) which displays stress-induced martensite behavior at body temperature. In particular, this limitation requires that the element be made out of a particular alloy that has specific properties, namely SIM behavior at body temperature. This is a physical parameter of the element; analogous to Applicant specifying that the shape memory alloy has a melting temperature at body temperature.

The second limitation referred to requires that the restraining portion of the claimed medical device holds the shape memory alloy element in a deformed configuration at a temperature less than the body temperature of the mammal. In particular, the claimed combination of the SMA element and the restraint must at

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some time be at a temperature less than body temperature, a limitation which helps distinguish over prior art dental arch wires. Using the same analogy above as with regard to the melting temperature of a shape memory alloy, this is akin to requiring that the medical device be provided at a temperature less than the melting point of the shape memory alloy.

Accordingly, the two limitations are not inconsistent, but rather are complimentary. The shape memory alloy must display SIM behavior at body temperature, but in addition the SMA element must be restrained by the restraint at some point in time at a temperature lower than body temperature.

It is believed that in view of this explanation, this rejection under 35 U.S.C. § 112 should be withdrawn.

# REJECTION RELATING TO A (90) TEMPERATURE

Claims 11-14, 17 and 18 were rejected under 35 U.S.C. § 112, and similarly the specification was objected to under 35 U.S.C. § 112 with regard to the reference to A(90) temperature.

It is believed that the rejection is based on the following grounds:

1. Reference to A(90) is not in the application as originally submitted, i.e., is new matter; and

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2. The specification as originally submitted did not state the criticality of the A(90) temperature and the preferred range.

This rejection is respectfully traversed for the following reasons.

First, the addition of the A(90) temperature in the specification is not new matter. This is in inherent property of the alloys described in the Quin patent, which is incorporated by reference into the present application. Provided herewith is a declaration by an expert in the field, Dr. John D. Harrison, where he concludes, based upon facts, that:

> "A(90) temperature is an inherent property of any alloy that can transform from the martensitic phase to the austenitic phase."

The Examiner's attention is directed to the declaration and the factual basis in the declaration for that conclusion.

Since we are dealing with an inherent property of the material, it is not new matter to add A(90) values to the specification and the claims. Both the Federal Circuit and the C.C.P.A. have held that an inherent property of a product that is not disclosed in the original specification can be recited in later filed claims without losing the benefit of the earlier

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Mar. 9, 1995

filing date. <u>Kennecott Corp. v. Kyocera Int'l, Inc.</u>, 5 U.S.P.Q.2d 1194, 1197, 835 F.2d 419 (Fed. Cir. 1987); <u>In re</u> <u>Nathan</u>, 140 U.S.P.Q. 601, 321 F.2d 1005 (C.C.P.A. 1964). In <u>Kennecott</u>, the Federal Circuit held that it is permissible to add claims directed to a ceramic body having a predominantly "equitaxed microstructure," even though the originally filed specification directed to a process for producing the ceramic did not disclose that the ceramic body had such structure. The Court allowed the later filed claims because the "equitaxed microstructure" was an inherent property of the ceramic material fabricated in the original patent. The Court stated:

> "The disclosure in a subsequent filed patent application of an inherent property of a product does not deprive that product of the benefit of an earlier filing date. Nor does the inclusion of a description of that property in later filed claims change this reasonable result."

<u>Id.</u> at 1198.

Similarly, in <u>Nathan</u>, the C.C.P.A. held that is permissible to include in later added claims, a limitation that a particular compound had an "alpha orientation," even though that was not in the original disclosure because the alpha orientation was "an inherent characteristic" of the claimed subject matter. The <u>Nathan</u> Court explained that "a subsequent clarification of,

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or a change in, an original disclosure does not necessarily make that original disclosure fatally defective." 140 U.S.P.Q. at 603.

Another C.C.P.A. case supporting this established rule of law is <u>In re Reynolds</u>, 170 U.S.P.Q. 94, 443 F.2d 384 (C.C.P.A. 1971). In <u>Reynolds</u>, the claimed product had an inherent function that was not expressly described in the specification. The issue was whether words describing this function could be added to the Specification by Amendment, or whether such description was "new matter." The Court held that the express description of the inherent property was not "new matter" and could be added to the Specification, without losing the original filing date. <u>Id.</u> at 98.

Accordingly, in view of this case law, and in view of Dr. Harrison's explanation that at least one and possibly more of the alloys described in the Quin example (which was incorporated by reference in this application) has an A(90) temperature "of not more than 0°C," it is clear that if the Examiner is suggesting the specification and claims create issues of new matter, this suggestion is incorrect.

It should also be noted that the Examiner's comment that an A(90) greater than 0°C is an inherent property of "any SIM material" is incorrect. All SIM materials do not have such an A(90). Dr. Harrison notes that "at least one, but definitely

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not all, of the alloys in the hexagonal area in Fig. 2 of the Quin patent inherently has an A(90) of not more than 0°C, when annealed under standard conditions" (paragraph 10).

With regard to the suggestion that Applicant needed to recognize the criticality of the limitation, this suggestion is respectfully traversed. It is believed the law does not support the contention. For example, in <u>Kennecott</u> and <u>Nathan</u>, cited above, the claims were amended to include a limitation that was not in the application as originally filed, but was inherent in the application as filed. Similarly here, Applicant is entitled to include in his claims a claim limitation which is inherent in the specification as filed.

A leading case on this issue is <u>In re Voss</u>, 194 U.S.P.Q. 267, 271-3 (CCPA 1977). In that case the CCPA held that:

> It is only required, for example, that the specification describe the invention sufficiently for those of ordinary skill in the art to recognize that the Applicant invented the subject matter he now claims. The PTO has the initial burden of presenting evidence or reasons why those skilled in the art would not recognize in the specification

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a description of the invention defined by the present claims. (citations omitted)

Applying this principle, the Court held that the Patent Office improperly rejected claims under 35 U.S.C. § 112 even though the expression "at least 50 percent" crystal content did not appear in the parent application. The Court noted that the application at issue incorporated by reference a discussion of glass-ceramic materials from another patent application, and even though that portion incorporated by reference did not include "at least 50 percent," the Court concluded that the "at least 50 percent" limitation "merely quantifies the percent of crystallinity one of ordinary skill in the art <u>at that time</u> would have attributed to the term 'glass-ceramic material'" (emphasis original).

Likewise, here, we have the necessary information incorporated by reference. Moreover, Dr. Harrison, in his accompanying declaration, concludes:

> A person of ordinary skill in the art, by reading the above-identified patent application, will recognize that the preferred shape memory alloy for use in at least some medical devices according to the present invention has an A(90) of not more than 0°C.

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Thus, based on <u>Voss</u>, the claims submitted are supported by the specification and should be allowed.

The Examiner's attention is also directed to the decision of the Patent and Trademark Office Board of Appeals in <u>Ex Parte Cure</u>, 215 U.S.P.Q. 567 (1982), which likewise is factually similar to the present application. In <u>Cure</u>, the claims at issued included a limitation "said shield being constructed of a metal which melts at 2800°F," but there was no disclosure in the application of that limitation. The only disclosure was that the shield could be made with low carbon steel. The Board concluded that based on the disclosure of low carbon steel, one having ordinary skill in the art would have understood that "other metals having a melting point less than 2800°F are suitable for use as shield materials in Appellant's invention." The Examiner should note that claims in <u>Cure</u> were not limited just to low carbon steel having such a melting point, but <u>any</u> metal.

Likewise here, as stated by Dr. Harrison, "one of ordinary skill in the art from the disclosure will recognize that the preferred shape memory alloy for use in at least in some medical devices claimed in the present invention has an A(90) of not more than 0°C." Accordingly, the rejection under 35 U.S.C. § 112 should be withdrawn in view of <u>Cure</u>.

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### CONDITIONAL NOTICE OF APPEAL

Accompanying this amendment is a conditional notice of appeal which should be entered in case the amendment does not place this application in condition for allowance.

### CONCLUSION

In view of the foregoing remarks, allowance of all the claims is respectfully requested.

Respectfully submitted,

SHELDON & MAK

) | q [ qi Date

Bν G. Je don

Reg. No. 27,953

225 South Lake Avenue, 9th Floor Pasadena, California 91101 (818) 796-4000

> I HEREBY CERTIFY THAT THIS CORRESPONDENCE IS BEING DEPOSITED WITH THE U.S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOPE ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D. C. 20231 ON

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IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of:

JAMES E. JERVIS

Serial No. 07/956,653

Filed: October 2, 1992

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS Group Art Unit: 3301

Examiner: KENEALY, D

Office Action Mailed: 10/31/94

### Pasadena, California

CONDITIONAL NOTICE OF APPEAL BEING DEPOSITED WITH THE U. S. POSTA

Honorable Commissioner of Patents and Trademarks Washington, D. C. 20231

Sir:

TRADEMARKS, WASHINGTON, D. C. 20231 C ard 9, 199 (DATE SIGNED)

SERVICE AS FIRST CLASS MAIL IN AN ENVELOF

ADDRESSED TO: COMMISSIONER OF PATENTS AN

Applicant conditionally appeals from the final rejection of October 31, 1994, rejecting claims 1-64.

This appeal should be entered only if the accompanying amendment does not place this application in condition for allowance.

Please charge the requisite \$280 (Large Entity) fee to our Deposit Account No. 19-2090 if this Notice of Appeal is entered.

The Commissioner is hereby authorized to charge payment of any additional fees associated with this communication or credit any overpayment to Deposit Account No. 19-2090.

Respectfully submitted,

Bv: Jeffrey G. Sheldon

Reg. No.: 27,953

Date

3 15191

SHELDON & MAK 225 South Lake Avenue, 9th Floor Pasadena, California 91101 (818) 796-4000 MM11396 04/04/95 07956653 19-209

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Mar. 7, 1995

99 14 1995 4 1995 4 1997 4 1977 4 197	0 - 148 V. Douglas P30 P438 MP0884 - US7 J-3-95 ATENT AND TRADEMARK OFFICE
In re application of:	) Group Art Unit: 3301
JAMES E. JERVIS	) Examiner: KENEALY, D
Serial No. 07/956,653 Filed: October 2, 1992	) ) ) Office Action Mailed: ) 10/31/94
For: MEDICAL DEVICES INCORPORA SIM ALLOY ELEMENTS	TING ) ) Pasadena, California
TERMINAL	DISCLAIMER RECEIVED
Honorable Commissioner of Patents and Trademarks	APR 4 - 1995
Washington, D. C. 20231	GROUP 330
Sir:	

I, Herbert G. Burkard, residing at 256 Prior Lane, Atherton, California 84027, represent that I am Corporate Counsel and Assistant Secretary of Raychem Corporation, a corporation having its principal place of business at 300 Constitution Drive, Menlo Park, California 94025, and that Raychem Corporation is the assignee of the entire right, title and interest to Application Serial No. 07/956,653, filed October 2, 1992, for a method for inserting medical devices incorporating SIM alloy elements.

I, on behalf of Raychem Corporation, hereby disclaim the terminal part of any patent granted on the above-identified application, which would extend beyond the expiration date of the full statutory term defined in 35 U.S.C. 154 to 156 and 173 as shortened by any terminal disclaimer filed prior to U.S. Patent No. 4,665,906, issued May 19, 1987, and hereby agree that any patent so granted on the above-identified application shall be enforceable only for and during such period that the legal title 090 BA 03/27/95 07956653 WP51\FC3\TEMP\9438TERM.DIS 1 148 110.00 CK Peb. 22, 1995

to said patent be the same as the legal title to the above referenced patent, this agreement to run with any patent granted on the above-identified application and to be binding upon the grantee, its successors or assigns.

In making the above disclaimer, petitioner does not disclaim the terminal part of any patent granted on the present application that would extend to the expiration date of the full statutory term as defined in 35 U.S.C. 154 to 156 and 173, as presently shortened by any terminal disclaimer, of the abovelisted application in the event that it later: expires for failure to pay a maintenance fee, is held unenforceable, is found invalid by a court of competent jurisdiction, is statutorily disclaimed in whole or terminally disclaimed under 37 CFR 1.321, has all claims cancelled by a reexamination certificate, is reissued, or is otherwise terminated prior to the expiration of its full statutory term as presently shortened by any terminal disclaimer filed prior to the grant of the patent.

The undersigned has reviewed all the evidentiary documents accompanying or referred to in the instant Terminal Disclaimer and it is certified to the best of the undersigned's knowledge and belief, title is in the assignee identified above.

The undersigned (whose title is supplied below) is empowered to act on behalf of the assignee.

I hereby declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false

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Feb. 22, 1995

statements and the like so made are punishable by find or imprisonment, or both, under Section 1001 of Title 18 of the United States Code and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

The fee of \$110 required by 37 C.F.R. § 1.20(d) is submitted herewith.

Respectfully submitted,

March / Er

RAYCHEM CORPORATION

By: Herbert G. Burkard

Assistant Secretary

I HEREBY CERTIFY THAT THIS CORRESPOND IS BEING DEPOSITED WITH THE U. S. POSTAL SERVICE AS FIRST CLASS MAIL IN AN ENVELOP ADDRESSED TO: COMMISSIONER OF PATENTS AND TRADEMARKS, WASHINGTON, D. C. 20231 ON

95 (DATE SIGNED)

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Feb. 22, 1995

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SHEL 225	DON & MAK S. LAKE AVENU	E - 9TH	FLOOR		ART UNIT	PAPER NUMBER
PASA	DENA, CA 9110	1			3301	
					DATE MAILED:	04/26/95
	Below is a comm	unication from	the EXAMINER in cha	rge of this appli	cation	
	COM	MISSIONER OF	PATENTS AND TRA	DEMARKS	*	
			ADVISORY	ACTION		
Ютн	E PERIOD FOR RESPON	NSE:				
a) 🔽	is extended to run	L mos	or continues to run		from the date of the	final rejection
		om the date of th	ne final rejection or as o	f the mailing date	of this Advisory Acti	on, whichever is later. In no
	The date on which the re	esponse, the pe the period of e:	tition , and the fee have stension and the corres	been filed is the conding amount of	date of the response of the fee. Any exter	conse and the appropriate fe a and also the date for the nsion fee pursuant to 37 CFF s set forth in b) above
	ppellant's Brief is due in ac		• •	ied statutory perio	ou for response of as	s set form in by above.
	oplicant's response to the f	inal rejection, fil	ed 3 13 95	has been consid	lered with the followi	ng effect, but it is not deeme
	place the application in co					
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	b. 🔲 They raise new iss	sues that would a	require further consider	ation and/or searc	ch. (See Note).	
	c. 🔲 They raise the issu	ue of new matte	r. (See Note).			
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Serial No. 07/956653 Art Unit 3301

David Kenealy

4/24/95

## Attachment To Advisory Action

With regard to the rejection relating to the A(90) temperature, examiner agrees that an A(90) temperature is inherent to SIM materials. Like the ceramic that inherently has an "equitaxed microstructure", SIM materials inherently have an A(90) temperature. However, a specific value for the A(90) temperature is not an inherent quality of all SIM materials. In fact, the specific value for the A(90) temperature is exactly what applicant Therefore it is difficult to see how bases patentability on. applicant can argue that the specific value is inherent without also conceding that the claim is obvious. Examiner concludes that the O degree value for the A(90) temperature is new matter that is unsupported in applicant's original specification and validly rejected under 35 USC § 112 1st paragraph.

Examiner is hesitant to allow claim 19 despite applicant's arguments and explanations. The claim that the device be at "body temperature" is still indefinite because different mammals have different body temperatures. Does the material exhibit SIM behavior at <u>and</u> below body temperature?

Finally, examiner also requests drawings to accompany the case since applicant's claim 19 requires structural entities to restrict a wire.

Any inquiry concerning this communication or earlier communications from the examiner should be directed to Exr. Kenealy whose telephone number is (703) 308-2680.

ROBERT A. HAFER

S.P.E. ART UNIT 331

9438\mp0884-US7

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of:	) Group Art Unit: 3301
JAMES E. JERVIS	Examiner: KENEALY, D
Serial No. 07/956,653	FAX COPY RECEIVED
Filed: October 2, 1992	JUN 7 1995
For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS	) Pasadena, California GROUP 3300

2 Same

### **REQUEST FOR EXTENSION OF TIME TO FILE APPEAL BRIEF**

Honorable Commissioner of Patents and Trademarks Washington, D.C. 20231

Sec. Sec. 2.

6/7/85

225 South Lake Street, Suite 900 Pasadena, California 91101 (818) 796-4000 (213) 681-9000

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06/22/95

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Sîr:

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Applicant requests an extension of time of one month to file an appeal brief in the above-identified application. A Notice of Appeal had been previously filed on March 9, 1995.

The Examiner is authorized to charge the requisite fee of \$110 to our Deposit Account No. 19-2090.

.1	Small Entity	"Large" Emity
First Month After Time Period Set	\$ 55.00	\$ 110.00
Second Month After Time Period Set	185.00	. 370.00
Third Month After Time Period Set	435.00	870.00
Fourth Month After Time Period Set	680.00	1,360.00

If any additional extension of time is required, such extension is hereby requested. The Commissioner is hereby authorized to charge any additional fees which may be required, or credit any overpayment, to Account No. 19-2090.

XO NOTON DIK

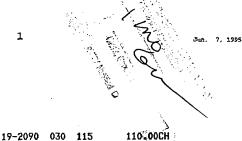
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 ${\bf \hat{s}}_{i,j}$ 

Respectfully submitted,

SHELDON & MAK

Sheldon, Reg. No. 27,953 Jeff/ ď



Edwards Exhibit 1033, p. 265

ROA MENDMENT COVER SHEET DOCKET NO. 9438/MP0884-US7 FILED: October 2, 1992 *NECEIVED NUL 1 4 1995 GROLIP 3300* IN RE APPLICATIO ES E. JERVIS SERIAL NO .: 07/956,6 FOR: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS HONORABLE COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231 Sir:

Transmitted herewith is a paper in the above-identified application. Any necessary extension of time period set for this paper is hereby requested.

[] No additional fee is required.

[] The fee has been calculated as shown below:

[X]EXTENSION FEE RATE RATE Non-Small Entity Small-Entity FEE FIRST MONTH AFTER TIME PERIOD SET 110.00 55.00 \$110 SECOND MONTH AFTER TIME PERIOD SET 360.00 180.00 \$ THIRD MONTH AFTER TIME PERIOD SET 840.00 420.00 \$ \$ FOURTH MONTH AFTER TIME PERIOD SET 1,320.00 660.00

[X] TOTAL EXTENSION FEE 110.00

[] FEE FOR EXTRA CLAIMS added by Amendment in this response:

#### Column 1 Column 2 Column 3

		Number of Claims after Amendment	Number Previously Paid for	Number of Extra Claims	RATE Non-Small Entity	RATE Small Entity	FEE
	TOTAL CLAIMS		MINUS **	* =	x 20	x 10	\$
	INDEPENDENT		MINUS ***	* ==	x 72	x 36	\$
ļ	First presentation of	of multiple depende	nt claim		+ 220	+ 110	\$

TOTAL FEE FOR EXTRA CLAIMS \$

- * If the entry in Column 1 is less than the entry of Column 2, write "0" in Column 3.
- ** If the number of Total Claims previously paid for is less than 20, write "20" in this space.
- *** If the number of Independent Claims previously paid for is less than 3, write "3" in this space.

[] Enclosed is the fee of \$_____ by Check No. _

[X] Please charge Deposit Account No. 19-2090 in the amount of \$110.00.

[X] The Commissioner is hereby authorized to charge payment of any additional fees, in particular the following fees, associated with this communication, or credit any overpayment to Deposit Account No. 19-2090:

Any filing fees under 37 C.F.R. § 1.16 for the presentation of extra claims Any patent application processing fees under 37 C.F.R. § 1.17

6/7/85 Date:

SHELDON & MAK yly J Adr y.G. Sheldon, Reg. No.: 27,953 By:

<u>CERTIFICATION OF FACSIMILE TRANSMISSION</u>: I hereby certify that this paper is being facsimile transmitted to the Patent and Trademark Office on the date shown below.

Date Transmitted: June 7, 1995

Date Signed: June 7, 1995

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SHELDON & MAK 225 SOUTH LAKE AVENUE, SUITE 800 PASADENA, CALIFORNIA 91101 (818) 796-4000 - Direct Line: (818) 356-1201

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r: //anfrailyn C. Faik

HEREBY CHERTIFY THAT TH BEING DEPOSITED WITH 250 TL SERVICE AS FIRST CLASS MAIL BOCOMMISSIONER MASHINGTON 25086 DEMARKS G 9 195-(DATE SIGNED)

JUN-07-95 15:25 FROM SHELDON & MAK ID:818 795 6321 PAGE 2/6 COPY RECEI JUN 7 199 IN THE UNITED STATES PATENT AND TRADEMARK ŐF Group Art Unit: 3301 In re application of: KENEALY, D JAMES E. JERVIS Examiner: VIA FACSIMILE Serial No. 07/956,653 (703) 305-359 Filed: October 2, 1992 For: MEDICAL DEVICES INCORPORATING SIM ALLOY Pasadena, Californ ELEMENTS CERTIFICATION OF FACSINGLE TRANSMISSION AMENDMENT I hereby certify that this paper is being facsimile transmitted to the Patent Office and Trademark Office on the date shown below. Honorable Commissioner of Patents and Trademarks lacilun Washington, D. C. 20231 ' 9S Sir: In response to the Advisory Action of April 26, 1995, please amend the above-identified application as follows: IN THE CLAIMS 0 Cancel claims 11-18, without prejudice. Applicant may present these claims in a continuation application. Please amend claim 19 as follows: اليويد (Fourth Amendment) A medical device which comprises: an element for use within a [mammalian] human body (a) or in such proximity to a [mammalian] human body that the device 1 PC3\PTO\AMD\9438-5.AMD Jun. 7, 1995

9438\MP0\$\$4-U\$7

is substantially at <u>human</u> body temperature, the element comprising a shape memory alloy which displays a stress-induced martensite behavior at body temperature; and

(b) a restraint holding the shape memory alloy element in a deformed configuration at a temperature less than the body temperature of the [mammal] <u>human</u> for positioning the shape memory alloy element within or in proximity to [a mammalian] <u>the</u> <u>human</u> body in its deformed configuration, the deformation occurring through the formation of stress-induced martensite; We wherein the <u>shape memory alloy</u> element is sufficiently deformed that <u>when the shape memory alloy</u> element is at human <u>body temperature</u> removal of the restraint from the shape memory [allow] <u>alloy</u> element, without change in temperature of the device, releases at least a portion of the <u>shape memory alloy</u> element from its deformed configuration.

### <u>REMARKS</u>

Applicant wishes to thank the Examiner for the courtesy shown to Applicant's attorney, the undersigned, during a telephone interview on May 16, 1995.

During the telephone interview, the amendments to claim 19 were discussed. The Examiner indicated that the amendments made herein to claim 19 would obviate the rejections under 35 U.S.C. § 112.

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Jun. 7, 1995



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With regard to the claims that contain the "A(90)" limitation, Applicant respectfully disagrees with the position of the Patent Office. Nevertheless, to expedite issuance of this application, the claims containing "A(90)" are cancelled by this amendment, thereby obviating the rejection as to those claims.

In the Advisory Action, the Examiner requested drawings to accompany the case. During the telephone interview, the undersigned advised the Examiner of the filing of Figs. 3 and 4, which are believed to meet the drawing requirements of 37 C.F.R.

Applicant wishes to advise the Examiner that claim 19 is not restricted to a wire. This comment is being made because the Examiner refers to a wire in the Office Action. Claim 19 only refers to a "shape memory alloy element" and a "restraint". Although one or more of these elements may be a wire, the claim does not so require.

In view of the amendments to the application and the above remarks, it is believed that this application is in condition for allowance. Such allowance is respectfully requested. Since Applicant's appeal brief was due on May 9, 1995, it is unclear to Applicant whether an extension of time is required for submitting this amendment. In an abundance of caution, such an extension is hereby requested, and an

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Jun. 7, 1995

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authorization to charge our Deposit Account 19-2090 for the extension fee is included. If for some reason the extension is not necessary, it is requested that the deposit account not be charged.

Applicant wishes to advise the Examiner that Applicant may file a continuation application that includes the "A(90)" claims, as well as a claim comparable to claim 19 that is not limited to humans. Applicant respectfully disagrees with the rejections for new matter and under § 112. However, to expedite issuance of this patent, Applicant has agreed to amend the application as indicated above.

Respectfully submitted,

SHELDON & MAK

6/7/97 Date

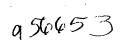
By Sheldon ev G.

Reg. No. 27,953

225 South Lake Avenue, 9th Floor Pasadena, California 91101 (818) 796-4000

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Jun. 7, 1995





# UNITED STATES DEPARTMENT OF COMMERCE Patent and Trademark Office

Address: COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231

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			DATE MAILED:	61
	j	EXAMINER INTERVIEW SUMMARY RECO		
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Edwards Exhibit 1033, p. 271

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UNITED STATES DEPARTMENT OF COMMERCE Patent and Trademark Office Address: COMMISSIONER OF PATENTS AND TRADEMARKS Washington, D.C. 20231

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	ILING DATE	FIRST NAMED APPLICANT	I ATTORNEY DOCKET NO.
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		F3M1/0706	EXAMINER
JEFFREY G.	SHELDON		
SHELDON & 1	MAK		
225 S. LAKE	E AVENUE - 9	TH FLOOR	ART UNIT PAPER NUMBER
PASADENA, (	CA 91101		
			07/14/95
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Note the attached INFOP	MATION DISCLOSURE	CITATION, PTO-1449.	
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ROBERT A. HAFER S.P.E. ART UNIT 331

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# Edwards Exhibit 1033, p. 272

Serial No. 956653 Art Unit 3301

An Examiner's Amendment to the record appears below. Should the changes and/or additions be unacceptable to applicant, an amendment may be filed as provided by 37 C.F.R. § 1.312. To ensure consideration of such an amendment, it **MUST** be submitted no later than the payment of the Issue Fee.

Authorization for this Examiner's Amendment was given in a telephone interview with Mr. Sheldon on 6/26/95.

The following changes have been made to the claims: <u>Claim 27:</u> At line 3 of subparagraph b), the phrase "the mammal" has been deleted and replaced with the phrase "a human".

At line 10 of subparagraph b), the term "mammalian" has been deleted and replaced with the term "human". Claim 28: At line 3 of subparagraph a), the term "human" has been inserted in between the term "about" and the term "body".

At line 5 of subparagraph b), the term "mammal" has been deleted and replaced with the term "human". <u>Claim 33:</u> At line 2 the term "mammalian" has been deleted and

replaced with the term "human".

At line 5 the term "human" has been inserted in between the term "about" and the term "body".

<u>Claim 35:</u> At line 1 the term "mammalian" has been deleted and replaced with the term "human".

At line 5 the term "human" has been inserted in between the term "about" and the term "body".

-2-

Serial No. 956653

Art Unit 3301

Claims 37-40 and 43: Have been cancelled. Claim 54: Has been cancelled.

<u>Claim 58:</u> At line 1 the term "mammalian" has been deleted and replaced with the term "human".

- 3 -

At line 5 the term "human" has been inserted in between the term "about" and the term "body". <u>Claim 59:</u> At line 1 of subparagraph a), the term "mammalian" has been deleted and replaced with the term "human".

At line 2 of subparagraph a), the term "mammalian" has been deleted and replaced with the term "human".

At line 4 of subparagraph a), the term "human" has qtbeen inserted in between the term "about" and the term "body".

At line 4 of subparagraph  $\mathbf{b}$ ), the term "mammalian" has been deleted and replaced with the term "human". <u>Claim 60:</u> At line 2 the term "mammalian" has been deleted and

replaced with the term "human".

At line 5 the term "human" has been inserted in between the term "about" and the term "body".

At line 12 the term "mammal" has been deleted and replaced with the term "human".

At line 15 the term "mammalian" has been deleted and replaced with the term "human".

<u>Claim 62:</u> At line 2 the term "mammalian" has been deleted and replaced with the term "human".

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Serial No. 956653 Art Unit 3301

At line 5 the term "human" has been inserted in between the term "about" and the term "body".

At line 12 the term "mammal" has been deleted and replaced with the term "human".

At line 14 the term "mammalian" has been deleted and replaced with the term "human".

Any inquiry concerning this communication or earlier communications from the examiner should be directed to David Kenealy whose telephone number is (703) 308-2680.

David J. Kenealy June 28, 1995

ROBERT A. HAFER

S.P.E. ART UNIT 331

Edwards Exhibit 1033, p. 275

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Edwards Exhibit 1033, p. 278



IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In re application of

) Group No.: 27,953

Kenealy, D.

9438

James E. Jervis

Serial No.: 07/956,653 Filed: October 2, 1992

Batch No. N17

Examiner:

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

### TRANSMITTAL OF FORMAL_DRAWINGS

Patent and Trademark Office Washington, D.C. 20231

Attention: Official Draftsman

8/14/95

Dear Sir:

Please find (2) sheets of formal drawing(s) for this application. Each sheet of drawing indicates the serial number and Group Art Unit on the reverse side of the drawing.

Respectfully submitted,

SHELDON & MAK, INC.

Date:

G. Sheldon Jef Req. No. 27,953

SHELDON & MAK 225 South Lake Avenue 9th Floor Pasadena, California 91101 (818) 796-4000

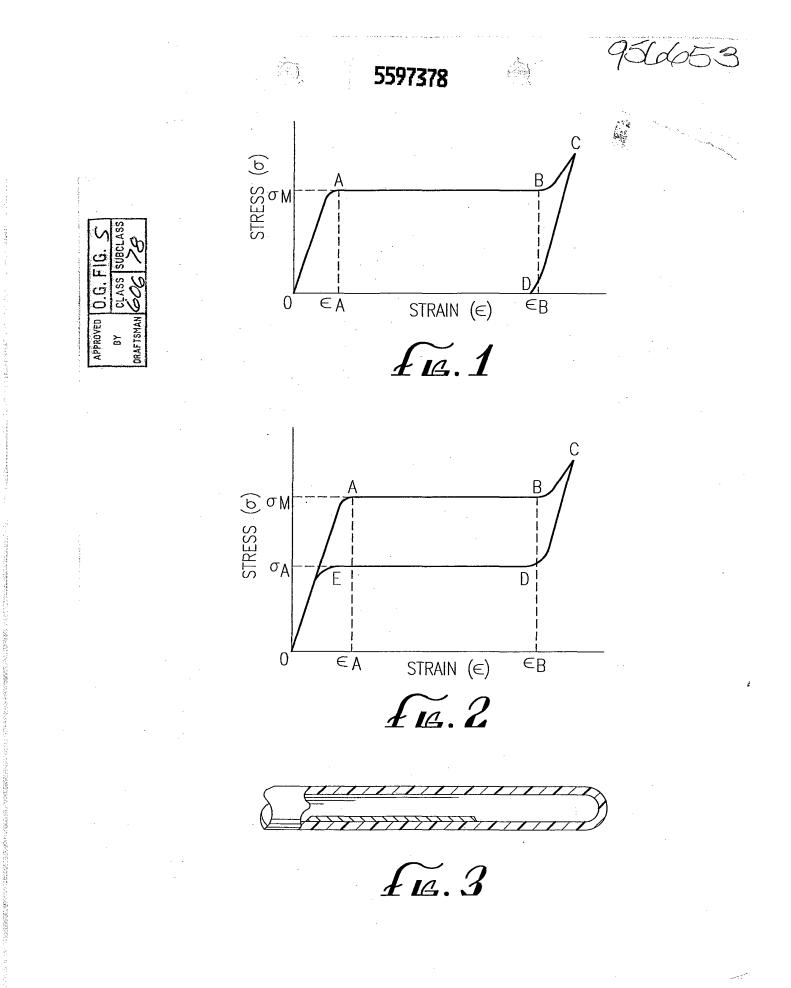
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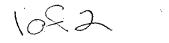
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Edwards Exhibit 1033, p. 280

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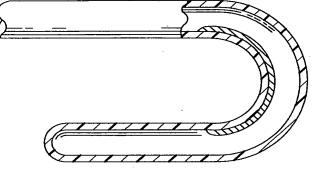


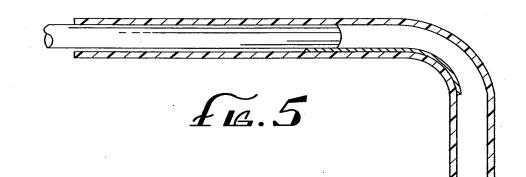
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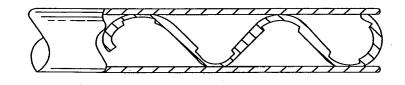




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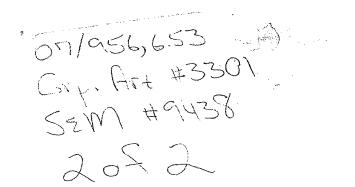






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Edwards Exhibit 1033, p. 282





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1421 PART B-ISSUE FEE TRANSMITTAL MAILING IN TRUCTIONS: This form should be us. a for transmitting the ISSUE FEE. Blocks 2 through 6 should be completed where appropriate entering in Block 1 unless you direct otherwise, by: (a) specifying a new correspondence address in Block 3 below; or (b) providing the PTO with a separate "FEE ADDRESS" for maintenance fee notifications with the payment of Issue Fee or thereafter. See reverse for Certificate of Mailing. #1 2. INVENTOR(S) ADDRESS CHANGE (Complete only if there is a change) 1. CORRESPONDENCE ADDRESS INVENTOR'S NAME ROOL Street Address -. ? City, State and ZIP Code F3M1/0706 REY G. SHELDON CO-INVENTOR'S NAME DON & MAK Street Address 25 S. LAKE AVENUE - 9TH FLOOR PASADENA, CA 91101 City, State and ZIP Code Check if additional changes are on reverse side EXAMINER AND GROUP ART UNIT DATE MAILED SERIES CODE/SERIAL NO. FILING DATE TOTAL CLAIMS FMEAL 07/06/ 07/95/ 0702792 First Named Applicant JAMES. 1 TITLE OF INVENTION MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS CLASS-SUBCLASS BATCH NO. APPLN. TYPE SMALL ENTITY FEE DUE DATE DUE ATTY'S DOCKET NO. 9438 606-078.000 N17 HTTI ITY MD \$1210. ΩĤ 10706 3. Correspondence address change (Complete only if there is a change) 4. For printing on the patent front 1 Jeffrey G. Sheldon page, list the names of not more than 3 registered patent attorneys or agents OR, alternatively, the name of a firm 2 SHELDON & MAK, INC. having as a member a registered attorney or agent. If no name is listed, no name will be printed. з DO NOT USE THIS SPACE 080 VM 08/28/95 07956653 1,210.00 CK 1 142 080 VM 08/28/95 07956653 1 561 30.00 CK 5. ASSIGNMENT DATA TO BE PRINTED ON THE PATENT (print or type) (1) NAME OF ASSIGNEE: Raychem Corporation check #5968 for \$1,240. 6a. The following fees are enclosed: K Advance Order - # of Copies 10 (2) ADDRESS: (CITY & STATE OR COUNTRY) K Issue Fee Menlo Park, California (ENCLOSE PART C) Advance Order - # of Copies 🗌 issue Fee A. 
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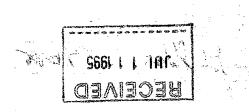
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Has received an application for a patent for a new and useful invention. The title and description of the invention are enclosed. The requirements of law have been complied with, and it has been determined that a patent on the invention shall be granted under the law.

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## United States Patent

Grants to the person or persons having title to this patent the right to exclude others from making, using or selling the invention throughout the United States of America for the term of seventeen years from the date of this patent, subject to the payment of maintenance fees as provided by law.

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Commissioner of Patents and Trademarks

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# IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

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In re the Application of:

JAMES E. JERVIS

Serial No. 07/956,653

Filed: Otober 2, 1992

For: MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS Group Art Unit: 3301

Examiner: Kenealy, D.

Raychem Corporation 300 Constitution Drive Menlo Park, CA 94025

February 1, 1996

## POWER TO INSPECT AND MAKE COPIES

Honorable Assistant Commissioner for Patents Washington, D.C. 20231

Sir:

This communication gives Shonda Reed-Baten and Jennifer Harris of Barbara Harris & Associates the right to inspect and make copies of our patent application, titled Medical Devices Incorporating Sim Alloy Elements, U.S. Serial No. 07/956,653, filed October 2, 1992.

If you have any questions or concerns, please feel free to call the undersigned at (415) 361-3338.

Respectfully submitted, mhad

Herbert G. Burkard Registration No. 24,500 Tel. No. (415) 361-3338

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# **INFORMATION ON HOW TO EFFECT DRAWING CHANGES**

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#### 1. Correction of Informalities-37 CFR 1.85

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File new drawing with the changes incorporated therein. The art unit number, serial number and number of drawing sheets should be written on the drawing in accordance with 37 CFR 1.84()). Applicant may delay filing of the new drawings until receipt "Notice of Allowability" (PTOL-37). If delayed, the new drawings **MUST** be Filed within the **THREE MONTH** shortened statutory period set for response in the "Notice of Allowability" (PTOL-37). Extensions of time may be obtained under the provisions of 37 CFR 1.136. The drawing should be filed as a separate paper with a transmittal letter addressed to the Official Draftsperson.

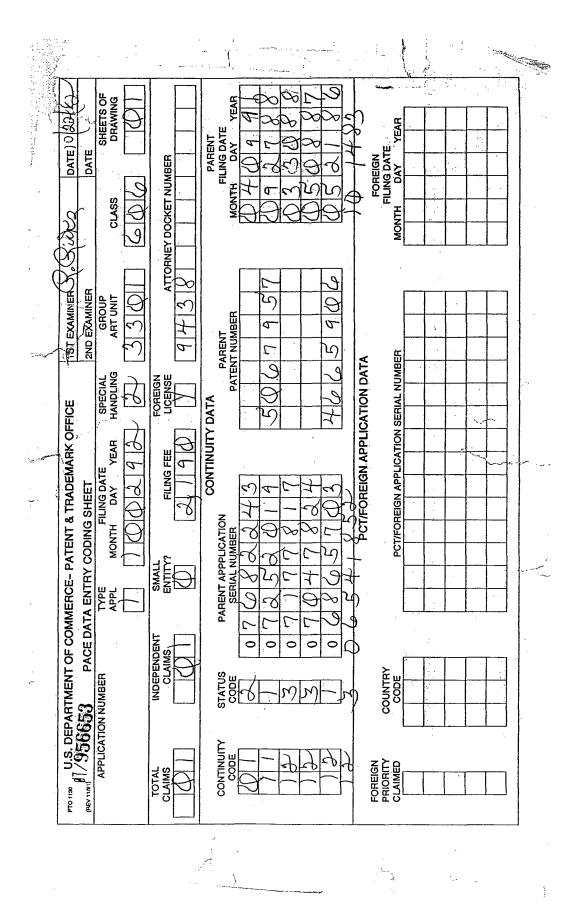
#### **Timing of Corrections**

Applicant is required to submit acceptable corrected drawings within the three month shortened statutory period set in the "Notice of Allowability" (PTOL-37). Within the three month period, two weeks should be allowed for review by the Office of the correction. If a correction is determined to be unacceptable by the Office, applicant must arrange to have acceptable correction re-submitted within the original three month period to avoid the necessity of obtaining an extension of time and paying the extension fee. Therefore, applicant should file corrected drawings as soon as possible.

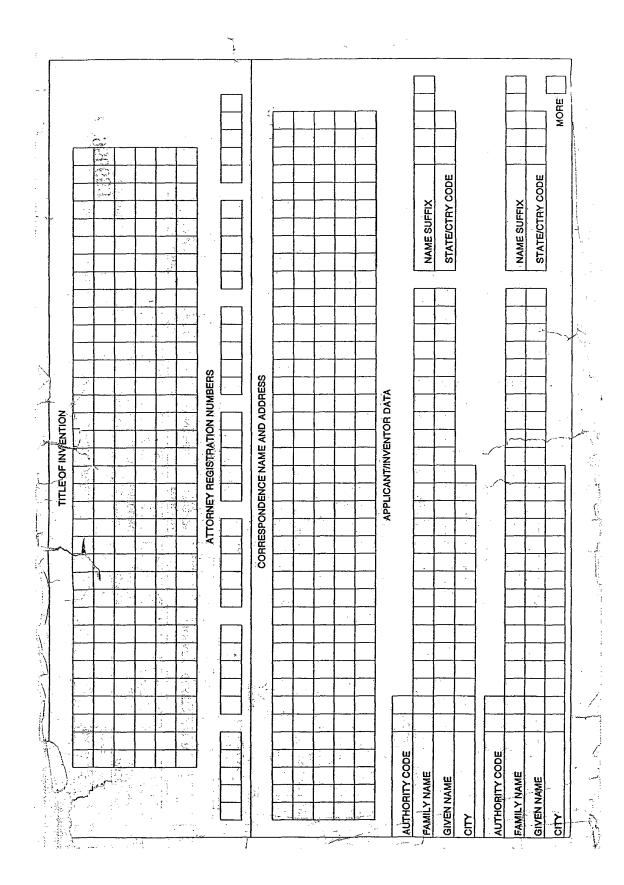
Failure to take corrective action within set (or extended) period will result in ABANDONMENT of the Application.

#### 2. Corrections other than Informalities Noted by the Draftsperson on the PTO-948

All changes to the drawings, other than informalities noted by the Draftsperson, **Must** be made in the same manner as above except that, normally, a red ink sketch of the changes to be incorporated into the new drawings **MUST** be approved by the examiner before the application will be allowed. No changes will be permitted to be made, other than correction of informalities, unless the examiner has approved the proposed changes.



Edwards Exhibit 1033, p. 291





# United States Patent [19]

# Jervis

[56]

## [54] MEDICAL DEVICES INCORPORATING SIM ALLOY ELEMENTS

- [75] Inventor: James E. Jervis, Atherton, Calif.
- [73] Assignce: Raychem Corporation, Menlo Park, Calif.
- [*] Notice: The portion of the term of this patent subsequent to May 19, 2004, has been disclaimed.
- [21] Appl. No.: 956,653
- [22] Filed: Oct. 2, 1992

#### **Related U.S. Application Data**

[62] Division of Ser. No. 682,243, Apr. 9, 1991, Pat. No. 5,190, 546, which is a division of Ser. No. 252,019, Sep. 27, 1988, Pat. No. 5,067,957, which is a continuation of Ser. No. 177,817, Mar. 30, 1988, abandoned, which is a continuation of Ser. No. 47,824, May 8, 1987, abandoned, which is a continuation of Ser. No. 865,703, May 21, 1986, Pat. No. 4,665,906, which is a continuation of Ser. No. 541,852, Oct. 14, 1983, abandoned.

[51]	Int. Cl.	б 	A61B 17/58

- [52]
   U.S. Cl.
   606/78; 606/76; 604/281

   [58]
   Field of Search
   606/78, 76, 77;

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# [45] Date of Patent: *Jan. 28, 1997

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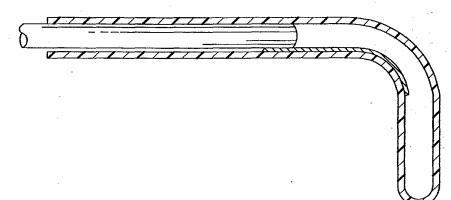
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Primary Examiner-Robert A. Hafer Assistant Examiner-David J. Kenealy Attorney, Agent, or Firm-Jeffrey G. Sheldon; Sheldon & Mak, Inc.

#### ABSTRACT

Medical devices which are currently proposed to use elements made form shape memory alloys may be improved by the use of stress-induced martensite alloy elements instead. The use of stress-induced martensite decreases the temperature sensitivity of the devices, thereby making them easier to install and/or remove.

# 40 Claims, 2 Drawing Sheets



[57]

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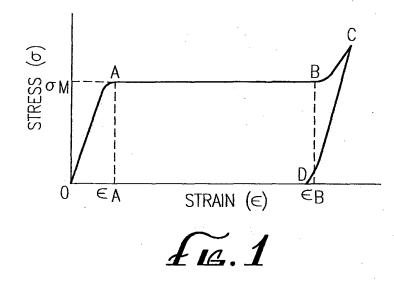
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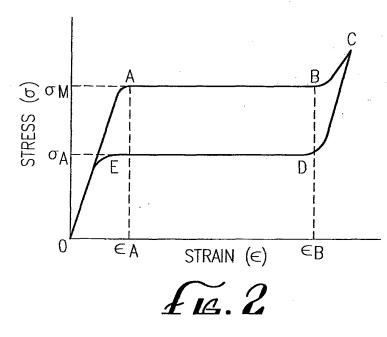
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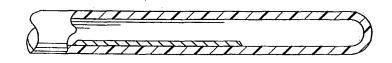
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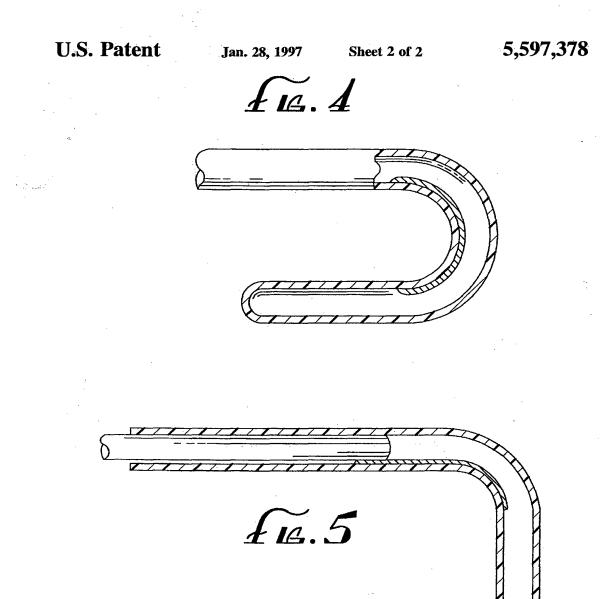


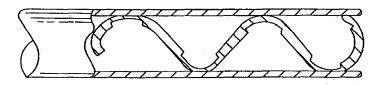






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# MEDICAL DEVICES INCORPORATING SIM **ALLOY ELEMENTS**

## CROSS-REFERENCE TO RELATED **APPLICATIONS**

This application is a divisional of application Ser. No. 682,243 filed on Apr. 9, 1991, now U.S. Pat. No. 5,190,546, which is a divisional of Ser. No. 252,019 filed on Sep. 27, 1988, now U.S. Pat. No. 5,067,957, which is a continuation 10 of Application Ser. No. 177,817 filed Mar. 30, 1988, now abandoned; which is a continuation of Application Ser. No. 047,824 filed May 8, 1987, now abandoned; which is a continuation of Application Ser. No. 865,703 filed May 21, 1986, now U.S. Pat. No. 4,665,906; which is a continuation 15 of Application Ser. No. 541,852 filed Oct. 14, 1983, now abandoned.

# BACKGROUND OF THE INVENTION

1. Field of the Invention

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This invention relates to medical devices incorporating shape memory alloys, and to improvements therein.

2. Introduction to the Invention

25 Materials, both organic and metallic, capable of possessing shape memory are well known. An article made of such materials can be deformed from an original, heat-stable configuration to a second, heat-unstable configuration. The article is said to have shape memory for the reason that, upon  $_{30}$ the application of heat alone, it can be caused to revert, or to attempt to revert, form its heat-unstable configuration to its original, heat-stable configuration, i.e. it "remembers" its original shape.

Among metallic alloys, the ability to possess shape 35 memory is a result of the fact that the alloy undergoes a reversible transformation from an austenitic state to a martensitic state with a change in temperature. This transformation is sometimes referred to as a thermoelastic martensitic transformation. An article made from such an alloy, for  $_{40}$ example a hollow sleeve, is easily deformed from its original configuration to a new configuration when cooled below the temperature at which the alloy is transformed from the austenitic state to the martensitic state. The temperature at which this transformation begins is usually referred to as  $M_s$  45 and the temperature at which it finishes Mr. When an article thus deformed is warmed to the temperature at which the alloy starts to revert back to austenite, referred to as As (Ar being the temperature at which the reversion is complete) the deformed object will begin to return to its original 50 configuration.

Many shape memory alloys (SMAs) are known to display stress-induced martensite (SIM). When an SMA sample exhibiting stress-induced martensite is stressed at a temperature above  $M_s$  (so that the austenitic state is initially stable), 55 but below  $M_d$  (the maximum temperature at which martensite formation can occur even under stress) it first deforms elastically and then, at a critical stress, begins to transform by the formation of stress-induced martensite. Depending on whether the temperature is above or below  $A_s$ , the behavior 60 when the deforming stress is released differs. If the temperature is below A_s, the stress-induced martensite is stable; but if the temperature is above As, the martensite is unstable and transforms back to austenite, with the sample returning (or attempting to return) to its original shape. The effect is 65 seen in almost all alloys which exhibit a thermoelastic martensitic transformation, along with the shape memory

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effect. However, the extent of the temperature range over which SIM is seen and the stress and strain ranges for the effect vary greatly with the alloy.

In copending and commonly assigned U.S. patent application to Quin, now U.S. Pat. No. 4,503,767 the disclosure of which is incorporated herein by reference, a nickel/ titanium/vanadium alloy having SIM over a wide temperature range is disclosed.

Shape memory alloys have found use in recent years in, for example, pipe couplings (such as are described in U.S. Pat. Nos. 4,035,007 and 4,198,081 to Harrison and Jervis), electrical connectors (such as are described in U.S. Pat. No. 3,740,839 to Otte & Fischer), switches (such as are described in U.S. Pat. No. 4,205,293), actuators, etc.

Various proposals have also been made to employ shape memory alloys in the medical field. For example, U.S. Pat. No. 3,620,212 to Fannon et al. proposes the use of an SMA intrauterine contraceptive device, U.S. Pat. No. 3,786,806 to Johnson et al. proposes the use of an SMA bone plate, U.S. Pat. No. 3,890,977 to Wilson proposes the use of an SMA element to bend a catheter or cannula, etc.

These medical SMA devices rely on the property of shape memory to achieve their desired effects. That is to say, they rely on the fact that when an SMA element is cooled to its martensitic state and is subsequently deformed, it will retain its new shape; but when it is warmed to its austenitic state, the original shape will be recovered.

However, the use of the shape memory effect in medical applications is attended with two principal disadvantages. First, it is difficult to control the transformation temperatures of shape memory alloys with accuracy as they are usually extremely composition-sensitive, although various techniques have been proposed (including the blending by powder metallurgy of already-made alloys of differing transformation temperatures: see U.S. Pat. No. 4,310,354 to Fountain et al.). Second, in many shape memory alloys there is a large hysteresis as the alloy is transformed between austenitic and martensitic states, so that reversing of the state of an SMA element may require a temperature excursion of several tens of degrees Celsius. The combination of these factors with the limitation that (a) it is inconvenient to have to engage in any temperature manipulation, and (b) human tissue cannot be heated or cooled beyond certain relatively narrow limits (approximately 0°-60° C. for short periods) without suffering temporary or permanent damage is expected to limit the use that can be made of SMA medical devices. It would thus be desirable to develop a way in which the advantageous property of shape memory alloys, i.e. their ability to return to an original shape after relatively substantial deformation, could be used in medical devices without requiring the delicacy of alloying control and/or the temperature control of placement or removal needed by present shape memory alloy devices.

#### DESCRIPTION OF THE INVENTION

Summary of the Invention

I have discovered that if, in a medical device containing a shape memory alloy element which uses the shape memory property of that alloy, an element which shows the property of stress-induced martensite is used instead, an improved device results.

Accordingly, this invention provides a medical device intended for use within a mammalian body, or in such proximity to a mammalian body that the device is substantially at body temperature, which device comprises a shape

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memory alloy element, the improvement in which comprises the substitution of an alloy element which displays stressinduced martensite at said body temperature for the shape memory alloy element.

# BRIEF DESCRIPTION OF THE DRAWING

FIGS. 1 and 2 illustrate the stress-strain behavior of an alloy which exhibits constant stress versus strain behavior due to stress-induced martensite.

FIG. 3 is a side elevation view of a partial section of a catheter of the present invention in a stressed configuration.

FIG. 4 is a side elevation view of the catheter of FIG. 3 in an unstressed configuration. 15

FIG. 5 is a tracheal catheter, which is curved in its unstressed configuration, partially straightened by a straight pin restraint.

FIG. 6 shows an IUD formed at least partly from a pseudoelastic shape-memory alloy being restrained in a 20 deformed shape by a restraining tube.

# DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

The invention will be discussed first by introducing the concept of stress-induced martensite and the effect achievable by its use, and then by examples showing how SIM alloy elements can be substituted for conventional SMA 30 elements in medical devices to achieve the beneficial effect of the invention.

The Figures illustrate the phenomenon of stress-induced induced martensite by means of stress-strain curves. In both FIG. 1 and FIG. 2, the alloy is at a temperature between  $M_{s-35}$ and  $M_d$  so that it is initially austenitic; and it will be assumed for the purposes of this discussion that Ms is equal to Mo and  $A_s$  equal to  $A_f$  FIG. 1 shows the case when the temperature is below A_s, so that any martensite formed by the applied stress is stable; while FIG. 2 shows the case where the 40temperature is above  $A_s$ , so that austenite is the only stable phase at zero stress.

In FIG. 1, when a stress is applied to the alloy, it deforms elastically along the line OA. At a critical applied stress,  $\sigma_M$ , the austenitic alloy begins to transform to (stress-induced) 45 martensite. This transformation takes place at essentially constant stress until the alloy becomes fully martensitic at point B. From that point on, as further stress is applied, the martensite yields first elastically and then plastically (only elastic deformation is shown at point C). When the stress is 50 released, the martensite recovers elastically to point D, at which there is zero residual stress, but a non-zero residual strain. Because the alloy is below As, the deformation is not recoverable until heating above A, results in a reversion to austenite. At that point, if the sample is unrestrained, the 55 original shape will be essentially completely recovered: if not, it will be recovered to the extent permitted by the restraint. However, if the material is then allowed to re-cool to the original temperature at which it was deformed (or a temperature where SIM behavior of this type is seen), the 60 stress produced in the sample will be constant regardless of the strain provided that the strain lies within the "plateau" region of the stress-strain curve. That is, for a stress between  $\epsilon_B$  and  $E_A$ , the strain will be  $\sigma_M$ . This means that a known, constant force (calculable from  $\sigma_{M}$ ) can be applied over a 65 wide (up to 5% or more for certain Ni/Ti alloys) strain range. Thus, though this resembles the conventional shape memory

effect, because the alloy shows SIM and is below A, a constant force can be achieved.

In FIG. 2, when a stress is applied to the alloy, it deforms elastically along line DA, then by SIM along line AB, and by deformation of the martensite to point C, just as in FIG. 1. However, the stress-strain behavior on unloading is significantly different, since the alloy is above A, and the stable phase is therefore austenite. As the stress is removed, the alloy recovers elastically from C to D: then, at a critical stress,  $\sigma_A$ , the alloy reverts to austenite without requiring a change in temperature. Thus reversion occurs at essentially constant stress. Finally if the stress is removed from the reverted austenite, it recovers elastically along line EO. The recoverable deformation associated with the formation and reversion of stress-induced martensite has been referred to as pseudoelasticity. While  $\sigma_M$  may be comparatively high, e.g. 50 ksi,  $\sigma_A$  is usually substantially lower, e.g. less than 10 ksi; thereby creating a constant-force spring with an effective working range of about 5% ( $\epsilon_B - \epsilon_A$ ). The shape change available in the SMA is thus mechanically, rather than thermally, actuated and controlled, permitting a greater control over a device incorporating it.

Suitable alloy for this invention i.e. those displaying stress-induced martensite at temperatures near mammalian body temperature (35°-40° C.), may be selected from known SMAs by those of ordinary skill in the art having regard to this disclosure by testing for the existence of the SIM effect at the desired temperature. A particularly preferred alloy is the nickel/titanium/vanadium alloy of U.S. patent application No. now U.S. Pat. No. 4,505,767 referred to previously.

The following table sets forth transformation temperature data for alloys disclosed in U.S. Pat. No. 4,505,767: TADIE

	Comp			
Ni	Ti	v	Ms	A(90)
49.50	43.50	7.00	-107	-88
50.00	44.00	6.00	-96	84
49.00	43.00	8.00	-83	-61
50.00	45.00	5.00	-42	-33
49.00	45.00	6.00	-35	-12
50.50	48.00	1.50	-32	-6
48.50	44.50	7.00	-30	-13
50.00	46.00	4.00	-11	7
48.50	45.00	6.50	-10	15
49.00	45.50	5.50	-10	14
48.00	44.25	7.75	7	8
48.50	45.50	6.00	-5	. 27
41.50	38.50	20.00	2	-86
46.50	43.50	10.00	1	50
36.25	33.75	30.00	0	42
49.50	46.00	4.50	6	35
48.00	46.00	6.00	12	36
47.75	45.75	6.50	20	54
47.50	45.50	7.00	26	58
48.50	46.50	5.00	27	58
45.00	45.00	10.00	30	71
47.50	46.50	6.00	32	71
46.50	46.50	7.00	34	70

The invention will now be discussed in detail by some Examples of the use of an SIM alloy.

## EXAMPLE I

Heart Valves

Akins, in U.S. Pat. No. 4,233,690, the disclosure of which is incorporated herein by reference, describes the use of a

shape memory alloy ring to hold a sewing cuff to the body of an artificial heart valve. The ring is made in the austenitic phase, cooled to the martensitic phase, deformed, placed around the valve body, and heated or allowed to warm to cause reversion to the austenitic phase and recovery of the 5 ring into engagement with the valve body.

However, this technique has not found commercial acceptance. Present medical technique requires that the valve body be capable of being rotated relative to the cuff, thereby enabling the surgeon to set the rotational orientation of the ¹⁰ valve after it has been sewn into place. This is desirable because the techniques used make it difficult to visualize or accomplish optimal orientation during initial placement.

In order to accomplish the desired torque control to permit the desired rotation and yet ensure a firm hold of the cuff on the valve body, precise control of the pressure exerted on the valve body by the ring is needed. This is difficult because there are substantial manufacturing tolerances in the valve body which may be made, for example, of pyrolytic graphite or ceramics, etc. Because the austenite stress-strain curve is extremely steep, it is not considered practical to use the simple shape memory technique proposed by Akins. Indeed, Akins does not even address the issue of rotation of the cuff with respect to the valve body.

However, if an SIM alloy is used instead of conventional shape memory, the process may be considerably simplified.

First, if the alloy has a stress-strain curve like that of FIG. 1, the alloy ring may be made just as for Akins. The ring is then expanded from its initial austenitic state by the formation of SIM. When the ring is placed about the valve body, it needs only to be heated above  $A_f$  and allowed to cool to its original temperature for the ring to engage the valve body with a constant force, even if the valve body has a deviation from the specified size. The torque may thus be controlled to the desired level despite manufacturing tolerances.

Second, if the alloy has a stress-strain curve like that of FIG. 2, the ring may be expanded, placed over the valve body, and the stress released all at the same temperature. Because the austenitic phase is stable, the stress-induced 40 martensite spontaneously reverts to austenite until recovery is restrained by the ring engaging the valve body. Because the reversion to austenite takes place at constant stress, a constant force (and hence constant torque) may be obtained regardless of manufacturing tolerances. Close temperature 45 control is not required, either; and the fact that the patient in a heart valve replacement operation is conventionally cooled as much as 15° C. or so below normal body temperature does not affect the operation of the ring.

To control the torque at a sufficiently low level, it may be ⁵⁰ desirable for the alloy ring to be other than a solid ring, such as, for example, a continuous helical spring, a flat zigzag spring, etc. Such variations permit the achievement of a greater range of movement with constant force and a reduction in the force exerted by the ring on the value body, since ⁵⁵ the ring recovers in a bending mode rather than in tension.

#### EXAMPLE II

Catheters and Cannulas

Wilson, in U.S. Pat. No. 3,890,977, the disclosure of which is incorporated herein by reference, discloses a catheter or cannula (both being included hereinafter in the word "catheter") made of, or containing, an SMA element to cause 65 all or a portion of the catheter to deploy in a useful form once introduced into a living body.

However, again this device has not been commercialized. Possible defects of the device which have prevented commercialization include (i) the inability to slowly emplace the catheter in a desired position when the transition temperature of the alloy is below body temperature (since the SMA element will attempt to revert to its original shape as it reaches body temperature), thus limiting the ability of the physician to place the device carefully and precisely; or alternatively, if the transition temperature of the alloy is above body temperature, the requirement that the device be heated to a temperature above body temperature to cause recovery and that the device be placed so as not to change shape again when it re-cools (since the body temperature is below the transition temperature); (ii) the inability to remove the device easily: and (iii) the need for controlled temperature storage to prevent premature reversion to austenite of the SMA, with consequent shape change.

The issue of removal of a catheter is especially significant, and not addressed by Wilson. Consider, for example, a tracheal puncture catheter. This should be straight for easy insertion into the trachea through a puncture into the front of the neck, but should curve after insertion so that the flow of air or oxygen through the catheter passes axially down the trachea rather than impinging on the surface of the trachea and damaging it. If a shape memory catheter is used as contemplated by Wilson, it would presumably become austenitic and bend after insertion (see FIGS. 1a and 1b, and corresponding text, of Wilson). But removal would require either cooling to below the transition temperature (which could easily mean cooling to so low a temperature that the tracheal tissue is damaged), removal in the bent shape (presumably damaging tissue), or forcing the austenitic SMA to straighten to permit direct removal (unlikely to be satisfactory since the austenitic alloys e.g. of Ni/Ti may have vield strengths of 100 ksi or more, and force sufficient to cause plastic deformation would be required).

If an SMI element is used instead, however, removal can be accomplished almost as easily as insertion. If the catheter is made in a bent shape (as in Wilson), it can be straightened by insertion of a straight pin down the catheter axis, the catheter deforming by the formation of stress-induced martensite. Insertion of the catheter into the trachea is accomplished while the catheter is straight, at whatever rate is desired (permitting easy and accurate placement), and the pin is gradually withdrawn to permit the catheter to take up its desired shape as the martensite reverts to austenite. [It is assumed here that the stress-strain curve of the alloy at the temperature of use is of the form of FIG. 2, so spontaneous reversion occurs on removal of the stress induced by the pin]. When removal is desired, it may be achieve simply by the gradual insertion of the pin, straightening the catheter and permitting easy withdrawal. Insertion of the catheter into the body and pin removal may, of course, take place simultaneously if desired, as may pin reinsertion and removal of the catheter from the body.

# EXAMPLE III

# 60 IUDS

Fannon et al., in U.S. Pat. No. 3,620,212, the disclosure of which is incorporated herein by reference, discloses an intrauterine contraceptive device (an IUD) proposed to be formed of a shape memory alloy. The device is suggested to be deformed in the martensitic phase (the transition temperature being below the temperature of the uterus), and the deformed device insulated with, e.g., wax and inserted. Removal is contemplated only by using two SMA elements in opposition, the higher temperature one being martensitic at body temperature but strong enough so that, if heated, it will overcome the lower temperature element and deform the IUD back to a removable shape. The heating contemplated is electrical. The storage problem discussed in Example II also exists here, so that the device must be stored below its transition temperature.

By the use of an SIM element, however, these disadvantages may be overcome. Again, assume that the alloy is SIM 10 psuedoelastic, i.e. that it has the stress-strain curve of FIG. 2. Then an IUD may be formed into the desired shape in the austenitic state, and deformed by compression into a tubular placement device (the deformation being such that the strain levels lie within the "plateau" of the stress-strain curve). The 15 IUD has a longitudinal dimension and a transverse dimension. When the placement device is inserted into the uterus, the IUD may be deployed by extrusion of the IUD from the placement device. Deployment is then controlled but immediate, so that the physician may satisfy himself with place- 20 ment. Removal is the reversal of placement: the placement device is inserted into the uterus, the IUD deformed by withdrawal into the placement device, and the placement device withdrawn. Temperature control is not required. 25

#### EXAMPLE IV

#### **Bone Plates**

Johnson et al., in U.S. Pat. No. 3,786,806, the disclosure of which is incorporated herein by reference, propose the use 30 of Ni/Ti SMA bone plates in fracture fixation. The plate is deformed in its martensitic state, screwed to the two ends of the bone it is desired to compress together, and warmed (or allowed to warm) to the austenitic state, when the plate contracts, compressing the bone ends together. The Johnson 35 et al. bone plate is of generally oblong configuration, overlaps a bone fracture and is secured by two screws to one portion of the bone and by two other screws to the other portion of the bone.

Because of the high elastic moduli of the austenitic shape ⁴⁰ memory alloys, it will be difficult to control the amount of force which may be applied by a bone plate of the type proposed by Johnson et al., and precision placement of the bone ends and elongation of the plate will be required.

If, however, an SIM pseudoelastic bone plate is used, it ⁴⁵ will be easily possible to elongate the plate and fasten it to the bone ends without requiring high precision. Because of the comparatively large (e.g. 5%) strain range at essentially constant stress, the force which will be put on the bone ends to compress them will be readily adjustable (by the size of ⁵⁰ the plate, for example) and will be insensitive to precise placement of the bone ends and/or elongation of the plate. Also, the recovery of the plate, since it is controlled by mechanical restraint, may be as gradual as desired, achieving excellent force and time control, and permitting the ⁵⁵ surgeon to make adjustments as desired.

#### EXAMPLE V

#### Marrow Nails

Baumgart et al., in U.S. Pat. No. 4,170,990, the disclosure of which is incorporated herein by reference, discloses the use of the two-way shape memory effect (where an SMA element exhibits a first shape in the austenitic state and a second in the martensitic state, and spontaneously changes 65 between the two shapes with a change in temperature) in, inter alia, implants such as marrow nails (see FIGS. 1*a*  through 1e, and corresponding text, of Baumgart et al.) Marrow nails according to Baumgart et al. comprise a tube of memory alloy which has been split along its longitudinal axis and which may have a circular, elliptical, clover-leaf on other rotation preventing cross section, which may also be variable along the axis of the nail. A prepared marrow nail having a reduced diameter is loosely inserted into a slightly, or not at all, pre-drilled marrow channel of a bone which has been broken or fractured. By means of a heating probe the marrow nail is heated and thus expands. This achieves a relative fixing of the two bone ends along the marrow channel axis. Compression of the fracture is effected by the available muscle tension. If it should be necessary, the marrow nail may also be additionally prestretched along its longitudinal axis so that it is additionally compressed in the longitudinal direction when heated. In this case it is necessary, however, to anchor the nail at both of its ends which anchoring can be effected, for example, by sprockets or teeth on the outer surface of the nail.

The method proposed, however, requires the use of a wide temperature range in order to cause the phase change which is the origin of the two-way shape memory effect ( $5^{\circ}$  C. to  $60^{\circ}$  C. for the water used to cool or heat the nail). In addition, it requires the manufacture of two-way shape memory elements, which is generally more complex than the manufacture of conventional shape memory elements; and precise control of the transition temperature is required.

However, if an SIM pseudoelastic alloy element is employed, these disadvantages may be overcome. If internal tangs, which may be gripped by an inserted tool, are provided within a marrow nail of the type shown in FIG. 1a of Baumgart et al., then the nail may be radially compressed by the application of stress by such a tool. When the nail is released by the tool, it will expand to fill the bone channel with a constant force (not readily available by Baumgart et al.); and it may be withdrawn by the reverse procedure.

#### EXAMPLE VI

# Dental Arch Wire

Andreasen, in U.S. Pat. No. 4,037,324, the disclosure of which is incorporated herein by reference, proposes the use of dental arch wires made of Ni/Ii alloys instead of conventional 18-8 stainless steel wires. The wires are stated to be of lower elastic modulus and higher elastic limit than stainless steel, which is stated to be advantageous. Heat recovery of an SMA wire is also suggested as a technique for orthodonture.

The technique of using the conventional shape memory effect is not believed to have found clinical application, possibly because such a technique would require rapid placement of the wire in its martensitic state to avoid premature recovery, and would result in rapid recovery with extremely high forces, which would be painful for the patient.

The use of a wire which displays lower elastic modulus and higher elastic limit than stainless steel has found some application, however. Otsuka et al. in Metals Forum, v. 4, pp. 142-52 (1981) have suggested that this behavior may be the result of elasticity enhanced by cold working and martensite-to-martensite psuedoelasticity in an alloy which has a transition temperature below body temperature. The alloy, then, is martensitic rather than austenitic in its undeformed state.

While the use of an enhanced elasticity wire may offer some advantages over the more usual stainless steel wire, it

remains the situation that the amount of motion in the teeth that may be produced by an arch wire without further adjustment is largely limited by the pain tolerance of the patient (since the force applied by the arch wire is proportional to the deformation of the wire). However, is an SIM 5 pseudoelastic wire is used, it can exert a relatively constant force (chosen by the dentist to be sufficient to cause tooth movement but not painful) over a strain range of up to 5%. The load may be applied mechanically, and is thus more readily established, and no precise temperature control of the 10 alloy is needed as would be required for the shape memory effect.

# EXAMPLE VII

# Coil Stents and Filters

The use of tubular coiled wire stent grafts has been discussed in the medical literature since 1969. Although the coils helped maintain patency of the vessels in which they were placed, they were difficult of insertion unless narrow ²⁰ enough to significantly narrow the lumen of the vessel. Recently it has been proposed, see Radiology, v. 147, pp. 259-60 and pp. 261-3 (1983), the disclosures of which are incorporated herein by reference, to use SMA wire to form these tubular coils. The wire, which has a transformation ²⁵ temperature below body temperature, is introduced through a catheter after being straightened in its martensitic state. When the wire is heated, the coil re-forms.

Because of the difficulty of controlling the transformation temperature accurately, it has proved necessary to cool the straightened wire during insertion and/or to heat the wire to form the coil after insertion. These procedures add to the complexity of the operation.

If an SIM pseudoclastic wire is used to form the coil, which is then isothermally deformed by loading into a catheter, then the need for temperature control is avoided. The wire remains straight when in the catheter, but re-forms the coil spontaneously when it is extruded from the catheter. Accurate placement is thus readily obtainable, since there is no urgency as might be required with a conventional shape memory effect element.

It has similarly been proposed to use SMA wire to form a filter for emplacement by catheter in the vena cava to trap blood clots. The filter is formed in the austenitic state, the wire straightened in the martensitic state and inserted, and the filter re-forms on warming. Just as for the coil stents discussed above, the use of an SIM pseudo-elastic wire would greatly simplify manufacture and insertion of such a vena cava filter, permitting accurate placement with no need for urgency or temperature manipulation.

#### EXAMPLE VIII

# Bone Staples, Clips, etc.

Bone staples are frequently used to hold fragments of fractured bone together when the fracture is fixed, and may be used in some cased as a replacement for bone plates in the same situation. Sometimes the staples are inserted into drilled holes, sometimes merely driven into the bone  $_{60}$  directly.

It would be desirable to have a bone staple which provided a controlled force between the tines which would tend to hold the staple in place. Shape memory alloys have been proposed for this application, but again the problem of 65 accurate placement while operating quickly enough to prevent the shape change associated with the martensite-toaustenite transition and/or the need for temperature control complicate their use.

If an SIM alloy is used, these disadvantages may be readily overcome. If the alloy is below  $A_{2}$ , it may be emplaced in the martensitic state. Brief heating will then be required to cause it to become austenitic, but on re-cooling to body temperature, a constant force can be achieved. If the alloy is above  $A_{2}$ , the staple can be held deformed by a moderate force, then released after insertion to also provide an accurately-known force. In either event, removal is easier than if the alloy is purely austenitic, as discussed above for Examples II and V, for example.

Similarly, SIM alloy (especially alloy which is pseudoelastic, above  $A_s$  at its utilization temperature) may be used to manufacture vascular clips, etc. The alloy element here acts as a constant force spring over a wide strain range (greater than conventional elastic metals), resulting in ease of use.

From the foregoing, it is clear that, in a situation where narrow temperature differences are available or preferable, as often is the case in medical applications, mechanically constrained shape change is a much more useful solution than heat actuated shape change. It offers a degree of control heat actuation does not, it offers easier alloy composition control, it eases mating part tolerance requirements, and it offers simple mechanical reversal at minimal stress levels, all without heating, cooling or insulation complications.

It will be obvious to those skilled in the art, having regard to this disclosure, that other variations on this invention beyond those specifically exemplified here, and other medical devices making use of stress-induced martensite, may be made. Such variations are, however, to be considered as coming within the scope of this invention as limited solely by the following claims. I claim:

- 1. A medical device which comprises:
- (a) an element for use within a human body or in such proximity to a human body that the device is substantially at human body temperature, the element comprising a shape memory alloy which displays a stressinduced martensite behavior at body temperature; and
- (b) a restraint holding the shape memory alloy element in a deformed configuration at a temperature less than the body temperature of the human for positioning the shape memory alloy element within or in proximity to the human body in its deformed configuration, the deformation occurring through the formation of stressinduced martensite;
- wherein the shape memory alloy element is sufficiently deformed that when the shape memory alloy element is at human body temperature removal of the restraint from the shape memory alloy element, without change in temperature of the device, releases at least a portion of the shape memory alloy element from its deformed configuration.

2. A device as claimed in claim 1, in which the restraint is hollow, and the shape memory alloy element is positioned at least partially within the restraint.

- 3. A device as claimed in claim 2, in which the restraint is a catheter.
- 4. A device as claimed in claim 2, in which the shape memory alloy element is an intrauterine contraceptive device.
- 5. A device as claimed in claim 2, in which the shape memory alloy element is a filter for a blood vessel.
- 6. A device as claimed in claim 1, in which the shape memory alloy element is tubular, and the restraint is positioned within the shape memory alloy element to deform it.

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7. A device as claimed in claim 2, in which the shape memory alloy element has a transverse dimension and a longitudinal dimension, and wherein the shape memory alloy element is deformed by its transverse dimension being reduced, and wherein the restraint prevents transverse expansion of the element.

8. The device of claim 1, wherein the shape memory alloy element is sufficiently deformed that removal of the restraint from the shape memory alloy releases at least a portion of the shape memory alloy element from its deformed configuration without change in state of the restraint.

9. A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising:

- (a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and
- (b) restraining means engaging and stressing the memory alloy element at a temperature less than the body temperature of a human and greater than the As of the alloy for positioning the memory alloy element within or in proximity to the mammalian body while the ²⁵ memory alloy element is in its deformed shape;
- wherein the alloy is selected so that removal of the restraining means from the memory alloy element at a temperature greater than the As of the alloy when the device is placed within or proximate to the human body, transforms at least a portion of the alloy from its stress-induced martensitic state so that the memory alloy element transforms from its deformed shape toward its unstressed shape, without any change in temperature of the restraining means or the memory alloy element being required for the transformation of the alloy.

**10.** A medical device for treatment of a mammalian body, the device comprising:

- (a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about human temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and
- (b) a hollow restraining member with the memory alloy element being within the restraining member, the 50 restraining member engaging and stressing the memory alloy element at a temperature less than the body temperature of the human and greater than the As of the alloy for positioning the memory alloy element within or in proximity to the mammalian body while the 55 memory alloy element is in its deformed shape;
- wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of 60 the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element. 65

11. The medical device of claim 10 wherein the restraining member is a tube and the memory alloy element is axially slidable within the tube, and wherein the memory alloy element is sufficiently long that relative axial movement between the tube and the memory alloy element extends at least a portion of the memory alloy element beyond the tube and thereby transforms the memory alloy element toward its austenitic state.

12. The device of claim 11 wherein the memory alloy element can be extruded completely out of the tube for deployment in the mammalian body.

13. A medical device for treatment of a mammalian body, the device comprising (i) a restraining member and (ii) a hollow memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

- the restraining member being within the hollow memory alloy element and engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;
- wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stress-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

14. The device of claim 13 wherein the memory alloy element is a tube and the restraining member is axially slidable within the tube, and wherein the tube is sufficiently long that relative axial movement between the tube and the restraining members extends at least a portion of the tube beyond the restraining means and thereby transforms the tube toward its austenitic shape.

15. A medical device for insertion into a human body, the device comprising (i) a restraining member and (ii) a hollow catheter formed at least party from a pseudoelastic shapememory alloy, the alloy displaying reversible stress-induced martensite at about human body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape when the alloy is in its austenitic state;

- the restraining member engaging and stressing the catheter at a temperature greater than the As of the alloy so that the catheter is in its easily inserted shape so that the catheter can be inserted into the mammalian body;
- wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from its stress-induced martensitic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the catheter.

16. The medical device of claim 15 wherein the catheter is a cannula.

17. A medical device for insertion into a human body, the device comprising (i) a straight pin and (ii) a catheter formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at

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about human body temperature such that it has a stressinduced martensitic state and an austenitic state, the catheter having (i) a straight shape when the alloy is in its stressinduced martensitic state and (ii) a curved unstressed shape when the alloy is in its austenitic state;

- the straight pin engaging and stressing the inside of the catheter at a temperature greater than the As of the alloy so that the catheter is in its straight shape;
- wherein withdrawal of the pin from the catheter at a temperature greater than the As of the alloy transforms 10 at least a portion of the alloy from its stress-induced martensitic state to its austenitic state so that the catheter transforms from its straight shape to its curved shape, and wherein the alloy is selected so that the transformation can occur without any change in tem- 15 perature of the pin or the catheter,

18. The medical device of claim 17 wherein the catheter is a tracheal insertion catheter.

19. The invention of claim 9 wherein the transformation of the alloy occurs without any change in the state of the restraining means.

20. The invention of claim 10, 13 or 15 wherein the transformation of the alloy occurs without any change in the state of the restraining member.

**21.** The invention of claim 17 wherein the transformation of the alloy occurs without any change in the state of the pin.  25 

22. The invention of claim 9, 10 or 13 wherein the memory alloy element exerts constant stress during its transformation.

23. The invention of claim 15 or 17 wherein the catheter exerts constant stress during its transformation. 30

24. The medical device of claim 9 wherein the removal of the restraining means from the memory alloy element causes at least a portion of the alloy to transform to its austenitic state.

25. The medical device of claim 10, 13, 15 wherein 35 relative movement of the restraining member and the memory alloy element causes at least a portion of the alloy to transform to its austenitic state.

26. The device of claim 24 wherein the alloy is selected so that engaging the restraining means with the memory 40 alloy element after removal results in the memory alloy element transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

27. The device of claim 25 wherein (i) the restraining 45 member can be completely disengaged and separated from the memory alloy element, and (ii) the alloy is selected so that engaging the restraining member with the memory alloy element after separation results in the memory alloy element transforming towards its deformed shape by reversion to its 50 stress-induced martensitic state.

28. The device of claim 15 wherein (i) the restraining member can be completely disengaged and separated from the catheter, and (ii) the alloy is selected so that reengaging the restraining member with the catheter after separation 55 results in the catheter transforming toward its easily inserted shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

29. The device of claim 17 wherein (i) the pin can be completely disengaged and separated from the catheter, and 60 (ii) the alloy is selected so that reengaging the restraining means with the memory alloy element after separation results in the catheter transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state. 65

30. The device of claim 36 wherein (i) the placement device can be completely disengaged and separated from the

catheter, and (ii) the alloy is selected so that reengaging the placement device with the catheter after separation results in the catheter transforming toward its deformed shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.

31. The medical device of claim 15 wherein the restraint externally engages the catheter.

32. The medical device of claim 15 wherein the restraint internally engages the catheter.

**33.** A medical device suitable for placement within or proximate to a mammalian body for treatment of the mammalian body, the device comprising (i) a restraint, and (ii) an element formed at least partly from a pseudoelastic shapememory alloy,

- the alloy displaying reversible stress-induced martensite by virtue of being above its  $A_s$  and above its  $M_s$  and below its  $M_d$  at about body temperature;
- such that it has a stress-induced martensitic state and an austenitic state, the element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;
- wherein the restraint is (i) stressing the element at a temperature less than the body temperature of the mammal for placement of the element in its deformed shape in or in proximity to the mammalian body and (ii) is capable of being at least partially removed from the element while the device is within or proximate to the body at the body temperature and the element is therefore at an operating temperature greater than the  $A_s$  and  $M_s$  and below the  $M_d$  of the alloy,
- such removal of the restraint causing at least a portion of the alloy to transform from its stress-induced martensitic state to its austenitic state so that the element spontaneously transforms from its deformed shape toward its unstressed shape.
- and such transformation can occur without a change in temperature of the restraint or of the element from the operating temperature.

34. A medical device for treatment of a human body, the device comprising:

- (a) a memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about human body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape; and
- (b) a hollow tubular restraining member with the memory alloy element being within the restraining member, the restraining member engaging and stressing the memory alloy element at a temperature greater than the As of the alloy so that the memory alloy element is in its deformed shape;
- wherein the memory alloy element is axially slidable within the tube, and wherein the memory alloy element can be extruded completely out of the tube for deployment in the mammalian body to transform at least a portion of the alloy from its stress-induced martensitic state towards its austenitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the memory alloy element.

35. A medical device which comprises:

- (a) a tubular element for use within a human body or in such proximity to a human body that the device is substantially at human body temperature, the tubular element comprising a shape memory alloy which displays stress-induced martensite behavior at body temperature; and
- (b) a restraint within the tubular element holding and deforming the tubular shape memory alloy element in a deformed configuration to allow it to be positioned ¹⁰ within or in proximity to a human body, the deformation occurring through the formation of stress-induced martensite;
- wherein the tubular element is sufficiently deformed that removal of the restraint from the tubular shape memory alloy element, without change in temperature of the device, releases at least a portion of the tubular element from its deformed configuration.

**36.** A medical device for treatment of a human body, the device comprising (i) a restraining member and (ii) a hollow memory alloy element formed at least partly from a pseudoelastic shape-memory alloy, the alloy displaying reversible stress-induced martensite at about human body temperature such that it has a stress-induced martensitic state and an austenitic state, the memory alloy element having (i) a deformed shape when the alloy is in its stress-induced martensitic state and (ii) a different unstressed shape;

- the restraining member being within the hollow memory alloy element and engaging and stressing the memory 30 alloy element at a temperature less than the body temperature of the human and greater than the As of the alloy for positioning the memory alloy element within or in proximity to the human body while the memory alloy element is in its deformed shape; 35
- wherein the restraining member and the memory alloy element are movable relative to each other to transform at least a portion of the alloy from its stressed-induced martensitic state at a temperature greater than the As of the alloy so that the memory alloy element transforms 40 from its deformed shape toward its unstressed shape, and wherein the alloy is selected so that the transformation can occur without any change in temperature of the restraining member of the memory alloy element.

37. The device of claim 36 wherein the memory alloy element is a tube and the restraining member is axially slidable within the tube, and wherein the tube is sufficiently long that relative axial movement between the tube and the restraining member extends at least a portion of the tube beyond the restraining member and thereby transforms the tube toward its austenitic shape.

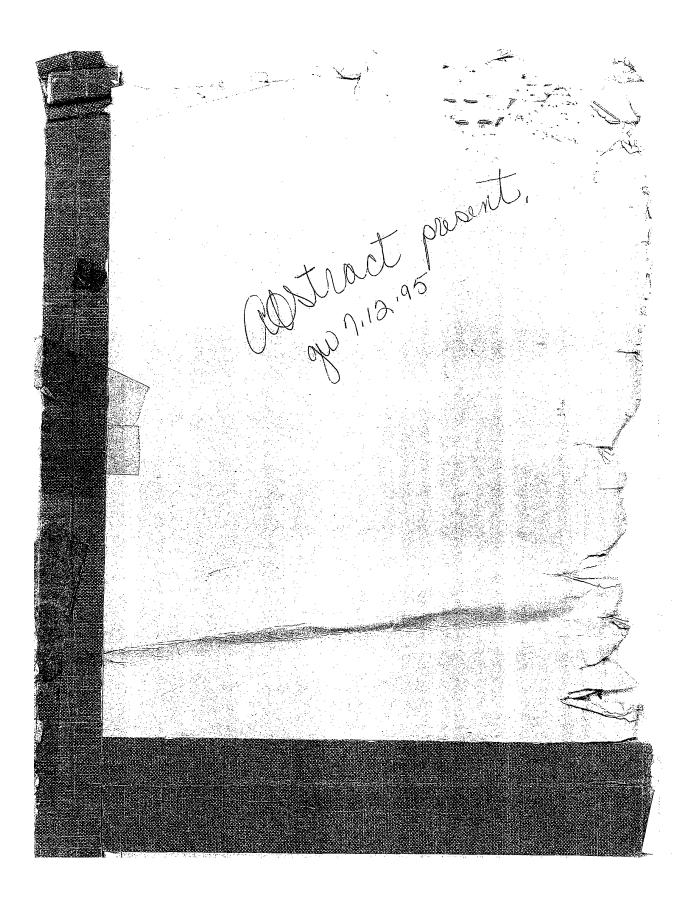
38. A medical device for insertion into a human body, the device comprising (i) a restraining member and (ii) a hollow catheter formed at least partly from a pseudoelastic shapememory allow, the alloy displaying reversible stress-induced martensite at about human body temperature such that it has a stress-induced martensitic state and an austenitic state, the catheter having (i) an easily inserted shape when the alloy is in its stress-induced martensitic state; the alloy is in its austenitic state;

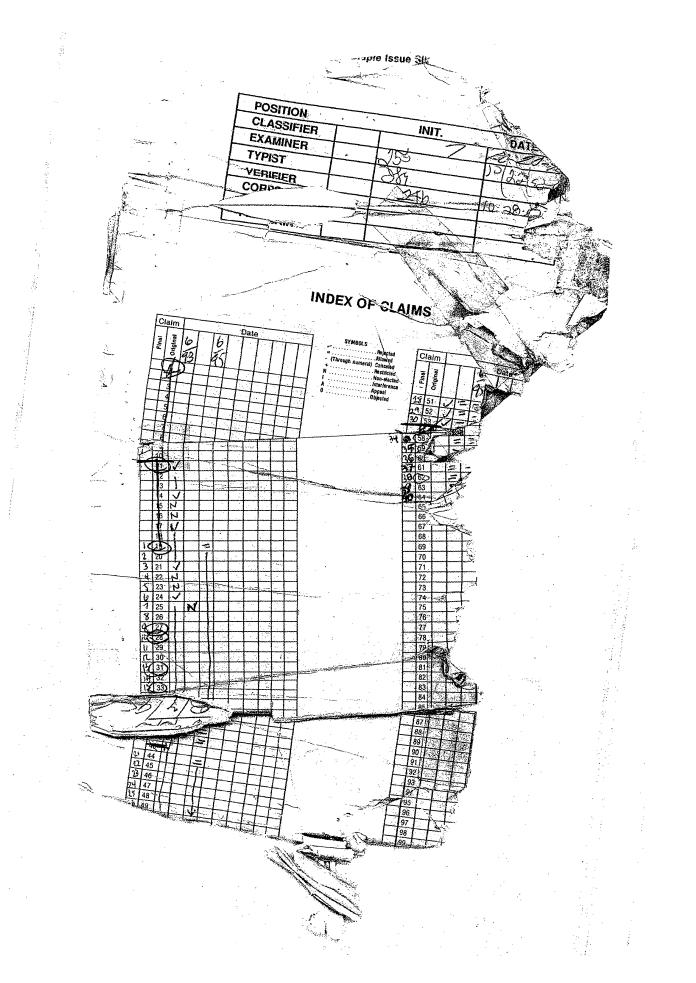
the restraining member engaging and stressing the catheter at a temperature less than the body temperature of the human and greater than the As of the alloy so that the catheter is in its easily inserted shape for inserting

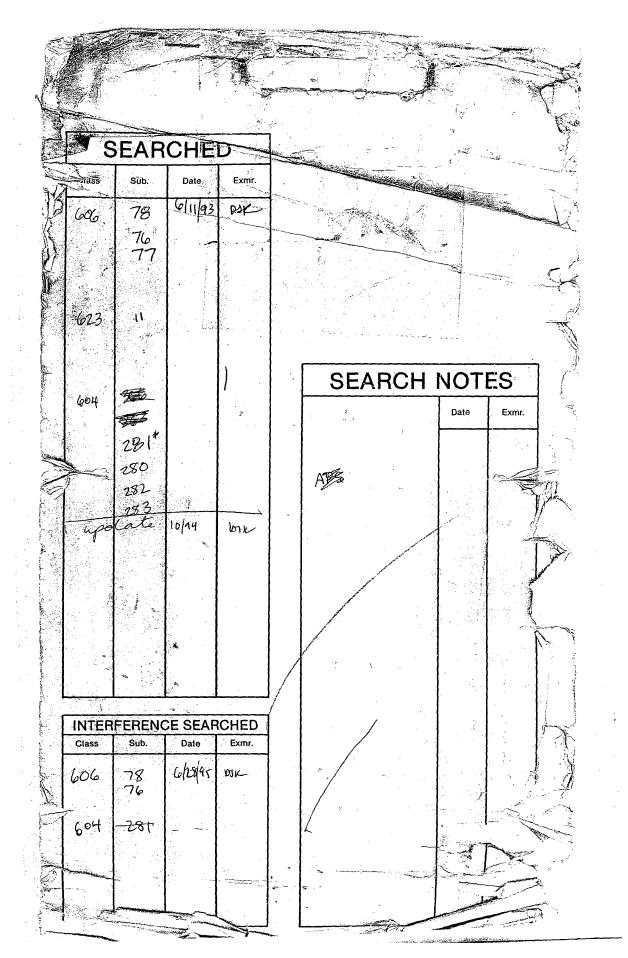
- the catheter into the human body; and wherein disengagement of the restraining member from the catheter at a temperature greater than the As of the alloy transforms at least a portion of the alloy from it stress-induced martenistic state to its austenitic state so that the catheter transforms from its easily inserted shape toward its unstressed shape, and wherein the
- alloy is selected so that the transformation can occur without any change in temperature of the restraining member or the catheter.

**39.** The invention of claim **34**, **36** or **37** wherein the transformation of the alloy occurs without any change in the state of the restraining member.

40. The device of claim 38 wherein the device is addpted so that (i) the restraining member can be completely disengaged and separated from the catheter, and (ii) re-engaging the restraining member with the catheter after separation results in the catheter transforming toward its easily inserted shape by reversion of at least a portion of the alloy from its austenitic state to its stress-induced martensitic state.







Edwards Exhibit 1033, p. 307

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